51st International School & Conference on the Physics of Semiconductors



BOOK OF ABSTRACTS

JASZOWIEC 2023 Szczyrk, Poland 17 – 23 June 2023

ORGANIZED BY











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WELCOMING ADDRESS

The 51st International School & Conference on the Physics of Semiconductors "Jaszowiec 2023" begins the second half-century of this conference, so it follows the long tradition of annual meetings of Polish physicists working in the field of semiconductors, in a beautiful mountain scenery.

We have 15 exciting invited lectures ahead, which will be delivered by 12 foreign and 3 Polish scientists. We will also listen to 36 oral presentations, carefully selected by the Scientific Committee from three times more submissions. Lectures and oral presentations cover many areas of novel physics, such as Fourier optics, 2D materials and Moiré heterostructures, semiconductor qubit architecture, topological photonics and ultra-wide bandgap materials. They are accompanied by 149 poster presentations, where apart from papers on photonic materials and 2D structures, you can also find reports on topological insulators, thermoelectrics, magnetic structures, solar cells and IV-VI materials. The conference is preceded by the School, a tutorial session addressed to students and young scientists.

I would like to thank the Program and Advisory Committees for their support in shaping the scientific program. I would also like to express my gratitude to the members of the Organizing Committee for their time and enthusiasm. Without your tremendous efforts the organization of "Jaszowiec 2023" conference would not be possible.

I wish you a successful conference, exciting discussions, striking up many acquaintances and sharing new ideas. If, in addition to participating in a rich scientific program, you find time for at least one mountain hike, I will consider that the conference has fulfilled its role.

Conference Chair

Elżbieta Guziewicz

Elite auxience

CONFERENCE OFFICE

Conference office will be open:

10: 30 - 11:30, 16:30 - 18:00

Conference telephone number:

+48 502 962 261

CONFERENCE VENUE

HOTEL KLIMCZOK ul. Poziomkowa 20 43-370 Szczyrk Hotel reception: mobile: +48 669 260 101 phone: +48 (33) 82 60 100

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Program of the 51st International School & Conference on the Physics of Semiconductors "Jaszowiec 2023" Szczyrk, Poland, 17 – 23 June 2023

School – Special Tutorial Session

SATURDAY, 17 JUNE 2023

15:30 – 15:45	Tomasz Jakubczyk – School Opening
15:45 – 17:30	<u>Michał Boćkowski</u> (Institute of High Pressure Physics PAS, Poland) <i>Bulk GaN growth</i> – Sa1
17:30 – 18:00	Coffee break
18:00 – 19:45	Heike Riel (IBM Research Europe, Switzerland) Prospects and challenges of quantum computing – Sa2
19:45	Barbecue Dinner

SUNDAY, 18 JUNE 2023

20:00	Concert and Welcoming glass of wine
18:45 – 20:00	Dinner
17:30 –18:30	Tutorial Session Quiz and Closing
13:15 – 14:15	Lunch
11:30 – 13:15	<u>Tim Schröder</u> (Humboldt-Universität zu Berlin, Germany) Experimental Quantum Control of Spin Defects in Solids – Su2
11:00 – 11:30	Coffee break
9:15 – 11:00	<u>Jagoda Sławińska</u> (University of Groningen, Netherlands) Computational design of spintronics materials – Su1

<u>Conference</u>

MONDAY, 19 JUNE 2023

8:45 – 9:00	Elżbieta Guziewicz – Conference Opening
Chair: Toma	sz Dietl
9:00 – 9:45	David Norris (ETH Zurich, Switzerland) <i>Optical and Electronic Fourier Surfaces</i> – Mol1
9:45 – 10:30	<u>Alexander Högele</u> (LMU München, Germany) Excitons in moiré heterostructures – Mol2
10:30 – 10:45	Dennis Kudlacik , C. Harkort, N.E. Kopteva, D.R. Yakovlev, M. Karzel, E. Kirstein, O. Hordiichuk, M. Kovalenko, M. Bayer Spin-Flip Raman Scattering on Electrons and Holes in Two-Dimensional (PEA) ₂ Pbl ₄ Perovskites – MoO1
10:45 –11:15	Coffee break
Chair: Witolo	l Bardyszewski
11:15 – 12:00	Sebastian Klembt (Chair for Applied Physics, University of Würzburg, Germany) <i>Topological photonics and topological lasers</i> – MoI3
12:00 – 12:15	Helgi Sigurðsson, J.D. Toepfer, L. Pickup, S. Harrison, T. Cookson, Y. Wang, A. Askitopoulos, S. Alyatkin, P.G. Lagoudakis <i>Networks of liquid light</i> – MoO2
12:15 – 12:30	Michał Kobecki Giant Photo-Elasticity of the Superlattice Polaritons for Detection of Coherent Phonons – MoO3
12:30 – 12:45	Dąbrówka Biegańska, M. Pieczarka, C. Schneider, S. Hofling, S. Klembt, M. Syperek Anomalous dispersion and dissipative coupling in AlGaAs exciton-polariton structure – MoO4
12:45 – 13:00	Amir Rahmani, M. Kędziora, A. Opala, M. Matuszewski Non-Hermitian synthetic lattice with light-matter coupling – MoO5
13:00 – 14:30	Lunch

14:30 – 16:00	Monday Poster Session A TMDs, van der Waals crystals and solar cells	
17:00 – 18:00	Quantum Design Spectrographs – MoE1	
18:00 – 19:30	Dinner	
Chair: Maria Kamińska		
19:30 – 20:15	Panaiot Zotev (University of Sheffield, UK) Van der Waals materials for nanophotonic applications – Mol4	
20:15 – 20:30	Arka Karmakar, T. Kazimierczuk, I. Antoniazzi, M. Raczyński, T. Taniguchi, K. Watanabe, A. Babiński, A. Al-Mahboob, M.R. Molas <i>Excitation Dependent Energy Transfer in 2D Heterostructure</i> – MoO6	
20:30 – 20:45	Alessandro Surrente, J. Jasinski, S. Palai, M. Smiertka, F. Gucci, A. Genco, S. Dal Conte, K. Watanabe, T. Taniguchi, M. Baranowski, P. Płochocka Dexter induced inverted valley polarization in monolayer WSe ₂ – MoO7	
20:35 – 21:00	Maciej Molas, T. Woźniak, Ł. Kipczak, A. Babiński, S. Sun, C. Xu, W. Ren Hot luminescence or Raman scattering in monolayers of MoSi₂N₄ – MoO8	
21:00 – 22:30	Monday Poster Session B Topological insulators, magnetic structures & termoelectrics	

TUESDAY 20 JUNE 2023

Chair: David Norris

9:00 – 9:45	Łukasz Cywiński (Institute of Physics Polish Academy of Sciences, Poland) <i>Coherent shuttling of spins between silicon-based quantum</i> <i>dots</i> – Tul1
9:45 – 10:30	<u>Maciej Bieniek</u> (University of Würzburg, Germany) Spin valley qubits in gated quantum dots in 2D – Tul2
10:30 – 10:45	Sarath Prem, M.M. Wysokiński, M. Trif Longitudinal coupling between electrically driven spin-qubits and a resonator – TuO1
10:45 – 11:15	Coffee break
Chair: Jacek S	zczytko
11:15 – 12:00	Doris E. Reiter (TU Dortmund, Germany) <i>Optical control of solid-state quantum emitters</i> – TuI3
12:00 – 12:15	Anna Musiał, M. Wasiluk, K. Roszak, M. Gawełczyk, P. Holewa, P. Wyborski, P. Podemski, M. Burakowski, H. Salamon, P. Mrowiński, M. Mikulicz, T. Smołka, K. Posmyk, A. Zielińska, J.P. Reithmaier, M. Benyoucef, G. Sęk, W. Rudno-Rudziński <i>Epitaxial InAs/InP quantum dots as emitters of single photons</i> <i>and entangled photon pairs for fiber-based quantum</i> <i>communication</i> – TuO2
12:15 – 12:30	Marcin Syperek, P. Holewa, E. Zięba-Ostój, D.A. Vajner, T. Gao, M. Lach, M. Wasiluk, B. Gaal, A. Sakansas, M. Burakowski, P. Mrowiński, B. Krajnik, M. Xiong, A. Huck, K. Yvind, N. Gregersen, A. Musiał, T. Heindel, E. Semenova State-of-the-art Quantum Dot-based Single-photon Sources for Quantum Key Distribution in Silica Fiber Networks at 1550 nm – TuO3
12:30 – 12:45	Marek Burakowski, P. Mrowiński, P. Holewa, A. Sakanas, E. Semenova, G. Sęk, M. Syperek Heterogeneously Integrated InAs/InP Quantum Dot Single- photo n Emitter with the Si Platform for On-chip Quantum Photonics – TuO4
12:45 – 13:00	Kacper Oreszczuk , A. Rodek, L. Zinkiewicz, J. Howarth, T. Taniguchi, K. Watanabe, P. Kossacki <i>Electrically gated single photon emitters in monolayer</i> <i>transition metal dichalcogenides</i> – TuO5

13:00 – 14:30	Lunch
14:30 – 16:00	Tuesday Poster Session A Wide bandgap materials (h-BN, Ga2O3, ZnO, CdMgO, GaN)
17:00 – 18:00	Quantum Design Detectors – TuE1
18:00 – 19:30	Dinner
Chair: Doris R	leiter
19:30 – 19:45	Nilesh Dalla, P.K. Kulboka, P.S. Sojewski, K.O. Oreszczuk, T.K. Kazimierczuk, P.K. Kossacki, R.J.W. Warburton, T.J. Jakubczyk <i>Realization of a High Finesse Platform for Functionalization</i> <i>of Emitters</i> – TuO6
19:45 – 20:00	Mateusz Kędziora, K. Łempicka-Mirek, M. Król, L.De. Marco, A. Coriolano, L. Polimeno, I. Viola, M. Muszyński, P. Morawiak, R. Mazur, P. Kula, W. Piecek, P. Fita, D. Sanvitto, J. Szczytko, B. Piętka <i>Electrical switching of a chiral lasing from polariton</i> <i>condensate in a Rashba-Dresselhaus regime</i> – TuO7
20:00 – 20:15	Magdalena Furman, A. Opala, M. Król, R. Mirek, K. Tyszka, B. Seredyński, W. Pacuski, J. Szczytko, M. Matuszewski, B. Piętka Natural Exceptional Points Appearing in Semiconductor Microcavity Systems – TuO8
20:15 – 20:30	Maciej Pieczarka, M. Gębski, A.N. Piasecka, M. Wasiak, J.A. Lott, T. Czyszanowski Driven dissipative Bose Einstein condensation of photons in a VCSEL – TuO9
20:30 – 20:45	Andrzej Opala, P. Oliwa, M. Muszyński, B. Piętka, J. Szczytko, M. Matuszewski Topologically protected interface states induced by spin-orbit coupling in liquid crystal microcavity – TuO10
20:45 – 21:00	Przemysław Oliwa, W. Bardyszewski, B. Piętka, J. Szczytko Quantum Mechanical-like Approach with Effective Hamiltonians Based on Resonant States in Optical Cavities – TuO11
21:00 – 22:30	Tuesday Poster Session B Photonic structures and photodetectors

WEDNESDAY 21 JUNE 2023

Chair: Czesław Skierbiszewski

9:00 – 9:45	Bart Kuyken (Ghent University, Belgium) III-V-on-Silicon-Nitride Mode-Locked Lasers – Wel1
9:45 – 10:00	Rafał Zdunek, M. Chlipała, G. Muzioł, C. Skierbiszewski, J. Łusakowski, H. Turski <i>Recovering rectangular band profile in polar InGaN-based</i> <i>heterostructures utilizing junction field</i> – WeO1
10:00 – 10:15	Piotr Kruszewski, J. Plesiewicz, V.P. Markevich, P. Prystawko, S. Bulka, M. Halsall, I. Crowe, L. Sun, A.R. Peaker Deep-level traps in as-grown and electron-irradiated n-GaN layers grown by MOVPE on Ammono-GaN substrate – WeO2
10:15 – 10:30	Tomasz Fąs, D. Yavorskiy, E. Pruszyńska-Karbownik, M. Sawicka, J. Kacperski, J. Wróbel, P. Nowicki, A. Kazakov, J. Polaczyński, A. Feduniewicz-Żmuda, Cz. Skierbiszewski, D. Schiavon, P. Perlin, T. Czyszanowski, J. Suffczyński <i>Bound states in continuum confined in GaN based</i> <i>subwavelength gratings</i> – WeO3
10:30 – 10:45	Antonina Bieganowska, C. Schneider, S. Höfling, S. Klembt, M. Syperek, M. Pieczarka Dynamics and Effective Interactions of a Photonic Condensate in an Optical Trap – WeO 4
10:45 – 11:15	Coffee break
Chair: Zbigniew Wasilewski	
11:15 – 12:00	Masataka Higashiwaki (Osaka Metropolitan University, Japan) Advances and Prospects of Gallium Oxide Material and Device Technologies – Wel2
12:00 – 12:45	<u>Katharina Lorenz</u> (Instituto Superior Técnico, University of Lisbon, Portugal) <i>Radiation Effects in Wide Bandgap Materials</i> – WeI3
12:45 – 13:05	Dinesh Soares "ACS Omega": A High Quality, Global, Open Access Journal from the American Chemical Society – WeS1
13:05 – 14:30	Lunch
14:30 – 17:00	Nextnano Tutorial Session – WeE1

19:30 – 3:00 Ban

THURSDAY 22 JUNE 2023

Chair: Piotr Kossacki

10:00 – 10:45	<u>Mark Greenaway</u> (Loughborough University, UK) Magnetophonon resonances in graphene – ThI1
10:45 – 11:15	Coffee break
Chair: Paw	veł Machnikowski
11:15 – 12:00	Sylvain Ravets (CNRS, France) Kardar-Parisi-Zhang universality in a one-dimensional polariton condensate – ThI2
12:00 – 12:45	Daniel Wigger (Trinity College Dublin, Ireland) Time-resolved nonlinear optical spectroscopy of basic excitations in 2D materials – ThI3
12:45 – 13:00	Leszek Bryja , J. Andrzejewski, J. Debus, C.H. Ho, J. Jadczak The study of resonant Raman scattering in semiconducting layered GeS – ThO1
13:00 – 14:30	Lunch
14:30 – 16:00	Thursday Poster Session A Theoretical platform/ CdTe and PbTe-based structures/Si- based NWs
18:00 – 19:30	Dinner
Chair: Torr	nasz Story
19:30 – 19:45	Kajetan Fijałkowski, N. Liu, P. Mandal, S. Schreyeck, K. Brunner, C. Gould, L.W. Molenkamp <i>Macroscopic Quantum Tunneling of a Topological</i> <i>Ferromagnet</i> – ThO2
19:45 – 20:00	Md Shahin Alam, F. Tafti, C. Autieri, M. Matusiak Anomalous transverse response in topological magnet CeAlSi – ThO3
20:00 – 20:15	Paweł Sidorczak, W. Wołkanowicz, K. Gas, A. Kaleta, S. Gierałtowska, R. Minikayev, S. Kret, M. Sawicki, T. Wojtowicz, D. Wasik, M. Gryglas-Borysiewicz, K. Dybko Superconductivity in PbTe/SnTe semiconductor heterostructure: a candidate for spin-triplet superconductor –

20:15 – 20:30	Kacper P. Kluczyk, K. Gas, M.J. Grzybowski, M.A. Borysiewicz, T. Fąs, J. Suffczyński, E. Łusakowska, W. Wołkanowicz, J.Z. Domagała, P. Skupiński, K. Grasza, A. Mycielski, K. Vyborny, M. Baj, M. Sawicki, M. Gryglas-Borysiewicz <i>Peculiar magnetic phase in antiferromagnet MnTe</i> – ThO5
20:30 – 20:45	Michał Szot, P. Wojnar, J. Korczak, W. Zaleszczyk, L. Kowalczyk, S. Chusnutdinow, R. Minikayev, W. Wołkanowicz, T. Story, G. Karczewski <i>Photoluminescence studies of the futuristic SnSe</i> <i>semiconductor</i> – ThO6
20:45 – 21:00	Mujeeb Ahmad , W. Wołkanowicz, S. Dad, P. Dziawa, T. Wojtowicz, K. Dybko Investigating of the topological phase transition in Pb _{1-x} Sn _x Te topological crystalline insulators by using nonlinear Hall effect measurements – ThO7

21:00 – 22:30 On-line Poster Session

FRIDAY 23 JUNE 2023

Chair: Wojciech Pacuski

9:00 – 9:45	Johannes Binder (University of Warsaw, Poland) Epitaxial Hexagonal Boron Nitride – Growth, Properties and Applications – Frl1
9:45 – 10:00	Piotr Tatarczak, J. Iwański, J. Binder, A.K. Dąbrowska, M. Tokarczyk, R. Stępniewski, A. Wysmołek <i>h-BN bubbles – a step towards deterministic activation</i> <i>of single photon emission –</i> FrO1
10:00 – 10:15	Krzysztof Korona, A.K. Dąbrowska, J. Iwański, T. Korona, A. Reszka, J. Binder, R. Stępniewski, A. Wysmołek Nature of UV (3 – 5 eV) Emission from Hexagonal Boron Nitride – FrO2
10:15 – 10:30	Aleksandra Łopion, A. Bogucki, Z. Śnioch, K.E. Połczyńska, W. Pacuski, T. Kazimierczuk, A. Golnik, P. Kossacki <i>Optically Detected Magnetic Resonance of p- and n-type</i> <i>Doped (Cd,Mn)Te QWs</i> – FrO3
10:30 – 10:45	Magdalena Birowska Magnetic signatures in structural, electronic and optical properties of 2D MPX ₃ crystals – FrO4
10:45 – 11:15	Coffee break
Chair: Andrze	j Golnik
11:15 –12:00	<u>Mateusz Dyksik</u> (Wrocław University of Science and Technology,Poland) Magneto-spectroscopy of excitons in two-dimensional layered perovskites – FrI2
12:00 – 12:15	Andrzej Nowok, M. Baranowski, M. Dyksik, A. Surrente, M. Mączka, P. Płochocka <i>Temperature Dependence of Effective Masses in Three-</i> <i>dimensional Methylammonium Lead Trihalides</i> – FrO5
12:15 – 12:30	Natalia Zawadzka, M. Grzeszczyk, Z. Chen, M.I. Katsnelson, M.R. Molas, M. Koperski <i>Spin dimensionality in chromium trihalides</i> – FrO6
12:30 – 12:45	Elżbieta Guziewicz - Closing conference
12:45 – 14:00	Lunch

MONDAY POSTER SESSION A

TMDs, van der Waals crystals and solar cells

14:30 - 16:00

Automated Search For TMDs Monolayers On Exfoliated Samples W. Kolesiński, M. Goryca
<i>Electrical Tuning Of exciton lifetime in monolayer MoS</i> 2 P. H. Lankappa, P. Mondal, A. Bid,J K. Basu
Fabrication and Strain Engineering of Transition Metal Dichalcogenides Monolayers on GaAs Nanomembranes E. Cybula, J. Jasiński, S. Palai, M. Śmiertka, A. Balgarkashi, V. Piazza, A. Fontcuberta i Morral, A. Surrente, M. Baranowski, A. Castellanos-Gomez, P. Płochocka
Controlled coherent-coupling and dynamics of exciton complexes in a MoSe ₂ monolayer A. Rodek, J. Howarth, T. Hahn, T. Taniguchi, K. Watanabe, M. Potemski, P. Kossacki, D. Wigger, J. Kasprzak
Comparison of optical properties of MoSe₂ monolayers; samples grown by MBE technique vs mechanically exfoliated one M. Raczyński, W. Pacuski, T. Kazimierczuk, P. Kossacki
Electron-phonon coupling in a WSe₂ monolayer unveiled by two-photon photoluminescence excitation spectroscopy M. Śmiertka, E. Cybula, K. Zalewska, K. Watanabe, T. Taniguchi, M. Dyksik, A. Surrente, M. Baranowski, P. Płochocka
Electrical Control of Interlayer Physics in Type-II TMD Heterostructures K. Sadecka, M. Bieniek, J. Pawłowski, A. Wójs, P. Hawrylak
Searching for moiré-induced quantum spin liquid - the DMRG and ED study of twisted TMD hetero-bilayers W.K. Pasek, J. Jastrzębski, M. Kupczyński, P. Potasz
Exploring neutral and charged exciton optical properties in a MoTe ₂ /SiO ₂ /Si-based metal-oxide-semiconductor device E. Zięba-Ostój, E. Rogowicz, A. Paralikis, C. Piccinini, B. Munkhbat, M. Syperek
Optimization of Contacts for van der Waals heterostructures by AFM ironing A. Zielińska, J. Eroms, D. Weiss, M. Ciorga
Computational prediction of structure and electronic properties of monolayer and few-layer 2D polyimide covalent organic frameworks M. Wlazło, A. Nagai

MoPA12	Tunable valley splitting in 2D MPX₃ crystals. M. Rybak, M. Birowska
MoPA13	Optical markers of magnetic phase transition in layered CrSBr M. Rybak, M. Birowska
MoPA14	Interplay between stacking and magnetic configurations in MPS₃ bilayers A.M. Leon, B. Guedes, T. Heine, T. Brumme
MoPA16	Exploring structural, electronic, and magnetic properties of 2D Cr, Fe, and Zr monoborides I.M. Arias Camacho, N. Gonzalez-Szwacki
MoPA17	Magnetic skyrmions of Ti2C MXenes doped with Cr, Mn, and Fe T. Kulka, J. Majewski
MoPA18	The effect of temperature and excitation energy on Raman scattering in bulk HfS2 I. Antoniazzi, N. Zawadzka, M. Grzeszczyk, T. Woźniak, J. Ibáñez, Z. Muhammad, W. Zhao, M.R. Molas, A. Babiński
MoPA19	Growth and characterization of metallic and oxidized forms of bismuthene K. Toczek, M. Rogala, P. Krukowski, M. Piskorski, M. Le Ster, I. Lutsyk, W. Kozłowski, P. Dąbrowski, R. Dunal, A. Nadolska, P. Przybysz, W. Ryś, K. Szałowski, D.A. Kowalczyk, P.J. Kowalczyk
MoPA20	Transport signatures of Van Hove singularities in mesoscopic twisted bilayer graphene A. Sanjuan Ciepielewski, J. Tworzydło, T. Hyart, A. Lau
MoPA22	<i>MBE grown heterostructures of ZnSe, CdSe, MoSe₂, and hBN</i> B. Tronowicz, J. Kucharek, R. Bożek, T. Taniguchi, K. Watanabe, W. Pacuski
MoPA23	Structural, Optical and Electrical Properites of Quaternary Hafnium-Aluminium Zinc Oxide Films Grown by Atomic Layer Deposition for Transparent Conductive Oxide Applications M. Krajewski, M. Tokarczyk, P. Wróbel, A. Drabińska, M. Kamińska
MoPA24	ALD Grown ZnMgO:Al on Si for Photovoltaic Applications: Effect of High Mg Alloying and Al Doping R. Schifano, J. Kurek, S. Gierałtowska, L. Wachnicki, U. Rehman, D. Budiakivska, S. Chusnutdinow, K. Kopalko, R. Minikayew, B. Witkowski, M. Godlewski, M. Pawłowski, C. Jastrzebski
MoPA25	Construction and studies of a cathode for photoelectrocatalytic hydrogen generation by water splitting A. Bohdan, K. Korona, M. Krajewski, J. Kucharek, W. Pacuski, M. Tokarczyk, P. Podsadni, A. Czewriński, K. Zarębska, M. Skompska, M. Kamińska

MoPA26	Degradation studies of MAPbl₃ perovskite using cathodoluminescence and microscopy techniques S. Piotrowska, A. Reszka, A. Wincukiewicz, M. Tokarczyk, B.J. Kowalski, M. Kamińska
MoPA27	Polymer solar cells thermal annealing of active layers – impact of annealing temperature on efficiency and ageing rate W. Mech, M. Kamińska, K.P. Korona
MoPA28	Towards Electrostatically Gated Perovskite Thin Films A. Szoła, W. Kolesiński, M. Kędziora, M. Goryca
MoPA29	The structure and properties of bilayer borophene S. Rakshit, N. Gonzalez – Szwacki
ΜοΡΑ30	<i>Towards Strain Engineering of High Quality 2D Perovskite Flakes</i> K. Zalewska, K. Posmyk, M. Śmiertka, M. Mączka,K. Watanabe, T. Taniguchi, M. Dyksik, M. Baranowski,P. Płochocka, A. Surrente

MONDAY POSTER SESSION B

Topological insulators, magnetic structures & termoelectrics 21:00 - 22:30

MoPB1	Surface and Structural Analysis of (110)-oriented Pb₁-xSnxTe Topological Crystalline Insulator S. Dad, P. Dziawa, M. Wojcik, S. Kret, J.Z. Domagala, E. Łusakowska P. Wojnar, J. Sadowski
MoPB2	Ferromagnetic topological crystalline insulator Sn₁ _* Mn _x Te in an inhomogeneous magnetic field A. Kazakov, V.V. Volobuev, M. Chojnacki, T. Wojciechowski, T. Wojtowicz
MoPB3	Signatures of Weyl Fermions in Magnetotransport of Topological Semimetal α-Sn J. Polaczyński, A. Kazakov, W. Zaleszczyk, B. Turowski, C-W. Cho, B.A. Piot, R. Rudniewski, T. Wojciechowski, T. Wojtowicz, V.V. Volobuev
MoPB4	Magnetic Circular Dichroism from Exchange Splitting in Intrinsic Magnetic Topological Insulator MnBi₂Te₄ SK. Bac, F. Le Mardelé, J. Wang, L. Riney, K. Yoshimura, I. Mohelsky, M. Ozerov, M. Zhukovskyi, T. Orlova, M. Orlita, X. Liu, B.A. Assaf
MoPB5	Non-trivial topological phases in transition metal rich half-Heusler Oxides B.R. Dhori, R.M. Sattigeri, P.K. Jha
MoPB6	Topological phase transition in SnTe topological crystalline insulator thin films S. Samadi, R. Rechcinski, R. Buczko
MoPB7	Scanning Gate Microscopy Probing of Electron-Hole Interference in a Normal-Superconductor Junction S. Maji, K. Sowa, M. Nowak
MoPB8	Non-local Transport Signatures of Topological Superconductivity in Planar Josephson Junctions D. Kuiri, M. P. Nowak
MoPB9	Quantum Anomalous Hall effect and axion insulator phase in HgTe material class C. Autieri
MoPB10	Geometry of the (001) surface and electron dispersion relations of topological IV-VI semiconductors A. Łusakowski, P. Bogusławski, T. Story
MoPB11	<i>Magnetic and electrical properties of CoRE₂W₂O₁₀ ceramic materials B. Sawicki, E. Tomaszewicz, T. Groń, M. Oboz, I. Gruszka, A. Guzik, P. Urbanowicz</i>
MoPB12	<i>Magnetic properties of semiconducting ZnCr₂Se₄:Re single crystals</i> I. Jendrzejewska, T. Groń, B. Sawicki, J. Kusz, J. Goraus, Z. Stokłosa

MoPB13	<i>Magnetic and electrical properties of CuCr₂Se₄ nanoparticles</i> E. Malicka, T. Groń, A. Gudwański, B. Sawicki, M. Oboz, M. Karolus
MoPB14	Magnetic and electric properties of Nd ³⁺ and Mn ²⁺ -co-doped calcium molybdato-tungstate single crystals T. Groń, E. Tomaszewicz, B. Sawicki, M. Oboz, J. Kusz, M. Berkowski
MoPB15	High-Quality MBE-grown Aluminum Layers for Experimental Realization of Majorana Bound States A. Elbaroudy, B. Khromets, E. Bergeron, T. Blaikie, Y. Shi, M.C. Tam, S. Sadeghi, F. Sfigakis, J. Baugh, Z.R. Wasilewski
MoPB16	<i>Majorana-Magnon Interactions in Topological Shiba Chains</i> P. Shen, V. Perrin, P. Simon, M. Trif
MoPB17	Temperature Dependence of Spin Pumping in Y3Fe5O12/Pt, Ni81Fe19/Pt and SnTe/Ni81Fe19 Thin Films A. Moosarikandy, V.S. Bhat
MoPB18	Size and curvature modulation of the magnetic properties of magnetite nanoshells B.C. Camargo, B. Semenenko, M. Gorke, M. Barasinski, G. Garnweitner, J. Szczytko
MoPB19	Towards Spintronics of Antiferromagnetic Semiconductors - Characterization of MnSe on GaAs C.J. Krasucki, J.Z. Domagala, M. Gryglas-Borysiewicz, W. Pacuski, M.J. Grzybowski
MoPB20	<i>Raman spectroscopy as an indicator of antiferromagnetic to paramagnetic phase transition in transition metal thiophosphates M. Peter, J. Kopaczek, M. Rybak, Z. Sofer, R. Kudrawiec</i>
MoPB21	Resonant Raman scattering of few-layers CrBr₃ Ł. Kipczak, A. Karmakar, M. Grzeszczyk, J. Pawłowski, A. Babiński, M. Koperski, M.R. Molas
MoPB22	Nagaoka ferromagnetism in phosphorene quantum dots T. Thakur, B. Szafran
MoPB23	<i>Magneto-phonon interaction in layered Cr₂Ge₂Te₀ ferromagnet</i> G. Krasucki, K. Olkowska-Pucko, M.I. Sturza, H. Kohlmann, A. Babiński, M.R. Molas
MoPB24	Investigating the Proximity Effect in h-BN/ 2D Magnet Heterostructures K. Ludwiczak, J. Binder, A.K. Dąbrowska, P. Tatarczak, J. Siticka, J. Jasiński, R. Stępniewski, A. Wysmołek
MoPB25	Thermoelectric properties of Pb _{1-x} Sn _x Te crystals doped with vanadium deep donors T. Andrearczyk, A. Królicka, J. Korczak, M. Szot, R. Minikayev, P. Dziawa, E. Łusakowska, A. Szczerbakow, K. Dybko, T. Story

MoPB26	Effect of Ni and Mn dopant on thermoelectric power generation performance of ZnO nanostructures synthesized via hydrothermal method U. ur Rehman, K. Mahmood, A. Ashfaq, M. F. Iqbal, K. ul Sahar, A. A. Khan, E. Hussain
MoPB27	Outstanding Thermoelectric Properties (ZT ≈ 5 – 6) of Functionalized 2D Molybdenum Nitrides (MXenes) J. Kołodziejczyk, J.A. Majewski
MoPB28	Photoemission study of the thermoelectric group IV-VI van der Waals crystals (GeS, SnS, and SnSe) A.K. Tołłoczko, S.J. Zelewski, M. Błaszczak, N. Olszowska, M. Rosmus, S. Tongay, R. Kudrawiec
MoPB29	Extension of rapid estimation of the flake thickness by color for a magnetic van der Waals 2D material R. Komar, M. Raczyński, T. Kazimierczuk, P. Kossacki

TUESDAY POSTER SESSION A

Wide bandgap materials (h-BN, Ga₂O₃, ZnO, CdMgO, GaN) 14:30 - 16:00

TuPA1	Epitaxial Boron Nitride Processing and Conductivity J. Rogoża, B. Furtak, J. Binder, A.K. Dąbrowska, R. Stępniewski, A. Wysmołek
TuPA2	Direct MOVPE growth of high-quality h-BN on the wafer-scale: the role of substrate off-cut M. Tokarczyk, A.K. Dąbrowska, G. Kowalski, R. Bożek, J. Iwański J. Binder, R. Stępniewski, A. Wysmołek
TuPA3	<i>Manipulating hBN Bandgap Properties With Aluminum</i> J. Iwański, M. Tokarczyk, K.P. Korona, P. Tatarczak, J. Pawłowski, A.K. Dąbrowska, J. Binder, R. Stępniewski, A. Wysmołek
TuPA4	<i>MBE growth of CdTe on hBN</i> A. Szczerba, J. Kucharek, J. Pawłowski, T. Taniguchi, K. Watanabe, W. Pacuski
TuPA5	Carbon-enriched Epitaxial Boron Nitride Layers as Contact Material for h-BN Devices B. Furtak, A. Wysmołek, J. Binder
TuPA6	Determining Strain Components in a Diamond Waveguide from Asymmetric ODMR Spectra of NV- Center Ensembles M.S. Alam, D. Wigger, M. Gawełczyk, P. Machnikowski
TuPA7	Post-implantation Defect Accumulation in Crystal Lattice of β-Ga2O3 Implanted with Yb ion M. Sarwar, R. Ratajczak, E. Guziewicz
TuPA8	The structural properties anisotropy of beta-Ga₂O₃ implanted with Yb R. Ratajczak, M. Sarwar, C. Mieszczynski, P. Jozwik, W. Wozniak, U. Kentsch, R. Heller, E. Guziewicz
TuPA9	Electron Paramagnetic Resonance of Beta-Gallium Oxide Modified by High-Energy Electron Irradiation A. Wołoś, J. Sitnicka, R. Grasset, M. Konczykowski
TuPA10	Effects of Si/N Ion Implantation on Defect formation and Doping of Gallium Oxide I.N. Demchenko, Y. Syryanyy, A. Shokri, Y. Melikhov, M. Chernyshova
TuPA11	PA-MBE Grown CdMgO:Eu Ternary Alloy on m- and c-Oriented Al₂O₃ Substrates J.K. Jadoon, A. Adhikari, A. Lysak, A. Wierzbicka, W. Lisowski M. Stachowicz, A. Kozanecki, E. Przeździecka

TuPA12	MBE growth of wide band gap wurtzite MgZnO superlattices doped with Eu for effective visible emission M. Stachowicz, J.A. Mathew, J.M. Sajkowski, R. Jakiela, Y. Zhydachevskyy, S. Magalhaes, E. Alves, A. Kozanecki
TuPA13	Initial optimization of the growth conditions of GaAs homo-epitaxial layers after cleaning and restarting the MBE reactor. D. Jarosz, M. Stachowicz, P. Krzemiński, M. Ruszala, A. Juś, P. Śliż D. Płoch, M. Marchewka
TuPA14	Geometrical selection during MBE growth of nanowires on polycrystalline substrates K. Olszewski, M. Sobańska, A. Wierzbicka, Z.R. Żytkiewicz
TuPA15	Insights into Raman spectra and doping concentration profiles of Eu-doped {ZnCdO/ZnO} superlattices I. Perlikowski, E. Zielony, A. Lysak, E. Przeździecka
TuPA16	Electrical and thermal transport properties of CdO and CdMgO alloys grown using plasma-assisted MBE technique A. Adhikari, Z. Adamus, M. Pawlak, A. Lysak, E. Przeździecka
TuPA17	Characterization of MBE grown {Zn(Mg)O/ZnCdO}m superlattices doped in-situ with Eu A. Lysak, E. Przeździecka, A. Wierzbicka, A. Reszka, M. Stachowicz, R. Jakiela, P. Dłużewski, A. Adhikari, A. Kozanecki
TuPA18	n-ZnO/ZnCdO/p-Si and n-ZnCdO/ZnO/p-Si diodes: studies on the influence of the junction interlayer on electrical properties and structural defects R. Szymon, E. Zielony, M.A. Pietrzyk, A. Lysak
TuPA19	An insight into the optical properties of GaN nanowires with Al₂O₃ and HfO₂ shells R. Szymon, E. Zielony, M. Sobańska, A. Wierzbicka,A. Reszka, S. Gieraltowska, Z.R. Żytkiewicz
TuPA20	Effect of strain and surface proximity on the acceptor grouping in ZnO:N O. Volnianska, S. Mishra, E. Guziewicz
TuPA21	Tailoring of the optical and electrical characteristics of ZnO:Ga heterostructures by a metallic buffer layer A. Hassan, A.A. Khan, M. Azam, U. Farooq, M. Zubair, Y. Cao
TuPA22	<i>Ferromagnetic Resonance Studies of (Ga,Mn)N</i> Y.K. Edathumkandy, K. Gas, D. Sztenkiel, K. Das, D. Hommel, H. Przybylińska, M. Sawicki
TuPA23	Impact of current flow direction on the distribution of carriers in multiple, color-coded InGaN quantum wells M. Chlipała, H. Turski, K. Nowakowski-Szudlarek, A. Feduniewicz-Żmuda, C. Skierbiszewski

TuPA24	Extreme InGaN growth conditions by plasma assisted molecular beam epitaxy M. Siekacz, M. Żak, K. Gołyga, M. Kryśko, A. Feduniewicz-Żmuda, H. Turski, C. Skierbiszewski, G. Muziol
TuPA25	Doping inhomogeneity at the nanoscale in GaN:Si studied by electrochemical etching A. Feduniewicz-Żmuda, M. Siekacz, N. Fiuczek, O. Gołyga M. Sawicka, K. Sobczak, C. Skierbiszewski
TuPA26	Visible luminescence of BGaN based structures: towards defects related LEDs M. Guziewicz, E.B. Możdżyńska, P. Ciepielewski, E. Dumiszewska, B. Stańczyk, M. Wzorek, K. Kościewicz
TuPA27	Modification of luminescence of the nitride nanowires by chemical treatment of their surface B.J. Kowalski, A. Reszka, M. Klepka, A. Wolska, M. Sobańska, Z.R. Żytkiewicz
TuPA28	Strain relation in GaN nanowires with HfO ₂ and Al ₂ O ₃ shells examined by X-ray diffraction and Raman spectroscopy techniques A. Wierzbicka, R. Szymon, E. Zielony, A. Reszka, M. Sobańska, S. Gieraltowska, P. Sybilski, Z.R. Żytkiewicz
TuPA29	Investigation of strain relation and lattice dynamics in {CdO/MgO} superlattices grown by plasma-assisted molecular beam epitaxy A. Wierzbicka, E. Przeździecka, I. Perlikowski, E. Zielony
TuPA30	Band parameters of group IV semiconductors in wurtzite structure J. Ziembicki, P. Scharoch, M.P. Polak, M. Wiśniewski, R. Kudrawiec

TUESDAY POSTER SESSION B

Photonic structures and photodetectors 21:00 - 22:30

TuPB1	Merons in Reciprocal Space as a Result of Photonic Spin-Orbit Interaction in Birefringent Microcavity M. Popławska, P. Oliwa, M. Król, W. Bardyszewski, B. Piętka, J. Szczytko
TuPB2	Numerical Approach to Determine the Fourier Plane of Light Propagation Through Cartesian-Oval Shaped Microlenses Z. Werner, J. Lewandowski, M. Popławska, M. Furman, P. Oliwa, A. Bogucki, Ł. Zinkiewicz, B. Seredyński, W. Pacuski, J. Szczytko, B. Piętka
TuPB3	Localization of light on defects in microcavities with built-in uniform lying helix J. Czapiński, M. Muszyński, E. Otón, P. Oliwa, R. Mazur, P. Morawiak, W. Piecek, P. Kula, B. Piętka, J. Szczytko
TuPB4	Self-Organizing Photonic Potential with Tunable Band Structures Coupled by Spin-Orbit Interaction in Birefringent Optical Cavity M. Muszyński, P. Oliwa, E. Oton, R. Mazur, P. Morawiak, W. Piecek, P. Kula, B. Piętka, J. Szczytko
TuPB5	Flow of an exciton-polariton condensate in optical microcavities with the natural structural disorder S. Świerczewski, R. Mirek, M. Król, M. Furman, M. Matuszewski, A. Opala, J. Szczytko, B. Piętka
TuPB6	Experimental Evaluation of the Indistinguishability of Single Photons at 1.55 μm Generated by InAs(P)/InP Quantum Dots D.A. Vajner, M. Wasiluk, P. Holewa, E. Zięba-Ostój, B. Gaál, A. Sakanas, B. Krajnik, M. Xiong, K. Yvind, N. Gregersen, M. Syperek, E. Semenova, A. Musiał, T. Heindel
TuPB7	Magneto-Optical Properties of Single Symmetric InAs/InP Quantum Dots Emitting in the Telecom C-band M. Wasiluk, A. Musiał, M. Burakowski, J.P. Reithmaier, M. Benyoucef, W. Rudno-Rudziński
TuPB8	Optimization of on-chip photonic structures for a hybrid InP/Si platform via deep learning approach P. Mrowiński, G. Sęk, M. Syperek
TuPB9	Optical Characteristics of Monolithic High-Contrast Gratings for Quantum-Dot Vertical-Cavity Surface-Emitting Lasers at 940 nm. B. Kamiński, J. Boniecki, M. Janczak, N. Heermeir, F. Laudani, S. Rodt, S. Reitzenstein, M. Gębski, T. Czyszanowski, G. Sęk, A. Musiał

TuPB10	Methods for Determining High Finesse in Open Cavity Systems P. Kulboka, N. Dalla, T. Kazimierczuk, P. Kossacki, T. Jakubczyk
TuPB11	Reflectivity spectra of GaAs/AlAs Distributed Bragg Reflector in the MIR Calculated by the Transfer Matrix Method H. Janowska, A. Zielińska, A. Musiał, G. Sęk
TuPB12	Optimization of the polarization state of an ensemble of quantum dots for chiral emission J. Rosiński, M. Gawełczyk, K. Tarnowski, D. Wigger, P. Karwat, P. Machnikowski
TuPB13	Controlled coherent coupling in a quantum dot molecule measured with nonlinear spectroscopy D.W. Wigger, J.S. Schall, M.D. Deconinck, N. Bart, P.M. Mrowinski, M.K. Krzykowski, K.G. Gawarecki, M.H. von Helversen, R.S. Schmidt, L. Bremer, F. Bopp, D.R. Reuter, A.W. Wieck, S.R. Rodt, J.R. Renard, G.N. Nogues, A.L. Ludwig, P.M. Machnikowski, J.F. Finley, S.R. Reitzenstein, J.K. Kasprzak
TuPB14	Diffusion of quantum well excitons measured with nonlinear spectroscopy K.P. Połczyńska, M.R. Raczynski, Z.S. Snioch, G.N. Nogues, W.L. Langbein, W.P. Pacuski, P.K. Kossacki, J.K. Kasprzak
TuPB15	Coherent imaging and dynamics of exciton complexes in MoSe ₂ monolayers epitaxially grown on a hexagonal boron nitride K.E. Połczyńska, S.L. Denmat, T. Taniguchi, K. Watanabe, M. Potemski, P. Kossacki, W. Pacuski, J. Kasprzak
TuPB16	Probing the Local Temperature Distribution in Electrically Pumped Broad-Area VCSELs A.N. Piasecka, M. Gębski, M. Wasiak, J.A. Lott, T. Czyszanowski, M. Pieczarka
TuPB17	Modeling InAs/GaSb Type-II Superlattices for Mid-Wavelength Infrared Photodetectors with the nextnano++ Software H.S. Mączko
TuPB18	Applications of InAs/InAs _{0.625} Sb _{0.375} Superlattice in Far-infrared Detection G. Hussain, G. Cuono, C. Autieri, T. Dietl
TuPB19	<i>Magnetic properties of TRS broken ferrovalley semiconductors</i> G. Hussain, G. Cuono, C. Autieri
TuPB20	Bulk And Contact Low Frequency Noise Investigated With Transmission-Line-Model: InAs case Ł. Ciura, J. Wróbel, P. Martyniuk

TuPB21	<i>Optical characterization and pump-probe measurements of InAs/GaSb and InAs/InAsSb type-II superlattices</i> M. Rygała, A. Bader, T. Smołka, F. Hartmann, G. Sęk, S. Höfling, M. Motyka
TuPB22	Modelling of Band Structure and Optical Properties of InAsP/InP Quantum Dots in Nanowires J. Boniecki, G. Bucci, V. Zannier, L. Sorba, A. Musiał, G. Sęk
TuPB23	Temperature Dependence of Refractive Indices of In _{0.53} Ga _{0.47} As and InP in the Mid-Infrared Spectra Range Determined by Fourier Transform Spectroscopy M. Mikulicz, M. Rygała, T. Smołka, M. Badura, W. Kijaszek, A. Łozińska, D. Radziewicz, B. Ściana, M. Motyka
TuPB24	<i>Carrier dynamics of type-II quantum wells emitting in mid-infrared</i> T. Smołka, M. Rygała, K. Ryczko, A. Bader, F. Hartmann, B. Petrovic, S. Höffling, M. Motyka
TuPB25	Influence of Mg ²⁺ codoping on excitation relaxation and emission properties of (Lu,Gd) ₃ (Ga,Al) ₅ O ₁₂ :Ce,Mg scintillators A. Solovjovas, S. Nargelas, M. Kučera, G. Tamulaitis
TuPB26	Characterization of the Microwave Strip Antenna in the Experiment of Optically Detected Magnetic Resonance Z. Śnioch, A. Łopion, A. Bogucki, K.E. Połczyńska, W. Pacuski, T. Kazimierczuk, A. Golnik, P. Kossacki
TuPB27	Hybrid Acousto-Optic Driving of Rabi Oscillations of Electron Spin in a Quantum Dot M. Kuniej, M. Gawełczyk, P. Machnikowski
TuPB28	<i>Noise effects on the resonance fluorescence from acoustically modulated quantum dots</i> R.A. Bogaczewicz, P. Machnikowski
TuPB29	Energy gap determination from photoacoustic spectroscopy measurements K. Misztal, J. Kopaczek, R. Kudrawiec
TuPB30	Magnetic field and temperature dependence of the Mn ²⁺ spin relaxation rate in a (Cd, Mn)Te/(Cd, Mg)Te quantum well A. Bogucki, Z. Śnioch, A. Łopion, K.E. Połczyńska, W. Pacuski, T. Kazimierczuk, A. Golnik, P. Kossacki

