

Thermoelectric properties of $\text{Pb}_{1-x}\text{Sn}_x\text{Te}$ crystals doped with vanadium deep donors

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PbTe and SnTe as well as their substitutional alloys $\text{Pb}_{1-x}\text{Sn}_x\text{Te}$ are IV-VI semiconductor materials well known for topological crystalline insulator (TCI) properties as well as for their applications in mid-infrared devices and thermoelectric generators. Optimal application of these materials requires controlling of both type and concentration of conducting carriers: from almost semi-insulating bulk materials desired in TCIs to heavily doped ($n, p = 10^{19} - 10^{20} \text{ cm}^{-3}$) thermoelectric elements. In $\text{Pb}_{1-x}\text{Sn}_x\text{Te}$ the electrically active native defects (metal vacancies) determine very high background hole concentration ($p = 10^{18} - 10^{19} \text{ cm}^{-3}$). Controlling electrical properties requires doping with extrinsic donors (Bi, J) or acceptors (Na, Ag) and/or thermal annealing. Thermoelectric properties of $\text{Pb}_{1-x}\text{Sn}_x\text{Te}$ can also be controlled by incorporation of 3d transition metals, like Cr, Fe or V. The solubility of these transition metals in IV-VI semiconductors is limited to heavily doping regime (typically below 0.5 at.%) but they form a special class of resonant donor centers with the mixed valence $3+/2+$ charge state depending on Sn content and Fermi level position.

In this work, we examine $\text{Pb}_{1-x}\text{Sn}_x\text{Te}:\text{V}$ bulk crystals with nominal Sn content of $x=0.2$ and V content of 1 at.% grown by the Bridgman method under varying thermal regime and doping procedure. The crystals studied cover the composition range corresponding to the expected in-gap position of $\text{V}^{2+/3+}$ donor level [1]. Electron microscopy SEM-EDX analysis of chemical composition confirmed the Sn concentration smoothly varying along an ingot from $x=0.13$ to $x=0.3$ and an indication for very small V content (below 0.5 at.%) increasing towards the end of an ingot. To examine electrical properties, we measured the electrical conductivity and the Hall effect in the temperature range $T=5-300 \text{ K}$ revealing only p -type metallic conductivity with carrier concentration $p \approx 7 \cdot 10^{18} \text{ cm}^{-3}$ and resistivity ratio $\rho(300\text{K})/\rho(5\text{K}) \approx 10$ due to the temperature dependence of mobility. We also measured, at room temperature, the thermoelectric power α and determined the key applicational parameter - thermoelectric power factor $P_\sigma = \alpha^2/\rho \approx 8 \mu\text{W}/\text{K}^2\text{cm}$ (comparable to observed in good PbTe-based thermoelectric materials). No in-gap pinning of Fermi level by $\text{V}^{2+/3+}$ donors and semi-insulating state was experimentally found, presumably because only small part of V centers introduced into $\text{Pb}_{1-x}\text{Sn}_x\text{Te}$ (estimated 0.1 at.%) becomes electrically active. As a result, the electrical conductivity of our $\text{Pb}_{1-x}\text{Sn}_x\text{Te}:\text{V}$ bulk crystals still remains dominated by native defects. This balance can be changed by further doping with V or reduction of vacancies concentration by annealing.

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[1] E.P. Skipetrov et al., *Semiconductors* **46**, 741 (2012); *Low Temp. Phys.* **37**, 210 (2011).