THURSDAY POSTER SESSION A

Theoretical platform/ CdTe and PbTe-based structures/Si-based NWs 14:30 - 16:00

ThPA1	Quantum anomalous Hall insulator in ionic Rashba lattice of correlated electrons M.M. Wysokiński, W. Brzezicki
ThPA2	Exciton Diffusion in an Ensemble of Self-Assembled Semiconductor Quantum Dots K. Kawa, P. Machnikowski
ThPA3	<i>Introducing the step Monte Carlo method for simulating dynamic properties</i> D. Sztenkiel
ThPA4	Novel Te-bearing half-Heusler phases and their potential thermoelectric performance K. Bilińska, M.J. Winiarski
ThPA5	Optimizing the cap thickness in fabrication of quantum dot photonic structures by focused ion beam M.J. Jaworski, A.C. Chudzyńska, P.M. Mrowiński, G.S. Sęk
ThPA6	Facile Synthesis of Semiconductor Nanostructures with Variable Plasmonic Properties Using Pulsed Laser Ablation Technique Yu.V. Ryabchikov
ThPA7	<i>Optical Properties of Silicon Nanowires-Based Composites</i> Yu.V. Ryabchikov
ThPA8	Non-equilibrium transport theory for dopant arrays in silicon M. Gawełczyk, G.W. Bryant, M. Zieliński
ThPA10	Elimination of Early Breakdowns in Fully-Depleted Detectors of Ionizing Radiation K. Kucharski, P. Janus, A. Panas, G. Głuszko, D. Tomaszewski, K. Krzyżak
ThPA11	Intervalley hybridization and optical properties of GeSn quantum dots K. Gawarecki, P. Scharoch, J. Ziembicki, N. Janik, R. Kudrawiec
ThPA12	Evaluation of Reaction Parameters in Polymeric Carbon Nitride Hybridized ZnO Nanocomposite for Improving Photocatalytic Activity using Thermal Synthesis N.N. Som, A. Opalinska, W. Lojkowski
ThPA13	Pressure-driven tuning of bandgap in methylammonium lead iodide A. Pieniążek, F. Dybała, M. Polak, Ł. Przypis, A.P. Herman, J. Kopaczek, R. Kudrawiec

ThPA14	Nanoprobe-based approach for electrical characterization of semiconductors nanostructures S. Tiagulskyi, R. Yatskiv, O. Černohorský, N. Bašinová, Š. Kučerová, J. Vaniš, A. Reszka, J. Grym
ThPA15	Young's Modulus and Microhardness of (Pb,Cd)Te Solid Solution Grown by SSVG and MBE Method S. Adamiak, W. Wołkanowicz, A. Juś, E. Łusakowska, R. Minikayev, D. Płoch, J. Korczak, A. Szczerbakow, W. Szuszkiewicz
ThPA16	Characterization of (Cd,Mn)Te and (Cd,Mn)(Te,Se) material and contact properties D.M. Kochanowska, A. Wardak, K. Gościński, P. Sybilski, A. Reszka, A. Sulich, A. Mycielski
ThPA17	Comparison of (Cd,Mn)Te and (Cd,Mn)(Te,Se) Compounds for Room Temperature X and Gamma Ray Detection: Optical Properties and Detector Response A. Wardak, D.M. Kochanowska, M. Szot, M. Kochański,M. Dopierała, A. Mycielski
ThPA18	Mn²+-V _{Zn} - charge transfer complexes in Zn _{1-x} Mn _x Te V.K. Le, A. Avdonin, A. Mycielski
ThPA19	Dynamics of direct and indirect excitons in (Cd,Mn)Te/(CdTe) double QW structure controlled by magnetic field V.Yu. Ivanov, S.M. Ryabchenko, V.I. Sugakov, G.V. Vertsimakha, G. Karczewski
ThPA20	Properties of Infrared Detectors Made of Pb _{1-x} Mn _x Te/CdTe Multilayer Composite S. Chusnutdinow, A. Kazakov, M. Szot, S. Schreyeck, R. Jakieła, K. Brunner, G. Karczewski
ThPA21	Terahertz Spectroscopy of Hyperbolic Metamaterial Based on CdTe/Cd _{1-x} Mg _x Te Multiple Quantum Wells J. Łusakowski, A. Frączak, E. Imos, A. Siemaszko, M. Zaremba, K. Karpierz, Z. Adamus, T. Słupiński, T. Wojtowicz
ThPA22	Near-band-gap Optical Properties of Modulation-doped CdTe/Cd _{1-x} Mg _x Te Multiple Quantum Well Structures J. Łusakowski, M. Grymuza, W. Solarska, K. Karpierz, Z. Adamus, T. Słupiński, T. Wojtowicz
ThPA23	<i>Terahertz Emission from CdTe-based Multiple Quantum Wells</i> D. Yavorskiy, R. Zdunek, K. Karpierz, Z. Adamus, T. Słupiński, T. Wojtowicz, J. Łusakowski
ThPA24	Simulation of Optical Properties of Terahertz Hyperbolic Metamaterials Based on Multiple Quantum Wells J. Łusakowski, T. Kazimierczuk

ThPA25	Diffusional dependent structures on the crystal surface M. Chabowska, M. Załuska-Kotur
ThPA26	Fitting parameters of atomic orbitals to reproduce optical properties of III-V systems K. Kwas, K. Gawarecki
ThPA27	Excited electronic states of Sr ₂ : ab initio predictions and experimental observation of the 2 ¹ Σ _u ⁺ state J. Szczepkowski, M. Gronowski, A. Grochola, W. Jastrzebski, M. Tomza, P. Kowalczyk
ThPA28	Excellent excitonic properties of novel hexagonal MA ₂ Z ₄ monolayers T. Woźniak, P.E. Faria Junior, U. Umm-e-hani, M.S. Ramzan, A.B. Kuc
THURSDAY POSTER SESSION B ON-LINE ONLY 21:00 - 22:30

ThPB1	Electrical Properties of Anisotype ZnO:Al/ZnSe/p-CdTe Heterostructures E.V. Maistruk, I.G. Orletskyi, M.I. Ilashchuk, I.P. Koziarskyi, D.P. Koziarskyi
ThPB2	Current Transfer Mechanisms in CuMoO₄/n-Si Heterostructure I.P. Koziarskyi, E.V. Maistruk, D.P. Koziarskyi
ThPB3	<i>Kinetics of Fe Island Films degradation in Air</i> K.A. Korotkov, A.M. Kasumov, A.I. Dmitriev, V.M. Karavaeva, A.I. levtushenko
ThPB4	Isolation of responsive elements of coordinate p-i-n photodiodes by p⁺-layer M.S. Kukurudziak
ThPB5	Damping of open circuit photovoltage in a Si homojunction B.A. Orłowski, K. Gwozdz, K. Goscinski, M. Chabowska, E. Guziewicz, B.J. Kowalski

Bulk GaN growth

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Application of gallium nitride (GaN) substrates in electronic and optoelectronic industries is constantly increasing. In order to fabricate wafers, GaN crystals of the highest structural quality and desired electrical (and sometimes optical) properties must be grown. Today, there are three main GaN crystallization methods: i/ halide vapor phase epitaxy (HVPE) with its derivatives: halide-free VPE and oxide VPE [1-3]; ii/ sodium-flux [4]; and iii/ ammonothermal. The last approach can be basic or acidic depending on what mineralizer is used to increase the solubility of GaN in the feedstock zone [5,6].

In this paper we will focus on HVPE and basic ammonothermal growth of GaN. Wherein, the HVPE as the best method for crystallization of drift layers necessary for construction of high-power vertical electronic devices (FET transistors, Schottky diodes) will also be discussed.

Recent results obtained in the basic ammonothermal growth will be shown. Structural, electrical, and optical properties of ammonothermal highly conductive (n-type) and semiinsulating (SI) GaN (Am-GaN) will be analyzed. Based on the presented results, main limiting factors of bulk Am-GaN crystallization and new ways for improving this process will be discussed. It will be shown: i/ how to eliminate lateral growth during crystallization in vertical directions; ii/ how to prepare the surface of a native seed in order to minimize residual stress in the growing crystal; iii/ how to obtain a uniform and constant supersaturation in the growth zone where more than fifty 2.2-inch crystals are placed; iv/ what kind of growth mode is required to obtain a uniform concentration of wanted and unwanted dopants in bulk Am-GaN. In the case of HVPE-GaN growth, the advantages of using native seeds will be demonstrated. All barriers for growing truly bulk HVPE-GaN will be analyzed. They will be compared to the limiting factors existing in Am-GaN growth. Morphology and its influence on uniform doping of bulk HVPE-GaN crystals will be discussed in detail. The main dopants used for growing n-type and SI crystals will be shown.

In conclusion, it will be explained what else we need to know about the crystallization of truly bulk GaN by the two methods described above. Some aspects of wafering of GaN crystals into substrates as well as seeds will also be summarized.

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Computational Design of Materials for Spintronics

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Exploring materials that exhibit spin-orbit-related phenomena holds great promise for advancing alternative computing devices beyond the traditional von Neumann paradigm. By leveraging the manipulation of spins through control of material symmetries, we can pave the way for next-generation electronic devices that minimize power consumption.

In this talk, I will provide a comprehensive overview of the computational design methods employed in the study of electronic materials. Delving deeper, I will discuss the remarkable properties of several recently (re-)discovered crystals that unveil intriguing spinorbit-related phenomena, along with various techniques for controlling spins. My focus will be on the utilization of low-symmetry crystals, which facilitate unconventional charge-to-spin conversion configurations, enabling the generation of highly efficient spin-orbit torques.

Furthermore, I will discuss the fascinating realm of chiral crystals, which exhibit collinear charge-to-spin conversion akin to the chirality-induced spin selectivity observed in molecules. Noteworthy examples include trigonal Te, Se, and TaSi2, which demonstrate high efficiency in charge-to-spin conversion and possess a remarkable attribute called the persistent spin texture, potentially leading to significantly prolonged spin lifetimes. Although the precise role of the persistent spin texture in protecting spin transport necessitates further elucidation, chiral materials with substantial spin-orbit coupling show immense potential in resolving one of the pivotal trade-offs in spintronics.

To conclude, I will explore the exciting prospects of high-throughput computational design in materials science, highlighting its potential in accelerating the discovery and development of novel materials with desirable properties.

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Experimental Quantum Control of Spin Defects in Solids

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Optically active spin defects in diamond have proven to be a promising resource for the implementation of quantum communication and have recently enabled the first demonstration of a 3-node network. In this presentation I bridge ongoing and anticipated work in the control, analysis and engineering of single spin defects for applications in quantum communication. I introduce a recent proposal in which we demonstrate theoretically that the realization of quantum communication over 1000 km becomes achievable with relatively few solid-state spin qubits that are efficiently coupled to an optical nanostructure. These coupled spin defects and their coherent control, however, have complex requirements that go beyond current state-of-the-art.

In our work, we focus on understanding the spin qubits' nano- and microscopic noise environment, on coupling to nanostructures, and on system control schemes. To illustrate these efforts, I outline how to coherently control diamond spin qubits, how to enhance qubit-tophoton coupling with nanostructures, and how to mitigate noise in such nanostructures. In particular, I discuss how to achieve single 'central' spin qubits with long coherence times through decoupling from spin bath noise, and how several of such quantum memories can be controlled simultaneously in a sub- diffraction volume. For enhanced qubit-to-photon coupling I introduce photonic nanostructures and methods for their fabrication, and demonstrate how such qubit-nanostructure devices facilitate flux control of electromagnetic radiation. One important (yet undesired) 'side-effect' of nanostructures is spectral diffusion of the spin system's optical transition frequency induced by time-varying electrostatic field noise. I introduce ways to investigate and mitigate the impact of this noise, and show that certain spin defects are immune to electric field noise to first order. Finally, I look into the near future and lay out how we plan to generate multi-qubit entangled states—an important step towards applying solid-state spin defects for the implementation of long-distance quantum communication and quantum networks.

Mol1

Optical and Electronic Fourier Surfaces

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Gratings and holograms are patterned surfaces that tailor optical signals by diffraction. Despite the long history of such structures, variants with remarkable functionalities continue to be discovered. Further advances could exploit Fourier optics, which specifies the surface pattern that generates a desired diffracted output through its Fourier transform. The required surface profile should contain a precise sum of sinusoidal waves, each with a well-defined amplitude, spatial frequency, and phase, to shape the optical wavefront. However, because fabrication techniques typically yield profiles with at most a few depth levels, complex "wavy" surfaces cannot be obtained, limiting the straightforward mathematical design and implementation of sophisticated diffractive optics. In this presentation, we will discuss a simple yet powerful approach to eliminate this design-fabrication mismatch by demonstrating optical surfaces that contain an arbitrary number of specified sinusoids [1]. Multicomponent linear gratings allow precise manipulation of the dispersion, stopbands, and coupling of electromagnetic signals. More broadly, we analytically design and accurately replicate intricate two-dimensional Moiré patterns, quasicrystals, and holograms, demonstrating a variety of previously impossible diffractive surfaces. Finally, we show that such patterns can be reduced to nanometer length scales, creating wavy Fourier surfaces for 2D electronics [2]. Therefore, this approach provides benefit for optical devices and emerging topics in photonics and 2D optoelectronics.

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Excitons in moiré heterostructures

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Van der Waals crystals of two-dimensional semiconducting transition metal dichalcogenides have evolved as an increasingly significant material platform in condensed matter research. With access to a variety of single-crystal monolayers, a wide range of van der Waals heterostructures can be assembled into rationally designed vertical stacks with emergent flat bands of correlated electrons and tailored optical properties of strongly bound excitons. In my talk, I will discuss our recent insight into the roles of layer constituents, atomic registry and twist angle for the formation of excitons with distinct spin-valley degrees of freedom and layer character. In particular, I will highlight the effect of charge carrier doping on moiré excitons in MoSe2-WS2 heterobilayers with canonical moiré superlattices [1] and present contrasting phenomena in MoSe2-WS2 heterostacks subject to mesoscopic reconstruction [2,3].

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Spin-Flip Raman Scattering on Electrons and Holes in Two-Dimensional $(PEA)_2PbI_4$ Perovskites

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The class of Ruddlesden-Popper type $(PEA)_2PbI_4$ perovskites comprises two-dimensional (2D) structures which are promising materials for photovoltaic and optoelectronic applications as their optical properties are determined by excitons with a large binding energy of about 260 meV. In 2D perovskites, a similar degree of optical spin control can be achieved as in conventional III-V and II-VI semiconductors, whose band structure is inverted compared to lead halide perovskites. We present our recent studies using spin-flip Raman scattering to measure the Zeeman splitting of electrons and holes in a magnetic field up to 10 T (see Fig. 1a)[1]. From the recorded data, the electron and hole Landé factors (q-factors) are evaluated (see Fig. 1b), their anisotropies are measured, and the hole sign is determined. The electron g-factor value changes from +2.11 out-of-plane to +2.50 in-plane, while the hole q-factor ranges between -0.13 and -0.51. Spin-flips of resident electrons and holes have been observed through their interaction with photogenerated excitons, as well as double spin-flip processes in which a resident electron and hole interact with the same exciton. Furthermore, we demonstrate the hyperfine holenuclei interaction in 2D perovskites by means of the dynamic nuclear polarization detected in corresponding changes of the hole Zeeman splitting (see Fig. 1c). Due to the small g-factor of the hole, we are able to achieve an Overhauser field value of $B_{\rm N,h} = 0.6 \,\mathrm{T}$.



Figure 1: (a) Spin-flip Raman spectrum in the anti-Stokes spectral range in cross polarizations for B = 10 T. The hole $E_{\rm h}$, electron $E_{\rm e}$, and their double spin-flip $E_{\rm e+h}$ lines are highlighted by arrows. (b) Electron and hole Raman shift as function of the magnetic field in out-of-plane geometry ($\mathbf{B} \parallel \mathbf{k}$). (c) Power density dependences of the energy splitting $\Delta E_{\rm N} = E^+ - E^-$ (the superscript indicates the excitation polarization) for the electron and hole shifts. Right axis gives the corresponding Overhauser field $B_{\rm N}$.

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Mol3

Topological photonics and topological lasers

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Topological Photonics is an emerging and novel field of research, adapting concepts from condensed matter physics to photonic systems adding new degrees of freedom. After the first demonstrations of topological photonic insulators [1,2], the field has moved on to study and exploit the inherent non-hermiticity of photonic systems and the interplay with their topological nature. In my talk, I will attempt to give an overview about the quickly emerging field of topological photonics. In this context, I will discuss topological lasers as a prime example of using topological concepts potentially for new technologies in the broad context of synthetic (photonic) matter. Examples will be given from novel photonic lattice devices resulting from the coupling of individual vertical III-V semiconductor microresonators.

Here, the so-called exciton-polaritons – hybrid states of light and matter – can emerge in the strong coupling regime. By choosing precise lattice geometries we are able to tailor optical band structures realizing novel photonic lattice. Here, the specific geometry as well as the hybrid light-matter nature allow for ways to break time-reversal symmetry and implement topologically non-trivial systems. Here, we were able to experimentally demonstrate the first exciton-polariton topological insulator, manifesting in chiral, topologically protected edge modes [3]. In order to study topological effects in combination with optical non-linearities, so-called topological lasers have been envisaged and realized. They exploit topological effects to efficiently couple and phase-lock extended arrays of lasers to behave like one single coherent laser. The major drawback so far is that the emission appears in the plane of the topologically protected light propagation, thus hindering light extraction. Here, we have presented the first experimental demonstration of a topological insulator vertical cavity laser array [4], using the so-called crystalline topological insulator model. Starting for the above mentioned examples. I will give an overview of the field of topological optical lattices and lasers and give an outlook on emerging novel materials beyond III-V semiconductors, such as organic materials, transition metal dichalcogenides and perovskites [5,6].

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MoO2

Networks of liquid light

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Recent years have seen a surge of advancements in optical manipulation over bosonic light-matter quasiparticles known as exciton-polaritons in semiconductor microcavities. These particles appear under strong-coupling conditions between cavity photons and embedded quantum-well excitons. Characterised by very high interaction strengths, nonlinearities, and picosecond timescales, these coherent light sources provide an exciting testbed to explore room-temperature nonequilibrium Bose-Einstein condensation in the optical regime.

In this talk, I will present results on all-optically engineered macroscopic networks of connected exciton-polariton condensates, which permit studies on fundamental emergent behaviours in complex nonequilibrium dynamical systems while subject to a drive and bosonic final-state stimulation. I will explain how a uniquely polaritonic feature gives rise to so-called "ballistic condensates" which, when spatially coupled, form a bosonic condensed matter analog of time-delay coupled oscillators that are ubiquitous in nature. I will present experimental and theoretical results on large-scale condensate networks displaying aforementioned emergent behaviors, including: spontaneous synchronization with unprecedented long-range spatial and temporal correlations [1,2], geometric frustration and formation of persistent superfluid currents [3], non-invasive optical control of the network coupling weights [4], synthesis of artificial lattices for studies of non-Hermitian topological physics [5], flatbands [6], and vortex frustration [7].



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Giant Photo-Elasticity of the Superlattice Polaritons for Detection of Coherent Phonons

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High-frequency coherent phonons in the GHz range have a potential for application in quantum technologies due to their nanometer wavelengths being comparable with the size of quantum nanodevices. Therefore, increasing strength of photon-photon coupling became of interest to the community [1,2]. In this work, we show that it is possible to reach quantum sensitivity of the phonon detection with a standard pump-probe set-up by exploiting giant photoelasticity of exciton-polariton resonance in a short period GaAs/AlAs superlattice (SL) [3].

For excitation and detection of the coherent phonon wave packet, we used a standard pumpprobe experimental set-up with a mechanical delay line and a lock-in amplifier. A coherent acoustic phonon wave packet is generated by absorption of the pump laser pulse by a metal transducer at the back of the sample. Strain pulse propagates into GaAs substrate and reaches the GaAs/AlAs SL at the front, where it gets detected. By sweeping the central wavelength of the probe in the vicinity of the polariton resonance we obtained signals for several detuning values of probe photon energy. The amplitude of the measured signal depends strongly on the photon energy of the probe.

Our experiments reveal giant photo-elasticity of polaritons and extremely high sensitivity to propagating coherent phonons. The strong dispersion of the dielectric permittivity in the vicinity of the polariton resonance results in a strong ultrafast response of the optical properties to dynamical strain which accompanies the coherent phonons. This discovery opens new possibilities of ultrasensitive measurements of extremely low-density phonon fluxes.

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Anomalous dispersion and dissipative coupling in AlGaAs exciton-polariton structure

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Strong light-matter coupling in semiconductor microcavities leads to the formation of quasiparticles called exciton polaritons, in which the energy is coherently exchanged between quantum-well (QW) excitons and bound photons. However, continuous photon loss due to the mirror imperfections, together with the exciton decay via radiative and nonradiative channels, makes it a highly lossy, dissipative system. It has been shown that dissipative light-matter coupling can be a result of various forms of decay channels [1], for instance when both resonances decay to the shared photonic environment [2]. Dissipative terms in Hamiltonian can greatly affect polariton dispersions, particularly when dissipative coupling is strong enough in comparison to the inherent coherent coupling. Then it may lead to an anomalous shape of the dispersion, with inverted lower state branch. Such energy-wavevector dependence has only recently been observed, and solely in microcavities containing transition metal dichalcogenides (TMDCs) [1,3]. With a negative particle mass, linked to its group velocity, such states have a great potential in dispersion engineering and in studies of non-Hermitian phases of quantum matter in a solid state.

In this work, we observe anomalous inverted dispersion of exciton polaritons in AlGaAsbased microcavity at low temperature. This material system is superior to microcavities with TMDCs in quality factors, linewidths, reproducibility, stability and ease of design, hence it has a large potential in reliable structure production and engineering. Our sample hosts both standard Γ -excitons in the QWs, coherently coupled to photons, but also lower-energy spatially and momentum indirect X-excitons, prone to dissipation. We show how the X-exciton resonance couples to the photonic mode in a wide range of exciton-photon detunings, leading to the inverted-mass eigenstates, observed in photoluminescence (PL) measurements. We present the evolution of polaritonic level attraction strength with the detuning, clearly showing a transition between monotonic (with a single maximum at k=0) and nonmonotonic (with maxima at finite wavevectors) |k|-dependence in lower branch dispersion. By implementing the non-Hermitian dissipative-coupling model, we characterize decay channels and interactions between resonances. Moreover, we present the temporal evolution of such states, as well as the dispersion of eigenstate lifetimes, showing a difference in branches' PL decay. Our work is the first realization of anomalous polariton states in a non-organic III-V semiconductor-based microcavity with wide detuning availability and with such a detailed characterization, which is crucial in understanding the unique nature of dissipative coupling in semiconductor microcavities and in future design of new non-Hermitian quantum phases.

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Non-Hermitian synthetic lattice with light-matter coupling

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We propose that light-matter coupling can be used to realize a synthetic lattice. In particular, we consider a one-dimensional chain of exciton-photon sites to create a synthetic comb lattice (see Figure 1) that exhibits a transition from a Lieb lattice flat band to a finite mass dispersion by tuning site-dependent light-matter coupling. Moreover, in a non-Hermitian system with gain and loss, the flat band phase is much more robust and the transition is accompanied by the appearance of exceptional points in the complex energy spectrum. We demonstrate that by engineering the light-matter coupling in the synthetic comb lattice, one can explore various phases in the polariton lasing regime. Our proposal paves the way for studying non-Hermitian systems in higher dimensions.



Figure 1: Scheme of our model. (Top) At lattice site j photonic (ψ_j^C) and excitonic (ψ_j^X) states are coupled. The neighboring photonic sites are coupled to each other with coupling rate J. A possible choice of a unit cell is marked by the green square. (Bottom) In practice the model can be implemented in a lattice of coupled micropillars.

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Automated Search For TMDs Monolayers On Exfoliated Samples

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Despite the constant development of the transition metal dichalcogenide (TMD) monolayer fabrication methods, the exfoliated TMD flakes still exhibit the best optical properties. However, by its nature, exfoliation is pretty much a random process, which results in a chaotic distribution of different multilayers on the substrate's surface. The search for monolayers suitable for further processing and measurements is often mostly done manually, which is tedious and time-consuming due to monolayers' size. It turns out that using simple optical properties and computer calculations can significantly speed up this process.

Thin TMD multilayers of different thicknesses deposited on a reflective substrate, such as a silicon wafer, reflect the white light differently due to internal interferences and absorption within the TMD material (Fig. 1). Because of this effect, it is possible to assess the number of layers in a given flake by eye test using an optical microscope with a white light source. By analysing images of silicon wafers covered in exfoliated TMDs, one can select monolayer flakes and, by further analysis of their size and shape, find samples usable for purposes such as fabricating heterostructures.

In this presentation computer program capable of such identification will be shown. The software can find monolayers from saved images of exfoliated TMDs. Statistical analysis of previously found monolayers allows the algorithm to locate monolayers on photos taken under different conditions, such as differing exposure time or light intensity. The scope of this program's usability is also easily expandable to other layered materials.

The main advantages of that approach over manual search are the significantly shorter time required to search through the entire surface of the wafer and the nearly complete elimination of the chance of missing a good flake. It also potentially outperforms other deterministic methods of finding monolayers, such as PLE measurement, because of accessibility, as it does not require any additional experimental setup other than a microscope with a camera.

The presentation will cover statistical analysis used to determine the colour of monolayers in a given image alongside the main coding ideas used in the software.



Figure 1: A microscope image of the exfoliated WSe_2 on the silicon substrate. Layers of different thicknesses have distinctively different colours. The orange and yellow are the thickest, while the dark blue spot near the centre of the image is a monolayer.

Electrical Tuning Of Exciton Lifetime In Monolayer MoS₂

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The recent studies of low-dimensional semiconductors and their hybrid structure have led to discovering of many fascinating properties that are absent in their bulk counterparts. [1] Here, we studied the dynamics of excitons, which have high binding energy at room temperature using steady-state photoluminescence (PL) and time-resolved photoluminescence (TRPL) by systematically controlling the carrier density. We observed neutral and charged exciton(trions) dominant regimes in some specific carrier density.[2] In pristine MoS₂ devices, we observed an exciton dominant regime with an exciton lifetime of 3 ns, when we doped electrostatically with holes. Interestingly, we observe a sharp decrease in exciton lifetime with an increase of the electron density by electrostatic doping, with a corresponding increase in negative trion population, as shown in Fig. 1(a). This indicates the formation of negative trion. With increased hole doping by a chemical method, the exciton lifetime decreases, but it remains almost constant with electrostatic doping, which is visible in Fig. 1(b). This might be due to the exciton-exciton annihilation mechanism. Further hole doping by a chemical method leads to a transition to a positive trion dominated regime, as shown in Fig(c). Our observations suggest that when calculating the lifetime of excitons, the exciton-to-trions formation and excitonexciton annihilation mechanisms should be considered.



Fig.1. Lifetime of A-exciton (red circles) and trions (blue circles) at different doping levels achieved via chemical treatment and electrostatic doping: (a) pristine sample(P); (b) medium-hole-doped sample(M); (c) high-hole-doped sample(H). The lifetime of the A-exciton depends strongly on doping level. (d) Schematic of the exciton interaction at different regimes. (e) Schematic showing the various channels available for excitons and trions.

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Fabrication and Strain Engineering of Transition Metal Dichalcogenides Monolayers on GaAs Nanomembranes

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Transition metal dichalcogenides (TMDs) possess an excellent mechanical robustness and flexibility withstanding strains as high as 10% before breaking. Therefore, strain engineering has become a particularly attractive tool to tailor their electronic and vibrational properties or to preferentially funnel photoexcited carriers towards defects, which act as single photon emitters [1]. One of the approaches to site-control these quantum light emitters is the deposition of TMD monolayers on patterned substrates [1], thereby imposing strain on the two-dimensional layers.

So far, patterned substrates consist mainly of SiO_2 nanostructures, which, however, suffer from surface charge instability. This provokes spectral wandering and blinking of single photon emitters. Here, we explore an alternative approach to deterministically imposing strain on TMD layers by making use of high-quality, epitaxially grown III-V semiconductor nanostructures.

We show a successful deposition of WSe₂ monolayers on 100 nm wide GaAs nanomembranes with 10 μ m pitch in Fig. 1a. By performing photoluminescence spectroscopy, we show that the TMD monolayers deposited on GaAs nanomembranes effectively funnel carriers to the potential minima corresponding to strain, which is demonstrated by photoluminescence and Raman spectroscopy. At low temperature, we observe a blue shift of the A exciton of WSe₂ monolayer measured on a nanomembrane, as compared to the same resonance measured on the flat part. (See Fig. 1b) Intensity on the nanomembrane is lower than on the flat part and some resonances are polarization dependent. In addition, we found that the most intense Raman mode splits and shifts to the higher frequency on the nanomembrane, compared to the flat part. (See Fig. 1c) This observation suggests the presence of compressive strain, differently than expected.

This confirms that the applied stress allows to efficiently funnel photocreated carriers to the potential minima. In perspective, we will investigated the single photon emission properties of localized states by photon correlation spectroscopy.



Figure 1. (a) Micrograph of the WSe_2 flake deposited on GaAs nanomembranes. (b) PL map energy of the A exciton. (c) Raman measurements on the nanomembrane (red) and on the flat part (black).

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Controlled coherent-coupling and dynamics of exciton complexes in a $MoSe_2$ monolayer

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Quantifying and controlling the coherent dynamics and coupling of optical transitions in solids is of paramount importance in fundamental considerations in condensed matter optics and for their prospective optoelectronics applications in quantum technologies. Here we perform Four-Wave-Mixing microspectroscopy of a charge-tunable MoSe₂ monolayer. The experiments show that the homogeneous linewidth and the population decay of exciton complexes hosted by this material can be directly tuned by an applied gate voltage, which governs the free carrier density.



By performing two-dimensional spectroscopy, we also show that the same bias tuning approach permits us to control the coherent coupling strength between different exciton species.

Our findings yield exciting prospects for forthcoming investigations of coherence phenomena in the context of recent discoveries of strongly correlated exciton phases in solids, optically probed quantum Hall states, Moiré superlatices, and magnetic two-dimensional materials.

Figure 1: Phase-referenced two-dimensional FWM spectroscopy of X and X^- (a) The pulse sequence used in the 2D FWM experiment. (b) Examples of 2D FWM spectra for different gate biases as given in the plots.

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Comparison of optical properties of $MoSe_2$ monolayers; samples grown by MBE technique vs mechanically exfoliated one.

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Semiconducting transition metal dichalcogenides (s-TMD) exhibit strong optical response within the limit of a single atomic layer. It was first observed for molybdenum disulfide, which exhibits strong photoluminescence due to direct bandgap.[1] The state of the art s–TMD monolayer samples are produced by the process of mechanical exfoliation, which became famous with the first graphene sample produced by this method. [2] The bulk crystals that are used for mechanical exfoliation can be grown with low defects concentration within their volume, especially when compared to defects concentrated on their surfaces. Therefore, the subsequent thinning of the bulk-grown sample with an adhesive tape allows for the exfoliation of atomically thin layered crystals (top-down approach). The amount of defects in such layers is similarly small as in the case of the original bulk crystal. For that reason – the surface of the sample is the sample – epitaxial techniques (bottom-up approach) were expected to be always inferior when compared to the manually exfoliated ones from the bulk crystal. The breakthrough happened with MoSe₂ monolayer sample grown by molecular beam epitaxy on high-quality substrate of hexagonal boron nitride.[3] It showed the possibility of producing TMD monolayer sample with an optical response comparable to the best – state-of-the-art – exfoliated ones. The development of TMD monolayer MBE–growth technique is crucial for industrial application due to its unprecedented spatial homogeneity and repeatability unachievable by the technique based on manual mechanical exfoliation. Moreover, MBE–grown MoSe₂ monolayers are folds- and ripples-free, and those are the intrinsic features of mechanically exfoliated samples that are very hard to get rid of.

In our work, we investigate the differences in optical properties of MBE-grown and mechanically exfoliated samples. We show that the photoluminescence intensity variation in MBE-grown samples can be explained by three major factors – the interference effects within the h–BN substrate, incomplete surface coverage by epitaxial monolayer (measured with atomic force microscope) and the radiatively non-active edge of MoSe₂ flakes. Additionally, we performed polarisation–resolved magnetospectroscopy and time–resolved measurements of exciton photoluminescence. We extract the exciton intervalley scattering time of ~16 ps that is universal for both types of samples – exfoliated and MBE–grown. We also show variations of the charged exciton formation time among different samples caused by the changes in intrinsic carrier density.

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Electron-phonon coupling in a WSe_2 monolayer unveiled by two-photon photoluminescence excitation spectroscopy

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The "all-surface" character of two-dimensional semiconductors leads to strong interacrions of the electronic and vibrational degrees of freedom of a monolayer semiconductor to its surroundings [1]. Here, we investigate this aspect by performing a detailed investigation of the two-photon photoluminescence excitation (PLE) spectrum of a high quality WSe₂ monolayer, which we show in Fig. 1(a).

The low-temperature photoluminescence (PL) and reflection contrast spectra, shown in Fig. 1(b), show resonances of neutral and charged exciton species, as well as higher order excitonic transitions, like the 2s state of the A exciton. Figure 1(c), is a false-color map of the two-photon PLE experiment, with the horizontal axis corresponding to the emission energy and the vertical axis corresponding to two times the excitation energy. From the PLE spectrum, we extract the intensity of the two-photon PL of the exciton and plot it in Fig. 1(b), where we see pronounced resonances related to the absorption of the 2p and 3p states. We show in Fig. 1(d,e) selected PL spectra and close-up views, excited at the energies indicated in Fig. 1(c), respectively. In Fig. 1(e), we note a peak, separated by 108 meV from the doubled energy of the laser. Its shifts by the exact same amount as the doubled energy of the laser is a distinctive sign of its phonon-related nature. Ongoing work is devoted to a more precise understanding of this phonon line.



Figure 1: (a) Micrograph of the hBN-encapsulated WSe2₂ flake. The monolayer region is framed by the red triangle. (b) PL and reflectivity contrast and two-photon PLE spectra of the WSe₂ monolayer. The relevant excitonic resonances are indicated by dashed vertical lines. (c) False color map depicting two-photon PLE spectra. (d) Two-photon PL spectra extracted at selected excitation energies. (e) Close-up of the PL spectra. The phonon line shifting with the excitation laser is indicated by arrows.

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Electrical Control of Interlayer Physics in Type-II TMD Heterostructures

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We describe the electronic and topological properties of MoSe₂/WSe₂ heterostructure using a combination of density functional theory (DFT) and tight-binding (TB) approximation. We start with determining the electronic structure of MoSe₂/WSe₂ from first principles, using the PBE parametrization of the GGA for the exchange-correlation potentials and including atomistic spin-orbit interaction. We obtain type-II band alignment and conduction band minima at Q points in agreement with previous works [1,2]. We include the analysis of Kohn-Sham wavefunctions allowing to describe the leading layer, spin and orbital contribution for a given band and given point in the reciprocal space, as well as to capture the interplay between two types of valleys present in the conduction band, i.e. K and Q. Following our previous work on monolayers we construct the *ab initio*-based TB model [3] for MoSe₂/WSe₂ heterostructure, which allows us to understand orbital contributions to Bloch states and will enable further studies of both twisted transition metal dichalcogenide (TMD) heterostructures and TMD-based multi-million atom nanostructures. To validate the model, we perform the analysis of topological properties of the system manifested in the Berry curvature. Finally, we perform a study of the effect of electric field on the interlayer properties of the MoSe₂/WSe₂ heterostructure towards the construction of gate-tunable spin-valley qubits [4] in TMD heterostructures.

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Searching for moiré-induced quantum spin liquid - the DMRG and ED study of twisted TMD hetero-bilayers

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Quantum spin liquid (QSL) is an exotic, highly correlated phase of matter characterized by the absence of long-range order and extremely large entanglement. The most intriguing property of QSL is its ability to support non-local excitations [1]. Such a phase occurs in frustrated triangular lattices and can be described by the Hamiltonian with four-spin terms [2].

Twisted hetero-bilayers of transition metal dichalcogenides are advantageous platforms for the study of correlated phases. Due to the mismatch between the lattice constants and a nonzero twist, a new periodicity called the moiré pattern arises in such materials, which can be effectively described by the Hubbard model on the triangular lattice [3,4]. For a range of twist angles, analyzed by us in previous work [5], moiré TMD materials at the half-filling can also be adequately described by the spin Hamiltonian.

In our work, we study finite structures craved from twisted TMD hetero-bilayers described by a spin Hamiltonian with four-spin terms. Using the DMRG method, we search for the range of parameters for which the QSL phase occurs and compare it with the results from the exact diagonalization analysis. In addition, we investigate the entanglement entropy of ground states.

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Exploring neutral and charged exciton optical properties in a MoTe₂/SiO₂/Si-based metal-oxide-semiconductor device

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Molybdenum ditelluride (MoTe₂) belongs to a broad family of transition-metal dichalcogenides (TMDs). In its 2H semiconducting phase, as a single-layer crystal, it has a direct optical bandgap at ~1.17 eV at 5 K, translated to ~1.0 eV at 300 K.[1] It makes this material the only TMD with optoelectronic device applicability in the near-infrared spectral range (>1 μ m photon wavelength). However, many properties of single-layer MoTe₂ crystals are still unknown, including those related to the control of optical bandgap, interactions and excitation dynamics. It is predicted that the excess carrier density, charge polarity and electric field in MoTe₂ should effectively control these properties.[2] In this contribution, we examine the role of excess carrier density in monolayer (ML) MoTe₂ and examine to what extent one can control the optical gap, interactions in excitonic complexes and excitation dynamics.

We fabricated a metal-oxide-semiconductor (MOS) device allowing control over the excess carrier density and carrier type, switching from electrons to holes, in ML MoTe₂. The device consists of large-area exfoliated ML MoTe₂ placed partially on a gold electrode prepatterned on a SiO₂/doped Si substrate. The Si plays a role of a second electrode. The MOS device can be biased with externally applied voltage (V_g), ranging from -30 V to +30 V and operates at cryogenic temperatures. Sweeping the voltage, we achieved a high range of charge tunability tracked by the optical response from the crystal at T = 5 K using the highspatially-resolved photoluminescence (µPL) and contrast reflectivity experiments (µCR). By increasing V_g, one can observe the conversion of neutral excitons (X) to trions\polarons (T) through the monotonous increase in the T transition-related µPL intensity at the cost of a decrease in the X transition intensity. At roughly +/-20 V, the X population is almost entirely drained, and the T intensity is saturated. This observation is confirmed in the μ CR, where at higher Vg, a new T-related resonance appears on a low-energy side of the always present X resonance. Interestingly, the voltage sweeping allows tuning energies of the X-related optical transition observed in µPL, showing a nearly linear increase from ~1.181 eV at ~0 V to 1.185 eV at -20 V and ~1.184 eV at +20 V. Under the same conditions, the T transition (when visible) remains almost unchanged at ~1.158 eV for negative bias and ~1.159 eV for positive. The trion/polaron binding energy $(E_{b,T})$ depends on the V_g polarity. For the negative Vg, $E_{b,T}$ reaches ~27 meV, whereas, for the positive Vg, it amounts to roughly 25 meV.

Finally, we tested the dynamics of X and T excitations using a non-degenerated pumpprobe transient reflectivity experiment. We observe at least two fast and slow recombination channels for each species. At Vg \sim 0 V, near the charge-neutrality condition, the X decays within ~9 ps and ~70 ps, whereas T decay with ~12 ps and ~90 ps. Interestingly, the decay channels can be controlled mainly by negative Vg, shortening the short component by a few ps and the long component by a few tens of ps.

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Optimization of Contacts for van der Waals heterostructures by AFM ironing

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Semiconducting transition metal dichalcogenides (TMDCs) like MoS₂, MoSe₂, WS₂ and WSe₂ offer many new possibilities of structure engineering from the spintronics' point of view and are, therefore, attractive for future novel electronic devices. Integration of any device with an external circuitry requires, however, an efficient contacting scheme, which is challenging for TMDCs. Between a metal and a semiconductor (SC) the Schottky barrier (SB) is typically formed, which is theoretically dependent on the difference between the work function of the metal and the electron affinity of the SC [1]. In practice, the Fermi level at the metal/SC interface is typically pinned as a result of metal-induced gap states. This problem can be ommitted by introducing a tunnel barrier in the form of an additional interlayer, e.g., hBN layer, between the metal and the TMDC, which limits the wavefunction spreading from the metal into the TMDC [2]. Transfer techniques commonly used to stack individual 2D layers into vdW structures can lead to inhomogeneities due to contaminants trapped between a 2D layer and a substrate or a metal layer [3]. One of the recently suggested methods to circumvent this problem is the so-called AFM ironing, when a contact-mode AFM scan is used to remove any possible imperfections like organic residues, bubbles or wrinkles, and improve adjacency of layers and, therefore, contact's resistance. This method has been used with success in many experiments [4-6].

Here, we investigate the effect of AFM ironing on the quality of Ti/Au contacts to MoS_2 . MoS_2 flakes were produced by means of mechanical exfoliation and deposited on Si/SiO_2 substrates. Apart from investigating the direct metal/ MoS_2 contacts, we studied also the influence of a thin hBN layer, acting as a tunnel barrier between the metal and the MoS_2 flake, on contact parameters. To study the influence of AFM ironing on the quality of such prepared contacts, the electrical and optical properties of the ironed and the non-ironed samples were compared using transport measurements and photoluminescence spectroscopy, respectively.

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Computational prediction of structure and electronic properties of monolayer and few-layer 2D polyimide covalent organic frameworks

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In contrast to similar and more established groups of materials, such as zeolites and metal organic frameworks, covalent organic frameworks (COFs) are distinctively built only from organic elements – carbon, oxygen, hydrogen, and often nitrogen or boron. They create light, mechanically rigid and chemically inert networks which have many advantages but also create issues with developing large-scale films or membranes because of low solubility and processability [1]. Together with a lack of strong coordination bonds, which direct the design and synthesis in case of frameworks that contain transition metal ions, it has slowed the progress in COF synthesis and characterization. Recently though, more robust design strategies have led, e.g., to first reports of bulk produced and single-crystal 2D COFs [2]. With the possibility in sight of controllable synthesis of monolayer and few-layer frameworks, they are on the path to becoming interesting candidates for integration in optoelectronic devices, similar to other layered 2D materials. Meanwhile, the rigidity and low solubility of COF that have hampered their development are becoming their valuable traits in device integration.

Imide COFs [3] in particular exemplify these characteristics by containing sp^2 carbon and a fair amount of enamine (-C-N-) instead of imine (-C=N-) bonds, both of which stabilize the crystal structure [1]. Here, we report the results of DFT calculations of such materials in the form of monolayers and AA and AB-stacked multilayers. We find the multilayers to be stable and with varying electronic properties. In particular, the finite band gap exhibited by every structure (monolayer and stacked) is sensitive to atomic arrangement of the atoms.



Figure 1: Formation energies (with respect to the flat monolayer), c lattice vector (out-of-plane) lengths, and PBE band gaps of AA and AB stacked polyimide COFs. The unit cell comprised of a monolayer relaxed towards a minimum with significant torsion (between 37 - 45 deg) between outer rings. The relaxed monolayer exhibits an energy lowering of -0.25 eV per formula unit (f.u., $C_{13}N_3O_4H_3$). The PBE band gap of the monolayer varies between 1.02 (flat) and 1.68 eV (relaxed).

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Tunable valley splitting in 2D MPX₃ crystals.

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Valleytronics has attracted a great interest in 2D materials, due to number of intriguing properties such as spin-valley locking and valley Hall effect [1]. Tailoring valley degrees of freedom offers unique playground to realize pecular phenomena and novel applications, Here, we have demonstrated that the monolayer of transition metal trichalcogenides $MnPX_3$ (X=S, Se), which are layered antiferomagnts [2,3], could be viewed as promising valley electronics materials. In particular, we have shown that the valley splitting at the K+/K- can be effectively control by the direction of the magnetic moments. In particular, the sizable valley splitting occurs for the out-of plane direction of the magnetic moments in monolayer of MnPSe3, resulting in valley dependent gaps, as it is presented in Fig. 1. On the other hand, the valley dependent gaps rely on the type of the chalcogen atoms. Moreover, the K+/K- valleys are spin degenerated, which might hinder their spintronic applications. Hence, we have also propose a novel way how the valley polarization can be achieved. We have suggested that particular deformation of the hexagonal lattice of MnPX₃ could lead to the spin resolved valley splitting. Our results give insight into the valley splitting realization in 2D antiferromagnets.



Fig. 1. Tunable Valley splitting upon the change of the direction of Mn spins from the inplane to out-of plane direction.

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Optical markers of magnetic phase transition in layered CrSBr.

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Two-dimensional (2D) magnetic semiconductors give the opportunity to take advantage of the electron charge and the electron spin at the same time, which gives the possibility of broadening modern semiconductor technology and spintronic devices. It can greatly expand the applications of ferromagnets (FM) in other devices such as transformers, electromagnets, high-density storage, and magnetic random access memory [1].

Here, we investigate the role of the interlayer magnetic ordering of CrSBr: antiferromagnetic van der Waals crystal whose single monolayer is intrinsic ferromagnetic (AFM:A), in the framework of <u>ab initio calculations</u>, and by using <u>optical spectroscopy [2]</u>. We have demonstrated that optically induced band-to-band transitions visible in optical measurement are remarkably well assigned to the band structure by the momentum matrix elements and energy differences for magnetic ground state (A-AFM). In addition, our study reveals significant differences in electronic properties for two different interlayer magnetic phases. When the magnetic ordering of A-AFM to FM is changed, the crucial modification of the band structure reflected in the direct-to-indirect band gap transition and the significant



splitting of the conduction bands are obtained.

Fig. 1 (a) Phonon dispersion for AFM, FM magnetic states (b) Experimentally measured raman peaks positions in function of temperature.

In addition, Raman measurements (Fig.(b)) demonstrate a splitting between the in-plane modes B_{2g}^2/B_{3g}^2 , which is temperature dependent and can be assigned to different interlayer magnetic states, corroborated by the DFT+U study (Fig.(a)). Finally, our results point out the origin of

interlayer magnetism, which has its origin in electronic rather than structural properties. Our results reveal a new approach for tuning the optical and electronic properties of van der Waals magnets by controlling the interlayer magnetic ordering in adjacent layers.

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Interplay between stacking and magnetic configurations in MPS₃ bilayers

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The two-dimensional transition metal thiophosphates MPS₃ (M = Mn, Fe, and Ni) have attracted a lot of attention recently because they exhibit a long-range antiferromagnetic (AFM) order and display interesting electronic-optical properties associated with their magnetic state [1]. In these materials, the AFM ground state is governed by the competition between direct M–M exchange and indirect M–S–M superexchange interactions within atomic layers, as well as the d-electron occupancy of the metal ion [2]. However, how the stacking between layers influences the electronic properties of MPS₃ is largely unknown. Here, we perform first-principles calculations to study the interplay between AFM and ferromagnetic (FM) configurations under different layer stacking (see Figure 1). Our results provide insight into understanding the interlayer AFM-FM coupling in those systems, revealing that different magnetic states can be stabilized through layer stacking. We suggest that these systems provide a unique playground for studying different aspects of magnetism in 2D systems.



Figure 1. Different high-symmetry stacking configurations in bilayer MPS_3 . The top view to the right of each structure indicates the stacking of the two hexagonal lattices of the transition metals. The low-right panel shows the interlayer exchange energy for each stacking.

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Exploring structural, electronic, and magnetic properties of 2D Cr, Fe, and Zr monoborides

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MBenes, a family of two-dimensional transition metal borides, has recently attracted significant attention [1]. Their elemental compositions, surface terminations, geometrical forms, and chemical and physical properties are unique and fascinating. These twodimensional compounds hold promising applications in various fields such as molecular sensing, Li- and Na-ion batteries, electrocatalysis, and magnetic devices. Also important are their exceptional electrical conductivity, high hydrophilicity, rich surface chemistry, and outstanding stability.

In this work [2], we present the results of density functional theory-based calculations for 2D Fe₂B₂, Cr₂B₂, and Zr₂B₂ hexagonal and orthorhombic MBenes. According to our calculations, the orthorhombic form of the 2D structures is the most energetically favorable for Fe₂B₂ and Cr₂B₂, whereas, Zr₂B₂ MBene prefers to adopt the hexagonal form. From phonon calculations, we can also deduce that the orthorhombic structure of Zr₂B₂ and the hexagonal Fe₂B₂ are also structurally stable. The magnetic properties of the two-dimensional systems were also studied and Fe₂B₂ and Cr₂B₂ are found to be ferromagnetic, whereas Zr₂B₂ is nonmagnetic. The Fe₂B₂ structure exhibits a magnetic moment of 3.73 Bohr magnetons per unit cell (see the DOS picture shown in Fig. 1) followed by Cr₂B₂, with 2.58 Bohr magnetons per unit cell. Transport computations reveal that the highest conductivity may be expected for Zr₂B₂ and the smallest for Fe₂B₂. A detailed comparison with available data in the literature will be provided.



Figure 1: Spin resolved density of states (DOS) of ferromagnetic Fe₂B₂. The upward and downward plots show the majority and minority states, respectively, and the dashed line shows the Fermi energy.

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Magnetic skyrmions of Ti2C MXenes doped with Cr, Mn, and Fe Teresa Kulka¹, and Jacek Majewski¹

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Two-dimensional titanium carbides Ti2C, MXenes, have been attracted scientific effort, mostly owing to their intriguing properties and promising applications [1]. The magnetic moments of the external metallic Ti layers are ferromagnetically or antiferromagnetically oriented to each other depending on the remnants from the technological processes that cover the surface. The possibility to tune magnetic moments of the external MXenes layers through suitable functionalization could be utilized in numerous spintronic applications [2].

We have investigated the change of the magnetic properties of MXenes caused by the substitutional doping of Ti layers with transition metal (TM) atoms such as Mn, Cr, and Fe. The introduction of TM atoms with different number of d-electrons (and therefore different magnetic moment) leads to breaking of the homogeneity of the spontaneous magnetization and formation of the skyrmionic-like patterns (see Figure below). Magnetic skyrmions are small swirling topological defects in the magnetization texture, which are regarded as novel information carriers in future spintronic devices [3].

Our studies are based on the first principles calculations within the Kohn-Sham realization of the density functional theory (DFT) employing the Quantum Espresso computer package. We take spin-orbit interaction into account and therefore our calculations have to be performed in the non-collinear magnetism regime. To facilitate eventual future comparison with experiments, we have undertaken an attempt to compute the spin tunnelling current between the non-uniformly spin-polarized electrons of the MXenes surface and magnetic tip of the scanning tunnelling microscope (STM), in close analogy to codes providing the STM images from the electronic density obtained in DFT calculations.



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The effect of temperature and excitation energy on Raman scattering in bulk HfS₂

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Raman scattering (RS) in bulk hafnium disulfide (HfS₂) is investigated as a function of temperature and excitation of several laser energies (fig. 1). The low-temperature quenching of ω_1 (134 cm⁻¹) and the emergence of a new mode at approx. 184 cm⁻¹, labeled Z, is reported (fig. 1.a). An unexpected blue-shift of the main Raman-active modes (A_{1g} and E_g) as temperature increases, was observed (fig. 1.b). The excitation-dependent RS is reported. The apparent quenching of the A_{1g} mode at T=5 K and of the E_g mode at T=300 K in the RS spectrum excited with 3.06 eV excitation is also observed. We discuss the results in the context of possible resonant character of light-phonon interactions. Analyzed is also a possible effect of the iodine molecules intercalated in the van der Waals gaps between neighboring HfS₂ layers, which inevitably result from the growth procedure.



Fig. 1 (a) The temperature evolution of RS spectra measured on HfS2 bulk under excitation of E_{exc} =1.96 eV. For temperatures lower than 100 K a new mode, called here Z, has been observed and the quenching of ω_1 . (b) Temperature evolution of the Raman shift of the A_{1g} and E_g modes under excitation of E_{exc} =1.96 eV. The Raman scattering spectra measured on Hf₂ bulk at (c) T =300 K and (d) T =5 K under a series of different excitation energies, indicated in the Figure.

Growth and characterization of metallic and oxidized forms of bismuthene

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Bismuth is in the XV^{th} group of the periodic table, with the electronic configuration $6s^26p^3$, and as a semi-metal it is the last non-radioactive element in the periodic table with unusual electronic properties.

These properties are of interest due to spin-orbital effects and surface and edge states.[1] It was in bismuth that many quantum phenomena such as, for example, magnetoresistance, quantum size effects and recently also topologically protected edge states were first detected. It was recently shown that bismuth can grow in two dimensional form material known as bismuthene.[1,2] It is extremely important to understand the growth and degradation paths of these new material.

We will present the properties and morphology of bismuth nanostructures growing epitaxially on a pyrolytic graphite (HOPG), as well as insulating materials such as mica and hBN, on which the growth of bismuthene islands (2D bismuth with black phosphorus structure) can be observed.

We will focus on the conducted experimental research using scanning tunneling microscopy and atomic force microscopy, thanks to which we were able to study the electronic and atomic structure of bismuth single layers. X-ray photoelectron spectroscopy was used to investigate and determine the chemical composition of the oxidized structures. Raman spectroscopy was used for further analysis of the oxidized and metallic structures.

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Transport signatures of Van Hove singularities in mesoscopic twisted bilayer graphene

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Magic-angle twisted bilayer graphene exhibits quasi-flat low-energy bands with Van Hove singularities close to the Fermi level. These singularities play an important role in the exotic phenomena observed in this material, such as superconductivity and magnetism, by amplifying electronic correlation effects. In this work, we study the correspondence of four-terminal conductance and the Fermi surface topology as a function of the twist angle, pressure, and energy in mesoscopic, ballistic samples of small-angle twisted bilayer graphene. We establish a correspondence between features in the wide-junction conductance and the presence of van Hove singularities in the density of states. Moreover, we identify additional transport features, such as a large, pressure-tunable minimal conductance, conductance peaks coinciding with non-singular band crossings, and unusually large conductance oscillations as a function of the system size. Our results suggest that twisted bilayer graphene close the magic angle is a unique system featuring simultaneously large conductance due to the quasi-flat bands, strong quantum nonlinearity due to the Van Hove singularities and high sensitivity to external parameters, which could be utilized in high-frequency device applications and sensitive detectors.

MBE grown heterostructures of ZnSe, CdSe, MoSe₂, and hBN

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The best optical properties of exfoliated transition metal dichalcogenides (TMDs) are obtained for monolayer materials encapsulated in hBN, so mechanically stacked in heterostructure hBN/TMD/hBN [1]. Part of this heterostructure can be reproduced using epitaxial methods, and indeed TMD grown on hBN exhibits much better optical properties [2] than TMD grown on other substrates. Still, the top hBN layer is missing in such a procedure. To finish heterostructure using large-scale epitaxial methods, the growth of hBN on TMD would be an ideal option. This however seems to be not accessible nowadays, because the very high growth temperature of hBN (over 1200C) makes it impossible to grow it on TMDs without destroying the previously deposited material. Consequently, other cap materials should be considered as candidates for protection of epitaxial TMDs.

In this work, we experimentally verify the possibility of covering TMDs with II-VI compounds based on selenides: ZnSe and CdSe. Materials are chosen due to chemical similarity to TMDs based on selenium, e.g MoSe₂, therefore there should be no risk of substitution of Se in TMD by other elements. Moreover, ZnSe is chosen due to its wide band gap and its transparency in the MoSe₂ emission spectrum. We expect that for TMDs, ZnSe will be just an isolator, similar to hBN. On the contrary, CdSe emission spectrum is close to the one of MoSe₂, so one can expect interactions in a formed heterostructure. Importantly, both ZnSe and CdSe can be grown at low temperatures (typically 300C) compared to temperatures used during the preparation of MoSe₂ (300-800C).

Heterostructures ZnSe/MoSe₂ and CdSe/MoSe₂ were grown using Molecular Beam Epitaxy (MBE) on exfoliated hBN deposited on SiO₂/Si substrates. Atomic force microscopy (AFM) and optical microscopy reveal the rich morphology of grown structures indicating the polycrystalline nature of II-VI material grown on MoSe₂. The results of low-temperature photoluminescence show that both ZnSe and CdSe strongly affect the optical spectra of MoSe₂. Moreover, the photoluminescence of CdSe is much stronger than the photoluminescence of MBE MoSe₂. Therefore research on MoSe₂ with II-VI layer present on top seems to be difficult. Still, we checked that the good optical properties of MoSe₂ grown on hBN can be restored by desorption of II-VI material from the surface of the heterostructure.

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Structural, Optical and Electrical Properites of Quaternary Hafnium-Aluminium Zinc Oxide Films Grown by Atomic Layer Deposition for Transparent Conductive Oxide Applications

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Transparent Conductive Oxide (TCO) films are characterized by having simultaneously exceptional electrical conductivity and high transmission in the visible range. Among many TCO materials, doped zinc oxide (ZnO) is one of the foremost candidates for next generation electronic devices, since it exhibits several desirable optoelectrical properties, in particular direct and wide bandgap (~3.37 eV), high carrier concentration exceeding 10^{19} cm⁻³ and very low resistivity in the range of 10^{-3} Ω cm. These properties may be finely tuned and enhanced by doping elements to create an oxide alloy. Despite the vast range of papers covering the properties of singly-doped ZnO, there is still undiscovered potential of simultaneous doping ZnO with two elements, creating a quaternary material. During our studies we fabricated a co-doped ZnO films with both aluminum and hafnium atoms (HAZO), utilizing atomic layer deposition technique (ALD). This method offers a precise control of the film thickness due to self-limiting character of the reactions during every half-cycle along with unraveled conformality and uniformity of the grown films. The optical, electrical and structural properties of the samples were investigated for varying Hf and Al concentrations to establish a comprehensive image of the studied material. The co-doped samples exhibit extremely low resistivity, close to $1 \times 10^{-3} \Omega cm$ – outperforming commonly used aluminum doped zinc oxide (AZO). We assume that the enhancement in carrier concentration after hafnium addition originates from the passivation of the deep oxygen vacancies, that act as electron trapping states along with decreasing the grain boundaries. Besides enhancing electronic properties, the co-doping opens a new possibilities of tuning both optical band gap and electrical properties. The proposed co-doped zinc oxide can found applications in the future electronic devices utilizing TCO materials such as flat panels, solar cells and variety of sensors.



Figure 1. A) Optical bandgap widening with doping steadily increases, mainly due to Bernstein-Moss effect. B) Resistivity for differently doped samples shows a minimum around 4% and 6% for AZO and HAZO respectively.

ALD Grown ZnMgO:Al on Si for Photovoltaic Applications: Effect of High Mg Alloying and Al Doping.

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Heterojunction solar cells based on ZnO/Si employed as an emitter and absorber, present a high interfacial defect density due to the ~40% lattice mismatch between the two materials. Despite that, it has been predicted by simulations that a potential theoretical efficiency of ~24% can be achieved if the conduction band gap misalignment (ΔE_C) between ZnO and Si is eliminated by Mg alloying [1]. Experimentally an increase in efficiency from ~4% to ~7% can be achieved by following this route [2]. However, it has been also found that further improvement in efficiency is hindered by the increase in resistivity of the ZnO based layer when the Mg content overcomes \sim 2-3 at.%. In the present work it is found that by keeping the Al content equal to ~2 at.% and by increasing the Mg content up to ~12 at.% into the ZnO layers, the films still exhibit excellent electrical properties: carrier concentration and mobility equal to $\sim 2 \times 10^{20}$ cm⁻³ and ~ 2 cm² V⁻¹ s⁻¹ as displayed in Fig.1(a) (experimental point with x_{Mg} content < 5 are without Al). A constant increase of the open circuit voltage of the test solar cells from ~0.34 V (without Al doping) to ~0.44 V as a consequence of the Mg and Al introduction is observed as shown in Fig.1(b). The realized test solar cells with the optimized Mg and Al content exhibit an open circuit voltage, short circuit current density, efficiency, and fill factor equal to ~0.42 V, ~30 mA cm⁻², ~7.5%, and ~60%, respectively. A realized device and I-V curve under standard illumination are shown in Fig.1(c).



Fig1 (a) Carrier concentration, mobility, and resistivity vs Mg content (b) Open circuit voltage (V_{OC}) dependence on the Mg content (c) I-V curve under standard illumination of a test solar cell with Mg~12 at.%. (In (a) and (c) experimental data with $x_{Mg} < 5$ are without Al).

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Construction and studies of a cathode for photoelectrocatalytic hydrogen generation by water splitting

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Hydrogen is one of the most sustainable energy sources – it can be stored, transported, and used in fuel cells or combustion engines without the generation of any pollutants. Therefore, replacing fossil fuels with hydrogen has received intense attention in the last few years. However, hydrogen on Earth is mainly found as a component of molecules such as water and organic compounds. Today, it is produced mainly through steam reforming/gasification of fossil fuels, which not only consumes a large amount of natural gas, naphtha or coal, but also generates CO_2 . In the pursuit of CO_2 – free hydrogen technology, electrolysis of water is a promising method.

In this work we present the results of the construction and studies of an innovative cathode for photoelectrocatalytic hydrogen production by water splitting. The oxides and $MoSe_2$ were selected so that the light-excited electrons had sufficient energy to ensure the hydrogen generation reaction. At the same time, the positions of the conduction bands of these materials relative to the vacuum level were such to ensure the transport of light-generated electrons to $MoSe_2$, which can thus act as a catalyst for the reaction.

Tuning the parameters of the oxide and MoSe₂ deposition processes allowed us to find a range of technological parameters for which a mechanically stable electrode with promising opto-electrical parameters was obtained. Wide structural and photoelectrical studies were performed. Raman scattering related to characteristic phonon vibrations and Energy-dispersive X-ray spectroscopy techniques were used to identify the deposited materials. SEM images proved successful deposition of the oxide and MoSe2 on the electrode. Photocurrent resulting from the excitation of the constructed cathode immersed in an aqueous solution with radiation from the solar spectrum was also observed. The research carried out so far shows that the constructed electrode is promising in terms of the possibility of using it for photoelectrocatalytic water splitting.



Fig.1. Raman spectra of an electrode (green curve) and carbon/MO structure (red curve). In the low-energy part of the spectrum, peaks from MO are visible.



Fig.2 SEM image of the deposited metal oxide layer onto a carbon.

Degradation studies of MAPbI₃ perovskite using cathodoluminescence and microscopy techniques

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In modern times, considering climate change, we desperately look for sustainable energy sources. Photovoltaic cells are one of the most popular solutions to this problem. The widely used type of solar cells are those made of silicon, however, they are not without drawbacks - they are still too expensive and close to their efficiency limit. Moreover, their production is not as ecological as one could wish. In recent years, there has been a strong development of new generations of solar cells made with various materials acting as active layers. One of the most promising candidates for future applications in photovoltaics are hybrid organic-inorganic perovskites. The research shows that they can outperform Si cells in terms of efficiency, the cost of their production is much lower, and there are many ways to produce them in a more eco-friendly way[1].

The biggest problem with perovskites is that they can easily deteriorate under exposure to oxygen, water, and light. It is speculated that the degradation process depends on the size of the crystallites and the number of grain boundaries. In this study, we examined the effect of the perovskite (MAPbI3, where MA stands for methylammonium) synthesis process, carried out at different rates, on the surface morphology of MAPbI₃, in order to obtain the largest possible grains, and we followed the degradation process. Scanning electron microscopy and cathodoluminescence (CL) techniques were applied for the study. CL allowed the tracking of the degradation process at a spatial resolution of tens of nanometers, as the luminescence of MAPbI₃ and its degradation product PbI₂ are spectrally well separated (710 nm versus 490 nm). In the preliminary studies, it was shown that indeed the degradation is preferential at grain boundaries (Fig. 1).



Fig. 1. (left) SEM image and (right) SEM-CL image detected at 490 nm (maximum of PbI₂ luminescence) and 710 nm (maximum of MAPbI₃ perovskite luminescence).

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Polymer solar cells thermal annealing of active layers – impact of annealing temperature on efficiency and ageing rate

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Polymer solar cells are a third-generation photovoltaic cells that attract a great research interest in the last years. Their most important part is an active layer responsible for solar light absorption and exciton separation. The active layer is a nanoscale blend of organic semiconductors: donor and acceptor materials forming of bulk heterojunction. Typically, it is polymer (e.g. PTB7-Th) and fullerene derivative (e.g. PC₇₀BM). Polymer solar cells have some advantages comparing to silicon solar cells: low cost and ecological production, low weight, flexibility, possible semi-transparency and ease of production of multi-junction cells.

There are also some important disadvantages: lower efficiency and weaker air stability. One of these cons, lower efficiency, has been improved by application of a non-fullerene acceptor materials in form of low band-gap organic molecules e.g. BTP-4Cl-12 [1]. Such new acceptors have higher absorption near maximum of solar light spectrum than regular organic acceptor e.g. PC₇₀BM with maintaining the same or better electron mobility. For the new acceptor with lower band-gap there were developed new donor materials like PBDB-T-2F with matching band-gap and energy levels what increased efficiency to over 17% in 2019 [1].

We present our results of research on two active layers of polymer solar cells: well known PTB7-Th:PC70BM and state-of-art PBDB-T-2F:BTP-4Cl-12. We tested annealing temperature impact on crucial parameters of the cells and on their long time stability. Thermal annealing not only removes solvents residues but also improves carrier mobility. The effects of annealing on the cell efficiency were showed in Fig.1. We obtained maximum cells efficiency 5.1% for PTB7-Th:PC70BM annealed at 50°C and 8.1% for PBDB-T-2F:BTP-4Cl-12 annealed at 130°C. Optical absorbance spectroscopy was carried out to check influence of thermal annealing on excitonic transitions. We observed clear BTP-4Cl-12 absorbance peak redshift after annealing (Fig. 2). The photocurrent spectra confirmed efficient current generation in both donor and acceptor materials. To investigate long-time stability of the cells a 4 months ageing measurements were performed. We have found out that optimal thermal annealing significantly improved cell stability.





Fig. 1. Average power conversion efficiency (PCE) vs. annealing temperature.

Fig. 2. Absorbance spectra of BTP-4Cl-12 annealed at different temperatures

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Towards Electrostatically Gated Perovskite Thin Films

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Perovskite semiconductors are widely studied in the context of solar light harvesting used for photovoltaic applications, but less in the context of their fundamental physical properties. As a medium exhibiting strong light-matter coupling, they may constitute an interesting platform for optoelectronics and opto-spintronics in particular.

Our long-term goal is to study spin dynamics in those materials to explore their potential applications in opto-spintronics. For that, we need to control the charge density in studied structures in a reliable way.

Here we present our efforts in the fabrication of electrostatically gated perovskite thin film encapsulated in hexagonal boron nitride - needed for the protection of the perovskite material from contamination and degradation. The electrostatic gating in the proposed van der Waals heterostructure is performed with the use of a graphene flake making and electrical contact with the perovskite film, as shown in Figure 1.

In our approach we modify the methods widely applied for the fabrication of transition metal dichalcogenide (TMD) structures, not only to obtain perovskite structures of high optical quality, but also to exploit the possibility to combine them with TMD materials in the future.



Figure 1: A schematic of perovskite thin film encapsulated between hexagonal boron nitride (hBN) slabs, electrostatically gated via graphene flake.

The structure and properties of bilayer borophene

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Borophene is a crystalline atomic monolayer of boron that exhibits several polymorphic forms [1]. In recent years, borophene has been identified as a rising star in materials science for the development of sensors, energy storage devices, and batteries.

Very recently it was experimentally demonstrated [2] the synthesis of an atomically well-defined borophene polymorph beyond the single-atomic-layer limit. The structure of the bilayer borophene is consistent with two covalently bonded α -sheets. While these sheets retained the highly desirable electrical and mechanical properties of single-layer borophene, it is proposed that the space between the layers could be used for energy or chemical storage. For example, the inclusion of a layer of lithium ions could enable the manufacture of 2D batteries.

In this work, we present the results of first-principles calculations for double-layer boron structures consisting of hexagonal, α -sheet, or honeycomb boron layers (see Fig. 1). Different arrangements between the layers have been considered. For the most stable structures, the electronic, elastic, and transport properties are studied. According to our calculations, the high metallic character of borophenes is, in most cases, preserved also for the double-layer structures. Also, the structural stability of the single layers is reinforced by the presence of interlayer bonds.



Figure 1: The atomic structure of double layer borophene. The single layers are boron honeycomb (hc) structures. The bottom panel shows the bonding between the layers.

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Towards Strain Engineering of High Quality 2D Perovskite Flakes

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While strain engineering is a well-established tool to tune the optoelectronic properties of graphene and transition metal dichalcogenides [1], this approach is still at its infancy for two-dimensional (2D) metal halide perovskites. Before systematically applying uniaxial or biaxial stresses to 2D perovskites, it is essential to introduce an exfoliation strategy which enables to obtain flakes of optical quality comparable to that of the parent crystal.

Here, starting from $(PEA)_2PbI_4$ 2D perovskite crystals shown in Fig. 1(a), we exfoliate and encapsulate in hBN thin perovskite flakes [see Fig. 1(b)]. The high quality of both crystal and exfoliated flakes is demonstrated by the sharp resonances visible in the reflectance spectra of Fig. 1(c,d), attributed to bright, fine-structure split excitons. To test the response to external mechanical stimuli, we transferred $(PEA)_2PbI_4$ flakes to the non-planar substrates shown in Fig. 1(e) [2], and measured the room temperature PL spectrum as a function of the strain. The variation of the PL energy with increasing strain suggests the presence of an effect of the mechanical deformation on the electronic properties of these flakes.



Figure 1: (a) Bulk $(PEA)_2PbI_4$ crystals used for the exfoliation of thin perovskite flakes. (b) hBN-encapsulated $(PEA)_2PbI_4$ flakes. Low temperature PL and reflectance spectrum of (c) the parent bulk $(PEA)_2PbI_4$ crystal and the exfoliated flakes. Relevant excitonic resonances are indicated. (e) Photograph of the curved substrates on which some flakes were transferred. (f) Room temperature PL energy of a flake transferred on substrates of different radii of curvatures.

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Solutions for spectroscopy and imaging

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Van der Waals materials for nanophotonic applications

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Nanophotonic structures enable a range of applications including optical waveguiding, Purcell enhancement of emission, low-threshold lasing and higher harmonic generation enhancement. Many research fields and technologies have benefited from nano-scale resonators and waveguides previously fabricated from noble metals [1] or high refractive index dielectrics such as silicon [2] and III-V materials [3]. While these offer a large range of opportunities for both research and technology, van der Waals materials may expand the possibilities of nanophotonics in the visible and near-infrared portion of the spectrum due to high, super-Mossian refractive indices $(n\approx 4)$ [4], a large range of transparency windows, and numerous advantages due to their weak van der Waals attractive forces. In order to inspire and facilitate nanophotonic structures fabricated from these layered materials, we extract the dielectric constants of a diverse set including transition metal dichalcogenides (TMDs), III-VI semiconductors, and magnetic layered materials. Employing well established techniques, we fabricate nanoresonators with a range of geometries from these materials and observe Mie resonances as well as strong coupling between the excitonic features of TMDs and anapole modes with Rabi splittings up to 140 meV. After the transfer of a monolayer of WSe₂ onto WS₂ nanoantennas, we observe room temperature Purcell enhancement of emission [5] and low temperature formation of single photon emitters with enhanced quantum efficiencies within a system fabricated entirely from layered materials. Due to the weak van der Waals interactions of the nanoresonators and the substrate, we were able to employ an atomic force microscopy (AFM) cantilever in the repositioning of double-pillar nanoantennas to achieve ultra-small gaps (≈10 nm) [5]. This post-fabrication technique enables applications such as stable, low-power optical trapping of quantum emitters with Purcell enhancement factors above 150.

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Excitation Dependent Energy Transfer in 2D Heterostructure

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Monolayer (1L) thick two-dimensional (2D) transition metal dichalcogenides (TMDs) exhibit strong light-matter interaction [1]. Vertical heterostructures (HSs) made by the stacking of different TMDs have already shown promises for future optoelectronic device applications [2]. Interlayer charge (CT) and energy transfer (ET) are the two main photocarrier relaxation pathways in the TMD HSs. The interlayer CT process mainly occurs due to the energy level offset between the two materials, which survives only up to few nm distance [3]. Whereas, the long-distance interlayer ET process can survive up to several tens of nm [4]. Our previous work [5] has shown that due to the strong overlap between the optical *bandgaps* of the two TMDs, interlayer ET process dominates over the traditional CT process. In the present work [6], we explore the effect of resonant *overlap* between the *high-lying* excitonic resonances in the long-distance ET process in HS made by the 1Ls of molybdenum disulfide (MoS_2) and tungsten diselenide (WSe_2). At, low temperature (8 K) resonant excitations from the lower bandgap WSe₂ B and D excitonic levels create more photocarriers at the higher bandgap MoS₂ layer, resulting an enhanced ($\sim 2\times$) MoS₂ PL emission from the HS area. With the increasing temperature the enhanced PL emission starts to destroy due to the increased electron-phonon scattering. Our work provides a new insight into the longdistance interlayer ET process from the lower-to-higher bandgap TMD material, which will be beneficial in developing TMD-based future optoelectronic device applications.

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Dexter induced inverted valley polarization in monolayer WSe_2

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A highly efficient, but not particularly explored, intervalley scattering mechanism in transition metal dichalogenides resembles a Dexter coupling in the reciprocal space. This mechanism couples exciton states with the same spin in different valleys [1].

Here, we demonstrate that the Dexter-type intervalley scattering affects not only the neutral exciton [1], but also other excitonic complexes of a monolayer WSe_2 . While the photoluminescence (PL) spectrum is co-polarized with the excitation laser when this is tuned far-off resonance with either A or B exciton, its degree of circular polarization (DCP) becomes negative when the laser light is tuned resonant with B exciton, as shown in Fig. 1(a). This vanishing or negative polarization of the excitation and biexciton complexes is confirmed by the polarization-resolved PL excitation (PLE) of Fig. 1(b,c,d). Pump-probe measurements shown in Fig. 1(e,f) demonstrate that immediately after an excitation pulse resonant with A exciton is absorbed, the DCP is negative, thus supporting the Dexter coupling as main mechanism driving the exciton dynamics in this condition.



Figure 1: (a) PL spectra of encapsulated monolayer WSe₂ with detection of the light co-polarised $(\sigma^+\sigma^+)$ and cross-polarised $(\sigma^+\sigma^-)$ to excitation. Excitation energy between A_{2s} and B_{1s} in top panel and resonant with B_{1s} exciton in the bottom panel. (b,c,d) Excitation energy dependent PL intensity of the co- and cross-polarised emission and calculated degree of circular polarisation (DCP) for A_{1s}, XX and XX⁻, respectively. (e) Photoinduced absorption of the B_{1s} exciton and (f) degree of circular polarization for an excitation resonant to A_{1s} exciton.

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Hot luminescence or Raman scattering in monolayers of MoSi₂N₄

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Monolayers (MLs) of semiconducting transition metal dichalcogenides (S-TMDs), *e.g.* MoS_2 and WSe_2 , have been demonstrated to carry the spin-like degree of freedom known as valley pseudospin due to the optical bandgap located at the K[±] points of their hexagonal Brillouin zone [1]. Recently, MLs of the MSi₂Z₄ family (M = Mo, W; Z = N, P, As, Sb), which form a new class of hexagonal non-centrosymmetric materials hosting extraordinary spin-valley physics, have been discovered [2].

In this work, we investigate the optical response of the $MoSi_2N_4$ ML, grown using chemical vapor deposition on Si/SiO₂ substrate [3], with the aid of photoluminescence (PL) performed in a wide range of temperature (5 – 300 K) and first principles calculations.

The relative PL spectra, calculated as the difference between the PL spectra measured on the ML and on the Si/SiO₂ substrate, as shown in the Figure. MoSi₂N₄ MLs are semiconductors with an indirect band gap of about 1.94 eV at 300 K. However, the excitation of the PL spectra with high energies of 3.06 eV or near to the band gap of 1.96 eV does not give rise to a measurable signal either at 300 K or at 5 K. Nevertheless, a significant PL is apparent under excitation close to the so-called A and B direct transitions in the K^{\pm} valleys [3]. The obtained relative spectra are combined of two types of



Figure Relative PL spectra of $MoSi_2N_4$ ML under different excitations. The insets show the spectrum at 5 K and the optical image of the studied ML.

emission, *i.e.* a broad band with linewidths of ~200 meV, on top of which a series of narrow peaks emerges. Consequently, it is difficult to decompose this emission to Raman scattering or hot luminescence, since the ratio between the resonantly enhanced Raman signal and the optical recombination at the K[±] points is unknown. At T=5 K, only the Raman peaks are seen, which were ascribed to phonon modes using the calculated phonon dispersion spectrum.

Our results are the spore of research devoted to the MLs of the $MoSi_2N_4$ family, which properties locates them as ideal candidates for valleytronics, in line with S-TMD MLs.

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Surface and Structural Analysis of (110)-oriented Pb_{1-x}Sn_xTe Topological Crystalline Insulator

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The novel class of semiconductors namely topological crystalline insulators (TCIs) is a rapidly developing research area of modern solid state physics. In these materials, surface electron states with Dirac dispersion relation are formed on certain high symmetry surfaces. Pb_{1-x}Sn_xTe alloy has been experimentally discovered as a member of TCIs family in the previous decade. It exhibits topological phase transition above a critical concentration of Sn ($x \ge 0.36$ at LHe temperature) [1]. Theoretical predictions of band inversion at high symmetry L point in the Brillouin zone, and experimental evidence for topologically protected states on (001) [2] and (111) surfaces [3] of Pb_{1-x}Sn_xTe alloy have already been proclaimed in literature. However, (110)-oriented Pb_{1-x}Sn_xTe has not been studied experimentally in this context yet. We have grown thin layers of Pb_{1-x}Sn_xTe solid solution by molecular beam epitaxy (MBE). To our knowledge the growth of Pb_{1-x}Sn_xTe in the (110) orientation has not been reported so far. We have found that direct growth on epi-ready

GaAs(110) substrates resulted in non-uniform layers with mixed (100) and (111) orientations. To get around this problem we have used hybrid substrates with thick CdTe(110) buffer grown in separate MBE system. The layers have been investigated by several characterization techniques such as energy dispersive X-ray fluorescence, atomic force microscope, X-ray diffraction and, high resolution transmission electron microscopy which reveals the (110)-orientation of $Pb_{1-x}Sn_xTe$. Angle-resolved photoemission experiments are underway to identify topologically protected surface states on the Pb₁₋ $_x$ Sn_xTe(110) surface.



Fig. $2\Theta/\omega$ scan of (Pb,Sn)Te/CdTe//GaAs(110). Both (Pb,Sn)Te layer and CdTe buffer show same out of plane <011> growth

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Ferromagnetic topological crystalline insulator Sn_{1-x}Mn_xTe in an inhomogeneous magnetic field

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The archetypical topological crystalline insulator (TCI) SnTe doped with manganese becomes ferromagnetic at low temperatures. In the ferromagnetic state, there is an additional contribution to the Hall resistance, the anomalous Hall effect (AHE, R_{xy}^a), which depends on the magnetization of the material. An inhomogeneous external magnetic field modifies the spin texture of the magnetic material and can lead to the so-called topological Hall effect. The inhomogeneous magnetic field can be created by placing magnetic or superconducting (SC) structures on top of the layer under investigation. In the case of an SC layer made of type II superconductor, the inhomogeneity is caused by a vortex lattice formed in the magnetic field.

Previously, the effect of an inhomogeneous magnetic field on magnetotransport was studied mainly in non-magnetic materials. In this work, we extend this approach to ferromagnetic topological insulators. We grew 30-50 nm thin layers of $Sn_{1-x}Mn_x$ Te with a Curie temperature of around 5 K by molecular beam epitaxy. These layers were then processed in two-section Hallbars. One of the sections was covered with a SC Nb separated from the epilayer by a thin (~20 nm) oxide layer. The other section of the Hall bar was left bare for comparison. The influence of the SC layer on the magnetotransport of TCI ferromagnet was probed by measuring AHE in the $Sn_{1-x}Mn_x$ Te layer. We found that the AHE hysteresis loop in the covered area, $R_{xy}^{a,c}$, was wider than in the uncovered area, $R_{xy}^{a,u}$, as seen in Fig. a,b. This widening is caused by the Meissner effect and is a direct consequence of the uncovered and covered areas gives profiles that strongly resemble the magnetization curves of a superconductor (Fig. c). We also observed that additional peaks appear on a $R_{xy}^{a,g} - R_{xy}^{a,u}$ curve, indicating the emergence of the topological Hall effect caused by a SC flux inhomogeneity.



(a,b) AHE contribution to the Hall resistance R_{xy}^a in Sn_{0.92}Mn_{0.08}Te measured as a function of the magnetic field B for various temperatures. AHE hysteresis in the Hallbar under the SC is much wider than in the bare epilayer. (c) The difference between AHE curves in covered and uncovered areas ($R_{xy}^{a,g} - R_{xy}^{a,u}$ resembles magnetization hysteresis. However, under certain conditions additional spike (marked with a magenta circle) appears, which can be considered as a signature of the topological Hall effect.

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Signatures of Weyl Fermions in Magnetotransport of Topological Semimetal α-Sn

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Topological semimetals have emerged as an attractive platform for both applied and fundamental research. This has been stimulated by their unique electronic properties, namely the presence of topological surface states (TSSs) and a linear, Dirac-like band structure in the bulk. Among the many compounds studied, α -Sn (grey tin) is of particular interest as a rare example of an elementary material with an inverted band structure. Nominally a zero-gap semiconductor, it can be turned into both Dirac semimetal (DSM) and topological insulator (TI) phases by strain [1]. Despite its rich topological phase diagram, most of reports focus on the properties of TSSs in the TI phase of grey tin.

Successful molecular beam epitaxy (MBE) on a hybrid, insulating CdTe/GaAs(001) substrates allows us to address a weakly explored part of the grey tin topological phase diagram of gray tin. With film thicknesses up to 200 nm and compressive in-plane strain introduced by the substrate, we ensure that our α -Sn layers are in the DSM phase. Here we will present the results of a thorough magnetotransport studies of our structures, focusing on signatures of 3D Dirac and Weyl fermions.

The π -Berry phase of Shubnikov-de Haas oscillations is found in magnetic fields both perpendicular (B_{perp}) and parallel (B_{paral}) to the current, which is a signature of the 3D nature of the observed carriers and their non-trivial topology. The low effective mass derived from the temperature-dependent amplitude of the oscillations, $m^* \approx m_0$, is consistent with previous reports. Importantly, we perform a detailed study of the negative longitudinal magnetoresistance (NLMR), present in B_{paral} up to 30 T. By ruling out alternative sources of NLMR, we are able to relate it to the chiral anomaly – a phenomenon expected for Dirac and Weyl semimetals. Finally, we analyse the anisotropic magnetoresistance (AMR) and the planar Hall effect (PHE), observed when the sample is rotated in a constant, in-plane magnetic field. We show that these effects exhibit an angular dependence consistent with anomaly-related magnetoresistance. Altogether, our work establishes the electronic properties of α -Sn as consistent with the 3D DSM phase, confirming previous theoretical predictions [1] and laying a solid foundation for further research.

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Magnetic Circular Dichroism from Exchange Splitting in Intrinsic Magnetic Topological Insulator MnBi2Te4

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MnBi2Te4 has attracted significant attention due to exhibit both its non-trivial topology and its magnetism, which give rise to interesting phenomena such as the quantum anomalous Hall effect [1-3] and the axion insulator effect [4]. However, most studies have focused on the properties of its surface states while its bulk-originated bands have remained relatively understudied. In this study, we report recent magnetooptical infrared spectroscopy measurements conducted on the intrinsic magnetic topological insulator MnBi2Te4. We carried out infrared absorption measurements under magnetic fields up to 34 T and observed an optical absorption in the mid-infrared that shifts to higher energy as the magnetic field increases. The energy of the transition versus magnetic field reflects the behavior of the anomalous Hall effect. The transition energy increases through the canted magnetic state of MnBi2Te4 and then saturates at 8 T when the system enters a ferromagnetic state. Then, the transition energy further increases when the magnetic field is above 25 T, where the magnetization of Mn antisite defects is likely to start flipping. Additionally, this absorption displays a strong magnetic circular dichroism, which likely originates from band splitting under the effect of magnetic exchange. Our studies not only reveal the infrared response of MnBi2Te4 but also provide insight into how the electronic structure changes through the phase diagram of this material, consistently following the behavior of the magnetization.

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Non-trivial topological phases in transition metal rich half-Heusler Oxides

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Topological Insulators with gapless surface states and insulating bulk in noncentrosymmetric cubic systems have been extensively explored following the discovery of two-dimensional quantum spin hall effect in zincblende HgTe.[1,2] In such systems the negative band inversion strength function E_{BIS} (= $E_{\Gamma 6}$ - $E_{\Gamma 8} < 0$) governs the robustness of the non-trivial topological states at ambient conditions.[3] Hence, realizing large negative values of E_{BIS} has been a guiding motivation of several investigations reported in literature. Here, we present a material design approach which can be employed to realize large negative values of EBIS in cubic materials such as Half-Heusler (HH) oxides with 18 valence electron configurations.[4] We explore, α -, β -, and γ -phase (by placing transition metal atom at different Wyckoff positions) in HH oxides XYO (X = Li, K, Rb; Y = Cu, Ag, Au) for their non-trivial topological phases. Off these three phases, we observed that, the α -phase of nine HH oxides (wherein the transition metal atoms occupy 4a Wyckoff positions in the crystal structure) is the most promising with non-trivial topological phase, governed by the Mass-Darwin relativistic effects which enhances the E_{BIS} function. Whereas the other phases are either trivial semi-conductors or conductors. We find that the E_{BIS} in the α -phase is as high as -2.37 eV which would remain robust under ambient conditions. We also find that under modest strain fields, the band inversion strength is robust retaining the non-trivial character. The non-trivial character of these compounds is further verified by analyzing the orbital characters across the Fermi level, ARPES-like surface state spectra, slab band structures and the Z_2 invariants. We believe that the α -phase of HH oxides presented here can be synthesized experimentally for diverse room temperature applications.

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Topological phase transition in SnTe topological crystalline insulator thin films

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The quantum spin Hall (QSH) phase was theoretically predicted in (111)-oriented thin films in a certain range of thicknesses in the SnTe class of topological crystalline insulators (TCIs). The energy gap changes sign and oscillates with increasing the number of layers due to intersurface hybridization. This leads to the emergence of the two-dimensional (2D) topological insulator phase classified by a \mathbb{Z}_2 invariant protected by time-reversal symmetry [1,2].

We study topological properties of symmetric (111)-oriented films with a twin plane (TP) in the middle of the structure. The TP forms a (111) mirror symmetry plane and can be either cationic or anionic type. Recently, we have shown that both a cationic and an anionic TP defines a 2D TCI with mirror Chern number $C_m = 2$ and $C_m = 1$, respectively [3]. Using tight binding approximation we calculate mirror Chern number and \mathbb{Z}_2 invariant for thin films with TPs for a wide range of thicknesses. We show that TP topology combined with finite-size effects can influence how the topological invariants change with growing film thickness.

In addition, we investigate phase transitions in (001) films by calculating the mirror Chern number and find that it oscillates between $C_m = +2$ and $C_m = -2$ with growing thickness. Using a realistic tight binding model we uncover that the oscillation is made up of two components — a slow one described in Ref. [4], and a fast one, due to valley mixing, which changes the sign of C_m with every two added layers. Following Ref. [5] we consider films with atomic steps located symmetrically on the two surfaces. Such steps constitute boundaries between areas of different film thickness. In agreement with Ref. [5], we find that there appear gapless step-edge modes protected by (001) mirror symmetry, whenever the thicknesses on both sides of the step correspond to different values of C_m .

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Scanning Gate Microscopy Probing of Electron-Hole Interference in a Normal-Superconductor Junction

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We theoretically investigate Scanning Gate Microscopy (SGM) probing of electron and hole trajectories in a two-dimensional Normal-Superconductor (NS) junction embedding a Quantum Point Contact (QPC). SGM is a well-known technique that has been used to visualize branched electron flow from QPCs [1]. In our work, taking advantage of recent progress in the realization of NS junctions in gated heterostructures [2], we propose to use this method for probing the paths of electrons and Andreev-reflected holes. We find that in an NS junction the conductance probed by the SGM technique exhibits oscillations that are due to the self-interference of electrons and holes. In contrast to ordinary SGM measurements, the interference occurs not only between the QPC and the SGM tip, but also between the tip and the NS interface. Most importantly, we show that for the measurements performed at a non-zero bias, the oscillations exhibit beating with the two periods determined by the two Fermi wavelengths that correspond to the electron and the hole wave-vectors. Finally, we show that at non-zero energy, the hole does not retrace the electron path due to the difference in the incident and reflection angles at the NS interference which, in turn, leads to a pronounced interference pattern in the conductance map [see Fig. 1].



Figure 1: Electron-hole interference in SGM conductance maps.

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Non-local Transport Signatures of Topological Superconductivity in Planar Josephson Junctions

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Finding signatures of topological superconductivity and the appearance of elusive Majorana-bound states has become one of the most significant challenges of modern solid-state physics. Josephson junctions, realized as hybrid nanostructures defined in a two-dimensional electron gas proximitized by a superconductor, have been considered promising candidates for the realization of well-controlled and scalable topological elements [1,2]. In this work, we propose and theoretically study a new method for probing the topological features of these systems. Instead of typical tunneling spectroscopy, which is sensitive only to states localized in the vicinity of the tunneling barrier [3] and prone to disorder-induced effects, we propose using a technique that probes the system in an open regime via non-local transport. In this configuration, the non-local conductance corresponds to the band structure of the system and enables elucidating the closing and reopening of the gap upon an increase in the phase difference, which is associated with the topological transition. This, however, happens only when the non-local signal reveres its sign when crossing zero energy and occurs due to the change of quasiparticle character of the bands at zero momentum. Moreover, we demonstrate that phase evolution is highly dependent on the strength of the indispensable Zeeman interaction, significantly changing the full phase diagram available for measurement in realistic experiments on hybrid Josephson junctions.

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Quantum Anomalous Hall effect and axion insulator phase in HgTe material class

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Dilute magnetic semiconductors have played a central role in the demonstrating and describing a strong and intricate sp-d exchange interactions, paving the way for the rise of magnetic topological insulators [1,2]. Recently, the exchange splittings of magneto-optical spectra in $Cd_{1-x}Mn_x$ Te and $Hg_{1-x}Mn_x$ Te have been described [3] and it has been demonstrated that superexchange dominates in magnetic topological insulators [4].

We study theoretically the interplay between magnetism and topology in HgTe-based systems in order to obtain the quantum anomalous Hall phase and the axion insulator phase.

To engineering the quantum anomalous Hall phase, we investigate the electronic and magnetic properties of the dilute magnetic semiconductors $Cd_{1-x}Cr_xTe$, $Hg_{1-x}Cr_xTe$, $Cd_{1-x}V_xTe$, $Hg_{1-x}V_xTe$ by using a density functional theory approach which goes beyond the standard functionals in order to correctly reproduce the topology and the band gap of these systems. We study the exchange couplings for all considered cases and we find that the coupling is ferromagnetic in case of doping with V, differently from the case of doping with Mn and Cr, where we find antiferromagnetic couplings. We conclude that the ferromagnetic coupling among V atoms in the insulating phase of topological HgTe can produce the quantum anomalous Hall phase[5].

To generate the axion insulator, we study the three-dimensional HgTe/MnTe superlattices stacked along the (001) axis. Our results show the evolution of the magnetic topological phases with respect to the magnetic configurations. An axion insulator phase is observed for the antiferromagnetic order with the out-of-plane Néel vector direction below a critical thickness of MnTe, which is the ground state amongst all magnetic configurations. Defining T as the time-reversal symmetry, this axion insulator phase is protected by a magnetic two-fold rotational symmetry $C_2 \cdot T$. The axion insulator phase evolves into a trivial insulator as we increase the thickness of the magnetic layers. By switching the Néel vector direction into the ab plane, the system realizes different antiferromagnetic topological insulators depending on the thickness of MnTe. These phases feature gapless surface Dirac cones shifted away from high-symmetry points on surfaces perpendicular to the Néel vector direction of the magnetic layers[6].

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Geometry of the (001) surface and electron dispersion relations of topological IV-VI semiconductors.

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In the electronic structure of topological crystalline insulators (TCI) from the IV-VI semiconductor family there are four Dirac points in the (001) surface Brillouin zone on $\overline{\Gamma} - \overline{X}$ directions. At these points, the energy band gap for surface states is zero due to mirror symmetries with respect to the (110) and ($\overline{1}10$) planes. However, in the experimental works [1-3] it was discovered that there are two pairs of the surface states, one with zero gap and the other with nonzero gap. The opening of the gap suggests that one of the above mirror symmetries is broken. Scanning Tunneling Microscope measurements [2] revealed that the anion and cation sublattices at the surface are displaced relative to each other along the [110] direction by a vector (0.3, 0.3) Å. Such a displacement leads both to breaking the mirror symmetry with respect to the ($\overline{1}10$) plane and the opening of the gap at two Dirac points.

Here, using the local density functional methods we analyse detailed atomic configurations at the (001) surfaces of PbTe, PbSe, SnTe and SnSe. Our results reveal rumpling, *i.e.* relative displacements of anion and cation sublattices in the [001] direction for the first few monolayers, and oscillatory character of interlayer distances between them. Both results are consistent with the previous theoretical and experimental data for the rock salt crystals. However, contrary to the claim of Ref. [2] we do not find any displacements of surface atoms in the (001) plane.

As expected, the rumpling does not open the band gap, because it does not break the mirror symmetry. We also show the dispersion relations for different Sn concentration in $Pb_{1-x}Sn_xSe$ slabs. The chemical disorder leads to the opening of the gap in TCI phase. For Sn concentrations corresponding to Weyl semimetal phase we observe an interval on the $\overline{\Gamma} - \overline{X}$ direction where the gap is vanishingly small.

Calculations are performed using a slab geometry. Optimization of atomic configuration is performed using OpenMX package [4]. Based on an initio results we construct a tight binding hamiltonian for thick slabs. Importantly, to obtain correct results it is necessary to take into account the dependence of tight binding parameters on the distance from the surfaces of the slab.

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Magnetic and electrical properties of CoRE₂W₂O₁₀ ceramic materials

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Divalent metals tungstates, MWO₄ (M = Mn, Co, Zn, Cd and Pb), have been successfully used in spectroscopic and radiometric devices and as heavy and fast scintillators [1]. Many of them, *i.e.* when ionic radius of M^{2+} is relatively small, adopted monoclinic wolframite-type structure [2]. In turn, trivalent rare-earth metals tungstates (RE₂WO₆) exhibit many structural types including monoclinic symmetry with space group C2/c (where RE = Pr–Dy) [3]. They have been used in diode-pumped crystal lasers, new generation lighting in optical telecommunications, lidars and other applications requiring narrow spectral sources [4]. Our earlier studies on MPr₂W₂O₁₀ (M = Mn, Co and Cd) and CdRE₂W₂O₁₀ (RE = Y, Nd, Sm, Gd–Er) tungstates obtained by high-temperature sintering of adequate MWO₄/RE₂WO₆ mixtures showed that these new compounds crystallize with orthorhombic or monoclinic structure and exhibit generally non-conductive and paramagnetic properties [5,6].

Samples of other new CoRE₂W₂O₁₀ (RE = Y, Dy, Ho, Er) tungstates were obtained by high-temperature solid-state reaction method using CoWO₄ and RE₂WO₆ as the starting materials [7]. These materials were pressed and sintered at 1050°C to a ceramic form. Powder diffraction measurements of as-prepared sinters were carried out on a PANalytical PW1050 diffractometer. Static dc magnetic susceptibility was measured within the temperature range of 2–300 K in the zero-field cooled (ZFC) and field cooled (FC) mode. For these purposes a Quantum Design MPMS-XL-7AC SQUID magnetometer (Quantum Design, USA) was used. Electrical conductivity $\sigma(T)$ was measured by the dc method using a KEITHLEY 6517B Electrometer/High Resistance Meter within the temperature range of 77–400 K. Broadband dielectric spectroscopy measurements were carried out using pellets, polished and sputtered with (~80 nm) Ag electrodes within the frequency range of $2 \cdot 10^2 - 2 \cdot 10^6$ Hz using a Hioki 3532-50 LCR HiTester and within the temperature range of 80–400 K.

The results of structural, magnetic and electrical measurements of $CoRE_2W_2O_{10}$ ceramics (RE=Y, Dy, Ho, Er) showed orthorhombic-type structure, paramagnetic state and insulating behaviour as well as a large value of relative permittivity above room temperature, which strongly depends on presence of $3d \text{ Co}^{2+}$ ions, but not on the type of $4f \text{ RE}^{3+}$ ones. The important property of ceramic samples is that they have a greater ability to accumulate charge in comparison with the micro-crystalline ones.

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Magnetic properties of semiconducting ZnCr₂Se₄:Re single crystals

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Seleno-spinels are promising compounds in the commercial use of thermoelectric devices [1] due to a fairly large cubic cell unit (~10 Å) and a strong covalent bond. Pure ZnCr₂Se₄, both in mono and polycrystalline form combines *p*-type semiconductor conductivity and helical antiferromagnetic (AFM) order below the Néel temperature $T_N = 20$ K with a strong ferromagnetic (FM) short-range interaction, as evidenced by the high positive Curie-Weiss temperature $\theta = 115$ K [2,3]. The magnetic order at T_N is accompanied by structural transformation from cubic $Fd\bar{3}m$ to tetragonal $I4_1/amd$ symmetry with a small contraction along the *c* axis [4]. Rhenium is a conductor and paramagnetic, it has hexagonal symmetry and crystallizes in $P6_3mmc$ space group. The unit cell parameters of rhenium crystal are as follows: a = 276.0 and c = 445.8 pm [5].

ZnCr₂Se₄ single crystals doped with rhenium ions with the content of 0.06, 0.07, 0.08, 0.09. 0.10 and 0.11 located in the tetrahedral sites of the spinel structure were grown from binary selenide ZnSe, elemental rhenium and selenium by chemical vapor transport with anhydrous CrCl₃ as transport agent. Dynamic AC magnetic susceptibility, χ_{ac} , was measured in the temperature range 2-300 K and at an internal oscillating magnetic field $H_{ac} = 1$ Oe with an internal frequency f = 120 Hz taken at external static DC magnetic fields $H_{dc} = 0$, 10, 20, 30, 40 and 50 kOe. Magnetization isotherms, M(H), were measured at 2, 4, 10, 20, 40, 60, and 300 K and in the static DC magnetic field up to 70 kOe. Specific heat measurements were made in the temperature range of 2-300 K. For these purposes a Quantum Design MPMS-XL-7AC SQUID magnetometer was used. The electrical conductivity $\sigma(T)$ was measured by the DC method using KEITHLEY 6517B Electrometer/High Resistance Meter in the temperature range of 77–400 K.

Magnetic (AC), specific heat, and electrical (DC) measurements showed semiconductor behavior, coupling of the phonon system and electron gas by a spin lattice determined by the Debye temperature of 295 K, two singularities on magnetization isotherms in critical fields of 12 and 60 kOe, AFM long-range order with the Néel temperature of $T_N = 21.7$ K, and a strong FM short-range interaction, as evidenced by an increase in the positive Curie–Weiss temperature, θ , from 75 to 83 K with increasing Re content in the sample. With the increase of the external DC magnetic field, a shift of T_N towards lower temperatures and θ towards higher ones was observed, as well as the appearance of broad peaks in the paramagnetic region at H_{dc} = 40 and 50 kOe, characteristic of spin fluctuations. Similar behavior was found in ZnCr₂Se₄:Ta single crystals [6].

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Magnetic and electrical properties of CuCr₂Se₄ nanoparticles

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Magnetic spinels with general formula CuCr_2X_4 (where X = S, Se and Te) are ferromagnetic [1] and metallic conductor at room temperature [2]. For this reason, these spinels have been extensively studied in terms of their potential applications in thermoelectric devices [3]. CuCr_2Se_4 obtained by solid phase synthesis has a normal cubic type structure with the symmetry of the Fd $\overline{3}$ m space group and the zero magnetic moment of the Cu ion in the tetrahedral site, whereas the three Bohr magnetons, corresponding with Cr^{3+} ions in the octahedral site [4]. Pure CuCr_2Se_4, both in mono and polycrystalline form has strong ferromagnetic long-range interactions below the Curie temperature of $T_C = 460$ K with strong ferromagnetic short-range interactions evidenced by a large positive Curie-Weiss temperature of $\theta = 465$ K [1] as well as the electrical resistivity of $3.8 \cdot 10^{-6}$ Ω m and the positive Seebeck coefficient of 20.5 μ V/K at room temperature [2].

Spinel nanocrystallites of CuCr₂Se₄ with a size of 32 nm were obtained as a result of five-hour grinding of single crystals of this compound (in order to avoid phase impurities) by high-energy ball milling. The static magnetic susceptibility was recorded both in zero-field-cooled and field-cooled mode. Dynamic magnetic susceptibility was measured at an internal oscillating magnetic field $H_{ac} = 3.9$ Oe with an internal frequency f = 1 kHz in the temperature range of 5–300 K. Magnetization isotherms were measured at 5, 20, 40, 60, and 300 K using a SQUID magnetometer in applied external fields up to 70 kOe. Electrical conductivity $\sigma(T)$ and thermoelectric power S(T) of the nanoparticles under study were measured by the DC method using a KEITHLEY 6517B Electrometer/High Resistance Meter and a Seebeck Effect Measurement System, respectively, within the temperature range of 77–400 K.

Magnetic and electrical studies have shown that the reduction of CuCr_2Se_4 single crystals to the size of nanoparticles leads to: 1) weakening of the long-range ferromagnetic interactions visible in the decrease of the Curie temperature from 460 to 196 K, 2) weakening of the short-range ferromagnetic interactions visible in the decrease of the paramagnetic Curie-Weiss temperature from 465 to 231 K, 3) lack of saturation of magnetization at 5 K and 70 kOe, 4) change of electrical conductivity from metallic to semiconductor, and 5) reducing the thermoelectric power factor σS^2 by three orders of magnitude at 400 K. The above results were considered in terms of magnetic susceptibility. Calculations of the exchange integrals and the band width of the mixed valency of chromium ions [Cr³⁺, Cr⁴⁺] showed that the decrease in the size of the crystallites causes a significant weakening of the mechanism of the superexchange magnetic ones, and the reduction of the band width [Cr³⁺, Cr⁴⁺] from 0.76 to 0.19 eV. Similar behavior was found for CuCr₂S₄ ananoparticles [5].

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Magnetic and electric properties of Nd³⁺ and Mn²⁺-co-doped calcium molybdato-tungstate single crystals

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Divalent metal molybdates and tungstates with tetragonal scheelite-type structure (space group $I4_1/a$, No. 88) both un-doped as well as activated with di- or/and trivalent ions are very interesting materials because of their excellent properties. Micro-/nanocrystalline as well as single crystals samples of these materials could be widely applied as efficient phosphors [1], solid state lasers [2], scintillators [3], and microwave dielectrics [4]. Calcium molybdate (CaMoO₄) and calcium tungstate (CaWO₄) as members of scheelite family are highly ionic compounds with a small contribution of a covalent bonding. Band gaps determined from the band dispersion calculations [5] indicated that both compounds are direct band gap insulators with the E_g values of: 3.41 eV for CaMoO₄, and 4.09 eV for CaWO₄. Our earlier magnetic and electric research of microcrystalline Ca_{1-3x-y}Mn_y Π_x Nd_{2x}(MoO₄)_{1-3x}(WO₄)_{3x} solid solution (when 0.0050 $\leq x \leq$ 0.2000 and y = 0.0200, where \Box denotes vacant sites) obtained by high-temperature annealing of ternary MnMoO₄/Nd₂(WO₄)₃/CaMoO₄ mixtures showed a paramagnetic behavior with long-range ferrimagnetic and short-range antiferromagnetic interactions in these materials as well as a weak *n*-type electrical conductivity with stronger activation of 1.0 eV above 300 K in the intrinsic region [6].

In this work, $Ca_{1-3x-y}Mn_y\Box_xNd_{2x}(MoO_4)_{1-3x}(WO_4)_{3x}$ molybdato-tungstate single crystals $(x = 0.005 \text{ or } x = 0.0099 \text{ and with constant } \text{Mn}^{2+} \text{ ions concentration}, i.e. y = 0.005)$ have been grown by the Czochralski method in an inert atmosphere and subjected to structural, magnetic and electrical studies. The static magnetic susceptibility of as-grown single crystals was measured in the temperature range of 2-300 K and recorded both in zero-field-cooled and field-cooled mode using a Quantum Design SQUID magnetometer. Electrical conductivity $\sigma(T)$ of doped single crystals was measured by the dc method using a KEITHLEY 6517B Electrometer/High Resistance Meter. Thermoelectric power S(T) was measured within the temperature range of 77-400 K using a Seebeck Effect Measurement System (MMR Technologies). Our studies have shown that replacing diamagnetic Ca^{2+} ions with paramagnetic Nd³⁺ ones with the content not exceeding of 0.02 and having a screened 4*f*-shell revealed a significant effect of orbital diamagnetism and Van Vleck's paramagnetism, residual electrical conductivity without intrinsic region and a change of sign of the Seebeck coefficient at ~230 K as well as the Fermi energy (~0.04 eV) and the Fermi temperature (~500 K), determined from the diffusion component of thermopower suggesting the presence of shallow acceptor and donor levels.

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High-Quality MBE-grown Aluminum Layers for Experimental Realization of Majorana Bound States

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The small effective mass, large Landé g-factor, and strong spin-orbit interactions (SOI) make high-quality InAs, and InSb quantum well (QW) structures particularly appealing for the experimental realization of Majorana bound states (MBS) in the pursuit of topological quantum computing. Besides such near-surface QWs, an essential element of the MBS device is a layer of superconductive material with a very large coherence length at low temperatures and sufficiently high critical value of the in-plane magnetic field. A natural choice for such a material is aluminum. Even though the critical magnetic field of bulk Al is only 0.01 T, this is substantially increased for very thin layers. However, the high surface mobility of Al in a UHV environment, even at room temperature, and its tendency for 3D nucleation make the deposition of thin (~10 nm) continuous layers challenging. Growth of smooth layers has been typically done in dedicated UHV chambers with the substrate actively cooled with liquid nitrogen during Al deposition. Here we report on the growth of high-quality continuous Al layers directly inside the standard Veeco GEN10 MBE reactor at temperatures above 0 °C. We (i) show that at sufficiently high Al flux, the transition from 3D to 2D growth mode takes place, (ii) discuss the possible origin of such transition, and (iii) show results of an extensive in-situ and ex-situ characterization.



AFM scans of two aluminum layers grown on $In_{0.75}Ga_{0.25}As/In_{0.53}Ga_{0.47}As$ (6nm/100nm) on the InP(001) substrate. In both cases, the nominal thickness of the Al layer was 10 nm.

a. The layer was deposited at 0.5 Å/s.

b. The layer was deposited at 3.0 Å/s.



Superconducting to normal phase transition signatures for three Al layers: G0972 was deposited at 3.0 Å/s (10 nm thick) G0973 was deposited at 2.0 Å/s (10 nm thick) G0965 growth rate decreased from 2.0 Å/s to

approximately 1.3 Å/s during the deposition and is thinner. Layers deposited at lower growth rates

Layers deposited at lower growth rates did not show superconductive behavior.

Majorana-Magnon Interactions in Topological Shiba Chains

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A chain of magnetic impurities deposited on the surface of a superconductor can form a topological Shiba band that supports Majorana zero modes and hold a promise for topological quantum computing. Yet, most experiments scrutinizing these zero modes rely on transport measurements, which only capture local properties. Here we propose to leverage the intrinsic dynamics of the magnetic impurities to access their non-local character. Specifically, we calculate the spin susceptibility of the magnetic chain and show that the uniform magnonic mode, which spreads over the entire chain, becomes imprinted with the parity of the ground state. We find that the visibility of the spin susceptibility associated with the two parities oscillates between -1 and 1 in the topological regime as a function of the magnon frequency, while it is arbitrary in the trivial regime. These two distinct patterns originate from interference effects and are robust against moderate disorder. Our approach offers non-invasive alternative to the scanning tunnelling microscopy techniques used to probe Majorana zero modes. Conversely, the magnons could facilitate the manipulation of Shiba states and the Majorana zero modes.



Figure 1: (a) A Sketch of the topological Shiba chain in the presence of magnons. (b) The phase diagram. (c) The spectrum and the Majorana zero mode.

Temperature Dependence of Spin Pumping in Y₃Fe₅O₁₂/Pt, Ni₈₁Fe₁₉/Pt and SnTe/Ni₈₁Fe₁₉ Thin Films

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Magnons(quanta of spin waves) are collective excitations in magnetic material and allow information processing without transferring charge [1]. Such no-charge-based devices are in great demand owing to information transfer without Joule heat loss. One of the crucial components of a magnonic device is the detection of a magnon signal by transferring it to an electrical signal. One prominent way to accomplish the goal of electric detection of magnon spin current is via a method called spin pumping [2] using inverse spin hall effect (ISHE) phenomena. When the heavy metal (e.g. Pt) or topological material is interfaced with the magnetic materials (FM), the spin current can be injected into the HM from FM via spin pumping. Precessing magnetic material transfer angular momentum from the FM to the FM/HM interface via coupling of the local magnetic moments of the FM with the conduction electrons of the HM. Recently, it has been reported via ferromagnetic resonance experiments that induced spin currents can be amplified instead of suppressed by adding a strong spin orbit coupled spin sink layers in a superconductor/ferromagnet hybrid structure [3] and is key for superconducting spintronics. Therefore, it is essential to study the low temperature spin pumping in Pt and topological materials, such as SnTe. We report broadband spin-wave spectroscopy and spin pumping experiments on $Y_3Fe_5O_{12}/Pt$, $Ni_{81}Fe_{19}$ (with Pt thickness varied from 7 nm to 75 nm), and SnTe/ Ni₈₁Fe₁₉ bilayer thin films for temperatures ranging from 300 K to 4 K. We observed a systematic shift in ferromagnetic resonance fields, amplitude, and linewidths as a function of frequency and temperature. For the in-plane and out-of-plane directions of the magnetic field, we observed distinct modes at lower temperatures. The spin mixing conductance, spin current density, and spin hall angle values showed strong temperature dependence. Our findings are key for the usage of Pt and SnTe as spin sink layers for the superconducting spintronics applications.

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Size and curvature modulation of the magnetic properties of magnetite nanoshells

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Magnetic domains in bulk materials are defined as regions in space where spin correlations establish long-range order. Their occurrence is usually dictated by the structuro-chemical properties of the materials considered, whereas their dynamics, disposition and distribution can be related to extrinsic parameters such as disorder.

As the thickness of a magnetic material is reduced, shape anisotropy effects kick into play and progressively disrupt the magnetic ordering seen in bulk. In the limit of two dimensions (2D), no magnetic ordering is allowed in isotropic systems. This is the consequence of the well-known Mermin-Wagner theorem, which prohibits long-range ordering in low dimensional systems. One possibility to re-enable order in this case is to introduce strains, or otherwise break the systems in-plane isotropy.

A viable approach to induce such a break of isotropy, while not suppressing twodimensionality, is to impose a curvature to an otherwise flat film. In doing so, magnetic centers confined to the plane still see their immediate surroundings as flat, but are actually canted in relation to farther neighbors. Such a canting between magnetic centers is akin to the introduction of an antisymmetric spin exchange coupling to the material, whose quantum mechanical equivalent is known as the "Dzyaloshinskii-Moriya Interaction" (DMI). The strength of such an interaction has the potential to trigger different spin textures, depending on the system considered. Formally, therefore, the careful control of the substrate curvature has also the potential to introduce different spin textures into the system under consideration, provided that the curvatures involved are steep enough.

Here, we employ magnetite nano-shells surrounding a non-magnetic silica core to explore the effects of curvature-induced magnetic textures in mesoscopic systems. Through a measurement of the magnetic response of nanoparticles of different sizes, we demonstrate the effective tuning of our samples with particle diameter. Such an approach allows a selective control of the blocking temperature of our devices, as well as the induction or suppression of vortex and onion states in the material. We compare the results of the core/shell system proposed with those of bulk magnetic nanoparticles, which are often described within the framework of a core/shell structure with a superparamagnetic core and a magnetically-active shell.

Towards Spintronics of Antiferromagnetic Semiconductors - Characterization of MnSe on GaAs

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Antiferromagnets are materials that have become popular in the field of spintronics due to their ability to control the spin axis orientation with electrical current. As a result, several magnetic imaging techniques have been developed to detect the reorientation of antiferromagnetic spins. The research on antiferromagnetic spintronics has mainly focused on semimetallic systems or insulating oxides, leaving optical band-gap semiconductors unexplored. In this study, we investigate manganese selenide (MnSe), which has rocksalt structure and potentially uncompensated (111) planes, making it an attractive material for spin axis detection sensitive to uncompesated surfaces. In addition, since it is a wide gap semiconductor, it can be interfaced with a heavy-metal layer with strong spin-orbit interaction (i.e. platinum) to take advantage of the spin Hall magnetoresistance.

In this work, we study the quality of the MBE grown MnSe/GaAs structures with respect to a key factor - temperature of growth. The main tool to determine the quality of the samples was X-ray diffraction (XRD). We measured samples of MnSe/GaAs(001) and MnSe/GaAs(111).

From $2\theta/\omega$ scan of MnSe/GaAs(111) we obtained a lattice constant $a \approx 5.46$ Å, what indicate to rocksalt crystal structure of grown MnSe layers. We compared FWHM of RC scans of MnSe, where the lowest value, indicating the best quality of layer, was $\Delta = 0.6^{\circ}$ for MnSe/GaAs(111) grown at T = 300 °C, and $\Delta = 0.3^{\circ}$ for MnSe/GaAs(001) grown at T = 350 °C. We observe crystal twinning of MnSe layer on the MnSe/GaAs(111) samples. Sample grown at T ≥ 450 °C degrade which appears in unreliable values of FWHM using XRD.



Figure 1: Long range $2\theta/\omega$ scan of MnSe/GaAs(111) sample grown at T = 300 °C.

MnSe/GaAs structures are a good base

for the studies of the growth of more complex structures with quantum well having one barrier made of MnSe, which is interesting both in terms of optical experiments as well as magnetotransport when thin layer of platinum is added on the top.

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Raman spectroscopy as an indicator of antiferromagnetic to paramagnetic phase transition in transition metal thiophosphates

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Over the years, while electronic and optical properties of two-dimensional (2D) semiconductors were extensively studied experimentally, magnetic properties have remained neglected. That situation was related to the view that ferromagnetic order could not survive in a 2D regime due to increased thermal fluctuations [1]. However, a few years ago, Lado and Fernández-Rossier experimentally showed for CrI_3 crystal that magnetic order could be stabilized and survive in a low-dimensional regime [2]. Since then, 2D magnets are drawing growing interest, but plenty of phenomena still require extensive studies. Among 2D magnets, transition metal thiophosphates (MPS₃ - where M is transition metal) are a large family of van der Waals crystals exhibiting various types of intrinsic magnetic ordering.

In this work, we focused on correlating changes in Raman modes behavior with phase transition from antiferromagnetic to paramagnetic order for MnPs₃, FePS₃, CoPS₃, and NiPs₃. Moreover, we have elucidated how the replacement of transition metal influences the position of a given vibrational mode and its coupling to the antiferromagnetic phase. It can be seen in fig. 1 that around Néel temperature, one of the Raman modes loses its intensity and disappears completely while a new one of different symmetry emerges. The mentioned changes in symmetry were confirmed by angle-resolved Raman spectroscopy measurements fig. 1 (c). Furthermore, our experimental findings were supported by density functional theory that let us identify vibrational modes.



Figure 1: The analysis of A_g Raman mode behaviour with (a,b) temperature and (c) polarization for MnPS₃ with a Néel temperature of 78 K.

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Resonant Raman scattering of few-layers CrBr₃

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Chromium bromide belongs to the family of magnetic layered materials with the general formula CrX_3 (X = Br, Cl and I). In bulk form, the intralayer and interlayer exchange couplings in CrB_3 are ferromagnetic, in contrast to $CrCl_3$ and CrI_3 characterized by the antiferromagnetic interlayer orders [1, 2].

In this work, we performed the Raman scattering (RS) experiment on thin CrBr_3 flake with thicknesses from 4 to 7 layers at low (T=5 K) and room (T=300 K) temperatures using 5 different excitation energies, *i.e.* 1.96 eV, 2.21 eV, 2.41 eV, 2.54 eV, and 3.06 eV. The thickness of the flakes was confirmed by atomic force microscopy.



Figure 1: (a) Raman scattering spectra of thin flakes of CrBr_3 with thickness ranging from 4 to 7 layers. (b) Thickness dependence of the energies of the RS peaks.

In order to find the best resonant conditions for RS, we compared the intensities of the Raman peaks of CrBr₃ measured under different excitations. We found that the most intense Raman signal is observed under 2.41 eV excitation at room temperature. However, a substantial enhancement of the RS signal of about 3 times as compared to the aforementioned case was obtained under 3.06 eV excitation at T=5 K. The latter laser energy was chosen to investigate the thickness evolution of the Raman spectra, see Fig. 1(a).

As was reported in Ref. [1], four phonon modes, assigned as A_g^1 , E_g^1 , E_g^2 , A_g^2 , are Raman-active in CrBr₃. Note that the top in-

dexes are additional numbering to resolve the peaks sharing the same notation. The thickness dependence of the observed phonon energies is presented in Fig. 1(b). A clear monotonic blueshift of the A_g^1 and A_g^2 peaks is seen when the number of layers increases with the relative energy difference between 4 and 7 layers of about 0.6 cm⁻¹. The corresponding evolutions of the E_g^1 , E_g^2 energies reveal a more complex behavior with the relative difference of around 0.4 cm⁻¹. Our results confirm that the Raman technique can also be very useful in determining the thicknesses of the flakes in magnetic layered materials.

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Nagaoka ferromagnetism in phosphorene quantum dots

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Phosphorene, a monolayer form of black phosphorus, is a 2D material with strong anisotropy of the effective masses [1,2] that has been investigated in recent years. We consider a 2x2 array of quantum dots in phosphorene to investigate Nagaoka ferromagnetism - a magnetic ordering promoted in a system where number of electrons are one less than the number of sites. Although the ordering of spins in such system was proved by Nagaoka earlier, an experimental demonstration of this physical phenomenon was shown only recently by Ref. [3] for quantum dots arranged in square array in GaAs. Given than the electrons in phosphorene have large masses as well as large anisotropy, it is not obvious that the magnetic ordering will persist for the square array system in the material. We investigate the robustness of Nagaoka ferromagnetism against the geometry of the arranged dots using a single-band effective mass Hamiltonian [4] to extract the ground state of the system. An exact phase diagram of magnetic ordering as a function of the rectangle parameters is shown and we find the accurate geometry for which the Nagaoka energy gap is the largest. We also discuss the robustness of this ground state against potential detuning and shifting of one quantum dot. We demonstrate that shifting the dot has different impact on the ground state in different directions due to anisotropy, and thus Lieb-Mattis theorem, not Nagaoka theorem, now describes the deformed configuration.

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Magneto-phonon interaction in layered Cr2Ge2Te6 ferromagnet

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Two-dimensional magnets with van der Waals (vdW) structures have recently been revisited due to their potential uses. This fascinating group of materials gives high hopes to improve our spintronics nanotechnology and understanding of magnetic effects in two dimensional structures. Among magnetic vdW materials, $Cr_2Ge_2Te_6$ is a particularly interesting material, as it is in the very rare class of ferromagnetic semiconductors with a Curie temperature (T_c) of about 61 K and exhibits thickness-dependent ferromagnetic transition [1].

In this work, we examine Raman scattering (RS) measured on 10-nm-thick $Cr_2Ge_2Te_6$ flake using an excitation energy of 1.96~eV in a wide temperature range (5-300 K) and with polarization resolution.

The low-temperature (T=5 K) Raman spectrum of the studied Cr₂Ge₂Te₆ flake is composed of 7 Raman peaks, see the Figure. Their assignment to two groups of symmetries (E_g

and A_g) was confirmed using polarization-resolved Raman experiments. Although most of the Raman peaks experience redshifts with increasing temperature, there are two Raman modes, E_g^2 and E_g^4 , marked by red arrows in the Figure, which are sensitive the transition through the T_c to temperature. The $E_g^2\ peak\ disappears$ from the spectra above the T_c temperature, while E_g^2 one experiences temperature nontrivial evolution. Interestingly, the temperature dependence of the E_g^4 mode was already reported in Ref. [2]. In our case, the E_g^4 peak disappears from the Raman spectra above the T_c temperature, while its opposite behavior is observed under excitation energy of 2.41 eV.



Figure False-colour map of temperature evolution of the RS spectra measured on $Cr_2Ge_2Te_6$ flake. The T_c value is indicated by horizontal red line, while the red arrows point abrupt transition for two peaks around T_c .

Our results demonstrate that the electron-phonon interaction, i.e. interplay between resonant and non-resonant conditions of Raman scattering, may be also important for the magneto-elastic coupling in the magnetic layered materials.

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Investigating the Proximity Effect in h-BN/ 2D Magnet Heterostructures

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Hexagonal boron nitride (h-BN) is a two-dimensional crystal widely used in layered materials research due to its flat surface, high chemical stability, and lack of dangling bonds. Recent advances in the study of color centers in h-BN have made it an excellent platform for investigating basic physical phenomena, including optically addressable spin defects, rendering it a promising candidate for quantum technologies [1].

We focus on the growth of high quality h-BN samples using Metalorganic Vapour Phase Epitaxy (MOVPE) technique. The samples are grown on large (\sim 2 inches) sapphire substrates and cover them uniformly [2,3]. This method allows us to engineer material parameters such as its thickness, structure and consequently - type and concentration of defects. We observe emission from colour centres in epitaxially grown h-BN at different energies, indicating the presence of various defects, like boron vacancies or carbon-related defect complexes.

In this study, we explore the proximity effect between h-BN and 2D magnets (e.g. chromium trihalides). Such an approach enables the investigation of paramagnetic centres in h-BN using locally applied magnetic fields. We studied the photoluminescence of h-BN/CrBr₃/h-BN heterostructures and found an additional peak in the spectrum at low (liquid helium) temperature. We believe that the optical response is correlated with the magnetic ordering in CrBr₃ crystals (Fig 1).

In the future, the creation of such heterostructures can become feasible using a subsequential growth of 2D magnets on top of the h-BN epitaxial layers [3]. Such an approach could be a reliable solution for scaling up ultrathin heterostructures and allowing their future development for real-life applications.





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Thermoelectric properties of Pb_{1-x}Sn_xTe crystals doped with vanadium deep donors

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PbTe and SnTe as well as their substitutional alloys $Pb_{1-x}Sn_xTe$ are IV-VI semiconductor materials well known for topological crystalline insulator (TCI) properties as well as for their applications in mid-infrared devices and thermoelectric generators. Optimal application of these materials requires controlling of both type and concentration of conducting carriers: from almost semi-insulating bulk materials desired in TCIs to heavily doped ($n,p=10^{19} - 10^{20}$ cm⁻³) thermoelectric elements. In $Pb_{1-x}Sn_xTe$ the electrically active native defects (metal vacancies) determine very high background hole concentration ($p=10^{18} - 10^{19}$ cm⁻³). Controlling electrical properties requires doping with extrinsic donors (Bi, J) or acceptors (Na, Ag) and/or thermal annealing. Thermoelectric properties of $Pb_{1-x}Sn_xTe$ can also be controlled by incorporation of 3d transition metals, like Cr, Fe or V. The solubility of these transition metals in IV-VI semiconductors is limited to heavily doping regime (typically below 0.5 at.%) but they form a special class of resonant donor centers with the mixed valence 3+/2+ charge state depending on Sn content and Fermi level position.

In this work, we examine $Pb_{1-x}Sn_xTe:V$ bulk crystals with nominal Sn content of x=0.2 and V content of 1 at.% grown by the Bridgman method under varying thermal regime and doping procedure. The crystals studied cover the composition range corresponding to the expected ingap position of $V^{2^{+/3+}}$ donor level [1]. Electron microscopy SEM-EDX analysis of chemical composition confirmed the Sn concentration smoothly varying along an ingot from x=0.13 to x=0.3 and an indication for very small V content (below 0.5 at. %) increasing towards the end of an ingot. To examine electrical properties, we measured the electrical conductivity and the Hall effect in the temperature range T=5-300 K revealing only *p*-type metallic conductivity with carrier concentration $p \approx 7*10^{18}$ cm⁻³ and resistivity ratio $\rho(300K)/\rho(5K)\approx 10$ due to the temperature dependence of mobility. We also measured, at room temperature, the thermoelectric power α and determined the key applicational parameter - thermoelectric power factor $P_{\sigma}=\alpha^2/\rho\approx 8\mu W/K^2$ cm (comparable to observed in good PbTe-based thermoelectric materials). No in-gap pinning of Fermi level by $V^{2+/3+}$ donors and semi-insulating state was experimentally found, presumably because only small part of V centers introduced into Pb_{1-x}Sn_xTe (estimated 0.1 at.%) becomes electrically active. As a result, the electrical conductivity of our Pb_{1-x}Sn_xTe:V bulk crystals still remains dominated by native defects. This balance can be changed by further doping with V or reduction of vacancies concentration by annealing.

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Effect of Ni and Mn dopant on thermoelectric power generation performance of ZnO nanostructures synthesized via hydrothermal method

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Abstract

In this article, we have presented a low-cost hydrothermal approach to enhance the thermoelectric performance of ZnO nanostructures via modulation doping. For this purpose, we have prepared a series of pure and X:ZnO (X= Ni & Mn) samples. The Seebeck value of the Mn-doped samples possesses the maximum Seebeck coefficient of -36 μ V^{/o}C compared to the pure and Ni-doped samples (-22 μ V/°C & -27 μ V/°C) at room temperature. The highest value of the Seebeck coefficient for the Mn-doped samples is related to the formation of mid-gap energy band states due to the substitution of Mn²⁺ with Zn²⁺. These mid-band states induce an imbalance in the DOS, by producing a spin polarization effect that leads to a high Seebeck value. In terms of electrical conductivity, the Ni-doped ZnO sample exhibits the highest electrical conductivity of about 122 S/cm, due to the incorporation of Ni metal ions inside the ZnO matrix (confirmed by XRD) and leads to a high carrier concentration. However, the highest Seebeck value for the Mn-doped sample results in the maximum thermoelectric power factor ~1.12×10⁻⁵ Wm⁻¹C⁻² at room temperature.

Keywords; ZnO, doping, XRD, Seebeck coefficient, Electrical conductivity, Power factor

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Outstanding Thermoelectric Properties (ZT ≈ 5 – 6) of Functionalized 2D Molybdenum Nitrides (MXenes)

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MXenes are relatively new family of low dimensional materials, which has been gaining more and more popularity in recent years. MXenes are mainly carbides and nitrides of early transition metals and they combine the properties of both components. Bare MXenes typically exhibit metallic behaviour and, therefore, are known to be good electric conductors. Interestingly, this property changes with functionalization of their surfaces. It occurs that functionalizing groups can change metallic MXenes into semiconducting ones, and not only open the band gap but also influence other properties, just opening the path towards many potential applications, e.g., in electronics, optoelectronics, and thermoelectricity.

In this communication, we present probably the first reported studies of geometry, stability, electronic structure and transport properties of bare and functionalized molybdenum nitrides Mo₂N (MXenes). The studies are based on first-principles calculations in the framework of density functional theory (DFT) employing pseudo-potentials and plane-wave basis as implemented in the QUANTUM ESPRESSO package. Here, we discuss the results for the bare, and functionalized with oxygen and fluorine Mo₂N layers. The bare Mo₂N is metallic with good electric conductivity. Functionalization of Mo₂N with F leads to opening of the band gap of 0.09 eV below the Fermi level, whereas the Mo₂N functionalization with O creates p-type semiconductor with energy gap of 0.49 eV.

The electronic transport properties were calculated using BoltzTraP2 code and the phonon transport properties were obtained by the Phono3py. It occurs that functionalization has significant impact on Seebeck coefficient and lattice thermal conductivity of these materials, quantities strongly determining the thermoelectric performance. The Seebeck coefficient is increased to 2.5×10^{-4} VK⁻¹ for Mo₂NF₂ and 1.2×10^{-3} VK⁻¹ for Mo₂NO₂. Functionalization decreases lattice thermal conductivity by about 6.5 times for Mo₂NF₂ and 10 times for Mo₂NO₂. These effects lead to the impressive thermoelectric properties of studied MXenes - with thermoelectric figure of merit in the range of 0.3 < ZT < 0.78 for Mo₂NF₂ and 2.25 < ZT < 5.65 for Mo₂NO₂, what outperforms all currently known materials.

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Photoemission study of the thermoelectric group IV-VI van der Waals crystals (GeS, SnS, and SnSe)

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Semiconducting group IV-VI van der Waals crystals (MX, where M = Ge, Sn, and X = S, Se) are receiving increasing attention as environmentally-friendly thermoelectric materials. Among them, SnSe is considered the most promising as it exhibits a remarkably high thermoelectric figure of merit (ZT), initially attributed to its low lattice thermal conductivity [1]. However, it has been shown that the electronic band structure plays an equally important role in thermoelectric performance [2,3]. A certain band shape, described as a "pudding mold" and characteristic for all MXs, has been predicted to significantly improve ZT by combining good electrical conductivity with high Seebeck coefficient [4]. We explore this subtle feature experimentally for GeS, SnS, and SnSe by means of angle-resolved photoemission spectroscopy. The technique also allowed us to determine the effective mass and Fermi level position of as-grown undoped crystals. Our findings are supported by *ab initio* calculations of the electronic band structure. The results greatly contribute to the general understanding of the valence band dispersion of MXs and reinforce their potential as high-performance thermoelectric materials, additionally giving prospect for designing systems consisting of van der Waals heterostructures.

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Extension of rapid estimation of the flake thickness by color for a magnetic van der Waals 2D material

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The color analysis of thin flakes exfoliated on Si/SiO_2 substrates is an efficient method for identification of flakes of a given thickness. It is well known since the beginning of studies on graphene and related thin materials (especially hexagonal boron nitride[1]). With this method, we can bypass the need of using atomic force microscopy (AFM) to measure the thickness, significantly increasing the speed of the assembly of more complex structures. Thus, we greatly reduce the exposure time of the sample to potentially degrading external agents, which would be especially beneficial for rapidly oxidizing materials, one of which is chromium sulfide bromide (CrSBr).

CrSBr is an example of a layered van der Waals material that combines both optical and magnetic properties. Such a combination usually requires stacking of two different materials, while here it is realized within a single flake. The coupling of magnetic and optical properties makes this material a potential system for spintronics applications. Additionally, the magnetic properties of these materials vary with the number of atomic layers, which can help tailor them for technological applications.

Have we extended the well known methods of fast optical estimation of the layer thickness on the group of magnetic materials. By careful comparison of optical color images and AFM scans, we calibrate this technique for such materials as CrSBr. An example of a flake snapshot from the optical microscope and AFM scan with thickness readings for CrSBr is shown in Fig. 1. Importantly, most of the steps within this procedure can be done in a glovebox within a very high purity inert atmosphere, reducing the chance of material degradation.





Figure 1: Snapshot and AFM scan (with thickness labels) of CrSBr flake.

The obtained data allowed us to map the range of the thicknesses corresponding to selected colors. We discuss also enhancement of the sensitivity possible with use of narrow color filters.

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Coherent shuttling of spins between silicon-based quantum dots

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Coherent communication between quantum registers separated by ~10 μ m distances is one of the requirements for scalability of architectures based on gated quantum dots [1]. Moving a spin qubit across this distance, e.g. along a chain of tunnel-coupled dots [2], is one of possible solutions. I will discuss our recent theoretical work [3,4] on such a dot-to-dot adiabatic charge transfer in presence of realistic amount of charge noise and electron-phonon coupling in Si/SiGe QDs, which are currently the best candidates for a scalable semiconductor qubit architecture. I will discuss the close relationship between non-adiabatic (and thus non-deterministic) character of charge transfer, and spin dephasing caused by dot- and state-specific g-factors. Finally, I will describe an alternative qubit shuttling scheme in which gates are used to create a moving quantum dot [5,6]. I will argue that this scheme is more robust to electrostatic disorder compared to the one based on pre-existing tunnel coupled quantum dots, and I will discuss how the valley-orbit coupling in presence of interface roughness will be the main source of spin qubit dephasing in this scheme [6].

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Spin valley qubits in gated quantum dots in 2D Maciej Bieniek^{1,2}

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With the development of advanced methods for fabricating 2D materials heterostructures, devices based on atomically thin semiconducting transition metal dichalcogenides (TMDs) are emerging as a promising platform in the field of quantum information science. Semiconducting TMDs typically exhibit a direct bandgap at the K-point in the hexagonal Brillouin zone. They have potential for realizing spin qubits with long coherence time due to spin-valley locking, caused by the strong spin–orbit interaction and the symmetries of their 2D lattice. These promising properties motivate our recent efforts to understand quantum dots in 2D materials [1-7]. In the first part of the talk I will review experimental progress in the field of gated TMD's quantum dots [6,7].

In the second part, I will present our theoretical approach to electronic structure of TMD's. Starting from general properties obtained using ab initio methods, I will discuss construction of the tight-binding model [1]. Then, I will describe an atomistic theory of electrons confined by metallic gates computed using a computational box with periodic boundary conditions including up to millions of atoms [2]. I will show twofold degenerate energy spectrum of electrons confined in the two nonequivalent K valleys as well as sixfold degenerate spectrum associated with Q valleys, the role of spin splitting and topological moments on valley electronic states. Next, I will focus on the strongly interacting system of electrons [3], where we predict the existence of valley- and spin-polarized broken-symmetry states tunable by the parabolic confining potential. The ground state in such system is formed by one of two phases, both spin and valley polarized or valley unpolarized but spin intervalley antiferromagnetic, which compete as a function of electronic shell spacing.

In the third part, I will focus on the theory of electron-spin-based qubits [4]. The qubits are identified with two degenerate spin and valley locked states in a gated quantum dot. I will describe the qubit operations necessary to rotate the spin-valley qubit as a combination of the applied vertical electric field, enabling spin-orbit coupling in a single valley, with a lateral strongly localized valley-mixing gate. I will also discuss the two-qubit system defined on valley isospins [5] of two electrons confined in a gate-defined double quantum dot, focusing on how to initialize, control, interact, and read out such valley qubits by only electrical means.

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Longitudinal coupling between electrically driven spin-qubits and a resonator

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At the core of the semiconducting spin qubits success is the ability to manipulate them electrically, enabled by the spin-orbit interactions. However, most implementations require external magnetic fields to define the spin qubit, which in turn activate various charge noise mechanisms. Here we study spin qubits confined in quantum dots at zero magnetic fields, that are driven periodically by electrical fields and are coupled to a microwave resonator. Using Floquet theory, we identify a well-defined Floquet spin-qubit originating from the lowest degenerate spin states in the absence of driving. We find both transverse and longitudinal couplings between the Floquet spin qubit and the resonator, which can be selectively activated by modifying the driving frequency. We show how these couplings can facilitate fast qubit readout and the implementation of a two-qubit CPHASE gate. Finally, we use adiabatic perturbation theory to demonstrate that the spin-photon couplings originate from the non-Abelian geometry of states endowed by the spin-orbit interactions, rendering these findings general and applicable to a wide range of solid-state spin qubits.

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Optical control of solid-state quantum emitters

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Solid-state quantum emitters play a vital role in quantum technologies as photon sources. To make them viable for actual applications, the emitted photons need to have close to perfect properties. This includes a high brightness, high purity, high indistinguishability and, in case of entangled photons, a high concurrence. Semiconductor quantum dots are at the moment the most promising candidates to fulfil these requirements as astonishing good values for such quantities have been reported in the last few years. However, there is still room for improvement.

One aspect is the preparation of the excited state, which then leads to the emission of the photons. The straight forward method of resonant excitation, making use of the Rabi scheme, suffers from the influence of phonons [1]. Furthermore, because the exciting frequency is the same as the frequency of the emitted photons, sophisticated polarization filtering results in a loss of half the photons. Off-resonant schemes like the phononassisted state preparation [1] are incoherent, which can be disadvantageous depending on the target state. Accordingly, new protocols are proposed to optimize the excitation of quantum emitters.

In this talk, I will review different optical preparation schemes to excite the quantum emitter. In addition to well established schemes like Rabi or phonon-assisted processes, I will introduce the Swing-UP of EmitteR Population (SUPER) scheme that uses two reddetuned pulses to excite a quantum emitter [2]. Besides displaying interesting physics, the SUPER scheme is coherent and truly off-resonant and, thus, spectral filtering can be applied [3]. This could boost the photon rate up to a factor of two, making the SUPER scheme highly interesting for applications in quantum technology.



Figure 1: Artistic view of the Bloch sphere under the SUPER scheme

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Epitaxial InAs/InP quantum dots as emitters of single photons and entangled photon pairs for fiber-based quantum communication

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Quantum dots (QDs) are an important resource for quantum communication, because of their capability of generating various non-classical light states. Compared to other physical systems, QDs provide high single-photon purity independent of the source brightness. They offer the possibility for on-chip integration and compatibility with well-established semiconductor technology, including deterministic fabrication of photonic devices. To fully exploit their application potential, emission in the telecom bands is required. InAs/InP material system is one alternative to provide emission in the telecom C-band [1].

Our work concerns InAs/InP QDs grown using ripening assisted molecular beam epitaxy technique [2]. They have been proven to be efficient sources of (triggered) single photons, with multiphoton emission probability below 1% [3] and spectrally broad (~60 nm) extraction efficiency for the QDs placed inside simple cylindrical photonic structures, at the maximum exceeding 13% [4]. Their thermal stability is better than for GaAs-based QDs emitting in this spectral range due to deeper confining potential, and it is mainly limited by escape of holes via the ladder of QD's excited states [5,6]. It enables maintaining single QD emission for temperatures as high as 150 K, which makes this system suitable for simple and cost-effective cooling methods, thus considerably increasing its application potential.

Additionally, these dots feature spin memory effects allowing for preferential initialization of electron spin state [6]. However, identification of trions is not straightforward and requires magnetooptical investigations [7], due to high in-plane symmetry of these dots, manifesting in low exciton fine structure splitting [2]. This, on the other hand, allowed achieving emission of entangled photon pairs (fidelity exceeding 0.55) generated from biexciton-exciton cascade at telecom C-band. It was realized under non-resonant excitation of a QD in a simple mesa structure and it can be improved by selected excitation of target states and placing QDs deterministically in microcavity structures. This enables application of such InAs/InP QDs in implementation of quantum teleportation schemes or quantum repeater architecture.

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State-of-the-art Quantum Dot-based Single-photon Sources for Quantum Key Distribution in Silica Fiber Networks at 1550 nm

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Optical fiber-based quantum network connecting remote nodes forms the backbone to implement ultra-secure data exchange protocols and schemes, utilizing a quantum state of a single photon. Notably, for such quantum networks existing silica-fiber-based infrastructure can be used, where the spectral window around 1550 nm provides a low-loss photon transmission channel for long-haul communication. Moreover, one can benefit from existing devices and classical signal management protocols, simplifying the preparation, encoding and transfer of quantum states. Devices producing single photon states on demand are considered key building blocks of future quantum networks. While several single-photon sources (SPS) operating in the telecom C-band, including sources based on non-linear frequency conversion or emerging materials, have been demonstrated to date, semiconductor quantum dots (QDs) are among the most promising candidates, enabling exceptional performance and engineered photonic devices.

In this presentation, we show a deterministically-fabricated optically-triggered SPS emitting in the telecom spectral range near 1550 nm. The SPS consists of a single pre-selected self-assembled InAs/InP QD embedded in a circular Bragg grating cavity for Purcell-enhanced emission of the embedded quantum emitter. The applied deterministic fabrication technique results in a high device yield in combination with excellent quantum-optical properties, paving the way for scalable applications in photonic quantum information technologies. The high Purcell enhancement with $F_p = 5$ realized in our device, shows potential for single photon generation at GHz rates. Moreover, the cavity's far-field emission pattern allows for efficient photon extraction close to 17% at the numerical aperture 0.4, which can be further optimized for the coupling to single-mode fibers. The device provides high single photon emission purity with $g^{(2)}(0) = 3.2 \times 10^{-3}$ at T = 4.2 K, while still supporting a good performance ($g^{(2)}(0) = 0.2$) at elevated temperatures (T = 50 K) accessible with compact Stirling-type cryocoolers. Finally, Hong-Ou-Mandel-type two-photon interference (TPI) experiments reveal record-high photon indistinguishabilities of 19% in the as-measured case and 99.8% with temporal post-selection.

Finally, we evaluated our SPS in a test environment for its potential implementation in polarization-encoded BB84 quantum key distribution (QKD) in silica fiber networks. A careful benchmarking of the performance of our source compared with previous single-photon QKD experiments reveals that transmission distances in excess of 100 km can be expected in full implementations.

Heterogeneously Integrated InAs/InP Quantum Dot Single-photon Emitter with the Si Platform for On-chip Quantum Photonics

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Photonics-based quantum technologies would remarkably advance the processing and transfer of information. Photonic integrated circuits (PIC) in which semiconductor-based components, like emitters, waveguides (WGs), and detectors, can all be integrated on a compact photonic chip, provide a better performance, and increase their functionality when assembled with silicon-based electronics.

We present modelling, fabrication, and experimental studies on a hybridized InP/Si WG system with InAs/InP quantum dots (QDs) as single photon emitters [1]. The structure is fabricated via the direct wafer bonding technique of InP and silicon-on-insulator chips, followed by the e-beam lithography and inductively coupled plasma etching steps resulting in the InP WGs formation. Before fabrication, the structure was optimized numerically, revealing the optimal architecture for the most efficient coupling between QDs and propagating modes in the InP/Si WG and, further, along the the Si/SiO₂ WG. A linear taper structure achieves the efficient light field coupling between the InP/Si and Si/SiO₂ WGs [2]. We demonstrate the light coupling between InP and Si parts and examine the quantum nature of photons generated by a QD and propagating through WGs. The Hanburry-Brown and Twiss interferometric experiment on photons emitted by a QD and collected at the cleaved facet of the Si/SiO₂ WG revealed a good suppression of multiphoton emission events with $g^{(2)}(0) < 0.2$. The outcomes of our study underscore the potential benefits of this technological approach for advancing the development of quantum on-chip photonics using single-photon sources which can be easily combined with external fiber optics networks.



Scanning electron microscope images of the fabricated outcoupler (a, b) and taper (c). Conceptual illustration of the PIC with a heterogeneously integrated QD emitter (d). Modes of transmission for the calculated waveguide (e).

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Electrically gated single photon emitters in monolayer transition metal dichalcogenides

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Monolayers of transition metal dichalcogenides (TMDs) draw much attention as semiconducting materials with robust optical properties, strong Coulomb interaction, and an optically accessible valley degree of freedom. While some of the questions related to excitation dynamics and other electronic properties of the pristine TMDs are still to be answered, much attention is drawn to the more specific system based on ultrathin layers of TMDs. In recent years, significant research efforts have been directed to quantum emitters formed on various carrier localization centers in TMDs. Such objects are characterized by small linewidth, long luminescence lifetime, and single-photon emission. Related sharp emission lines are visible among or energetically below the band of relatively broad non-localized excitonic states. Such objects can be observed in monolayers or bilayers of several different TMDs. Single photon emitters (SPEs) are interesting from the point of view of photonics. Their properties, combined with the extraordinary properties of the free excitonic states in atomically thin TMDs, can open up perspectives for many advances in photonics and optoelectronics. However, these possibilities are currently hindered by a need for a more thorough understanding of the excitonic structure of the SPEs.

In this work, we identify and study SPEs created on different types of localized centers, particularly point defects and strain centers. Following Ref. [1], we expect that the Zeeman shift amplitudes should strongly depend on the nature of the trapping state, allowing identification of the SPEs localized on different types of defects. We investigate the SPE states in the TMD monolayers and determine the statistically most prevalent mechanism of carrier localization. We find, that in case of the WSe₂ monolayer the hole trapping is the most prevailing mechanism, although emitters based on electron trapping can also be observed. Furthermore, control over the Fermi level facilitated by the electrical gating in the sample allows for estimating the energy of the doping states responsible for the localizing center in the TMD layer.

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Epitaxial Boron Nitride Processing and Conductivity

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Hexagonal boron nitride (h-BN) is a very promising candidate for optoelectronic applications in the deep ultraviolet spectral range due to its exceptional physical properties, such as high chemical stability, thermal conductivity and wide bandgap energy. A large cross section for neutron capture renders BN an outstanding candidate for neutron detectors.

In our experiments, we used h-BN grown on two inch sapphire wafers. Growth was performed via metalorganic vapour-phase epitaxy (MOVPE) [1,2]. High-resolution SEM images, X-ray and Raman studies indicate a very good quality of the obtained hBN layers. However, obtaining micro-scale devices on epitaxial h-BN layers is a challenge due to wet delamination that may occur during typical processing stages.

To address this issue, we present a novel preparation method of as grown h-BN layers prior to optical lithography. Our technique relies on the controlled delamination and deposition of the epitaxial layer that prevents unwanted delamination from the sapphire substrate during etching and metal coating processes and produces fewer wrinkles in the processed h-BN.

It enables multiple-stage processes, including etching and sputtering, on thin layers of epitaxial h-BN. Fig. 1 shows an example of a h-BN Hall bar processed with reactive ion etching and coated with metallic contacts made of an AuPd alloy.



Fig 1. Optical image of a h-BN Hall bar structure with AuPd electrodes. The structure was created with a 2-stage lithographic process.

This novel preparation method allowed to study the effect of annealing of epitaxial h-BN layers in nitrogen atmosphere in the MOVPE reactor as well as the effect of irradiation with electrons in a scanning electron microscope (SEM). The electrical conductivity was measured before and after annealing and electron irradiation. The results show a significant increase in electrical conductivity after both treatments. These findings demonstrate the potential for controlling the electrical properties of h-BN, which is crucial for electronic and optoelectronic applications of epitaxial h-BN.

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Direct MOVPE growth of high-quality h-BN on the wafer-scale: the role of substrate off-cut

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Hexagonal boron nitride (h-BN) is an attractive 2D material for possible applications in electronic and optoelectronic devices based on van der Waals heterostructures, but direct growth of high-quality hBN on the wafer-scale is still the bottleneck for future successful implementation of hBN in industry. Although the growth of h-BN by MOVPE has already been reported [1, 2], there is a fundamental lack in understanding of many basic aspects of growth. In particular, the role of the substrate off-cut, although known to be of major importance for the growth of other materials, has not been fully addressed so far.

In this communication, we present a study that addresses the influence of the sapphire substrate off-cut angle on the final growth of h-BN obtained in a two-step growth procedure [2]. The main process starts with a self-limiting continuous growth (CFG) of a BN buffer followed by flow-modulated epitaxy (FME) in the second step and is used to study substrates with different off-cuts angles for three different CFG times.

Based on results obtained x-ray diffraction and reflectometry, Raman and Fourier-

transform spectroscopy, atomic force microscopy scanning electron and microscopy (SEM), we find that a substrate with off-cut angle of 1° clearly yields the highest quality of h-BN layers. Samples with this off-cut have the lowest amount of debris on the surface (see fig. 1), intense most x-ray diffraction signal, minimal Raman phonon linewidth thinnest and amorphous BN part.

A detailed analysis of the crystallographic structure of the hBN



Figure 1. SEM images of the two stage samples grown with 5 min, 30 min and 60 min of the CFG buffer and with a similar 60 min FME final process. a) S_5 sample with five off-cuts, b) S_{30} sample with five off-cuts, c) S_{60} sample with five off-cuts.

layers and mechanisms responsible for the existence of an optimal off-cut angle will be discussed.

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Manipulating hBN Bandgap Properties With Aluminum

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Research on artificial light sources led to the development of semiconductor emitters that have a number of advantages, e.g. small size, long lifetime, high efficiency. To obtain shorter-wavelength emitters one usually alloys GaN-based structures with aluminum to increase the bandgap of the material (to 6.2 eV for pure AlN). However, by using this technology one can so far obtain only low luminescence efficiencies in the deep UV (DUV, 4.4-6.2 eV). This spectral region is of high importance for the sterilization of water, air and surfaces since light in the DUV can distrupt DNA of pathogenes.

In our research we focus on filling the efficiency gap in the DUV spectral range. We test an approach based on hexagonal boron nitride which despite of an indirect bandgap has a photoluminescence (PL) intensity of the band-edge transition that is more than two orders of magnitude higher than that of an AlN epilayer which has a direct bandgap [1]. Because the conduction band minima responsible for direct and indirect emission are close to each other in hBN, the emission efficiency can be potentially further enhanced when electronic states are properly manipulated [2].

In this work we study boron nitride layers grown with metalorganic vapor phase epitaxy (MOVPE) on 2-inch sapphire substrates [3-4]. The studied samples were alloyed with aluminum using different flows of trimethylaluminium (TMAl). Based on results obtained by UV-Vis, Raman and Fourier-transform spectroscopies, X-ray diffraction, scanning electron microscopy, atomic force microscopy and photoluminescence we will discuss how different flows of aluminum precursor affect the properties of the final epitaxial hBN layer.

As presented in figure 1, the absorbance spectra change with the amount of TMAl. The spectra revealed two peaks which energies coincide with direct (higher energy) and indirect (lower energy) transitions in hBN. The intensity ratio and the position of those peaks vary between the samples. This observation suggests that we are capable of manipulating the boron nitride bandgap by alloying with aluminum. The possible reasons and consequences of such a behavior will be discussed.



Figure 1: Absorption coefficient measured at room temperature for the samples grown with different TEB:TMAl ratio. Absorption coefficient uncertainty is estimated to be 10^5 cm⁻¹.

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MBE growth of CdTe on hBN

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Hexagonal boron nitride (hBN) has been shown to be a perfect substrate for the growth of high optical quality 2-dimensional materials such as MoSe₂ [1]. The main advantages of hBN compared to other semiconductor materials is a very high energy gap (5.9 eV) and ultra-low roughness in the case of exfoliated hBN. The motivation of the present work is based on the observation that such properties of hBN could be also useful for the epitaxy of 2D structures formed from 3D materials, such as e.g. CdTe quantum wells (QWs). In such a case hBN could act both as a flat substrate and barrier material with a high energy gap.

The growth was realized by Molecular Beam Epitaxy on exfoliated hBN flakes deposited on Si (100) substrates covered by 90 nm of SiO₂. Despite the use of relatively small hBN flakes, the Reflection High-Energy Electron Diffraction (RHEED) was applied in order to observe the evolution of the surface during the growth of the structure. We tested the growth of various tellurides: CdTe, ZnTe, (Cd,Mg)Te, and their heterostructures. After the growth, samples were examined with Atomic Force Microscopy (AFM) and low-temperature photoluminescence (PL) was investigated.

Several samples were grown in various temperatures of the substrate during growth. For typical growth temperatures of tellurides (250-350°C) there was no observable deposition of material on hBN. The best results were obtained for the lowest temperatures, below 200°C. The AFM scans of hBN covered by II-VI materials revealed a polycrystalline structure with two different areas: with and without the influence of the electrons from RHEED. In the photoluminescence spectrum of samples with CdTe quantum wells with (Cd,Mg)Te barrier a sharp peak of a CdTe quantum well was observed in the area influenced by RHEED. Interestingly, in the area without the influence of RHEED PL of quantum wells was not observed by far. Also presence of hBN for the formation of high optical quality QWs is important, because areas where CdTe was deposited directly on SiO₂ (no on hBN flakes) revelaed only PL of (Cd,Mg)Te barriers, not CdTe quantum wells.

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Carbon-enriched Epitaxial Boron Nitride Layers as Contact Material for h-BN Devices

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One of the most prominent members of the 2D materials, hexagonal boron nitride (h-BN), is the subject of extensive research thanks to their wide potential use in optoelectronics. However, the difficulty in producing high quality layers as well as the lack of ohmic contacts limits its practical use, despite excellent properties such as chemical stability, thermal conductivity and wide bandgap.

In this communication we study the effect of variable carbon content in h-BN layers, grown by Metaloorganic Vapor Phase Epitaxy (MOVPE)[1], on its optical and electrical properties in attempt to obtain a conductive contact material. Using nitrogen instead of hydrogen as carrier gas, as well as adjusting the ratio of gaseous precursors and temperature allows for the manipulation of carbon content in the grown layers.

The differences in sample compositions were estimated using EDX spectroscopy. Optical and electrical measurements were also performed. Raman and UV-Vis spectra confirm differences in both structure and bandgap.

Electrical measurements were performed on a Hall-bar structure, manufactured by plasma etching and photolithography processes. Results show a reduction in resistivity in samples with high carbon content in comparison to pure h-BN.

Optimizing the growth process in order to obtain high quality conductive layers is a promising step towards finding suitable contacts for future h-BN devices.

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Determining Strain Components in a Diamond Waveguide from Asymmetric ODMR Spectra of NV⁻ Center Ensembles

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The negatively charged nitrogen-vacancy center in diamond (NV⁻) has shown great potential in nanoscale sensing and quantum information based on their rich spin physics. Efficient fluorescence detection and interface with light, in general, is crucial for realizing these applications in experiments. Laser-written waveguides in diamond have recently been used to improve the coupling between NV⁻ centers and light [1]. The NV⁻ spin states are detected by the optically detected magnetic resonance (ODMR) spectra. However, in the waveguides, these ODMR spectra are found to be consistently asymmetric.

Here, we theoretically simulate the experimentally obtained ODMR signals and find that strain, that could have been introduced in the process of laser writing, can explain the ODMR asymmetry even without invoking built-in electric fields, which differs from previous studies [2-3]. By fitting the theory to experimental signals gathered from different positions on the sample, we quantify all three relevant strain tensor components for full profiles across the waveguide.

Our model consists of NV⁻ centers interacting with a resonant or near-resonant microwave and the static strain field. We exploit the translational symmetry of the waveguide, which reduces the number of relevant strain components to three. Since yellow diamond samples have a high density of NV⁻ centers with a uniform distribution along all four possible defect orientations in the crystal, our modeled signal comprises equal contributions from all four possible NV⁻ center orientations. We numerically calculate individual NV⁻ centers' eigenstates and corresponding excitation probabilities for microwave-driven transitions. Based on this, we compose the simulated ODMR spectrum from pairs of weighted Lorentzians for each NV⁻ orientation. Finally, we extract the strain components by fitting the model to the experimental data.

Our results provide reasonable predictions for quantifying the strain distribution in the studied waveguide structures.

Acknowledgements:

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Post-implantation Defect Accumulation in Crystal Lattice of β-Ga₂O₃ Implanted with Yb ion

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Gallium oxide (Ga_2O_3) is an ultra-wide bandgap semiconductor of interest for many applications, including optoelectronics. Undoped Ga_2O_3 emits light in the UV range, that can be tuned to the visible region of the spectrum by rare earth dopants. Ytterbium may further enhance the span of applications due to emission in the infrared region of the spectrum. Ion implantation is a renowned doping technology used for the alteration of material properties. Despite of many advantages of this technique, it could lead to lattice damage and defect accumulation which need to be studied thoroughly as these factors strongly influence the material properties.

Our previous work, where we studied structural defects and recovery of the crystal lattice of β -Ga₂O₃ after Sm implantation and post-implantation annealing, indicates that the damage of the crystal lattice after implantation is not typical and needs further investigations [1]. In the present study, the (-201) orientated β -Ga₂O₃ single crystals were implanted with Yb ions fluences ranging from 5×10^{11} to 5×10^{15} at./cm². Channeling Rutherford backscattering spectrometry (RBS/c) was used to study modification of the crystal lattice induced by ion implantation and annealing. The RE depth profile was analyzed by the SIMNRA calculations, while the McChasy simulations were used to quantify the defect distributions. The obtained accumulation curve reveals the two-step damage process. From the fluence of 5×10^{13} /cm², the damage of the Ga₂O₃ lattice starts to increase very fast, reaching the amorphization level for the Yb ion fluence of 1×10^{14} /cm². The new form of defect that is created on the highest level is found to be annealing resistant.



Random and aligned RBS/c spectra for β -Ga₂O₃ implanted with Yb. The solid line represents the McChasy simulations.



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The structural properties anisotropy of β-Ga₂O₃ implanted with Yb

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Nowadays doped and un-doped gallium oxide is one of the most exciting materials in research and technology, and one of the most important tasks in this issue is the development of controlled methods of doping this material. Rare earth-doped Ga_2O_3 seems to be attractive for future optoelectronic devices such as phosphors, displays, and LEDs with emission in the visible spectral range. The ion implantation technique is one of the attractive ways to produce such systems. However, even though ion implantation is the most common semiconductor doping method, the studies of this process for Ga_2O_3 with RE are still at the initial stage.

In our recent research, we paid particular attention to the issue of the Ga₂O₃ anisotropy (in its thermodynamically stable β -Ga₂O₃ phase), which has not been theoretically predicted, but experimentally observed in optical and electrical investigations. In this work, the studies on the structural properties of β -Ga₂O₃ crystals, (010) and (-201) oriented, irradiated with different fluencies of Yb ions and subsequently annealed, have been performed. The crystal lattice damage, structure recovery, as well as Yb depth profiles, and Yb ions location in the β -Ga₂O₃ crystal lattice, were studied by Rutherford Backscattering Spectrometry in the channeling direction (RBS/c) and supported by computer simulations. Our studies reveal the strong influence of anisotropy on structural properties, with a significantly lower damage level for (010) oriented β -Ga₂O₃ crystals, and a small number of substitutions of gallium sites by Yb atoms as well. Interestingly, contrary to the common opinion, is that the β -Ga₂O₃ has strong radiation resistance, our results show that it is very easy to make this material amorphous. However, the crystal lattice recovery is also much easier than in other wide bandgap materials.

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Electron Paramagnetic Resonance of Beta-Gallium Oxide Modified by High-Energy Electron Irradiation

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Several electron paramagnetic resonance (EPR) signals have been identified in β -Ga₂O₃, among which there is an intrinsic acceptor – gallium vacancy, both doubly ionized V²-_{Ga} with spin ¹/₂ and singly ionized V⁻_{Ga} with spin 1 [1, 2] and an intrinsic donor – oxygen vacancy O²⁻ trapping single electrons and exhibiting strong motional narrowing [3]. These EPR signals can be used to monitor defect formation and evolution of electronic properties of β -Ga₂O₃ modified by high-energy electron irradiation.

 β -Ga₂O₃ single crystals doped with Sn were irradiated by a 2.5 MeV electron beam at 20 K at the Sirius facility, Ecole Polytechnique, Palaiseau. The electron irradiation produces a spread of Frenkel pairs, vacancy-interstitial. Interstitials are typically highly mobile at room temperature and annihilate at the surface, thus the expected effect of irradiation comes usually from vacancy-type induced defects. Indeed, in the EPR experiment after the irradiation, a signal characteristic to doubly ionized gallium vacancy, V²⁻Ga, is observed (Figure 1), indicating a lowering of the Fermi level to the band gap. In contrast, pristine samples show the presence of shallow donor signal specific for conducting samples. The characteristic parameters of the resonance line, its linewidth, g-factor, and amplitude, evolve with temperature indicating a transition from states localized at donors to the delocalized states in the conduction band. The evolution of signals after subsequent annealing, thus removing the irradiation defects, is also discussed.

The presented study aims to shed light on the possibility of tuning the Fermi level towards the β -Ga₂O₃ valence band. This still unsolved issue is crucial for the practical implementation of this ultra-wide band gap material in electronics and opto-electronics.



Figure 1. EPR spectra of pristine β -Ga₂O₃ doped with Sn (a) and irradiated with 2.5 MeV electrons (b).

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Effects of Si/N Ion Implantation on Defect formation and Doping of Gallium Oxide

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A new generation of power devices based on the wide-bandgap (WBG) semiconductor, Ga₂O₃, is expected to revolutionize the power electronics industry. Many promising β -Ga₂O₃-based devices have already been reported, such as Schottky barrier diodes, MOSFETs, and various solar-blind photodetectors [1 - 4]. So far, the development of Ga₂O₃-based transistors has been focused mainly on utilizing lateral geometry. On the other hand, the vertical geometry would allow for higher current drives without having to enlarge the chip size, simplified thermal management, and far superior field termination. However, for this geometry to be realized, both *n*-type and *p*-type doping should be achieved in Ga₂O₃, with *n*-type readily available [5, 6]. Contrarily, *p*-type doping in Ga₂O₃ is fundamentally challenging to be achieved due to the very flat valence band maximum (VBM) of O 2*p* in nature and self-compensation [7 - 9].

It was recently shown that *p*-type conductivity can be achieved even in undoped Ga_2O_3 layers with annealing in an oxygen atmosphere due to the β -Ga₂O₃ particular point defect chemistry and the large formation energy of oxygen vacancies [10, 11]. Such annealing increases the free hole concentration by three orders of magnitude due to the formation of VGa(-) - Vo (++) complexes, being a shallow acceptor center. However, the mobility was found to reduce, probably due to a greater number of acceptor scattering centers compared to the as-grown sample.

Another approach would be to use nitrogen as *p*-type dopant with subsequent post-annealing treatment as was shown in [6] where nitrogen and silicon were used to engineer a Ga_2O_3 transistor using ion implantation. Consequently, there is a need for a better understanding of the origin of the observed complicated behavior of Si and N elements in the Ga_2O_3 matrix, which results in different transport properties.

In this work, we aimed to identify structural defects caused by Si(N) implanted into the Ga_2O_3 matrix, their transformations and possible interaction with host intrinsic defects employing various experimental techniques: RBS/c, PIXE, HR XDR, XANES, TEM. The crucial task is the identification of the local surrounding of the mentioned above dopants and understanding the driving forces for phase transformations taking place upon implantation and subsequent annealing.

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PA-MBE Grown CdMgO:Eu Ternary Alloy on m- and c-Oriented Al₂O₃ Substrates

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Oxides based on cadmium (Cd) and magnesium (Mg) due to their properties can find an extensive range of potential applications in photovoltaics, light-emitting diodes, and other optoelectronic devices. In this work, a series of CdMgO:Eu ternary alloys were grown on mand *c*-oriented Al₂O₃ with varying Mg concentrations using plasma-assisted molecular beam epitaxy. The crystal structure and topographical features of grown ternary alloys were analyzed using X-ray diffraction and atomic force microscopy (AFM). X-ray photoelectron spectroscopy (XPS) was used to determine the chemical state and composition of Eu and it confirmed Eu³⁻ europium oxidation states. The optical properties of as-grown and annealed samples were examined with UV-visible spectrometry, PL, and CL measurements. Transmission spectra revealed that the alloys are highly transparent in the visible region, where transmittance increases with incident wavelength (λ) and over 500 nm approaches nearly 90% for all samples Fig. 1. (a-b). Utilizing diffuse reflectance spectra, the dielectric constants, refractive index, and extension coefficient were also evaluated. The band gap energy (Eg) was estimated using the standard Tauc plot and the Kubellka-Munk model. The increasing trend was observed in (Eg) under Mg-rich conditions. From transmittance spectra, the highest energy gap values for *m*- and c-oriented alloys are estimated to be 2.866 eV and 3.011 eV, which are due to the high Mg concentration as consistent with literature [1]. CL confirmed the presence of Eu-related peaks in the red spectral range for annealed samples. The surface topography of CdMgO:Eu alloys are homogeneous and uniform; rms values are at the level below 1 nm and increase with Mg concentration.



Fig. 1. (a-b) Transmittance vs λ for *m*- (1M-5M) and *c*-oriented (1C-5C) alloys samples, (c) Direct and indirect bandgap energy (Eg) of CdMgO:Eu alloys on *m*- and *c*-oriented Al₂O₃ from transmittance spectra.

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MBE growth of wide band gap wurtzite MgZnO superlattices doped with Eu for effective visible emission

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Zinc oxide became of increasing technological importance, in particular for the development of light emitting diodes and lasers operating in the visible and ultraviolet region. Such devices contain heterojunctions and active areas based on quantum structures. Rare earth elements can be introduced into semiconductors either via ion implantation or by in-situ doping for extension of spectral range of emission of those. Epitaxial growth and in-situ doping allows to avoid problems with incomplete reconstruction of the crystal lattice which are common problems in the case of implantation method. Molecular Beam Epitaxy is perfectly suited for the production of thin crystalline layers, complex multilayer structures and for selective doping of carefully chosen layers, such as quantum wells. It can also be expected that some RE dopants can be stabilized in various oxidation states, including less common states like for example Ce^{4+} , Yb^{2+} , Eu^{2+} and other lanthanide ions [1]. Such a possibility is particularly interesting in the case of Eu, which when incorporated into the same host material, often coexists in two ionic states: 2+ and 3+ [2].

Wurtzite MgZnO alloys in form of superlattice structures have been grown by molecular beam epitaxy on variously oriented sapphire substrates on high-quality ZnO buffer layers. The widened band gap energy of utilized MgZnO layers was large enough to play a role whether as a barrier or quantum well selectively doped with Eu ions. The concentrations and distributions of Eu atoms in the respective hosts were analysed using Secondary Ion Mass Spectrometry, which confirmed successful doping around the level of similar to 1%. Optical properties of the samples were investigated by room temperature photoluminescence and Photoluminescence excitation measurements. The results obtained with Secondary Ion Mass Spectrometry were verified with Rutherford Backscattering spectrometry, which included not only random spectra but also angular measurements of main channels in order to shed light on eventual strains present in the structures and acting upon Eu ions. All performed analysis suggest the co-existence of 2+ and 3+ ionic states of Eu in the SL structures, and only 3+ ionic state after thermal treatment. Futhermore, optical measurements revealed strong enhancement of ${}^5D_0 \rightarrow {}^7F_I$ intra-4f-shell transitions of Eu³⁺ in MgZnO based superlattices.

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Initial optimization of the growth conditions of GaAs homo-epitaxial layers after cleaning and restarting the MBE reactor.

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The Molecular Beam Epitaxy (MBE) technique is renown as the most suitable for growth of high quality crystalline materials and nanostructures such as GaAs. Although, once established, optimal growth parameters required for repeatability of top quality structures may be easily lost as MBE is highly sensitive to any changes in the system. Especially, routine servicing procedures, which include any activity which requires unsealing of the growth chamber are devastating for developed growth parameters and force the necessity of recalibration. In this work, we present the process of growth parameters pre-optimization for obtaining homoepitaxial GaAs layers after servicing and restarting MBE system. Namely we present, how each step of reestablishing optimal growth condition influence various characteristics of obtained GaAs layers. Those include in situ, structural, and spectral measurement techniques.



Diagram 1. The dependence of the growth rate of GaAs layers on the gallium flux for the constant arsenic flux $7.46e^{-6}$ Thor and the substrate temperature of 580 °C.

Diagram 2. The dependence of the growth rate of GaAs layers on the arsenic flux for the constant gallium flux $6.02e^{-6}$ Thor and the substrate temperature of 580 °C. 3 areas are marked in the diagram:

Red - gallium rich condition, Blue - arsenic rich condition, Green - arsenic rich optimal condition.

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Geometrical selection during MBE growth of nanowires on polycrystalline substrates

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If semiconductor nanowires (NWs) are to be applied in electrically driven devices they should be well-oriented relative to the substrate normal. This requirement is easily fulfilled if NWs are formed by self-assembly on amorphous substrate. Then there are no epitaxial constraints and NWs grow perpendicular to the substrate surface. If, however, NWs are strongly linked to a polycrystalline substrate, their random orientation is expected due to an arbitrary arrangement of substrate grains, eventually impeding application of such NW array.

In this work we show that unidirectional supply of material fluxes in molecular beam epitaxy (MBE) favors growth of well-oriented NWs even despite random orientation of grains on which epitaxially linked NWs nucleate. We present a geometrical model describing an impact of MBE system geometry on the orientation of the NWs. Specifically, we calculated effective area of the nanowire's top facet Seff that is in the line of sight of an effusion cell as a function of nanowire tilt relative to substrate normal (α) and substrate rotation angle (ω) (see Fig. a). The S_{eff} value is equivalent to the number of atoms incorporated on the NW top facet which, in the framework of the model, controls axial growth rate of the NW. For example, this is the case of metal-limited MBE growth of A_3B_5 NWs. Fig. b shows normalized value of S_{eff} as a function of α and ω . As seen, the maximum of Seff corresponds to the NW facing directly in the direction of the effusion cell ($\alpha = 40^\circ$, $\omega = 0^\circ$) as expected. However, during rotation of the substrate the fastest growing nanowires are the ones that are perpendicular to the surface (Fig. c). To verify the model we grew GaN NWs on ZrN polycrystalline buffers. Without substrate rotation the NWs grew uniformly arranged at $\alpha = 40^{\circ}$, $\omega = 0^{\circ}$ (Fig. d), i.e. exactly as predicted by our model. Standard growth with rotation of the substrate led to densely packed NWs with significantly limited tilt dispersion (Fig. e). We attribute this behavior to the shadowing of strongly tilted NWs by preferentially grown perpendicular ones.



Fig. (a) geometry of the system; (b) effective area S_{eff} of the NW top facet as a function of α and ω angles; (c) average NW growth rate after entire substrate rotation as a function of the NW tilt α . SEM images of GaN NWs grown on ZrN buffer without (d) and with (e) substrate rotation.

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Insights into Raman spectra and doping concentration profiles of Eu-doped {ZnCdO/ZnO} superlattices

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ZnO is a commonly known semiconductor compound that has found application in various types of optoelectronic devices, such as lasers, diodes, and photodetectors. In photovoltaic cells it is used as a transparent conducting film, due to its wide band gap (ca. 3.37 eV at room temperature [1]) and low resistivity [2].

In high-efficiency thin film solar cells absorber layer can consist of several layers, each of them suited to absorbing a specific energy range of photons. In this case, ZnO could be in control of UV light conversion. To expand its possibilities towards shorter wavelengths, we propose ZnCdO alloy, as CdO has an energy gap of 2.24 eV [3]. The remaining key component of the presented samples is europium (Eu). Both ZnO and CdO has been successfully doped with this rare-earth metal, giving rise to a huge improvement in carrier concentration and mobility, which is desirable in terms of future applications [4, 5].

Eventually, we decided to use superlattice (SL) structure type grown by molecular beam epitaxy to strictly control composition of the final material. Moreover, in SLs it is possible to modulate the energy gap simply by changing thickness of a sublayer, which is also a promising feature [6]. Thus, in this work we would like to present first Eu-doped {ZnCdO/ZnO} SL structures. Raman spectroscopy (Fig. 1.) has revealed additional modes, dopant-related ones, forbidden in bulk ZnO, that occur in doped ZnO films. Furthermore, post-growth treatment analysis showed that annealing cause significant modifications in Raman spectra associated with changes of impurity content. Additionally, p-n junctions with the aforementioned SL



Fig. 1. Raman spectrum of one of the samples with various ZnO modes. Si substrate modes are marked with an asterisk.

structures were prepared on silicon substrates to be able to perform electrical measurements. Current-voltage characteristics shown the obtained samples are functioning diodes with a rectifying p-n junction. Capacitance-voltage measurements helped to determine dopant concentration and its distribution within the junction depletion region. To the best of our knowledge, there are no results of studies reporting observation of Eu dopants in Raman spectra and doping concentration profiles of Eu-doped {ZnCdO/ZnO} superlattice structures.

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Electrical and thermal transport properties of CdO and CdMgO alloys grown using plasma-assisted MBE technique

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CdO is one of the oldest transparent conducting oxides (TCOs) that attracts considerable attention for optoelectronic applications because of its high electron concentration, high electron mobility, and high transparency in the visible spectrum. Our recent reports on MBEgrown CdMgO alloys show that the lower optical bandgap of CdO (Eg=2.3 eV) can be overcome by alloying with wide bandgap MgO (Eg=7.5 eV) which is useful for light detection from visible to deep UV regimes [1,2]. However, to obtain maximum energy absorption, it is important to understand the relation between Cd and O₂ parameters during the growth process which influences the stoichiometry of the CdO and related ternary alloys. In the present studies, we investigated plasma-assisted molecular beam epitaxy (PA-MBE) grown CdO and CdMgO ternary alloys. Temperature-dependent Hall measurement revealed the maximum mobility of 352 cm²V⁻¹s⁻¹ achieved at room temperature with a carrier concentration of about 4.5 x 10^{19} cm⁻³. The thermal transport properties, such as thermal conductivity, thermal diffusivity, and thermal boundary resistance, were measured using photothermal infrared radiometry. We believe that our findings may provide valuable insight for a better understanding of CdMgO alloys which can open up new prospects for designing novel optoelectronic devices.



Figure 1. Bandgap shift as a function of carrier concentration (Total shift $\Delta = \Delta E_{BM} - \Delta E_{BGN}$ is denoted by + sign)

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$Characterization \ of \ MBE \ grown \ \{Zn(Mg)O/ZnCdO\}_m \ superlattices \ doped \\ in-situ \ with \ Eu$

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Zinc oxide (ZnO) is a well-known n-type semiconductor with a wide, direct bandgap - 3.37 eV at room temperature and a high exciton binding energy (60 meV) and potentially has wide applications in optoelectronics [1]. The synthesis of Zn(Cd,Mg)O ternary alloys and quantum structures (heterostructures, multiquantum wells and superlattices) enables bandgap engineering in a wide spectral range from ultraviolet to green [2]. The red luminescence can be obtained in oxide structures by doping with Eu. ZnO nanostructures doped with rare earth elements are synthesized by various techniques, however, a method for obtaining quantum structures with a predetermined location of Eu has not yet been developed. The molecular beam epitaxy (MBE) technology makes it possible to place Eu either in barriers or/and in quantum wells.

In this work we present the properties of in situ Eu-doped $\{Zn(Mg)O/ZnCdO\}_m$ shortperiod superlattices (SLs) grown on sapphire substrates (Al₂O₃) by MBE. The thicknesses of the ZnMgO and ZnCdO:Eu layers are 15±3 nm and 2±1 nm, respectively. Eu impurity was introduced into quantum wells. The obtained structures were characterized by X-ray diffraction (XRD), scanning electron microscopy (SEM), transmission electron microscopy (TEM) and secondary-ion mass spectrometry (SIMS). The optical properties were tested by photo- / cathodoluminesence (PL/CL) measuments and UV-Vis spectroscopy. The XRD pattern of Eu-doped $\{Zn(Mg)O/ZnCdO\}_m$ SLs indicated hexagonal crystal structure. The formation of good quality SLs were confirmed by TEM and XRD measurements. Studies of the CL spectra for as grown in situ Eu-doped $\{Zn(Mg)O/ZnCdO\}_m$ SLs showed at room temperature emission bands at ~617 nm, due to the ${}^5D_0 - {}^7F_2$ intra-4f-shell transition of Eu³⁺ ions (Fig. 1). We have determined the optimal conditions for amplifying the red emission. All



structures were annealed for 1 minute in an O_2 environment at various temperatures. The highest intensity of the ${}^5D_0 - {}^7F_2$ peak was observed after annealing at 700°C. The decrease in intensity after annealing at high temperatures can be associated with the destruction of the superlattice structure due to Cd and Mg diffusion [3].

Fig. 1. RT-CL spectra of as-grown and annealed {ZnMgO/ZnCdO:Eu}₂₂ SLs.

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n-ZnO/ZnCdO/p-Si and n-ZnCdO/ZnO/p-Si diodes: studies on the influence of the junction interlayer on electrical properties and structural defects

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Zinc oxide (ZnO) remains competitive in the wide-bandgap semiconductor family, convenient for ultraviolet and blue optoelectronics: light-emitting diodes (LEDs) and lasers. Among others, its advantages are low price and ability for bandgap engineering. Indeed, the zinc-cadmium-oxide alloy (ZnCdO) is an example of such technology, providing bandgap narrowing [1]. However, alloying ZnO with CdO is problematic in the full composition range due to the solid solubility of Cd which is limited to ~40%, as well as different stable crystal structures - wurtzite (ZnO) and rock salt (CdO). Therefore, optimisation of ZnO and ZnCdO-based structures is still challenging at hand, as well as their n-type conductivity nature is discussed.

In our work [2], we studied the pair of bilayers of n-ZnO/ZnCdO and n-ZnCdO/ZnO on a p-Si substrate grown by molecular beam epitaxy. We carried out an in-depth investigation of these heterojunctions with their electrical properties based on: current-voltage (IV) and capacitance-voltage (CV) measurements in the broad temperature range of 40 to 300 K, as well as deep-level transient spectroscopy (DLTS). Research was supported by the micro-Raman scattering for study of their structural properties and lattice dynamics.

The IV characteristics confirmed the rectifying properties of the heterojunctions, with the electrical parameters obtained: rectifying factor, ideality factor, built-in voltage, series and

junction resistance. The further study of IV plots in log-log scale gave insight into the current transport mechanisms of diodes, pointing out the multi-tunneling capture-emission process and space charge limited current, which are a consequence of the presence of defects. Next, the dopant concentrations were calculated from CV characteristics with their dependences on temperature and junction depletion region width. Finally, the DLTS results confirmed the presence of typical donor-like traps (see Fig.). It was noticed that cadmium atoms affected the electrical properties and energies of trap present in the studied junctions.

Our research let us suppose that the Cd-related defect complexes, such as: Zn_i -Cd_{Zn}-V_O or Cd_{Zn}-V_O, are the most appropriate candidates for n-type conductivity of ZnCdO alloys.





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An insight into the optical properties of GaN nanowires with Al₂O₃ and HfO₂ shells

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Recent development of gallium nitride-based heterostructures to a great extent is based on nanostructures, which help to overcome the limits of heteroepitaxy. Among others, nanowires (NWs) arrays bring many advantages since they eliminate global stress, increase the photoluminescence efficiency, and provide new opportunities for the construction of modern devices by changing the NWs composition, architecture or dimensions. This may lead to the next generation of ecological and highly efficient light emitters and detectors.

In this work, we have studied self-assembled GaN nanowires of ~90 nm diameter and 2 μ m height, grown by plasma-assisted molecular beam epitaxy on a Si(111) substrate. The NWs were covered with oxide shells of Al₂O₃ and HfO₂ prepared by atomic layer deposition process. The nominal thicknesses of the shells were 5, 10 and 20 nm. The structural parameters of the NWs, such as diameters and fill factors, were derived from SEM images. The core-shell architecture was verified in the SEM transmission mode.

The influence of shells on the optical properties of GaN NWs was studied in optical reflectance and photoluminescence measurements. First, the reflectance spectra were measured and examined within the wavelength range of 300-1800 nm at room temperature. The effective medium approximation model was used to fit the observed interference patterns [1] (Fig. (a)). Based on the reflectance measurements, spectral dependence of effective refractive index was calculated. It has been shown that the experimental data precisely follow the theoretical model. Moreover, the fill factors were estimated with the Bruggeman model [1]. Their values were found to agree very well with those obtained from SEM images.

Finally, the optical properties of the NW samples were studied in the temperature range of 10 to 300 K. The photoluminescence signal (Fig. (b)) was observed in the ultraviolet region with strong emission bands around 3.45 eV, 3.42 eV and 3.18 eV [2, 3]. Their origin and dependence on oxide shell thickness will be discussed.



Fig. (a) Reflectance spectra (points) with theoretical fits (lines), and (b) PL spectra at 10 K of GaN/Si NWs with various oxide shells.

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Effect of strain and surface proximity on the acceptor grouping in ZnO:N

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The experimentally observed two types of photoelectron spectra of different crystallites in ZnO:N indicate the grouping of acceptor and donor complexes in separate domains [1]. Density Functional Theory (DFT) studies suggest that the acceptor domains can involve zinc vacancy (V_{Zn}) and $-(NHx)_0$ groups and provide complexes-related acceptor states near the valence band maximum. However, the formation energies of these complexes are relatively high, and the acceptor-related sample properties in general depend on a number of factors such as growth or annealing conditions [2]. As the formation of defects involves distortion of the crystal lattice, it may be assumed that micro-strain plays an important role here.

DFT study of the effect of strain and surface proximity on the formation of (V_{Zn}) -(NHx)_O complexes was performed using QUANTUM-ESPRESSO package [3] within the generalized gradient approximation (GGA) with the Hubbard-like +U term describing the on-site Coulomb interactions. The formation of complexes require the migration of the constituent defects and was computed using the Nudged Elastic Band (NBE) approach under (a) tensile and compressive biaxial strains in planar plane and along z-axis, (b) hydrostatic pressure, (c) local lattice distortion provoked by uncontrolled impurities as CH_x groups, and finally (d) surface proximity. In the last case, defects were introduced into the 0D nanocrystal object, as quantum dot (QD). DFT results indicates a strong strain effect on the electronic structure and



and dissociation energies, respectively.

activation of migration energy (E_A). In particular, 4% compressive biaxial strain reduced migration barrier of zinc vacancy moving around (NH)_O from 0.65 eV to 0.24 eV. Moreover, the presence of a CH_x group near the complex lowers this value to zero. The effect of strain explains lowering of acceptor formation energy and might be also responsible for grouping of acceptors which can be formed only in crystallites showing compressive strain. In support to the DFT results, low-temperature cathodoluminescent CL maps reveal more intensive acceptor CL for the higher average micro-strain values.

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Tailoring of the optical and electrical characteristics of ZnO: Ga heterostructures by a metallic buffer layer

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Abstract:

Due to its excellent transmittance, high UV emission, higher environmental and thermal stability, abundance of resources, and inexpensive cost for use in many sophisticated applications, nano-dimensional ZnO has received a lot of attention. In this regard, we present information on the photoluminescence and structural characteristics of thin films of Ga-doped ZnO made using pulsed laser deposition. A methodical presentation of the significant impact of Au and Ag as an interlayer on Ga-doped ZnO thin films has been made. According to X-ray diffraction examination, the generated films' crystalline microstructure was correctly aligned along (002) lattice planes with a hexagonal wurtzite structure. A surface analysis using a field emission scanning electron microscope revealed the films' surfaces to have very little surface roughness. With the addition of metallic interlayers, these films demonstrate an improvement of twofold photoluminescence behavior in the vicinity of band edge emission. Using an IR spectrophotometer, the resulting films' band gap energies ranged from 3.29 to 3.52 eV. Ga-doped ZnO offers a novel technique to replace other UV and IR photodetectors since reflection spectra show less reflectance (16%) and increased absorption in the UV-NIR region.

Ferromagnetic Resonance Studies of (Ga,Mn)N

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Dilute ferromagnetic semiconductors, in particular (Ga,Mn)N predicted to have an exceptionally high Curie temperature (T_c), have attained great research importance due to their unique ability to combine the properties of semiconductors and magnetic materials [1]. Moreover, GaN being a wide band gap semiconductor has been dominating the photonics [2] and high power electronics. So it is important to make an effort to understand the underling magnetic properties of (Ga,Mn)N.

We report ferromagnetic resonance (FMR) studies of a series of (Ga,Mn)N layers grown by molecular beam epitaxy [3,4]. All investigated samples showed ferromagnetic signatures, as evidenced by SQUID magnetometry, with $T_{\rm C}$ ranging from 3 to 12 K. A broad angularly dependent FMR signal appears only at higher temperatures, closer to and above $T_{\rm C}$, with intensities roughly scaling with magnetic susceptibility of the material, as shown in Fig, 1.

However, apart from a very weak paramagnetic signal of Mn^{2+} , no ferromagnetic resonance is observed below where 7 K, i.e., such ferromagnetic features as the hysteresis of magnetization the remnant curves and moment are the strongest. We counterintuitive relate this lack of low temperature FMR signal inhomogeneous to broadening caused by nonuniform distribution of magnetic ions and thus inhomogeneities in coupling strengths influencing the local anisotropies magnetic of Mn³⁺ ions.



Fig.1 Comparison of magnetic susceptibilities determined by SQUID at fields of 1 and 10 kOe with that determined from FMR signal intensity at fields about 2 kOe. The inset shows the angular dependence of the resonance fields at 12 K

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Impact of current flow direction on the distribution of carriers in multiple, color-coded InGaN quantum wells

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Unique properties of group III-nitride semiconductors allow for the design of highefficiency and high-power optoelectronic devices. However, not all of these properties are advantageous. The fact that the internal quantum efficiency (IQE) tends to drop for high current densities, caused by significant Auger recombination and decreased injection efficiency is one of its negative characteristics [1]. The straightforward solution to this problem is to increase the number of quantum wells (QW), but III-nitrides pose strong built-in, spontaneous and piezoelectric fields $(\vec{F_P})$. This, together with the low mobility of holes in comparison to electrons, leads to the asymmetric carrier distribution and injection in the active region with multiple QWs, where in standard structures the QW closes to the p-type region tends to dominate [2].

To assess this implication, two LEDs with current flow direction (\vec{J}) parallel to built-in polarization (Fig. 1 c-d), and two with \vec{J} anti-parallel to $\overrightarrow{F_P}$ (Fig1. a-b) are compared. This is achieved on bulk Ga polar substrate by using the buried tunnel junction (TJ) [3,4]. In addition, all LEDs consist of double 2.6nm QW where the position of the blue- and green-emitting active region was varied with respect to the p-type region, resulting in structures where electrons firstly flow to the shallow blue QW, for LEDs in Fig. 1 a, d, or deeper, green QW, for LEDs in Fig. 1 b, c.

By measuring the relation between emission peaks from green- and blue-emitting QW in a wide range of current densities we investigated the impact of the polarization on the carrier distribution in double, color-coded InGaN-based QW LEDs.



Fig. 1 Schematic structures and band structures at ~100 A/cm² for LEDs with (a-b) p-type down, (c-d) p-type up.

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Extreme InGaN growth conditions by plasma assisted molecular beam epitaxy

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The crucial challenge in high quantum efficiency nitride based devices like light emitting diodes (LEDs) or laser diodes is the growth of InGaN layers. In this work we study properties of InGaN layers grown by plasma assisted MBE (PAMBE) for very high nitrogen fluxes – 2.8 μ m/h. These conditions are unusual for PAMBE technology since available high N flux is limited by e.g. MBE pumping speed to maintain enough high vcuum (less than 10⁻⁴ torr) during process. The high nitrogen flux prevents InGaN decomposition. This allows us to explore growth conditions of InGaN that were previously not achievable in PAMBE – growth at high temperatures. The InGaN layers were grown in In-rich conditions, with different Ga fluxes with constant N flux (2.8 um/h) and a very high growth temperature T_g=710 C.

The obtained results are analyzed within phenomenological model of InGaN growth assuming growth on nonequivalent atomic steps [1]. In Figure 1(c) the InGaN content as a function of Ga flux is shown according this model prediction. As the Ga flux is reduced, we observe saturation of In content due to the decomposition of InGaN (Nitrogen loss) - it is observed as stabilization of photoluminescence emission wavelength (around 430 nm). The behavior of emission wavelength for low (0.04 um/h) and high (2.5 um/h) Ga fluxes also follows model prediction of In content.



Fig. 1. The intensity of photoluminescence from InGaN layers (a), the wavelength (b) and the indium composition as a function of Ga flux (In content calculated from theoretical model [1]).

We found that the photoluminescence intensity strongly depends on the Ga flux (see Fig.1 (a)). It can be correlated with the surface morphology. We will discuss the photoluminescence efficiency, surface morphology and indium content of InGaN layers.

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Doping inhomogeneity at the nanoscale in GaN:Si studied by electrochemical etching

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Electrochemical etching (ECE) has recently emerged as a very useful method to fabricate porous GaN of a decreased refractive index [1] e.g. for cladding layers in laser diodes [2] or for removal of sacrificial layers and structure lift-off [3]. Due to high selectivity against n-type doping, ECE technique can also serve as a tool to detect inhomogeneity in Si doping in GaN grown by plasma-assisted molecular beam epitaxy (PAMBE) [4]. We showed that surface morphology – atomically flat or step-bunched – impacts the Si incorporation and leads to doping inhomogeneity at the nanoscale.

In this work we focus on the extremely high Si doping of GaN up to $1 \cdot 10^{20}$ cm⁻³ and its consequences for the surface morphology and the electrical properties within the layer. We study GaN:Si⁺⁺ layers grown by PAMBE on GaN/Sapphire templates. First, we observe the enhanced growth rate along the a-direction (1120) that causes the emergence of the hillocks with star-shaped arms, that we call "nanostars", see Figure 1(a). Second, we investigate local Si content and measure only a 10% Si composition decrease in the nanostar by energy-dispersive X-ray spectroscopy (EDX) and third, we use ECE to study electrical properties at the nanoscale. Because the nanostars are less conductive than the surrounding layer, they are not etched during ECE, see Figure 1(b). We discuss the origins of the decreased nanostar conductivity and attribute it to an increased compensation that is most likely caused by the complexes of Si substitutional and vacancies.



0 5 nm

Figure 1 (a) Surface morphology of GaN:Si layer $n=1 \cdot 10^{20}$ cm⁻³, (b) SEM after ECE at 2.5V.

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Visible luminescence of BGaN-based structures: towards defects-related LEDs

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Boron containing Me^{III}N nitrides are currently of interest because boron can affect the physical properties of GaN-based ternary (BGaN, BAIN) and quaternary (BInGaN, BAIGaN) alloys. Theoretical studies indicate that boron can provide the ability to tune the bandgap width, lattice parameters and dielectric constant of the compounds. Therefore, BMe^{III}N compounds can be useful be useful for production of new electronic or optoelectronic devices.

We fabricated and characterized BGaN-based epitaxial structures on sapphire substrates with various buffer and active layers grown by MOCVD for LED applications. Photoluminescence and Raman spectra were measured to demonstrate red lighting and the residual stress within the LED structures. Crystals structure and morphology of the layers were characterized by XRD and TEM methods, and electrical parameters by Hall measurements. As was recently revealed, the B concentration in BGaN depends on growth temperature [1]. It was shown that the B concentration in BGaN determined to be about 2.5% at., decreases with increasing growth temperature. Such layers are highly insulating with a sheet resistance of $8 \times 10^{10} \Omega/sq$. A typical photoluminescence appears in the broad red range, which can be attributed to a defect center with an activation energy of 1,35 eV [2]. In this work, we intentionally doped BGaN layers with Si or Mg, using appropriate precursors, to form n- or p-type semiconductor. Additionally, p-n junctions with BGaN were produced in the form of multilayer epitaxial stacks. Because of many structural defects formed in BGaN



Fig. 1. PL spectra of LED structures with BGaN or BInGaN epilayers grown on sapphire (0001).

grown at temperature close to 900°C or higher, indium was incorporated into the layer to attenuate the contraction of the BGaN lattice parameters. Such an Ininduced contraction can lead to latticematched crystal growth. Fig.1 shows the PL spectra with red luminescence over a wide range, including the yellow component, while the bandgap edge emission is almost eliminated. The yield of red luminescence is higher for B-rich BGaN:Mg grown at 800°C, but for BGaN:Si the yield is higher for samples grown at 1050°C. Comparative studies of BGaN and BInGaN layers suggest next steps to improve structural and electrical parameters of the layers required for red defect- related LEDs.

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Modification of luminescence of the nitride nanowires by chemical treatment of their surface

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Although the influence of surface conditions on transport and optical properties of semiconductor nanowires (NWs) has been identified, the methods of controlling it and inhibiting its detrimental impact on the NW characteristics still remain an important subject of research. As the depth of the depletion or accumulation layer induced by surface band bending may become comparable with the diameter of the NW, its part available for transport, charge carrier injection and light generation can be markedly reduced. This applies also to NWs made of the group III nitrides. Therefore, we carry out complementary studies of luminescence of GaN NWs vs. chemical and electronic conditions on their surfaces.

The NWs are grown catalyst-free on in-situ nitradated Si (111) substrates by plasmaassisted molecular beam epitaxy (MBE). The morphology of the samples is assessed by scanning electron microscopy (SEM). The luminescence of individual NWs is investigated by SEM-based cathodoluminescence spectroscopy and mapping. X-ray photoelectron spectroscopy (XPS) enables us to measure the shifts of the valence band maximum with respect to the Fermi level (i.e. to deduce changes in surface band bending) as well as assess changes in the chemical composition of the surface.

We report the results of CL and XPS experiments on the GaN NWs modified by etching their surfaces in KOH and HCl solutions. Both substances are used as media that etch, deoxidize or passivate nitride surfaces [1-3]. Therefore we applied them to modify surface chemical composition and charge accumulation in view to optimize luminescence of the GaN NWs.

The CL technique enabled us to reveal changes induced in luminescence of individual NWs as well as of the ensemble of them. Cathodoluminescence line-scans helped us to follow changes in the intensity and spectrum of luminescence along the NW. The XPS experiments, carried out in an oblique configuration to probe a substantial part of the side NW surface, allowed us to follow the changes in the emission from the GaN atoms, surface contaminants (oxygen and carbon) and residues of the etching solutions. So, changes in the surface chemical composition could be assessed. Shifts of the valence band edge vs. the Fermi level showed the changes in the surface band bending. The correlation of the results acquired by both techniques evidenced the possibility to control GaN NW luminescence by surface etching.

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Strain relation in GaN nanowires with HfO₂ and Al₂O₃ shells examined by X-ray diffraction and Raman spectroscopy techniques

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Recent years have exhibit the dynamic development of integration photonics with electronics on a single platform. GaN nanowires (NWs) are often studied as a potential building blocks of such systems since NWs grow virtually free of strain and defects even on substrates with large lattice mismatch [1, 2]. Usually side surfaces of NWs are passivated by oxide shells. Thus, the question arises about the strain induced by such shells.

In this work we studied self-assembled GaN nanowires of ~100 nm diameter and 2 μ m height, grown by plasma-assisted molecular beam epitaxy on Si(111) substrates. The NWs were covered with shells of HfO₂ and Al₂O₃ using atomic layer deposition. The nominal thicknesses of the shells were 5, 10 and 20 nm. The dimensions of the NWs, such as diameters and shell thicknesses, were measured from SEM images. The uniformity of shells covering GaN NWs was analyzed in the SEM transmission mode.

Structural properties of GaN core-shell NWs were examined using X-ray diffraction (XRD) and Raman spectroscopy techniques. In particular, the values and the type of strain were investigated, together with their dependence on the shell thickness. First, the XRD curves and high resolution XRD maps were measured to calculate the accurate values of lattice parameters of GaN core parts. $2\theta/\omega$ scans of 0002 GaN reflections (see Fig. 1(a)) revealed changes in interplanar distances in the GaN core with the magnitude depending on the thicknesses of shell parts. Calculations of *a* and *c* lattice parameters allowed to find the inplane (ε_{xx}) and out-of-plane (ε_{zz}) strain values for the studied samples. Next, the Raman spectroscopy was used to study the lattice vibration mechanisms, crystalline quality and strain in GaN/HfO₂ and GaN/Al₂O₃ core-shell structures. The comparison of in-plane strain values (ε_{xx}) obtained by these two techniques is presented in Figure 1 (b). The origin of strain and influence of oxide shell thickness on strain distribution in GaN/oxide core-shell nanowires will be discussed within this work.



Figure 1. (a) XRD $2\theta/\omega$ scans of 0002 GaN reflection for GaN NWs with HfO₂ shell of different thickness, and (b) the values of ε_{xx} strain component calculated from XRD and Raman measurements of GaN/oxide NWs of various shell thickness.

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Investigation of strain relation and lattice dynamics in {CdO/MgO} superlattices grown by plasma-assisted molecular beam epitaxy

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Cadmium oxide (CdO) is a promising material for transparent conducting oxide (TCO) thin films due to its excellent performance regarding conduction [1]. It is characterized by high transparency in the visible spectrum, low resistivity and high carrier mobility [1]. In the TCO family, II-VI ternary oxide alloys have attracted considerable interest due to the possibility of modulating their band gaps in a wide range of values [1], what opens up the possibility of new applications in modern optoelectronic devices.

In this work we have studied a series of $\{CdO/MgO\}\$ superlattices (SLs) grown on *a*-, *c*-, *m*- and *r*-plane sapphire substrates by plasma-assisted molecular beam epitaxy (PA-MBE). The samples were measured by X-ray diffraction (XRD) and Raman spectroscopy techniques to obtain the strain distribution in SLs system as well as to study the lattice dynamics in the fabricated structures.

Figure 1 (a) shows the high resolution X-ray diffraction map (HRXRD) for selected sample grown on *r*-plane sapphire. It clearly reveals the periodic structure of the superlattice. Moreover, HRXRD maps allow to designate the accurate values of *a* and *c* lattice parameters for the series of samples. Furthermore, we can calculate the in-plane (ε_{xx}) and the out-of-plane (ε_{zz}) strain values.

Next, the Raman measurements were performed to examine lattice vibration mechanisms and crystalline quality of {CdO/MgO} SLs. The typical spectrum is shown in Figure 1 (b). First-order Raman modes are forbidden in the case of rocksalt compounds such as CdO and MgO. However, lattice mismatch between layers it the SLs induce strain in the structure. Disorder introduced by the strain may be the reason of the appearance of TO MgO mode.



Figure 1. High resolution XRD maps measured for 111 CdO/MgO SL peak (a) and Raman spectra with an analysis of selected spectrum range carried out by fitting the data with Lorenz function (b) obtained for SLs on *r*-plane sapphire.

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Band parameters of group IV semiconductors in wurtzite structure

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Metastable crystal phases of group IV semiconductors present in low aspect ratio nanostructures exhibit great potential to overcome current limitations in semiconductor industry. Their chemical and structural similarity to diamond structure is a key factor in incorporation such structures in current group IV technology, at the same time leaving much room for tailoring physical properties by means of size effect and polytypes mixing. Usual pathway for modeling heterostructures and low dimensional systems are **kp** and tight binding models which are computationally efficient and require minimal set of parameters. It is known however, that the accuracy of these models relies strongly on their parametrization and the stable phase values, present in the literature, cannot be transferred to other crystal phases. On the other hand *ab initio* methods provide great accuracy without empirical parameters but are computationally more expensive which limits their use in complex systems modeling.

The goal of presented work is the parametrization of six band \mathbf{kp} model for lonsdaleite crystal phase which is the second most common phase after diamond. We use high accuracy *ab initio* methods to derive physical properties of group IV materials and then transfer our results to \mathbf{kp} model. In particular the effects of strains on crystal and band structures are investigated in detail and described by elastic tensor and absolute deformation potentials respectively. Our parameters set together with already known parameters for diamond structure allows for efficient modeling of heterostructures containing different polytypes.

Solutions for spectroscopy and imaging

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Realization of a High-Finesse Cavity Platform for Functionalization of Emitters

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The race for the idealization of a quantum-computing device has led to the demand for an efficient single-photon generation device with optimized attributes of purity, indistinguishability, and brightness. The two latter parameters can be improved by incorporating single-photon emitters inside a photonic cavity. Here, we present a cavity platform, where the top mirror consists of a concave Gaussian-like shape, while the bottom mirror is planar. This cavity geometry achieves high finesse values since the upper

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Figure 1. Schematics of the cavity design.

wavefront of the Hermite-Gaussian beam matches the mirror geometry and is reflected into the cavity mode with minimal losses. It also facilitates achieving low-mode volume since the beam waist rests at the bottom mirror. Both factors are crucial for high-Purcell Factors. The bottom mirror rests on a piezo stage while the top rests on a tunable kinematic mount. The elements are connected by a stiff and lightweight cage mount to shift the mechanical resonance frequencies above the typical acoustic and seismic environmental noise frequencies.

The cavity length was scanned to characterize the cavity setup with an oscillating voltage applied to the piezo, which holds the bottom mirror. Two lasers, one acting as a reference and the other tunable, were shined upon the cavity. The resulting distance between the two corresponding peaks for a cavity mode was utilized to provide information about the slopes of the cavity modes. With that, we could determine the broadening of the cavity modes in the wavelength domain, which led to the information about the Q-factors of the cavity. Since, for a cavity without internal losses, the O-factor varies linearly with the mode numbers thus, we were able to calculate the finesse parameter. For our

100000 634.8 nm Linear Fit 90000 80000 Q=qF+Q₀ Q-Factor 43.510 43 616 43.52 70000 -0.002 0.000 0.002 60000 0.004 0.006 50000 0.008 Data 0.010 40000 3 2 5 6 Relative Mode Index

Figure 2. Q-Factors for different modes. In subset, photodiode voltage with cavity length detuning.

mirrors with 500 ppm transmission, we were able to achieve Q-Factors up to 10^5 and Finesse of $6 \cdot 10^3$, which is comparable to the theoretical values for the mirrors.

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TuO7

Electrical switching of a chiral lasing from polariton condensate in a Rashba-Dresselhaus regime

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Efficient optical processing of classical and quantum information imposes on light novel requirements: chirality and low threshold non-linearities. In this work we demonstrate a chiral lasing from an optical modes due to emerging photonic Rashba-Dresselhaus spinorbit coupling. For this purpose we developed a new electrically tunable device based on an optical cavity filled with birefringent liquid crystal (LC) and perovskite crystals [1]. Our novel method for the growth of single crystals of CsPbBr₃ inorganic perovskite in polymer templates allows us to obtain crystals with controlled size and thickness. Such crystals we place in a microcavity filled with bifefringent LC. We reach a strong light-matter coupling regime between perovskite excitons and cavity modes, and induce polariton condensation at room temperature on cavity chiral modes [2].

In our device, the emission from the condensate can be spectrally tuned by 29 meV, thanks to the sensitivity of the liquid crystal to an external voltage. We use birefringence of the liquid crystal to induce the Rashba-Dresselhaus spin-orbit coupling. We show chiral polaritonic modes below condensation threshold and simultaneous emission from the polariton condensate in two, tilted beams exhibiting polarization of opposite handedness above the threshold (Figure 1). We make use of different condensation thresholds for different regimes of spin-orbit interactions to create electrically controlled switch.



Figure 1. Polariton condensation in the Rashba-Dresselhaus regime. (a) Scheme of the sample and graph of a circularly polarized polariton condensation. Angle-resolved photoluminescence spectra in both circular polarizations and resulting the S_3 parameter: (b–d) below and (e–g) above the condensation threshold. (h) Emission intensity, spectral linewidth and energy blueshift for increasing excitation power.

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Natural Exceptional Points Appearing in Semiconductor Microcavity Systems

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In quantum mechanics various systems can be represented within a Hermitian Hamiltonian description. In the case of a physical system in which gain and dissipation play a significant role, this description is no longer valid and the non-Hermitian character of the system must be taken into account.

An example of a phenomenon that can only occur in non-Hermitian systems is the appearance of the so-called exceptional points (EPs). These specific singularity points arise in parameter space, where at least two eigenvalues are degenerate (both the real and imaginary parts of the energies are equal). The EPs are characterized by an exotic topology of the eigenvalue surfaces near an exceptional point singularity [2], which can be demonstrated while encircling them in the parameter space in which they appear. Performing a full loop around the EP does not lead to the initial state, but results in switching to the second eigenstate. In this work, following the theoretical model, we observe such a behavior experimentally in a semiconductor microcavity.

In our work, we demonstrate the emergence of the exceptional points associated with the transition of the system from the strong to the weak light-matter coupling regime. We perform angle-resolved experiments in which we additionally vary the Rabi splitting by the power of the incident laser beam. Specifically, we encircle the EP in parameters space of a wave vector–coupling strength, as shown in the Figure. The presence of the EP is evidenced by the specific exchange between the eigenstates, which is in excellent agreement with the proposed theoretical model.



Figure: Encircling the exceptional point in a space of parameters: Rabi splitting and wave vector change, (a) real and (b) imaginary part of the energy.

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Driven-dissipative Bose-Einstein condensation of photons in a VCSEL

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Photons are the first ever particles considered in the Bose-Einstein statistics and the most abundant bosons in nature. Surprisingly, photonic gases were among the latest to demonstrate Bose-Einstein condensation (BEC) [1], despite many analogies developed over the years between laser physics and BEC. Recently, semiconductor lasers have been proposed to be a promising direction in photon BECs at room temperature [2].

Here, we observe signs of a driven-dissipative BEC phase transition in a broad-area oxide-confined vertical-cavity surface-emitting laser (VCSEL) designed for 980 nm optical data communication. We tested devices with different cavity mode-quantum well energy detunings, following the conclusions of the photon BEC physics based on Rhodamine 6G-filled cavities. For negative detuning, we observed ground-state condensation, followed by a thermalised distribution of excited states, in agreement with the Bose-Einstein distribution. Remarkably, the photon gas thermalises to effective temperatures much lower than those of the device. We also extract the effective chemical potential from the spectrum and compare it to the theory recently developed in the ultra-fast thermalisation limit [3], which suggests the driven dissipative nature of the nonequilibrium BEC in the semiconductor resonantor. In contrast, for a positively detuned device, we observed a standard broad-area VCSEL behaviour with multimode lasing above the threshold current, with an absence of Bose-Einstein distribution. Furthermore, we discuss the issue of an accurate description of the density of states of a realistic device and the actual shape of the confining potential under different driving conditions.

Our results show a new perspective on semiconductor VCSELs, which can be used to test the nonequilibrium BEC physics in table-top room-temperature devices. Moreover, it opens the possibility to observe effective photon-photon interactions and collective superfluid phenomena in photon BECs, using the well-known ultrafast semiconductor nonlinearities, which has been impossible in rhodamine-based photon BECs limited by the slow thermo-optical nonlinearity. This would eventually link the worlds of polariton and photon superfluid physics into one device.

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Topologically protected interface states induced by spin-orbit coupling in liquid crystal microcavity

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Liquid crystal optical microcavities are a novel platform for achieving adjustable synthetic Hamiltonians with Rashba-Dresselhaus spin-orbit interactions [1].

In this work, we study the topological properties of light within a one-dimensional photonic lattice featuring spin-orbit coupling. To investigate the topological properties of our system, we propose a two-band tight-binding Hamiltonian in momentum space, treating spin-orbit coupling as a perturbation. We prove that topological properties of the considered system are characterised by a winding number \mathcal{W} . This topological invariant counts the number of times the eigenstates wind around the origin in momentum space when the wave vector runs through the Brillouin zone, $k = 0 \rightarrow 2\pi$. Compared to the well-known Su-Schrieffer-Heeger (SSH) model of a topological insulator, our one-dimensional lattice, allows for both negative and positive values of the topological invariant. This property enables us to investigate topologically protected states at the interface between lattices with a different sign of a winding number, schematically illustrated in Figure 1. We demonstrate that this leads to qualitatively new topological interface states that are absent in the SSH model.

Our theoretical analysis provides a promising perspective for realising a new class of topological lasers based on liquid-crystal optical microcavities. Furthermore, our results can be extended to a broad range of one-dimensional topological Hamiltonians with spin-orbit coupling or internal degrees of freedom [2].



Figure 1 – Schematic illustration of a topologically protected interface state in the basis of vertical and horizontal polarization eigenstates.

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Quantum Mechanical-like Approach with Effective Hamiltonians Based on Resonant States in Optical Cavities

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Optical microcavities are extensively studied in solid-state physics due to the interaction between confined photons and excitons in the active layer leading to a strong coupling regime and even a Bose-Einstein condensate. Alternatively, a cavity with birefringent and active material enables the creation of synthetic fields for photons, where light polarization acts as the spin for electrons. The interaction between orthogonally polarized modes leads to various effects of solid-state physics, including the Rashba-Dresselhaus interaction [1], merons and antimerons (Bloch and Neel skyrmions) [2], persistent spin helix [3], and annihilation of exceptional points [4]. Although the Berreman [5] and Schubert [6] transfer matrix methods are well established (though time-consuming), many researchers prefer to use a simpler 2×2 effective Hamiltonian based on ideal resonator (Fig. 1 (a)) to describe each of the aforementioned effects.

Our work introduces a method based on resonant state and perturbation theory to derive formulas for the 2×2 effective Hamiltonians for various types of cavity coupling. Firstly, we replace the standard stack of distributed Bragg reflectors with δ -mirrors (Fig. 1 (b)), and, using



Figure 1: Schematic of the idea of presented theoretical approach. The imaginary part (i.e. finite lifetime) of polarized photons is presented.

Green's function, we obtain a simplified formula for the electric field. We adopted the perturbation theory for Green's function to obtain the multimode Hamiltonian, which can then be further simplified to the 2×2 effective Hamiltonian (Fig. 1 (c)) via the next perturbation method, which includes interactions with other states (Fig. 1 (d)). In the last phase, we proposed a formula for polarization patterns, which is essential for their accurate determination, and relies on the rotation of basis via perturbation theory (Fig. 1 (e)). The application of a solid-state approach to solve optical problems not only simplifies calculations, but also allows better understanding of various problems like topological systems or non-hemitian physics.

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Merons in Reciprocal Space as a Result of Photonic Spin-Orbit Interaction in Birefringent Microcavity

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Studies of light trapped in liquid crystal optical microcavities (LC MC) offer a possibility to observe a wide variety of quasiparticles and spin-related phenomena. In our work, we present analysis of experimental data and computational methods based on effective 2×2 Hamiltonians, that led to spotting in the system specific polarization patterns.

In this study, we investigate the microcavity filled with liquid crystal, which acts as tunable birefringent material. Different effective refractive indices for light polarized along and perpendicular to the LC anisotropy direction lead to splitting between the horizontally and vertically polarized cavity modes. Through applying external voltage to ITO electrodes across the microcavity, we were able to tilt the molecules around the chosen axis and, therefore, smoothly tune the energy of horizontal mode. For a certain value of applied voltage, we coupled two modes with orthogonal polarization and different parities, which resulted in an analog of the Rashba-Dresselhaus (RD) spin-orbit coupling [1]. We studied polarization of the cavity modes in reciprocal space at positive detuning with respect to RD resonance and observed the appearance of two points with



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Figure 1: Theoretical results for polarization of light in momentum space calculated on resonant states in outgoing wave boundary condition for RD regime in LC MC. Parameters S_1 and S_2 are denoted as arrows $\vec{S} = [S_1, S_2]^T$ and S_3 in a color scale.

purely circular polarization. Winding of the polarization states around those points in momentum space forms a pattern known as a meron [2,3].

Theoretical descriptions of these systems can be performed by well-known and highly time-consuming Berreman [4] and Schubert [5] methods, but in this work we use our simple theory, which allows us to describe these effects with satisfactory precision. Figure 1 shows the polarization structure in momentum space for the RD regime in LC MC obtained for the cavity with δ -mirrors in the outgoing wave boundary condition.

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Numerical Approach to Determine the Fourier Plane of Light Propagation Through Cartesian-Oval Shaped Microlenses

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Exciton-polaritons are quasiparticles emerging as a result of strong coupling between photons in optical microcavity and excitons confined in quantum wells. Exciton-polaritons, due to their bosonic nature, can undergo a non-equilibrium phase transition to the condensate state [1]. This process occurs as a result of the relaxation of excitons with high momentum to the minimum of the lower polariton branch. Unfortunately, the observation of the polariton relaxation processes is experimentally challenging due to the numerical aperture (NA) of the optical system that limits the observable range of in-plane momenta.

Our idea was to design a novel shape of oval microlenses with high numerical aperture that would allow for the detection of so far unavailable ranges of polariton in-plane momenta and the direct probing of the excitonic reservoir. Our idea is based on the recent realization of aspherical solid immersion microlenses that allowed to shape the light emitted form nanostructures into low-divergence beam [2]. Our design allow not only to extract high emission angles but additionally, the Fourier plane is formed into a flat surface that can be easily projected onto a planar surface of a CCD.

We considered microlenses with various sizes and shapes to optimize imaging of the reciprocal (Fourier) space. The simulations were based on geometrical optics. We started from setting three point sources of light within the cavity and propagated from them beams at a different emission angles. Fourier plane is plotted based on the positions where the beams were the closest to cross each other. Repeating the procedure for different emission angles, we could simulate the geometry of the full Fourier plane as



Figure 1: (a), (b) Fourier plane (red) determined for two different shapes of microlenses, (c) polariton dispersion relations obtained for the simulations of planar optical microcavity (green) and microcavity with printed microlens (orange) from (a).

shown in Figs. 1(a) and 1(b). We found out that Cartesian-oval shape of the microlens results in a plane which is approximately planar for a wide range of the emission angles [Fig. 1(a)]. In addition, we compared the typical dispersion relation of exciton-polaritons in planar microcavities [Fig. 1(c)] obtained by standard imaging optics composed of microscope objective of NA = 0.55 (green) with the dispersion relation that would be realized by our microlens (orange). We prove that our microlens allows to access much wider range of energies and in-plane momenta and enables the observation of excitonic reservoir.

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Localization of light on defects in microcavities with built-in uniform lying helix

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Uniform lying helix (ULH) is a type of cholesteric liquid crystal arrangement, where the helix axis is oriented parallel to the substrate surface. The rotation of liquid crystal molecules results in the appearance of a periodic structure of refractive indices in the form of sripes (see Fig. 1). Molecules in each stripe are aligned parallel to each other, In our study, we placed such a structure inside a 3 micron optical microcavity made of two distributed Bragg reflectors (DBR), fabricated by the method of vapor deposition out of six-pairs of SiO₂ and TiO₂ grown on transparent Indium tin oxide electrodes. The center of the stop band of the DBRs was 530 nm.

The periodic modulation of refractive index of the ULH is responsible for the formation of photonic bands, similar to bands formed in a 1D photonic crystal. The introduced bandgaps are of around 40 meV, depending on period of the stripes, which slightly varied across the sample.



Figure 1. A defected state of ULH. The pitch of the stripe in the middle of the spiral is smaller than the pitch of the surrounding ULH.

In the study we focused mainly on on certain imperfections in the ULH periodicity, where the helix wound up on itself (see Fig. 1.) forming an elongated spiral. The optical defect state inside of this spiral was found to have a different energy compared to the regular ULH around it. We performed statistical analysis to determine a correlation between the energy of the optical defect state and its size. It was found that if the pitch of the defect was smaller than the pitch of the regular ULH, the energy of the defect state lay inside the energy bandgap generated by the regular ULH. If the helix pitch inside the spiral was bigger, the defect's energy lied beneath the minimum energy of ULH state.

smaller than the pitch of the By applying the electric field to electrodes built-inside the cavity we found that the pitches, as well as energy states of the

optical defect and the regular ULH structure were tunable up to 3V of external voltage, above that level, the structure deteriorated into a nematic and periodic structure disappeared. The presented work introduces a new, interesting system, where the 1D photonic crystal can be controlled and tuned by external electric field.

Self-organizing photonic potential with tunable band structures coupled by spin-orbit interaction in birefringent optical cavity

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Photonic crystals with optical anisotropy possess degrees of freedom related to the polarization of light and specified directions allowed by dispersion. The coupling of photonic modes of different parity – which is possible due to birefringence – may introduce chirality [1]. If an additional periodic potential is present, the resulting photonic band gaps and eigenstates of light inherit the polazation properties of the whole system [2].

Here, we demonstrate a dye-doped optical microcavity with a built-in uniform lying helix (ULH) structure that was induced in a chiral nematic liquid crystal matrix. As a result of the presence of ULH, a one-dimensional self-organized periodic lattice of refractive index modulation was formed. We observed two separate photonic band structures with orthogonal linear polarizations of light (Figures a, b). The inclination of the helix with respect to the plane of the cavity additionally induced a Rashba-Dresselhaus spin-orbit interaction for light, which couples these two bands together. As a result of the spin-orbit coupling, new chiral states with distinct circular polarizations are formed (Figure c).

The proposed platform has several unique technological advantages. A self-assembled one-dimensional well-oriented photonic potential is formed over a macroscopic area. The period of ULH can be tuned by external electric field, thickness of the cavity and temperature. In addition, due to doping the structure with light emitters, the system can exhibit non-linear effects like the laser light emission from photonic bands, which opens new possibilities for the photonic control of light-emitting devices.



Figure 1 – Dispersion relation measured for the orthogonal linear (a) horizontal H and (b) vertical V polarization of the detected light. (c) The Stokes parameter S_3 shows circularly polarized states of light caused by Rashba-Dresselhaus spin-orbit coupling.

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Flow of an exciton-polariton condensate in optical microcavities with the natural structural disorder

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Exciton-polaritons are coherent quasiparticles that emerge from the strong interaction between cavity photons and quantum well excitons in semiconductor microcavities. The hybrid nature of these quasiparticles leads to strong nonlinear interparticle interactions and excellent transport properties. Recently, experimental studies have reported the observation of nontrivial dynamical effects in microcavities with a structural disorder, including relaxation oscillations and the generation of nonlinear waves [1]. The observed effects may be used in optical information processing in the future. Nevertheless, the precise control of polariton flow in a microcavity with inhomogeneous potential is still challenging.

Here, we investigate the formation of exciton-polariton condensate currents in a microcavity with a built-in structural disorder Fig. 1 a). We focus on forming polariton condensates under nonresonant excitation in an asymmetric potential trap. We demonstrate that the flow and counterflow of condensates inside a double quantum well can be precisely controlled by the intensity of laser sources acting on each potential minimum Fig. 1 b). This control is made possible by the optical anti-trapping potential created by an incoherent excitonic reservoir. Our theoretical model is based on the generalized open-dissipative Gross-Pitaevskii equation coupled with the exciton-polariton reservoir of excitons and hot carriers. Our analysis accurately reproduces the experimental results and provides insights into the fully optical, precise control of exciton-polariton networks. Our study can be helpful in the development of novel optoelectronic devices based on exciton-polaritons.



Fig. 1 – Optically controlled polariton condensate flow between two potential minima occurring due to natural structural disorder a). Emission intensity from simultaneously excited condensation sites when P_2 is constant b).

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Experimental Evaluation of the Indistinguishability of Single Photons at 1.55 µm Generated by InAs(P)/InP Quantum Dots

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Epitaxial quantum dots (QDs) have great potential as non-classical light sources in quantum technologies [1,2]. For advanced quantum cryptographic key exchange algorithms, single indistinguishable photons, generated on demand, are needed [3]. An appealing technology platform in this context are QDs emitting at 1.55 μ m, showing prospects for long-haul low-loss optical transmission in a standard silica fiber [4,5].

In this contribution we investigate quantum optical properties of single photons emitted by InAs(P)/InP QDs heterogeneously integrated with silicon. The QDs are grown by metalorganic vapor-phase epitaxy in the Stranski-Krastanow mode and exhibit low areal density $(3.1 \times 10^8 \text{ cm}^2)$ and emission at ~1.55 µm [6]. A metallic mirror beneath the QDs increases photon extraction efficiency to about 10% [7]. It enables imaging of the QD emission, letting localize a specific QD with ~50 nm accuracy, allowing for engineering of QD photonic environment.

In our study, we confirm single-photon emission from the QDs via measurements of the second-order autocorrelation function $g^{(2)}(\tau)$, revealing $g^{(2)}(0) = (3.2 \pm 0.6) \times 10^{-3}$ under quasi-resonant pulsed excitation. To investigate the degree of indistinguishability of emitted photons and gain insight into underlying dephasing mechanisms, Hong-Ou-Mandel-type two-photon interference (TPI) experiments are conducted as a function of the temporal delay δt between consecutively emitted photons. The TPI visibility, quantifying the degree of indistinguishability, was comparatively analyzed using both, integration of the raw experimental data and applying a fitting model, revealing the raw (i.e. 'as measured') and the post-selected TPI visibility. For $\delta t = 4$ ns, we determine the visibility by evaluating the coincidences around $\tau = 0$ for co- and cross-polarized photons and compare the result with the method based on evaluating the areas of adjacent coincidence maxima in a characteristic five-peak pattern [8]. We obtain raw (post-selected) TPI visibilities of up to 20% (99%) and comparatively discuss the experimental uncertainties.

In summary, our work provides important insights and shows progress in generating telecom C-band single indistinguishable photons for applications in quantum information technologies.

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Magneto-Optical Properties of Single Symmetric InAs/InP Quantum Dots Emitting in the Telecom C-band

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Self-assembled epitaxial semiconductor quantum dots (QDs) have been the subject of study due to both – their interesting fundamental properties and potential practical applications. Discrete atom-like energy structure makes them ideal as an active region for non-classical light sources, e.g., single photon sources or sources of polarization entangled photon pairs [1, 2]. The latter require high in-plane symmetry of the nanostructure and/or external tuning to minimize the exciton fine structure splitting. Such sources can be utilized in quantum communication, e.g., to exchange cryptographic key or as building blocks of quantum networks, in particular, quantum repeater architectures. In this case, emission in the telecommunication windows, compatible with existing optical fiber infrastructure, is needed. QDs have also potential in spintronic applications, where individual spins of electrons and holes confined in the dots are used to store information (quantum memory) or as an active part of switching devices depending on the carrier g-factor. In this case, spin coherence properties are of interest.

Magneto-optical polarization-resolved measurements in Faraday and Voigt configurations enable to understand the nature of excitonic complexes in QDs. In particular, extension of the wave function can be deduced from the diamagnetic coefficient, g-factor of carriers/excitonic complexes can be determined based on the Zeeman splitting allowing also for observation of the fine structure of excitonic complexes and identification of emission lines in the single QD spectrum as originating from certain carrier configurations [3]. Finally, spin coherence time can be determined based on depolarization (Hanle) curve obtained with quasi-resonant excitation with circularly polarized laser beam [4].

In this study, we determine experimentally the fundamental magneto-optical properties of the new generation of symmetric, and low-density $(2 \times 10^9 \text{ cm}^{-2})$ InAs/InP QDs grown by ripening process-assisted molecular beam epitaxy [5]. These QDs fulfill the requirement of telecom spectral range emission (1.55 µm). Photon extraction efficiency was enhanced by growing the QDs on top of a distributed Bragg reflector, resulting in 6.8% photon extraction efficiency for planar sample [6] and 13.3% for QDs in cylindrical mesas [7]. High single-photon purity (g⁽²⁾(0) < 0.01) [8] and low fine structure splitting makes them promising candidates for generating non-classical light.

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Optimization of on-chip photonic structures for a hybrid InP/Si platform via deep learning approach

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The range of desired on-chip photonic components performing particular tasks in integrated circuits based on a single material platform like silicon-on-insulator is broad and often infeasible when it contains a well-defined quantum emitter. However, a heterogeneous integration with other platforms offers to merge both utilities, those which are well known in silicon platform and those which offer efficient light sources at telecom wavelengths. We aim to study numerically the photonic structures realized in InP/Si combined systems such as a ridge waveguide nanobeam cavity, ring resonator, directional coupler, and multimode interference (MMI) beam splitter as vital components for future quantum-integrated photonic chips. We used finite-difference time-domain (FDTD) commercial software to study the photonic device's performance in a broad spectral range within a single-shot simulation. First, we study a ring resonator which can be used as an interference filter for the propagation along the waveguide resulting in a free spectral range of ~50 nm and finesse of 20. Such high performance is found using a specific taper system near the ring structure. Next, using a grating structure design of nanoholes along an InP waveguide, we designed various systems that can enhance the dipole emission by a Purcell factor of 25 and 75 % of bidirectional transmission using only 7 periods of the grating design. It also allows a directional enhanced emission for asymmetric configuration of 10/5 periods resulting in a Purcell factor of 14 and 90 % of transmission within a 5-nm-wide spectral window. Then, we optimized the directional coupler and MMI beam splitter to get the highest transmission and a splitting ratio close to 50 %. Both geometries require specific tapering along the propagation direction for efficient light transfer.

Last, we realized an optimization algorithm for a 1D nanobeam cavity based on deep learning (neural network) [1,2,3] to achieve the optimized transmission along the waveguide from a single dipole source at 1.55 μ m photon wavelength. In this step, we used inversed, fully-connected network with 3 hidden layers, which can learn from our 3D FDTD results. The trained network can predict the exact geometry of a photonic system based on spectral data, which serves as an input set of parameters to maximize the device's performance.

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Optical Characteristics of Monolithic High-Contrast Gratings for Quantum-Dot Vertical-Cavity Surface-Emitting Lasers at 940 nm.

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Vertical-cavity surface-emitting lasers (VCSELs) have a wide range of unique properties that distinguish them from conventional semiconductor lasers [1]. However, they still face some challenges, for instance, they require innovations in reduction of the thickness of the cavity mirrors which are typically distributed Bragg reflectors (DBRs). Monolithic high-index contrast gratings (MHCGs) are a very attractive alternative for DBRs as well as externally-fabricated high-contrast gratings (HCGs). MHCGs has been already proven as efficient mirrors for VCSELs [2-5]. They bring unprecedented simplification of the device designs and offer flexibility in the material choice, opening new prospects for realization of VCSELs emitting at wavelengths hardly achievable in the past. MHCGs are able to reflect light, as well as can classical subwavelength HCGs, but without the requirement to be sandwiched between low-refractive-index layers [6].

In contrast to standard DBRs, the MHCG enables resonant wavelength setting of every laser in the on-chip array separately, by design of MHCG geometrical parameters. As a result, the MHCG VCSELs array chip can be fabricated in a cost-efficient way on a single wafer, reducing greatly the costs of production and device footprint.

In this work, we characterize optically MHCGs with different geometries by performing microreflectance measurements. Gratings design has been numerically optimized for high reflectivity (> 90%) in the 930-955 nm spectral range required for VCSELs in the optical water vapour detection systems. To achieve this, the grating period, filling factor and trench depth were varied. Three sets of gratings differing in etching depth have been fabricated by means of electron-beam lithography on a double-polished GaAs substrate. Within each set filling factor and trench width was varied to cover theoretically predicted high reflectivity area. The experimental characteristics are compared with numerical results to provide feedback for both fabrication and modelling. Obtained results are an important step in development of QD VCSELs with MHCGs as a top mirror.

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Methods for Determining High Finesse in Open Cavity Systems

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Open microcavities enhance the interaction of light and quantum emitters, which is used to obtain bright sources of indistinguishable on-demand single photons [1] and is expected to enable advanced measurements of emmitters' coherent and noncoherent dynamics [2]. Microcavities with low internal optical losses, characterized by several thousand and higher finesse factors, are of particular interest. Measuring such high finesse values is an experimental challenge that can be approached in several ways.

Traditionally, finesse is defined as $F = FSR/\delta v$ [3], where FSR (Free Spectral Range) is the frequency spacing to the next longitudinal mode, and δv is the resonance linewidth. However, this definition is hard to use for high-finesse microcavities based on distributed Bragg reflectors (DBRs) as the FSR can be comparable to, or even larger, than the stopband of the DBR structure. Additionally, measuring the resonance linewidth requires very high mechanical stability of the cavity.

To overcome the challenges mentioned above, a dynamic setup[4] can be used in which the length of the cavity changes periodically in time, minimizing the influence of acoustic noise. In addition, this allows finesse measurements to be performed using single or multiple light wavelengths, and resonances are observed as the length of the cavity changes.

In this work, we applied several methods for determining high finesse values in a planar-concave open cavity system. We discuss the advantages and limitations of each method and compare the results.

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Reflectivity spectra of GaAs/AlAs Distributed Bragg Reflector in the MIR Calculated by the Transfer Matrix Method

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Photonic structures play an important role in tailoring the emission properties according to the requirements of a specific application. There are applications demanding vertically emitting lasers operating in mid-infrared (MIR), as e.g., some of the optical gas sensing schemes. In that context distributed Bragg reflectors (DBRs) are typically used. The DBRs constitute of many pairs of alternating layers of two high-refractive-index contrast dielectric/semiconductor materials, which in epitaxial structures need to be lattice matched to the substrate to avoid strain accumulation. For this purpose it has been proposed a standard and technologically mature material - GaAs/AlAs. It is technologically demanding as the DBR need to be much thicker than for the shorter wavelengths resulting in longer growth process during which precise control of layer thicknesses needs to be maintained. In this case, only the thickness of each layer needs to be up-scaled for the longer wavelengths.

Our work aims at calculating reflectivity spectra of DBR structures designed for the MIR range by employing the transfer matrix method. The structure contains GaAs/AlAs DBR. The transfer matrix method is based on considering the amplitudes of electric field during the propagation of an electromagnetic wave through the structure and includes propagation within a material of a given refractive index and thickness, as well as reflection and transmission at each of the interfaces. By using the common notation, the light propagation in the whole structure can be represented by a 2D array called a transfer matrix [1-4]. The method was implemented in Python and used to calculate the reflectivity spectra of the DBR. Through such simulation, we are able to match the values of thicknesses of layers and types of materials of such a structure before fabrication in order to optimize its design. This is a crucial contribution to the self-consistent design-fabrication-measurement of optical properties loop and allows for broad range parameter study to understand the sensitivity of the reflectivity spectrum on the structural parameters.

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Optimization of the polarization state of an ensemble of quantum dots for chiral emission

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Controlling the interaction of light and matter is the basis for many applications ranging from light technology to quantum information processing, many of which are based on nanophotonic structures. The confinement of light in such nanostructures imposes an inherent link between its local polarization and its propagation direction, known as spin-momentum locking of light [1]. This modifies the elementary processes of light-matter interaction: the emission and absorption of light of an appropriately polarized emitter becomes chiral, i.e., propagation direction-dependent. Placing a solid-state quantum emitter in a photonic crystal waveguide is promising for future applications because a very high fraction of the generated light can be injected into the waveguide guided mode [2]. Here, we focus on maximizing the directionality of chiral light emission from a quantum dots (QDs) placed within half core region of the photonic waveguide, as quantified by the fraction D of light emitted in the desired direction.

Most research so far has focused on placing the single emitter with circularly polarized transitions at a point of perfect circular polarization (a C point). These points are scarce. The photonic crystal structures support circular polarization at a few accessible locations, resulting in small area with high directional contrast D_{90} (top left panel in Fig. 1). That would require a precise embeding of QD in a waveguiding structure. To avoid this deposition requirement we consider ensemble of quantum dots and calculete net directionality. We find an optimal elliptical polarization of the QD transition dipole that maximizes the net directionality of the whole planar QD ensemble. We also calculate the Purcell factor and determine the area where |D| > 0.9 to explore the advantage of optimized elliptical polarization with respect to circular one. It is noticeable that optimizing the elliptical polarization state broadens the area of the waveguide core that may be used for unidirectional coupling with a single



Figure 1: The Purcell enhancement factor (right) and area where $|D| \ge 0.9$ (left) for an optimized elliptical polarization state (bottom) compared to the circular one (top) as a reference.

quantum emitter, while maintaining a high Purcell factor. We also perform $\mathbf{k} \cdot \mathbf{p}$ calculations to find geometrical parameters of the QD leading to the optimized polarization of the emission. This proves that optimal polarization is feasible if a small magnetic field of 0.2 T is applied.

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TuPB13

Controlled coherent coupling in a quantum dot molecule measured with nonlinear spectroscopy

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Semiconductor quantum dot molecules are considered as promising candidates for quantum technological applications due to their wide tunability of optical properties and coverage of different energy scales associated with charge and spin physics. While previous works have studied the tunnel-coupling of the different excitonic charge complexes shared by the two quantum dots by conventional optical spectroscopy, we here report on the first demonstration of a coherently controlled inter-dot tunnel-coupling focusing on the quantum coherence of the optically active trion transitions. We employ ultrafast four-wave mixing spectroscopy to resonantly generate a quantum coherence in one trion complex, transfer it to and probe it in another trion configuration. With the help of theoretical modelling on different levels of complexity we give an instructive explanation of the underlying coupling mechanism and dynamical processes.

Diffusion of quantum well excitons measured with nonlinear spectroscopy

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Optical spectra of semiconductors display narrow resonances below the absorption edge, which are attributed to the excitons, i.e. pairs of electrons and holes, bound by the Coulomb interaction. We here perform coherent nonlinear spectroscopy of excitons in epitaxially grown CdTe-based quantum wells. By performing heterodyne four-wave mixing (FWM) microscopy, we find that the exciton absorption is dominated by homogeneous broadening on areas exceeding a few hundred microns squared. The photon-echo sequence of the FWM reveals no traces of the inhomogeneous broadening, indicating that excitons experience no localization due to the microscopic potential fluctuations. Under such conditions, the excitons with a large in-plane momentum of their center of mass (out of the light cone), can freely roam across the quantum well. After reaching mesoscopic distances, within $10\,\mu m$ range, they can relax back to the radiative cone and recombine radiatively. In these novel experiments, we take advantage of the microscopic configuration of the FWM to probe the diffusion of the exciton density and coherence. This is achieved by measuring FWM as function of the spatial separation between pumps, inducing density or coherence of the excitons, respectively, and the probe converting them into the FWM. An extra care is taken to distinguish between FWM originating from diffusive excitons, i.e. those traveling between the pump and the probe, and the FWM signal created directly by the scattered light impinging the probe. This proof-of-principle experiment demonstrates that heterodyne FWM can be employed to infer propagative effects of excitons in semiconductor nanostructures.



Figure 1: a) A geometry of the spatially-resolved FWM experiment, FWM generated at the probe position E_3 , is detected via heterodyne spectral interference with the reference Ref. b) build up of the exiton diffusion with increasing the delay τ_{23} , c) signatures of the diffusive excitons in the delay dependence: with increasing the pump-probe the maximum of the FWM shifts to larger delays τ_{23}

Coherent imaging and dynamics of exciton complexes in MoSe₂ monolayers epitaxially grown on a hexagonal boron nitride

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Technology of preparing heterostructures made of semiconducting transition metal dichalcogenides (TMDs) is based on exfoliation of thin films from van der Waals bulk crystals. While the nonscalability of the exfoliation top-down approach is not an issue in the fundamental research, for which the proof-of-principle demonstrations are essential, it is a major roadblock on the academia-industry pathway of this field. In order to merge these novel materials with the semiconductor microelectronics, strainfree monolayer samples homogeneously covering wafers of a few inch diameter are required.

Here we present studies of MoSe₂ grown by molecular beam epitaxy on the silicon substrate with exfoliated hBN flakes. We performed four-wave-mixing (FWM) imaging and spatially correlated the obtained amplitude of the nonlinear optical response of the studied flakes with the layer thickness obtained from AFM measurements as shown in the fig. 1. It allowed us to precisely characterise the signal originating from the epitaxiallygrown monolayers with respect to the surface morphology. Furthermore, by measuring the FWM signal in the temporal domain we were able to determine the dephasing dynamics of exciton complexes and ascertained its temperature dependence.



Figure 1: Epitaxial $MoSe_2$ monolayer grown on hBN flake at the SiO_2 on the silicon substrate. FWM imaging in comparison to AFM amplitude.

Our results show that these epitaxial monolayers, while opening the prospect of being compatible with the semiconductor optoelectronics industry, display excellent optical response, providing they cristalize on atomically flat surfaces, here provided by hBN flakes. The quality and intensity of the produced signal is comparable with their non-encapsulated counterparts obtained via exfoliation. As such, our findings fortify the viability of the MBE growth method for the production of high quality TMD monolayers. Furthermore, its inherent versatility opens up alluring new research venues for these particular family of 2D materials.

TuPB15

Probing the Local Temperature Distribution in Electrically Pumped Broad-Area VCSELs

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Bose-Einstein condensation of photons in an optical microcavity filled with rhodamine 6G has been reported [1]. The following recent study showed that a photon gas in a state of thermal equilibrium with its surroundings trapped in semiconductor optical microcavity can display signatures of Bose-Einstein condensation [2]. However, the effects of ununiform local temperature on Bose-Einstein distribution in semiconductor lasers have not been examined yet.

The emission spectrum of semiconductor electrically pumped vertical-cavity surfaceemitting lasers (VCSELs) is directly influenced by the local temperature and current density distribution. The effective inhomogeneous confining potential is the reflection of the effective local width changes in the cavity. This inhomogeneity effect changes the density of states function and influences extracted values of the Bose-Einstein distribution. The assumption that the density of states is purely 2D and has a fixed value over the whole VCSEL resonator may be an insufficient approximation. The most commonly used measurement method to characterize the thermally induced shift in the lasing wavelength provides only information on the average temperature distribution across the entire microcavity surface [3]. Here, we present a novel technique for measuring local spectral temperature distribution in wide aperture oxideconfined VCSELs by locally fitting the Boltzmann distribution to the high-energy tail of the spontaneous emission spectrum. Furthermore, we characterized the local potential change of the microresonator, measuring the local fundamental mode energy by filtering the spontaneous emission in the wavevector space close to the normal incidence $k_{\parallel} \approx 0$. The resulting data can be compared to theoretical simulations of the influence of current and temperature on the local potential shape.

Studies on the thermalised boson gas of trapped photons in electrically pumped VCSEL microresonators is a new promising direction in Bose-Einstein room-temperature condensation. Therefore, understanding the details of wide aperture semiconductor VCSELs may provide devices that act as Bose-Einstein condensate of photons, allowing for new research on Bose-Einstein photon condensation physics.

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Modeling InAs/GaSb Type-II Superlattices for Mid-Wavelength Infrared Photodetectors with the nextnano++ Software

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Type-II superlattices (T2SLs) are characterized by: i) flexible tunability of the energy band gap by adjusting thicknesses of consisted layers without changing their content, ii) carrier separation suppressing nonradiative Auge recombination, and iii) high effective masses reducing band-to-band tunnelling current [1]. In contrast to type-I superlattices where effective mass approximation often is applied, modeling T2SL may cause additional difficulties originating in strong hybridization of S and P orbitals. Such conditions occur when T2SLs consist of thick layers resulting in small energy gap or even energies of some electron states lower than energy of first hole state, within the picture of effective mass approximation. As both electrons and holes are described in such cases by mutually overlapping wavefunctions, additional interaction is inevitable and hybridization effects must be considered. Eight-band k·p model can be applied for this purpose [2].

In this poster, I present a simple analysis of InAs/GaSb T2SL using the nextnano++ software [3-5]. The InAs/GaSb T2SLs are examples of promising heterostructures with potential to outperform currently state-of-the-art mercury cadmium telluride in applications for mid-wavelength infrared photodetectors [1]. I show how to effectively design these super lattices and how to obtain information about electronic band structures, relevant optical transitions, wavefunction envelopes, and hybridization effects. Presented approach can be easily applied and followed for basic analysis of any superlattices, that might come very helpful for groups focusing on optical spectroscopy or growth of numbers of such structures.



Figure 1: Basic design and analysis process of T2SL with the nextnano++ software

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Applications of InAs/InAs_{0.625}Sb_{0.375} Superlattice in Far-infrared Detection

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ABSTRACT

Using first-principles calculations, we calculate the electronic and optical properties of InAs/InAs_{0.625}Sb_{0.375} superlattices (SL). To accurately approximate the electronic and optical properties, the modified Becke-Johnson exchange-correlation functional is pondered. After analyzing the electronic and optical characteristics of bulk InAs and InSb, we then investigated InAs/InAs_{0.625}Sb_{0.375} SL. The electronic and optical properties of the InAs/InAs_{0.625}Sb_{0.375} SL are studied with three lattice constants of the bulk InAs, GaSb and AlSb, respectively. It is observed that the electronic and optical properties strongly depend on the lattice constant. A considerable decrease in the energy gaps and the effective masses of the heavy-holes in the k_x - k_y plane compared to the bulk phases of the parent compounds can be observed. We demonstrate that the electrons are s-orbitals delocalized in the entire superlattice, while the holes have mainly 5p-Sb character localized in the In(As,Sb) side of the superlattice. In the superlattice, the low-frequency absorption spectra greatly increase when the electric field is polarized orthogonal to the growth axis allowing the applicability of III-V compounds for the long-wavelength infrared detectors.

Magnetic properties of TRS broken ferrovalley semiconductors

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ABSTRACT

The intrinsic ferromagnetism in 2D materials and the magnetic anisotropy energy (MAE) are importants yardstick for nanoscale applications. We employed first-principles scheme to investigate the electronic band structures, the strain dependence of MAE in pristine VSi₂Z₄ (Z=P, As) and its Janus phase VSiGeP₂As₂ and the evolution of the topology as a function of the Coulomb interaction. It is observed that MoSi₂P₄ monolayer show equal bandgaps at K/K' points of brillouine zone due to the the preserved time-reversal symmetery (TRS). On the other hand, all the vanadate based 2D structures exhibit unequal bandgaps at K/K' with ferromagnetic ground state ordering owing to broken TRS, which is known as ferrovalley semiconductors. A large value of coupling J is obtained, and this, together with the magnetocrystalline anisotropy can produce a large critical temperature. We found an out-of-plane (in-plane) magnetization for VSi₂P₄ (VSi₂As₄), while in-plane magnetization in VSiGeP₂As₂. Furthermore, we observed a correlation-driven topological transition in the Janus VSiGeP₂As₂. These ferrovalley semiconductors possess inherent spontaneous valley polarization induced by intrinsic ferromagnetism and, thus offer the possibility to address the challenges of valleytronic materials.

Bulk And Contact Low Frequency Noise Investigated With Transmission-Line-Model: InAs case

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The Transmission-Line-Model (TLM) is a well-known technique that can be used for obtaining parameters of the material or device under consideration, i.e., the sheet resistance, the metal-semiconductor contact parameters, e.g. contact resistance, transfer length, etc. The method relies on resistance measurements between pairs of contacts separated by different distances *L*. This work reports experimental studies on TLM samples made from n-type InAs doped with Si $(N_d=2\times10^{16} \text{ cm}^{-3}, N_d=2\times10^{17} \text{ cm}^{-3}, \text{ and } N_d=2\times10^{18} \text{ cm}^{-3})$ and with various metallization (Au, Ti/Au, Ti/Pt/Au) and contact formation procedure (annealing).

Besides the standard characteristics measured for a TLM, the low frequency noise was also measured. The preliminary results are shown in Fig. 1, where the resistance to TLM (inset) and the normalized low frequency noise $S_U (f=1\text{Hz})/U^2$ are shown as functions of the spacing of the contacts L. From this characteristic, the origin of observed low frequency noise can be identified by comparing characteristics' slope with a theoretical slope for two different models that assume bulk or contact origin of low frequency noise.

The scientific goal of the work is to investigate the influence of InAs doping, contact metallization, and their formation (annealing) on the total noise and separate bulk and contact noise contributions.



Fig. 1. Resistance and relative low frequency noise as a function of the distance between pairs of contacts for n-type InAs, doped with $N_d=2\times10^{16}$ cm⁻³, transmission-line-model sample with Au contacts.

Optical characterization and pump-probe measurements of InAs/GaSb and InAs/InAsSb type-II superlattices

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Modern optical gas detection systems utilize the technique of tunable diode laser absorption spectroscopy (TDLAS) for different applications in science, manufacture, or medicine. [1] Through tuning the emission wavelength, the presence and concentration of trace gases are possible to detect in real-time with their limit in single ppb range. Optical gas detection systems employing mid-infrared spectral range often comprise of active structures grown within so called 6.1 Å family of semiconductors, i.e. InAs, GaSb, AlSb alloys. Superlattice structures within this material system with type-II alignment have the potential to exceed the working parameters of widely used optoelectronic devices, such as HgCdTe infrared photodetector. [2]

Optically active type-II superlattices were obtained through molecular beam epitaxy and then characterized utilizing Fourier-transform infrared spectrometer (FTIR) and pump-probe system. The measurements were performed on two sets of structures – type-II InAs/GaSb superlattices and type-II Ga-free InAs/InAsSb superlattices. The parameter varying the samples within the first set was the approach to interface engineering – the samples were either "Sb-soaked" or an InSb monolayer was inserted between each SL period [3] which was also considered in our previous works. [4] Samples within the second set, on the other hand, varied through SL periodicity of 5, 6 and 8 nm which allowed to shift the main transition from 5 to 9 μm at room temperature. [5]

Photoluminescence spectra were obtained for all samples in 10 to 300K temperature range and then complemented with photoreflectance measurements for characteristic temperatures to increase the sensitivity of the measurement for less optically active transitions. In addition, pump-probe measurements were performed to investigate the dynamics of carrier relaxation and recombination processes in proximity of transition energies observed in previous experiments.

Presented findings could provide valid insights in the fundamental properties such as carrier lifetimes within InAs/GaSb and InAs/InAsSb SLs to further optimize and improve the working parameters of possible future applications in the field of infrared optoelectronics.

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Modelling of Band Structure and Optical Properties of InAsP/InP Quantum Dots in Nanowires

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Chemical beam epitaxy (CBE) is known as efficient technological approach to grow quantum dots (QDs) in nanowires (NWs). In such a case the lateral size of the dot is limited by the width of the NW following the size of the seeding Au droplet, whereas the height of the nanostructure is controlled by the amount of deposited material. Compared to more common epitaxial self-assembled QDs embedded in a semiconductor matrix, CBE-grown QDs in NWs offer greater engineering capabilities for tailoring the energy levels and optical properties to meet requirements of specific application [1]. In addition, NWs geometry limits emission into the free-space modes in-plane and guide the emission in the vertical direction, which translates into increased light extraction efficiency resulting in brighter source with larger photon flux [2]. This is particularly important for nonclassical light sources, such as single photon emitters. However, a kind of drawback of this growth method is a deposition of a thin layer of QD material onto the bare substrate, i.e. in the areas beyond the NWs. This thin layer is expected to emit light which can overlap spectrally with emission from the QDs for certain thicknesses and compositions. Additionally, carriers could be trapped in it affecting overall carrier dynamics - both relaxation to the QDs and their emission. Therefore, it is crucial to better understand energy states of such a layer and its effect on QD emission.

The main objective of conducted research is to study theoretically the energy band structure of InAs_xP_{1-x} QDs in InP NWs grown by CBE technique on InP substrate with different crystallographic orientations (1,0,0) and (1,1,1), which results in crystallization of the nanostructures in zinc blend or wurtzite structure, respectively. In particular, it is of interest if the thin $InAs_xP_{1-x}$ layer formed naturally during CBE growth of QDs, overlaps spectrally with QDs emission and what is the electronic structure of the system and its optical response. To calculate these properties versus various nanostructure geometries a commercial software Nextnano++ [3] is employed, i.e. an implementation of the 8-band k^p in three dimensions. Simulations are performed for realistic compositions and geometries following the structural data for several sets of samples: different QD heights and As content. As a result, the confined levels and probabilities of optical transitions are derived (without the effect) compared experimental excitonic and with results obtained from microphotoluminescence measurements.

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Temperature Dependence of Refractive Indices of In_{0.53}Ga_{0.47}As and InP in the Mid-Infrared Spectra Range Determined by Fourier Transform Spectroscopy

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The main motivation behind the determination of refractive indices of InP and InGaAs in mid-infrared spectral range is to enable more accurate and effective analysis and design of materials and optical components for a variety of applications e. g. growth of active regions for mid-infrared sources of light such as quantum cascade lasers (QCLs) and detectors. QCLs are semiconductor lasers that emit mid-infrared light, and have a wide range of applications in chemical sensing, medical diagnostics, and environmental monitoring [1]. InP substrates provide a suitable platform for the epitaxial growth of QCLs with high output power, high efficiency, and good thermal stability [2]. Moreover, it was shown, that InP and InGaAs can be used to grow efficient Distributed Bragg Reflector (DBR) designed for near-infrared spectral range [3]. By controlling the reflectivity and transmission of infrared light, DBRs can help improve the performance and functionality of infrared devices. In our case we will show that functionality of mentioned concept InGaAs/InP DBR can be further exploited for longer wavelengths i.e. 4 μ m.

Refractive index of InP was determined by transmittance measurements of double sided polished undoped substrate in wide spectral range performed for temperatures ranging from 10 K to 300 K and through the analysis of measured interference pattern [4]. The result in near-infrared spectral range was verified by common spectroscopic method for measuring refractive index - ellipsometry, which measures the polarization of light reflected from a sample at various angles of incidence and allow us to adjust the curve for longer wavelength with high accuracy [5]. This knowledge was used to determine the refractive index of In_{0.53}Ga_{0.47}As in the same wide range of temperatures by modelling the experimentally obtained spectra of reflectance measured on In_{0.53}Ga_{0.47}As/InP DBR with Lumerical software. The modelling of experimental spectra for 300 K required adjusting the nominal values for layer thicknesses of about 2% which finally made the refractive indices of In_{0.53}Ga_{0.47}As only unknown parameter to generate DBR reflectance spectrum for lower temperatures. This approach provides an accurate and non-destructive means of refractive index determination for semiconductors in wide range of temperatures and broad mid-infrared spectral range.

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Carrier dynamics of type-II quantum wells emitting in mid-infrared

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Interband cascade lasers (ICLs) are a type of semiconductor lasers that operate in the mid-infrared region of the electromagnetic spectrum. They have unique properties that make them essential for a variety of applications, such as gas sensing, spectroscopy, medical diagnostics, and communication [1]. ICLs have a cascade structure of multiple quantum wells, which allows them to achieve high efficiency, low threshold current, and continuous-wave operation at room temperature [2,3]. Nevertheless, further optimization of the active region can be achieved. In this approach we will examine influence of tensile and compressive strain on the electronic structure [4] and carrier dynamics.

This study presents transient absorption (TA) pump-probe measurements of type-II WQW InAs/GaInSb/InAs and InAsSb/GaAsSb/InAsSb structures designed for mid-infrared interband cascade lasers emitting close to $3.5 \ \mu m$. The aim of this study is to investigate the fundamental carrier dynamics in this material systems, which is essential for their further possible applications in e.g. mode-lock lasers. TA measurements were performed in order to determine influence of the compressive or tensile strain application in quantum wells on carrier lifetimes regarding fundamental transitions and other interface-related ones.

Additionally, photoreflectance (PR) and photoluminescence (PL) measurements were used to investigate the band structure and optical properties of the samples. Both the PR and PL spectra showed strong signals near $3.3 \mu m$, which corresponds to the fundamental type-II transition of the QW and additional signals (observed in PL) related to possible interface-atom intermixing processes. The temperature-dependent PL measurements revealed a strong temperature dependence of the PL intensity, with a significant reduction in the intensity at higher temperatures. The energy shift of the transitions were also studied through temperature-dependent PR. Observed optical transitions and their energies were in good agreement with theoretical calculations realized within 8kp formalism.

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Influence of Mg²⁺ codoping on excitation relaxation and emission properties of (Lu,Gd)₃(Ga,Al)₅O₁₂:Ce,Mg scintillators

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Scintillation materials have currently a very wide application field from medical imaging and high energy physics experiments to the homeland security systems and nuclear safety monitoring. The successful growth of Ce-doped gadolinium aluminum gallium garnet (GAGG) single crystals that are prospective for the detection of ionizing radiation due to their high density, high light yield, and the absence of intrinsic radioactivity encouraged an intensive development of mixed garnet-type scintillators for efficient and fast scintillation detectors. The codoping by small amounts of divalent alkali-earth ions was proved to be a successful method to achieve faster scintillation, whereas the influence of codoping on the emission properties at the codoping levels comparable to that of the activator ions still remains a subject for investigation [1].

In this work, we focus on Ce-doped and Mg-codoped $(Lu,Gd)_3(Ga,Al)_5O_{12}$:Ce,Mg (LuGGAG) multicomponent single crystalline epitaxial layers grown using liquid phase epitaxy (LPE) method. LPE technique has an advantage in the preparation of Ce-doped garnets at lower temperatures allowing noticeable reduction of the concentration of antisite and vacancy-related defects acting as effective traps for electrons. We report our results of a study of electronic excitation relaxation and emission properties in a set of Ce-doped LuGAGG films containing different content of Mg^{2+} ions (from 0 to 2200 ppm). The linear and transient absorption, steady-state and time-resolved photoluminescence spectroscopy have been used to study the influence of Mg^{2+} ions on excitation relaxation and emission properties of Ce³⁺ activator ions in the 80-600K temperature range.

The dominant excitation relaxation pathways were examined exploiting selective photoexcitation of the first and second excited levels of Ce^{3+} ions. The study of photoluminescence temperature dependence under selective excitation of the lowest excited Ce^{3+} level 5d₁ revealed emission thermal quenching due to increasing thermal escape of electrons to the conduction band and subsequent nonradiative recombination. The slight increase of photoluminescence intensity from low-to-room temperature after the selective excitation of 5d₂ level indicate the localization of electrons during their transfer to emitting 5d₁ level due to bandgap fluctuations caused by compositional disorder of mixed garnet crystal matrix. The additional decrease of photoluminescence intensity and increased emission decay rate was found in the LuGAGG films containing more than 300 ppm of Mg. The later observation is in line with the formation of nonradiative activator-codopant related excitation relaxation route [2].

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Characterization of the Microwave Strip Antenna in the Experiment of Optically Detected Magnetic Resonance

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Optically Detected Magnetic Resonance (ODMR) is a powerful technique that allows for local and selective sampling of magnetic properties. The intensity of the measured ODMR signal crucially depends on how effectively microwave radiation is provided to the sample.

Typically, the magnetic resonance signal, e.g., in Electron Paramagnetic Resonance (EPR), is measured with a single specific frequency of the RF field. The single-frequency restriction is imposed by the resonant cavity employed to amplify the strength of the RF field. In order to perform measurements in a wide range of frequencies, a totally different approach is needed. Our research aimed to characterize a new type of broadband microwave antenna and its role as an element of the experimental system.

The microwave antenna is designed to act as a long line: a transmission line, which length is comparable to the wavelength of the propagating microwaves. The transmission lines are the metalized tracks on the dielectric support plates that encircle the sample. The antenna's construction allows it to work in low temperatures, such as 1.5 K, and with a wide range of microwave frequencies. The actual antenna is shown in the picture below.



In this research, the signal was measured on the reference sample – a quantum well (Cd, Mn)Te with manganese content of 0.3%. The characterization was made based on measurements of various microwave radiation power and various environmental conditions, such as the temperature and pressure of helium surrounding the sample, affecting its dielectric constant. Obtained results are compared with the results of numerical simulations.

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Hybrid Acousto-Optic Driving of Rabi Oscillations of Electron Spin in a Quantum Dot

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Performing coherent Rabi rotations is a necessary step toward complete control of a qubit. For a quantum bit based on spin states, it can be achieved with microwaves [1]. However, such control with macroscopically long waves causes problems when considering the miniaturization of integrated circuits. While for an electron confined in a semiconductor quantum dot (QD), state control has been achieved with optical pulses [2], such methods are unavailable for many other promising solid-state systems (like NV centers or superconducting qubits). A promising solution is to use acoustic waves that couple to all mentioned systems and allow chip miniaturization.

In this contribution, we propose a hybrid (acousto-optic) approach to controlling the spin state of an electron in a QD. We theoretically show Rabi oscillations for the Zeeman-split ground state of a QD-confined electron modulated using the acoustic field. The acoustic field amplitude sets the rotation axis, while spin rotations are controlled by pulsed off-resonant optical coupling to the trion state. Thus, we take advantage of orders of magnitude shorter lengths of acoustic fields compared to microwaves and rotate the spin with two nanoscale-compatible fields.

Although direct coupling between spin states and deformations is negligible in such systems, our hybrid control leads to controllable Rabi oscillations with a frequency that depends on the amplitudes of both acoustic and optical fields. The ground state doublet and the trion level form a configuration known as a Λ -system [3]. In our model, the acoustic field modulates the trion level, to which both spin states are coupled via optical pulses. This leads to the second-order coupling between the spin states.

To better understand the evolution of the system, we compare direct numerical calculations with the results of approximate analytical derivation. To this end, we model the system using a Hamiltonian in the dipole and rotating wave approximation. We derive an effective Hamiltonian of the ground state doublet by eliminating the trion level in the quasidegenerate perturbation theory [4]. This effective description allows us to predict the evolution of the spin qubit. Importantly, due to the specific form of effective coupling, we find a way of varying the spin-rotation axis. Lastly, we study the destructive processes accompanying the evolution of the system. For this, we use the Lindblad master equation and show that trion recombination has negligible effects for short optical pulses.

We anticipate our work to be a starting point for designing and creating a controllable qubit and its extension to a two-qubit gate defined on tunnel-coupled QDs.

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Noise effects on the resonance fluorescence from acoustically modulated quantum dots

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In the field of quantum information processing, one of the current goals is the high scale integration of miniaturized devices. Photons are not always an ideal carrier of information, hence hybrid quantum acousto-optic systems have been proposed, which use surface phonons as the information carrier within the device [1]. A quantum dot (QD) can be used as a transducer between the acoustic and optical signals on the quantum level.

In this work we theoretically study the spectrum of the low-excitation resonance fluorescence (RF) for a single two-level QD, whose transition energy depends on time due to two simultaneous factors: First, the QD is modulated by a surface acoustic wave (SAW). Second, it is affected by environmental noise.

We iteratively solve the equations of motion for the density matrix of a two-level system driven by a weak, monochromatic, classical laser light. Then, the two-time correlation function is found using the quantum regression theorem. The time-integrated RF spectrum is obtained in the standard way as its Fourier transform, including a finite instrumental resolution. Thus, we extend our previous analysis of the noise effects in RF [2] to acoustically modulated systems [3,4].

The noise-free RF spectrum shows a series of unbroadened peaks from elastic scattering, shifted by a multiple of acoustic frequencies from the laser frequency [3,4]. Whitenoise environmental fluctuations leads to the appearance of additional peaks from inelastic scattering. They are located relative to the transition frequency and their width depends on the exciton decay rate and phase diffusion coefficient. We show that the environmental noise modifes the relative intensities of the spectral lines in the spectrum of a periodically modulated system. We study also the acoustic frequency mixing effect [4] and show that noise affects the quantum interference between various contributions to the scattering process, decreasing the contrast of spectral oscillations as a function of the relative phase of the two acoustic waves of different frequencies.

We consider also the case of telegraph noise environmental fluctuations, which differs from the white-noise case by the structure of spectral lines.

Our study presents a comprehensive characterization of noise effects on the resonance fluorescence spectra of acoustically modulated systems and may be useful both for interpreting experiments as well as for determining the limits of quantum acousto-optic transduction.

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Energy gap determination from photoacoustic spectroscopy measurements

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Photoacoustic spectroscopy (PAS) is a powerful technique to investigate the basic physical properties of semiconducting materials, e.g., thermal diffusivity, heat capacity, electron-trapping sites, energy gap, and so on. Moreover, PAS is a method that allows obtaining an optical absorption based on a sound wave emitted by a semiconducting sample. That sound wave is formed due to a periodic heat generation in the sample, which subsequently causes pressure changes in a surrounding gas detected by an acoustic transducer [1]. This method does not require any special sample preparation and allows examining materials even in powdered form, which is not possible for other absorption-like methods. However, so far, no in-depth studies comprehensively approach such issues as how sample size and thickness influence the photoacoustic signal. Here we address the questions mentioned above. All the experiments were carried out on molybdenum disulfide (MoS_2) since its properties are well-known, and it is easy to exfoliate MoS_2 crystal to obtain thinner samples.

In the first step, we investigated how the thickness of MoS_2 influenced the photoacoustic signal. Surprisingly when the sample is thinner, more signal is generated during PAS measurements (see Fig. 1a). Moreover, it is clearly visible in Fig. 1a that the energy gap determined from the knee method (see gray lines) is blue-shifted, going from the thick to thin samples. For the thickest MoS_2 crystals, we observed that the obtained energy is close to the indirect band gap at 1.2 eV, while for the thinnest MoS_2 , the energy is around the lowest direct optical transition (~1.8 eV). The second issue on which we focused is the sample size. In Fig. 1b, we showed photoacoustic amplitude for different sample sizes. All values of the PAS signal were taken at 1.9 eV (saturation area). It can be seen that signal slowly increases when reducing the sample size and then starts to drop quicker, following a linear trend as shown by a straight cyan line (inset of Fig. 1b). As mentioned, below certain sample size signal decreases, which is expected since the less light is absorbed and the signal from smaller volume is generated. In our studies, we will provide a tentative explanation for observed behavior.



Figure 1. (a) PAS signal for MoS_2 of different thicknesses. Inset shows PA amplitude taken from 1.9 eV (blue line), gray lines shows knee, and (b) PAS signal for a samples of various size. Inset shows PA amplitude against area of sample/light spot taken at 1.9 eV (blue line). The Cyan line in inset represents linear fit.

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Magnetic field and temperature dependence of the Mn²⁺ spin relaxation rate in a (Cd, Mn)Te/(Cd, Mg)Te quantum well

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The spin-lattice relaxation rate is an important parameter that determines whether a magnetic ion can be considered for information processing applications. More precisely, the dependence of the relaxation rate on parameters such as temperature and magnetic field can provide clues as to what type of mechanism is responsible for the energy dissipation.



Figure 1 The spin-lattice relaxation rate of the Mn^{2+} ions in CdTe quantum well vs temperature can be described with the power function. The relaxation rate dependence of the neutral exciton (X) is dominated by the direct mechanism. In contrast, the charged exciton (CX) dependency suggests multiphonon processes.

In this work we present the results of the spin-lattice relaxation rate measurements of a (Cd, Mn)Te/(Cd, Mg)Te quantum well. For this purpose we utilize the time-resolved optically detected magnetic resonance (ODMR) technique known for its sensitivity [1]. The spin-lattice relaxation time of Mn²⁺ ions in the (Cd, Mn)Te/(Cd,Mg)Te quantum well was measured as a function of magnetic field and temperature (see Fig. 1).

Obtained results can be explained in the frame of a modified model described in [2]. The modified model takes into account that energy levels of manganese are not equally spaced. The distance between the Mn²⁺ energy levels depends on the magnetic field and strain in the sample. The present observed temperature and magnetic field dependencies are well described by proposed model. Our results suggest that the dominant mechanism of energy dissipation in the measured system is the so-called 'direct process', in which a phonon is emitted or absorbed.

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III-V-on-Silicon-Nitride Mode-Locked Lasers

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In this work, we highlight our recent results on heterogeneously integrated III-V-on-siliconnitride (Si3N4) modelocked lasers [1, 2]. In particular, we demonstrate a III-V-on-Si3N4 mode-locked laser with a sub-GHz repetition rate and unprecedented noise performance [1] as well as energetic pulsed lasers. In contrast to earlier reported III-V-on-Si mode-locked lasers [3], silicon nitride waveguides can routinely achieve ultra-low losses close to 1 dB/m and do not suffer from nonlinear absorption, paving the way for a new generation of improved onchip mode-locked lasers

The low-noise comb laser employs a ring cavity and fabry-perot geometry, consisting of long silicon nitride spirals on top of a patterned silicon-on-insulator wafer, as shown in Fig. 1. Here, the 330 nm thick silicon nitride waveguides are defined using deep-UV lithography, permitting low-cost and high-volume wafer-scale manufacturing. We enable heterogeneous integration, by defining a recess locally defined in the 4.2 μ m SiO2 top cladding using dry etching techniques. Furthermore, given the large refractive index difference between the silicon nitride waveguide and the III-V gain waveguide, an intermediate silicon taper is introduced to ensure efficient evanescent coupling.

This process relies on the kinetically controlled adhesion of an elastomeric stamp to pick devices from the source InP wafer and print them on the silicon nitride target chip. In contrast to bonding techniques, microtransfer printing allows for integrating a III/V coupon in a recess. Moreover, this approach supports massively parallel integration, enabling wafer-scale fabrication. After transfer printing, the coupon is post-processed to isolate a saturable absorber. Furthermore, vias are etched to access the n-InP layer and electrical contacts are added to enable biasing of the device.



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Recovering rectangular band profile in polar InGaN-based heterostructures utilizing junction field

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III-nitride heterostructures are most commonly obtained along polar and semipolar directions. As a consequence built-in polarization, both spontaneous and piezoelectric, has to be taken into account when simulating and designing nitride devices.

We studied single InGaN quantum well (QW) heterostructures closely surrounded by p- and n-type layers with different doping levels. Band profiles of two exemplary structures simulated using 1D DDCC simulation software developed by Yuh-Renn Wu et. al [1] are shown in Fig 1.



Figure 1: Conduction and valance band profile for In_{0.17}Ga_{0.83}N/In_{0.03}Ga_{0.97}N QW surrounded by (a) no intentional doping and (b) $N_A = 6 \cdot 10^{19} \text{ atoms/cm}^3$, $N_D = 3 \cdot 10^{20} \text{ atoms/cm}^3$.

We used plasma-assisted molecular beam epitaxy to obtain very sharp doping profile to investigate optical transitions in Ga-polar QWs with varied total electric field. Change in doping concentrations within the cladding layers affect the depletion width what results in different value of the total electric field in the QW (FQW) (see Fig 1.). By reaching doping concentrations above 10^{20} $atoms/cm^3$ we were able to obtain p-n junction field high enough to completely abolish built-in electric field and recover rectangular-like band profile typical to nonpolar heterostructures (Fig 1(b)).

We analyzed influence FQW on

the photoluminescence (PL) spectra as a function of pumping power. As expected from simulations, for different doping levels we obtained substantially different shift of the emission peak. For structure without p-n junction we observed 9 nm shift, while for the highest used doping level the emission wavelength was independent of pumping power indicating total electric field in QW close to zero.

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Deep-level traps in as-grown and electron-irradiated n-GaN layers grown by MOVPE on Ammono-GaN substrate

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In this paper, the results of junction spectroscopy measurements on deep-level defects in Metal-Organic Vapor-Phase Epitaxy n-GaN samples grown on highly doped Ammono-GaN and subjected to 1.5 MeV electron irradiation are compared. It is found that in addition to the commonly observed deep-level traps in n-type GaN, such as E1 (0.25 eV) and E3 (0.59 eV), 1.5-MeV electron irradiation introduces two other electron traps, EE1 and EE2, with electronic levels at about 0.14 and 0.98 eV below the conduction band edge ($E_{\rm C}$), respectively. In the case of the EE1 level, a strong influence of the electric field (E) on the electron emission rate (e_{em}) is observed which suggests a donor type character of this trap level. Further, we have observed that strong electric field, as high as 2×10^5 V/cm, results in lowering the activation energy of electron emission from the EE1 level down to the value of 0.095 eV. The strong $e_{em}(E)$ dependence for the EE1 trap can explain the wide variation in electronic signatures of this trap reported in previously published publications. The analysis of the EE1 trap concentrations in the electron irradiated samples allowed us to estimate the average production rate of this trap by 1.5 MeV electrons as 0.125 cm⁻¹ for n-GaN material grown on Ammono-GaN substrate. Possible origins of the detected deep-level traps are discussed.



Fig. 1. On the left, the conventional DLTS spectra for both: reference sample (black curve) and 1.5 MeV electron-irradiated sample to dose of 7.4×10^{14} cm⁻² (blue curve). On the right, the dependencies of the rate of electron emission from the E1, E3, and EE1 traps versus electric field strength in the depletion regions of reverse-biased Schottky diodes.

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Bound states in continuum confined in GaN based subwavelength gratings

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Subwavelength gratings are photonic structures made of stripes of subwavelength thickness spaced periodically in one direction. They are a canonical example of an open optical system that can confine optical bound in the continuum (BIC) states. BIC arises from the symmetry of the system due to destructive interferences, resulting in the confinement of the mode. The quality (Q) factor of BIC states approaches infinity and they coexist with a continuous spectrum of unbound states. In real systems, due to finite dimensions, one observes socalled quasi-BIC, with a finite Q-factor. Use of gallium nitride (GaN) in photonics is highly advantageous due to its negligible absorption in visible spectrum and durability. Despite well established growth technology, nano-structuration still requires development.

We present two sets of GaN subwavelength gratings produced by e-beam lithography followed by dry-etching. One set was fabricated from MOVPE-grown GaN on sapphire substrate. The second one was etched in MBE-deposited GaN on porous GaN substrate made by electrochemical etching. Figure 1a) shows an example of the angle resolved reflectivity map of the grating on sapphire. Presence of vanishing reflectivity features at photon momentum k = 0 confirms the existence of quasi-BIC state. In addition, we detect a rarely observed interference based quasi-BIC for k at the border of the Brillouin zone. The Qfactors of quasi-BICs in both sets are determined to be above 10^4 and their linewidth goes below measurement resolution of 0.1 nm. Polarization resolved measurements reveal the presence of vortexes in reflected light, which further supports existence of quasi-BIC [1].

In summary, thanks to developed structuration technology we demonstrate modes of very high Q-factor in different types of GaN-based subwavelength gratings, as well as the existence of quasi-BIC states with non-zero wavevector values.



Figure 1: a) Angle resolved reflectivity measurements of GaN on sapphire subwavelength grating. BIC states are observed at k = 0 (blue circle) or $k \neq 0$ (green rectangle). b) Polarization vortex of a mode involving BIC at k = 0.

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Dynamics and Effective Interactions of a Photonic Condensate in an Optical Trap

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The bosonic nature of both polaritons and photons allows them to achieve Bose-Einstein condensation (BEC), forming a macroscopically coherent state. Polariton and photon condensates are intensively studied systems, where the nature of equilibrium and nonequilibrium BEC is explored. In high density regime, the strong coupling between excitons and photons is lost, and the system undergoes a crossover into the weak coupling, where one can observe a photonic condensate. On one hand, exciton polaritons interact with each other because of the excitonic component. On the other hand, interactions within the photonic condensate may originate from nonlinearities of the active material itself, e.g. via nonlinear refractive index, which has not been reported yet, especially using the geometry of an optical trap. These interactions are expected to manifest as collective Bogoliubov spectrum of excited states.

In this contribution, by using a GaAs-based microcavity sample, we experimentally study dynamics of photonic condensate in an optically induced trap and the effects of photon-photon interactions by measuring the momentum dispersion using the time-resolved spectral tomography on a streak camera. The sample is excited with the pumping pulsed laser shaped into a ring to obtain an effective potential created by the photo-excited reservoir. In the weak coupling regime, the effective potential results from local changes of the cavity's refractive index induced by excessive carriers, which act at the same time as the photon gain [1]. At strongest pumping powers - which correspond to the highest particle densities in the system - modification of the photon condensate dispersion is clearly visible, which can be a signature



of effective photon-photon interactions. The photonic condensate is drivennature, hence dissipative in in the experiment we observe shape a resembling the dissipative Bogoliubov spectrum, Figure 1 [2]. Our results are important in further understanding of the nature of photonic condensates as well as in their undiscovered dynamics.

Figure 1. (a) Measured dispersion of a photonic condensate at highest pumping power at time after pulse corresponding to the highest photon density. Theoretical photon and polariton curves are marked with Cav and LP. (b) Extracted energies from (a) (squares), with blueshifted theoretical cavity mode (black line) and fit with the dissipative Bogoliubov dispersion (red line).

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Advances and Prospects of Gallium Oxide Material and Device Technologies

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Gallium oxide (Ga₂O₃) is not a novel material and has a long history of more than 60 years. Especially, it has been well recognized as a native oxide formed on GaAs and GaN. However, as a semiconductor material, single-crystal Ga₂O₃ had been largely ignored by a great majority of researchers and engineers, even though it possesses attractive material properties such as an extremely large bandgap energy of 4.5 eV, a projected large critical electric field of over 7 MV/cm, and availability of large-diameter single-crystal wafers produced from melt-grown bulks. This situation changed in 2011, with our demonstration of first single-crystal Ga₂O₃ field-effect transistors (FETs) [1]. Since then, Ga₂O₃ has attracted much attention as an emerging semiconductor, especially due to high expectations for applications to future power switching and harsh environment electronics [2, 3]. Now, Ga₂O₃ has been widely recognized as a key material among ultrawide bandgap semiconductors, which are defined by bandgap energy exceeding those of SiC and GaN.

In this talk, after a brief introduction of material properties of Ga_2O_3 , state-of-the-art Ga_2O_3 material and device technologies will be introduced, based on world-wide recent research activities conducted by both our group and external research organizations. In addition, a brief outlook on Ga_2O_3 device applications will be given.

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Radiation Effects in Wide Bandgap Materials

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Wide bandgap semiconductors (WBS) enable electronic devices to operate at high power, high frequency and high temperature. Moreover, many WBS are considered more radiationresistant than conventional semiconductors such as Si or GaAs, making them interesting for applications in extreme radiation environments, such as space. However, radiation effects in WBS are usually very complex due to the diversity of the lattices of these compound crystals which often have nonequivalent lattice sites, mixed bonding schemes or different polymorphs. Moreover, the high mobility of point defects can promote the formation of extended defects.

Benefiting from the remarkable radiation resistance of GaN, we developed radiation sensors based on single GaN microwires [1]. Their potential for the detection of ionising radiation as well as their degradation upon high fluence ion irradiation will be discussed. Interestingly, radiation defects are not always a nuisance. Two examples will show how ion beams can be used to tune the electrical and optical properties of WBS. In MoO₃, the electrical conductivity can be controlled over several orders of magnitude by oxygen implantation [2]. In Ga_2O_3 , irradiation defects interact with chromium dopants, promoting their optical activation [3, 4]. Such defect engineering strategies will allow novel processing techniques and device designs.

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Dinesh Soares

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Magnetophonon resonances in graphene

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Resonant magnetoresistance measurements on monolayer graphene FETs under "ohmic" low current conditions have been used to study a wealth of novel physics. Recently, we observed magnetophonon resonance (MPR) in graphene, where peaks, periodic in inverse magnetic field, emerge in the magnetoresistance due to inelastic acoustic phonon scattering of electrons between Landau levels [1]. MPR provides a measurement of the transverse (TA) and longitudinal (LA) acoustic phonon velocities and gives insight into the nature of electron-phonon coupling.

At high currents (up to 100 Am⁻¹), electrons are driven far from equilibrium so that their kinetic energy can exceed the thermal energy of the phonons. We observe three non-equilibrium phenomena in monolayer graphene at high currents [2]: (i) a "Doppler-like" shift and splitting of the frequencies of the TA phonons, revealed by shifts of the MPR resonant peaks in the differential resistance, r_y (see Fig. 1a); (ii) an intra-LL Mach effect with the emission of TA phonons when the electrons approach supersonic speed (~1.4 x 10⁴ ms⁻¹), revealed by a broad magnetic field-independent peak in r_y at $I \sim 1$ mA (~ 70 Am⁻¹) and (iii) the onset of elastic inter-LL transitions and an associated resonance maximum in r_y at a critical carrier drift velocity (see Fig 1b). The third phenomenon is analogous to a "superfluid" Landau velocity and the formation of magneto-excitons involving a quantum jump, h/m, in the electron LL circulation. All three phenomena can be unified in a generic resonance equation. They offer avenues for research on out-of-equilibrium phenomena in other two-dimensional systems.



Fig. 1. (a) Plot of the dependence of differential resistance $r_y(I)$ on current for magnetic fields, *B*, between 0 and 2 T in 0.04 T intervals, measured at a carrier density $n = 3.16 \times 10^{12}$ cm⁻². Curves are offset by 1.5 Ω and the dashed lines highlight the shift of the positions of the magnetophonon resonance peaks. The green markers close to $I = \pm 1$ mA show the position of the *B*-independent peak in $r_y(I)$ when the electrons approach supersonic speed. (b) Plot of $r_y(I)$ for values of *B* between 0 to 1.2 T. Black arrows highlight the shifts of the resonant peaks in r_y due to elastic inter-LL transitions that arise when the electron velocity approaches the Landau velocity.

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Kardar-Parisi-Zhang universality in a one-dimensional polariton condensate

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Cavity polaritons are hybrid exciton-photon quasi-particles emerging from the strong coupling regime between photons confined in an optical cavity and excitons confined in quantum wells. They present physical properties reflecting their mixed nature. From the photon part, they inherit a small effective mass and can be confined in lattices with typical dimensions of the order of a few microns. Their excitonic part endows them with inter-particle interactions resulting in a giant Kerr non linearity. Cavity polaritons exhibit fascinating properties such as Bose Einstein condensation at elevated temperature, superfluidity, multistability... Importantly, the system is driven-dissipative in nature: the driving field maintains an out-of-equilibrium steady-state by compensating the constant leakage of photons.

Recent theoretical works have shown that the phase dynamics of out-of-equilibrium condensates obeys the celebrated Kardar Parisi Zhang (KPZ) equation [1-3]. In this talk, I will present our recent experimental observation of universal KPZ scaling laws in the first order coherence of a 1D polariton condensate [4]. I will first explain how we generated highly elongated polariton condensates in a 1D polariton lattice. I will then describe our measurements of the condensate coherence. Finally, I will show that data points lying within a well-defined spatio-temporal window collapse onto a single scaling function, characteristic of the KPZ universality class.

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Time-resolved nonlinear optical spectroscopy of basic excitations in 2D materials

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Since their discovery in 2010, the study of optically active 2D materials in the form of transition metal dichalcogenides, has grown to a distinguished area of research in solid state optics. In these materials tightly bound excitons dominate the optical response and lead to rich multi-particle physics based on the interplay with light, phonons, single charges, and other excitons. While microscopic descriptions, e.g., based on semiconductor Bloch equations, reveal many effects arising from the different coupling mechanisms, we have employed a handy mean-field extension of the optical Bloch equations. We take a local field coupling and excitation-induced dephasing into account, which account for exciton-exciton interactions [1]. The appeal of this approach lies in the fact that it allows to derive analytic expressions for optical signals even for complex nonlinear wave-mixing experiments in the limit of ultrafast pulses [2]. The local field model has proven to reproduce pump-probe [3], four-wave mixing [4, 5], and even six-wave mixing features [4] surprisingly well, despite its mathematical simplicity. In the first part of my talk I will explain the basic consequences of the local field model and present our recent results on four-wave mixing spectroscopy on gate-controlled MoSe₂ monolayers, which reveal the coherence properties and couplings of different exciton complexes when changing the electronic density in the system [5].

In the second part I will move to the van der Waals insulator hexagonal boron nitride (hBN), which is recently gaining significant attention because it can host bright colour centres operating as single photon emitters at room temperature. These quantum emitters also stand out due to their remarkably strong coupling to optical and acoustic phonons. In the second part of my presentation I will discuss our recent coherent control study of phonon-assisted excitation of single hBN colour centres [6]. Through this Ramsey–interferometry-like method we were able to determine the coherence times of bare and optical–phonon-assisted excitations and demonstrate the coherence loss induced by the emission of acoustic phonon wave packets.

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The study of resonant Raman scattering in semiconducting layered GeS

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Semiconducting layered group IV monochalcogenides such as black phosphorous and germanium sulfide with an anisotropic puckered crystalline structure in each layer have recently attracted much attention due to their unique optical and electronic properties and antic-



Fig. 1 False-color map of lowtemperature PLE/RRS spectra of GeS as a function of the excitation energy E_{exc} .

ipated applications in optoelectronics [1]. Resonant Raman scattering (RRS) is an efficient tool for studding of vibrational properties as well as electronphonon interactions. Here, we report on investigation of low temperature (T = 7 K), polarizationresolved, resonant Raman scattering experiments on GeS flakes in the range from 90 to 720 cm⁻¹. The RRS measurements are performed in back scattering configuration using a DCM dye laser with a tunable wavelength from 670 to 690 nm (Fig. 1). In order to determine the energy and optical polarization of the neutral exciton (X) complementary photoluminescence (PL) and reflectance (R) experiments are performed. In non-resonant RS spectra four Raman active modes A_{g}^{2} , A_{g}^{3} , A_{g}^{4} and B_{1g}^{2} are observed, however, when the excitation energy is tuned towards to the energy of the X we can resolve in the Raman spectra 18 peaks, among which 14 have not been reported previously in the back scattering configuration. Intensity of all the new Raman features

significantly increases as they are brought into resonance with the neutral exciton. Additionally, they exhibit almost the same polarization dependence as the X. Analysis of the origin and assignment of new phonon modes are based on their energies, polarization-dependent intensities, numerical calculations and comparison with previous experimental and theoretical studies of the Raman scattering in GeS and resonant Raman scattering in other materials. Accordingly, new features in the RRS spectra are attributed to infrared active and second-order phonon modes.

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Quantum anomalous Hall insulator in ionic Rashba lattice of correlated electrons

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We propose an exactly solvable two-dimensional lattice model of strongly correlated electrons that realizes a quantum anomalous Hall insulator with Chern number C = 1. We show that the interplay of ionic potential, Rashba spin-orbit coupling and Zeeman splitting leads to the appearance of quantum anomalous Hall effect. Moreover, we calculate in an exact manner Chern number for the correlated system where electron-electron interactions are introduced in the spirit of Hatsugai-Kohmoto model using two complementary methods, one relying on the properties of many-body groundstate and the other utilizing single-particle Green's function, and subsequently we determine stability regions. By leveraging the presence of inversion symmetry we find boundaries between topological and trivial phases on the analytical ground. Notably, we show that in the presence of correlations onset of topological phase is no longer signalled by a spectral gap closing. We provide a clear understanding of this inherently many-body feature by pinpointing that the lowest energy excited states in the correlated system are no longer of the single-particle nature and thus are not captured by a spectral function.

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Exciton Diffusion in an Ensemble of Self-Assembled Semiconductor Quantum Dots

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Electron-hole kinetics is at the heart of semiconductor physics. Coupled electronhole pairs, which are the fundamental excitations in a semiconductor, are able to move. In particular, exciton transfer occurs in self-assembled semiconductor quantum dot (QD) ensembles [1]. Despite the possible tunneling of carriers between QDs (Dexter-like transfer), the experiment [1] revealed enhanced transport in (two-dimensional) ensembles with a low spatial density of QDs where tunneling is inefficient. This suggests Förster resonance energy transfer (FRET) [2] as the transfer mechanism, that is, the photon-assisted process by which an exciton is annihilated in one dot and simultaneously created in another.

Therefore, we theoretically study exciton diffusion due to Förster transfer in an inhomogeneous ensemble of QDs randomly distributed on a circular mesa. QDs are assumed to be quantum emitters with one ground- and one excited state selected by the light polarization. The analytical formula of the Förster coupling is known for the general ensemble of quantum dipole emitters [3, 4]. It has the form of a sum of three power-law terms decreasing with distance, each multiplied by an oscillating factor. The inhomogeneity is given by a random distribution of the fundamental transition energy in each QD. In addition, a finite exciton lifetime is included. We derive and solve the equation of motion for the density matrix using the stochastic simulation method with a given exciton decay rate or by exact diagonalization for idealistic nondissipative systems.

Diffusion is described by time-dependent mean square displacement (MSD) and exciton density distrubution in space and time. Exciton diffuses in three subsequent time stages. First, there is a ballistic motion followed by a normal diffusion, which ends at saturation. The exciton density reveals a similar time dependence as MSD and a powerlaw localization in space. Furthermore, stochastic simulation allows us to trace the time dependence of the intensity of photoluminescene.

Finally, we present an approximate analytical solution, developed by us in [5, 6], which covers most of the effects predicted by the simulations, especially the triple-stage diffusion. The relatively small Förster coupling compared to the energy inhomogeneity of the ensemble validates the first-order approximation in which only direct transfer to remote QDs is important and allows one to express the MSD and exciton density analytically.

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Introducing the step Monte Carlo method for simulating dynamic properties

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The Monte Carlo (MC) approach is now an essential ingredient in many quantitative studies and branches of science including physics, finance-economics, biology, engineering, etc. In the field of condensed matter physics and materials science MC can be used to study, among many others, system of classical particles [1], classical spin systems [2], percolation



Fig. 1. Schematic illustration of the step Monte Carlo (sMC) method for the case of magnetization process (one Monte Carlo move). The magnetization direction \mathbf{m} (bottom panels) can take any point on the unit sphere with energy $E = f(\mathbf{m})$ (top panels). \mathbf{m}_{init} and \mathbf{m}_{final} are initial and final (trial) direction of magnetic moment with energy E_{init} and E_{final} respectively. In the sMC approach the probability of accepting the final random state \mathbf{m}_{final} depends on the activation energy $E_{AE} = E_{max} - E_{init}$, not on the relative energy $E_{final} - E_{init}$.

and fractals problems. The MC algorithm is the natural choice for studying the static properties of a system, where dynamical effects are not required. Then, the advantages of Monte Carlo are the relative ease of implementation and the rapid convergence to steady state.

In the modified MC approach proposed here (similarly to the kinetic MC approach) the probability of accepting the final state depends on the activation energy, not on the relative energy between the final and initial state [3]. However, the barrier height is calculated on an ongoing basis, by generating intermediate states with a predefined step Δ . Therefore, we name this method step Monte Carlo (sMC) [3]. Importantly, the sMC method correctly takes into account the presence of various local barriers and it obeys the detailed balance condition, even if the system is not in equilibrium. As a result, the appropriate dynamics of the tested system is simulated. The details of sMC algorithm are explained for the case of magnetization process. To test the correctness of sMC, we compare its results with those obtained by stochastic Landau-Lifshitz-Gilbert (sLLG) approach.

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Novel Te-bearing half-Heusler phases and their potential thermoelectric performance

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Theoretical predictions from first principles were conducted on nine half-Heusler tellurides $(M^{IV}M^{VIII}Te, where M^{IV} = Ti, Zr, Hf, and M^{VIII} = Fe, Ru, Os)$, each possessing 18 valence electrons. Among the compounds investigated, eight were found to be novel and mechanically stable according to the Born criteria for elastic constants. The MBJGGA approach predicted eight of the examined compounds to be semiconductors, with band gaps ranging from 0.395 eV (ZrOsTe) to 1.247 eV (ZrRuTe). Detailed analysis was conducted on the band gaps, spin-orbit splitoffs of heavy- and light-hole bands, elastic constants, and effective masses in the half-Heusler tellurides. Chemical trends were observed for the transition metal ions considered, with Fe-bearing compounds exhibiting the widest band gaps. The compounds ZrOsTe and HfRuTe are expected to have potential applications in thermoelectric devices due to Mahan's '10 k_B T rule'. The thermoelectric performance was discussed based on transport calculations, the deformation potential theory approximation for the relaxation time of carriers, and Slack's approximation for the lattice thermal conductivity. ZT (figure of merit) values as high as 2.76 and 4.14 at 1100 K were predicted for ZrOsTe and HfRuTe, respectively. This comprehensive investigation of half-Heusler tellurides presents interesting systems for further experimental research.

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Optimizing the cap thickness in fabrication of quantum dot photonic structures by focused ion beam

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In recent years, much attention has been paid to the fabrication and investigation of various photonic structures, such as micro-mesas, micro-lenses, micropillar cavities or photonic crystal slabs integrated with quantum dot emitters to optimize the performance in terms of non-classical light sources for quantum technology applications. There exists different fabrication methods of such structures. In this work, we use a focused xenon ion plasma beam (Xe-PFIB), which on one hand can suffer from a possible destructive influence on the crystal structure due to ion implantation, thus reducing the internal quantum efficiency of the source, i.e. intensity of photoluminescence from quantum dots.

Therefore, the goal of this study is to find the optimal range of cap layer thickness which assures efficient emission. The processes have been made on an exemplary QD material system of InGaAs/GaAs with the emission in the application-relevant range of the 2nd telecommunication window. By using wet chemical etching process, first we prepared samples with the cap thickness in the range from above 600 nm down to approx. to 50 nm. Next, a series of cylindrical micro-mesas of 4 μ m in diameter were fabricated using the Xe-PFIB. The ion beam energy of 30 keV and the beam current of 1nA were selected according to our preliminary tests. The fabricated pillar-like photonic structures in function of the cap thickness were then characterized by low-temperature micro-photoluminescence (μ -PL) measurements. Based on the single dot emission intensity dependence on the decreasing of the cap layer thickness, we found that the minimum cap thickness lies between 200 nm and 150 nm, for the given beam parameters and the chosen material system. Last, we considered further optimization steps, which relied on the carbon layer deposition to protect from the unwanted ion implantation, could potentially allow for the Xe-PFIB processing for even thinner cap layers.

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Facile Synthesis of Semiconductor Nanostructures with Variable Plasmonic Properties Using Pulsed Laser Ablation Technique

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Different semiconductor nanostructures such as silicon or carbon nanoparticles are promising nanomaterials in many various applications such linear or nonlinear bioimaging, optical nanothermometry, gas sensing, radio-frequency-induced hyperthermia etc. However, their simultaneous applications in different applications can be considerably limited due to the absence of specific features related to noble metal or magnetic metals. Thus, merging of semiconductor nanostructures with various metallic elements in the form of one nanoparticle by designing of multielement nanocomposites can significantly enlarge the application area of semiconductor nanomaterials.



Figure 1. Laser ablation time-dependent plasmonic properties of fluorescent carbon nanocomposites with tuneable plasmonic properties.

In this work, the synthesis of various composite semiconductor-based nanoparticles with variable both efficiency and spectral position of their plasmonic properties was carried out by pulsed laser ablation in liquids (PLALs) techniques providing a facile formation of multi-element nanostructures of high chemical purity. In particular, carbon and silicon nanoparticles with strong linear nonlinear photoluminescence, and respectively, used for were the development composite of nanostructures. performed laser-The induced structural modification of semiconductor nanomaterials provides the strong plasmonic properties whose

efficiency can easily be controlled by the laser irradiation time (Figure 1). The structural properties of the synthesized nanocomposites were investigated by different X-ray techniques such as X-ray diffraction (XRD) and X-ray photoelectron spectroscopy (XPS) as well as by electron paramagnetic resonance (EPR). Moreover, an impact of the chemical composition on the emission properties of fluorescence carbon nanodots was studied by the steady-state photoluminescence showing the modification of the fluorescence spectra of carbon nanodots upon the laser ablation time. The performed laser-induced change of the semiconductor nanostructure performance reveals new applications of silicon and carbon nanostructures in the field of nanoplasmonics related to the sensing of different molecules using surface-enhanced effects due to the presence of noble metal elements.

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Optical Properties of Silicon Nanowires-Based Composites

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Nanostructured silicon represent a great class of nanomaterials including various nanostructures such as porous silicon, silicon nanowires, silicon (oxy-)nitride, silicon carbide, silicon nanoparticles formed by pyrolysis, chemical etching or pulsed laser ablation techniques. Formation of low-dimensional materials reveal new unique structural and optical properties as compared to bulk silicon significantly depending on a type of a silicon wafer, used treatment procedure, nanostructuring conditions etc.





Figure 1. SEM image (on the top) and Raman shift (on the bottom) of Si NWs with and without Ag nanostructures at different laser power (488 nm).

One of the perspective candidates is silicon nanowires (Si NWs) formed by metal-assisted chemical etching of a bulk silicon wafer possessing highly aligned structure of silicon nanocrystals contrary to chaotic one in porous silicon films. It leads to more homogeneous and reproducible structural and optical properties of Si NWs that can be employed for photovoltaic, sensing or biomedical applications. Moreover, highly ordered porous structure of Si NWs makes them a nice matrix for introducing of different kinds of fluorophores, including other types of Sinanomaterials, in order based to develop multifunctional nanoplatforms.

In this research, Si NWs were formed using different time of chemical etching (from 15 s to 30 min) and their properties were investigated. In particular, Si NWs possess considerable Raman shift at 520 cm⁻¹ using various excitation wavelengths at the visible spectral range followed by some signal distortion with the increase of the laser power making Si NWs perspective for optical nanothermometry. Moreover, Si NWs also reveal some PL variable in the range of 600 - 700 nm depending on the used excitation wavelength (343 - 515 nm). Furthermore, introducing of

silicon or silicon carbide nanoparticles formed by the chemical etching leads to some spectral shift of the maximum position of Si NWs photoluminescence as well as its intensity.

Thus, optical properties of Si NWs can be varied not only by changing of the synthesis conditions but also by introducing of different silicon-based nanostructures in pores of Si NWs matrix forming silicon composite material.

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Non-equilibrium transport theory for dopant arrays in silicon

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The recent development of precise phosphorus donor placement in silicon [1,2] has attracted attention to the chains and arrays of such sites. Donor arrays have already proven well suited for quantum simulation of the Extended Hubbard [3] and SSH [4] models and are of major interest for future quantum-information devices.

In the experiment, systems of a few sites are gated and studied electrically in terms of so-called stability and Coulomb-blockade diagrams obtained by measuring the current while sweeping gate voltages. The theory aims then to simulate such diagrams and provide a two-way correspondence between the physical system and the simulated model. Thus, the need for a non-equilibrium transport theory of dopant systems arises.

Our approach based on combining exact diagonalization with non-equilibrium Green's functions allows for calculating both terminal and local currents in the arrays and other observables characterizing the system. Typically, by resorting to Green's functions, one loses the information on the underlying eigenstates. Our methodology lets us preserve this knowledge and back-trace states responsible for features of interest. This gives us a way to gain more information about the transport processes than can be obtained directly from transport measurements. Thus, we can, e.g., characterize the many-body configurations that contribute to the current, determine current magnitudes for different channels, and visualize how the electron flow circumvents disturbances in the form of array randomness. We study some of these effects to better understand transport studies of experimentally realized arrays [3].

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Elimination of Early Breakdowns in Fully-Depleted Detectors of Ionizing Radiation

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Reverse-biased P-N junctions are basic elements of many silicon detectors of electromagnetic radiation. They work under high reverse bias conditions (nominally 40 V). Therefore, a so-called early breakdown is a key parameter that determines whether given detector is good or bad. The term early breakdown denotes a rapid increase of the I-V characteristics of the reverse-biased P-N junction below an avalanche breakdown. Only detectors without early breakdowns can be candidates for real applications depending on the value of a so-called "dark" current. This work presents results of our technological experiments aimed at elimination of the early breakdowns in P-N junctions.

Circular diodes (diameter 4 mm) were fabricated on 4-inch high resistivity silicon wafers using different processing sequences. Next, electrical measurements of the junctions were done along the wafer diameter. The following technology variants were taken into consideration:

- devices after so-called "spray" n-type implantation (low dose and energy) of front side of the wafer;
- devices with different methods of doping of contact areas at the back side of the wafer: P implantation or diffusion from the POCl₃ source;
- devices with two thicknesses of poly-Si getter at the at the back side of the wafer: $1.1 \ \mu m$, $1.6 \ \mu m$;

Results of the experiment were as follows:

- "Spray" n-type implantation of front sides and back side contact implantation eliminated efficiently early breakdowns (Fig. 1). However, such doping methods can lead to degradation of "dark" currents. In such a case "dark currents" can be effectively improved using a poly-Si getter at the wafer back sides.
- Using the standard (thickness 1.1 μm) gettering process without implantation did not eliminate early breakdowns. However an increase of the getter thickness to 1.6 μm allowed to eliminate early breakdowns. Obviously, this method required a longer wafer processing.



Fig.1 Effect of type of ohmic contact at the back side: a) $POCL_3$, b) P-implant.; in both cases standard, 1.1 μ m-thick phosphorus doped poly-Si getter at the back side used.

Intervalley hybridization and optical properties of $Ge_{1-x}Sn_x$ quantum dots

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The $\text{Ge}_{1-x}\text{Sn}_x$ alloy attract much attention due to its band gap dependence on the composition. At some critical Sn content (of a few percent), the bulk changes its character from the indirect- to the direct band gap [1, 2]. The other properties of $\text{Ge}_{1-x}\text{Sn}_x$, like high carrier mobility, makes this material interesting from the point of view of applications for optoelectronic devices [3].

We present a theoretical, atomistic study of electron and optical properties of $\text{Ge}_{1-x}\text{Sn}_x$ nanostructures. Within the sp³d⁵s^{*} tight-binding model, we calculate the electron and hole states in $\text{Ge}_{1-x}\text{Sn}_x$ cubic boxes and colloidal quantum dots. We investigate the interplay of confinement effect and composition-induced indirect-direct band gap transition. We demonstrate an intervalley hybridization of the ground electron state and identify the relevant regimes of sizes and QDs compositions, where they are optically active. We also calculate the absorbance spectra of experimentally-relevant colloidal quantum dots [4].

The simulations are performed within the sp³d⁵s^{*} tight-binding model [5] combined with the valence force field model to account for strain in the system. We find the parameters via fitting the target Ge, α -Sn and zincblende GeSn band structures calculated using DFT approach with MBJLDA functional [2].

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Evaluation of Reaction Parameters in Polymeric Carbon Nitride Hybridized ZnO Nanocomposite for Improving Photocatalytic Activity using Thermal Synthesis

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Recently, semiconductor photocatalyst has garnered a lot of interest, as it has significant promise for environmental restoration and the generation of green hydrogen energy. The polymeric carbon nitride (PCN) and ZnO have been investigated in various catalytic reactions such as organic pollutant degradation and water splitting due to their relative inexpensiveness, moderate band gap (2.7 -3.0 eV), high thermal conductivity, and chemical stability [1]. However, the PCN is found to exhibit less applicability as a photocatalyst for water splitting due to the low surface area, high charge recombination probability, and weak visible light absorption, which is also dependent on the synthesis methodology. Since then, many efforts have been dedicated to the enhancement of the photocatalytic activity of PCN. Previously, in our group, we had successfully obtained PCN-AlOOH-ZrO2 nanocomposite, having a higher specific surface area of about 79.5 m²/g and a band gap of 3.0 eV [2]. We have also shown that the size of ZnO particles can be controlled with the water concentration and achieved the highest specific surface area (59.4 m²/g) [3].

Primarily, our goal is to present the development of nanomaterial synthesis to obtain the repeatable homogenous nanostructure, having a high specific surface area with narrowing the band gap which will enhance its photocatalytic activity for hydrogen evolution reaction. We have investigated the structural morphology and its stability through SEM, XRD, and ATR-FT-IR. This PCN nanocomposite was obtained by two steps: 1) microwave hydrothermal synthesis of ZnO followed by drying and 2) mixing the ZnO with different proportions of melamine and annealing in air at 550° C using a muffle furnace for 3hrs. The synthesized ZnO nanoparticle has an average grain size of 25 nm with BET_{SA} 45.483 m²/g and a density of $5.22\pm0.0.04$ g/cm³. We observed the strong XRD peak of PCN (at 28.33°) and 31.75°, 36.25° and 62.87° XRD peaks for ZnO hexagonal phase, which confirms the PCN-ZnO formation as shown in Figure 1, except 70% of ZnO. In near future, we will investigate other fundamental properties and photocatalytic activities with both experimentally and with density functional theory methodology.



Figure1: XRD diffraction pattern of PCN, ZnO and PCN-ZnO nanocomposite.

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Pressure-driven tuning of bandgap in methylammonium lead iodide

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Hybrid organic-inorganic perovskites of the composition of AMX₃, where A stands for an organic cation, M is a metal and X is a halide, are promising materials for a wide range of applications, including solar cells, light-emitting devices, photodetectors, and lasers due to low costs and simplicity of preparation, strong light absorption and long diffusion lengths facilitated by long carrier lifetimes and low nonradiative recombination rates.

Hydrostatic pressure can alter bond lengths and valence angles without changing chemical composition, tuning the electronic landscape responsible for basic properties of photovoltaic materials, like the energy bandgap. Moreover, combined with the density functional theory (DFT), it helps recognize structural factors affecting electronic band structure with atomic-level understanding.

In the following, the pressure-generated optical and structural changes of the low-temperature orthorhombic phase (OP) of methylammonium lead iodide were studied experimentally by photoluminescence (PL) and theoretically within DFT. The abnormal bandgap evolution of the OP under pressure with redshift-blueshift-redshift trend observed in both experiment (Figure 1) and theoretical predictions is explained by the competition between Pb-I bond shrinkage and PbI₆ octahedral tilting. Bandgap pressure coefficient within the OP determined from the redshift range of the PL energy is found to be negative and dependent on the temperature (α_{40K} = -36.3±0.1 meV/GPa, α_{80K} =-29.8±0.1 meV/GPa, α_{120K} =-13.3±0.1 meV/GPa). Negative deformation potential results from antibonding character of the bottom of the conduction band and the top of the valence band while dependence on the temperature is related to changes in Pb-I bond length as the atomic configuration approaches temperature-induced phase transition as well as increased transverse acoustic phonon contribution to octahedral tilting at higher temperatures.



Figure 1. Pressure response of the PL peak positions at (left) 40 K, (middle) 80 K, and (right) 120 K.

Nanoprobe-based electrical characterization of semiconductors nanostructures

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The miniaturisation of electronic components is running its natural course during the development of the modern electronic industry. Moving to the nanoscale requires specific methods of investigation since macroscale properties of semiconductor materials could not be directly extrapolated to the nanoscale. Here, we present a method of electrical characterisation on the nanoscale that overcomes the drawbacks of conventional characterisation techniques and helps shed light on the fundamental properties of nanoscale semiconductor structures and nanostructured materials.

To study the electrical properties of the nanostructures, we use two nanomanipulators mounted in the chamber of a multifunctional scanning electron microscope (SEM) that is additionally equipped with the focused ion beam (FIB) and the gas injection system (GIS). The FIB and GIS are used for ion beam milling and local deposition of Pt conductive layers with the aim of building more complex nanoscale structures without additional steps (transfer to other substrates, lithographic deposition of contacts, etc.). The manipulators in SEM allow us to measure the current-voltage characteristic of a single nanostructure with desirable morphology controlled by SEM imaging simultaneously with electrical characteriation. Moreover, the vacuum chamber reduces the impact of gas adsorption and light irradiation, which both affect the properties of structures under the test.

We demonstrate examples of the implementation of the nanoprobe-based approach for the electrical characterisation of ZnO/GaN single nanorod heterostructures [1,2], GaN/AlGaN nano-LEDs [3], patterned graphene/ZnO [4] and graphene/Ga₂O₃ interfaces, nanoscale ZnO nanorod/graphene Schottky diodes [5], and single nanowire metal/semiconductor/metal structures [6].

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Young's Modulus and Microhardness of (Pb,Cd)Te Solid Solution Grown by SSVG and MBE Method

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The PbTe, both alone or as a constituent of several systems (solid solutions, lowdimensional structures, composites etc.), attracts an increasing interest. The reasons are related to the new physical findings e.g., such as topological crystal insulators or dynamic local symmetry breaking and important applications, mostly in the harvesting energy area. The solids containing PbTe and CdTe, intensively investigated the last dozen years are one of the systems of today's active research. In particular, a significant hardening of the (Pb,Cd)Te solid solution with an increasing CdTe content could be an attractive property for selected applications of such crystals (see, e.g. [1]).

Recently, it was suggested that selected mechanical property, the microhardness, of PbTe bulk crystals and MBE-grown layers differ substantially [1]. On the other hand, analogous difference seems to be much smaller in case of GaAs [2] and not observed in (Cd,Hg)Te [3]. Therefore the question arises whether this effect exist also in (Pb,Cd)Te solid solution crystals.

All presently investigated samples were obtained at the Institute of Physics PAS. The bulk, single (Pb,Cd)Te solid solution crystals containing a few percent of CdTe were grown by the SSVG method. The samples oriented along three principal crystal directions by the X-ray diffraction were selected for further studies in order to check the microhardness anisotropy. In case of (001) orientation, sample with a natural surface was used, while the (011)- and (111)-oriented samples were etched in a bromine methanol solution. A few micrometers thick, (001)-oriented MBE layers with an analogous chemical composition were grown on GaAs substrates for the comparison. The samples characterized by XRD and AFM were investigated by the nanoindentation method at the University of Rzeszów. The applied load varied from 0.2 mN to 10 mN. The microhardness and Young's modulus, presently determined and compared with our previous results [1,2] and available literature data are given and discussed.

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Characterization of (Cd,Mn)Te and (Cd,Mn)(Te,Se) material and contact properties

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The basic parameters of materials used for semiconductor X-ray and gamma-ray detectors are carrier mobility (μ) and their lifetime (τ). Radiation generates hole-electron pairs in semiconductor detector material, an amount proportional to the energy of the incident radiation. Minimalization of carriers concentration results in increasing detector sensitivity. That leads to the creation of a semi-insulating material. The use of special material doping leads to material compensation and results in material resistances of the order of 10^{10} - $10^{11} \Omega$ cm. However, obtaining repeatable contacts for such highly resistive materials is very difficult. We have developed a method for obtaining nearly-ohmic contacts using amorphous layers based on CdTe, ZnTe, and (Cd,Mg)Te. The best I-V characteristics were obtained for materials with (Cd,Mg)Te:Sb contact layers and metallization. Properties of these contacts were investigated using scanning electron microscopy (SEM) and X-ray diffraction (XRD). We use current-voltage characteristics to examine the metal/amorphous layer/detector material interface.

For materials with resistivity of about 10^8 - $10^9 \Omega$ cm we made Schottky contacts. This type of contact is achieved by selecting appropriate contact materials that act as barriers for injecting electrons from one side and holes from the other side of the detector material. Creating a depletion layer in the sample by applying voltage leads to an increase in the signal produced by charge carriers generated by radiation.

In this work, we present such contacts made on (Cd,Mn)Te and (Cd,Mn)(Te,Se) crystals doped with V, In, or V and In, respectively. We measured I-V characteristics, Time of Flight (TOF), and photocurrent as a function of the applied field (PC-V) for selected samples [1]. Data were collected as a function of temperature in the range of 0 to 70°C. The measurements allowed us to obtain mobility values for electrons ($\approx 880-980 \text{ cm}^2 V^{-1} \text{s}^{-1}$) and holes ($\approx 30-50 \text{ cm}^2 V^{-1} \text{s}^{-1}$). From measurements of parameter $\mu \tau$ ($\mu_e \tau_e \approx 1 \cdot 10^{-3} \text{ cm}^2 V^{-1}$), we calculated an electron lifetime of the order of 10^{-6} s .

TOF measurements allow for the estimation of trapping and de-trapping processes. The I-V data enable us to determine barrier heights at the junction [2] and the activation energies of some states in the material's energy gap.

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Comparison of (Cd,Mn)Te and (Cd,Mn)(Te,Se) Compounds for Room Temperature X- and Gamma-Ray Detection: Optical Properties and Detector Response

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Semi-insulating CdTe-based compounds are being tested as materials for room temperature X- and gamma-ray detectors. More recently, selenium has been added to CdTe to increase the hardness of the compound as fewer harmful subgrains have been found in harder materials [1]. We have checked that compounds alloyed with Se are indeed harder. However, this application particularly requires good charge carrier transport, which is ensured by a low concentration of trapping defects.

In this work, two Bridgman-grown (Cd,Mn)Te-based materials were tested in order to compare their optical properties and the ability to detect X and gamma radiation. Our studies covered $Cd_{0.95}Mn_{0.05}Te$ and $Cd_{0.95}Mn_{0.05}Te_{0.98}Se_{0.02}$ crystals – as-grown samples as well as samples annealed in cadmium or selenium vapors. Low-temperature photoluminescence (PL) studies allowed us to discuss the presence of defect states. In both materials, two donor-acceptor pair transitions (DAP) exist. Shallow (s) and deep (d) DAP transitions are about 70 meV and 200 meV below the exciton lines, respectively. Bridgman-grown (Cd,Mn)Te has a high concentration of Cd vacancies, which are acceptors. Annealing in Cd vapors was carried out to reduce their concentration. This process eliminates or reduces the intensity of DAP^d line in (Cd,Mn)Te, whereas in (Cd,Mn)(Te,Se) even double annealing in Cd vapors does not influence the DAP^d line. In PL, the DAP^d line is observed in (Cd,Mn)(Te,Se) up to higher temperatures than in (Cd,Mn)Te. The possible interpretation of the differences between the two compounds might be the existence of a complex containing the Se vacancy in (Cd,Mn)(Te,Se). In principle, it should be a deep donor (analogy to Te vacancy) [2].

Then, the detector response of these two materials was compared. Pixelated detectors were created by the sputtering of metal layers. The room-temperature spectroscopic performance of the detectors was checked using a Co-57 point source. The (Cd,Mn)Te detector distinguishes 122 keV gamma-rays of Co-57 with an energy resolution of ~16%. On the other hand, the (Cd,Mn)(Te,Se) detector recognizes only X-rays of Co-57 at 7 keV with an energy resolution of ~45%, and a trace of gamma-rays at 14.4 keV. We associate the poor response of the (Cd,Mn)(Te,Se) detector with the presence of a deep trap involved in the luminescence of DAP^d. Therefore, (Cd,Mn)Te is a more promising material for use in X-ray and gamma-ray detectors.

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$Mn^{2+}-V_{Zn}$ charge transfer complexes in $Zn_{1-x}Mn_x$ Te

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 $Zn_{1-x}Mn_x$ Te dilute magnetic semiconductor (DMS) has recently attracted much attention due to the possibility of creation of the Mn-Mn ferromagnetic coupling [1] and fabrication of highly spin polarized red-green light emitting devices [2]. It is commonly believed that in $II_{1-x}Mn_x$ VI DMSs Mn^{2+} ions are electrically neutral. Recently, we have reported that in $Zn_{1-x}Mn_x$ Te doped with phosphorus (P) the Mn^{2+} ions strongly couple with P-acceptors giving rise to the compensation of the P-acceptors and the creation of Mn^{3+} ions [3].

In this communication, we report the results of the Hall effect, photoluminescence (PL), magneto-PL and electron paramagnetic resonance measurements performed on the undoped

ZnTe and Zn_{1-x}Mn_xTe crystals, which show a strong interaction between Mn²⁺ ions and singly negatively charged zinc vacancy defects V_{Zn}^{-} resulting in the formation of the Mn²⁺- V_{Zn}^{-} charge transfer complexes. The Mn²⁺- V_{Zn}^{-} charge transfer complexes. The Mn²⁺- V_{Zn}^{-} charge transfer complexes. The Mn²⁺- V_{Zn}^{-} coupling causes the transformation of the 0.047 eV $V_{Zn}^{-/0}$ shallow acceptor level, present in ZnTe, into a 0.138 eV $V_{Zn}^{--/-}$ deep acceptor level [see Fig. (a)]. This points to



the compensation of the $V_{Zn}^{-/0}$ shallow acceptor level by Mn^{2+} ions. We have suggested a charge transfer mechanism which transforms the $Mn^{2+}-V_{Zn}^{-}$ complexes into $Mn^{3+}-V_{Zn}^{--}$ ones,

$$Mn^{2+}(3d^{5},4s^{2}) + V_{Zn}^{-} \to Mn^{2+}(3d^{5},4s^{1}) + V_{Zn}^{--} \to Mn^{3+}(3d^{4},4s^{2}) + V_{Zn}^{--}.$$

The PL spectra measured under the 2.21 eV excitation for $Zn_{0.96}Mn_{0.04}$ Te sample exhibit two PL bands locating at 1.96 eV and 1.44 eV, which are attributed to the intrashell transitions of Mn^{2+} and Mn^{3+} ions, respectively.

The coexistence of the mixed valence Mn^{2+} and Mn^{3+} states leads to local ferromagnetic (FM) double exchange interaction between Mn^{2+} and Mn^{3+} ions by the Zener double exchange mechanism. The competition between the $Mn^{2+}-Mn^{3+}$ FM double exchange interaction and the intrinsic $Mn^{2+}-Mn^{2+}$ antiferromagnetic superexchange coupling gives rise to the paramagnetic–spin glass phase transition at $T_f = 3.2$ K in $Zn_{0.94}Mn_{0.06}$ Te sample [see Fig. (b)].

We have observed that the compensation of the acceptor impurities or cation vacancy defects by Mn^{2+} ions is the inherent property not only for $II_{1-x}Mn_xVI$ DMSs but also for $III_{1-x}Mn_xV$ alloys. This finding calls for a new strategy for the efficient *p*-type doping of the Mn-based DMSs.

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Anomalous behavior of the indirect excitons in (Cd,Mn)Te/(Cd,Mg)Te/CdTe double QW structures

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Interest in the study of double quantum well (DQW) semiconductor nanostructures is often related to the attempt to create a condensed state of so-called indirect excitons, which have a long exciton lifetime due to the localization of the exciton electron and hole in different QWs. This communication reports a new phenomenon in the photoluminescence (PL) of DQWs constructed from CdTe and CdMnTe QWs separated by a CdMgTe barrier. It has been found that the energy of indirect excitons (IX) in DQWs in the samples studied depends significantly on the energy of the excitation photon. To the best of our knowledge, such an effect has not been discussed previously.

Here we present the results of a study of exciton systems of CdMnTe/CdTe DQW structures consisting of a bound 15 nm width DMS QW, a non-magnetic CdTe QW with width varied in the range 6-10 nm, separated by a 2 nm width CdMgTe barrier. The stationary and time-resolved spectra of PL and PL excitation (PLE) were studied over a wide range of magnetic fields and optical excitation energies and intensities. Fig.1 shows magnetic field dependencies excitons energies at different powers of excitation (a), PL spectra of excitons from DQWs at different excitation energies in the magnetic field (b) and the emission energy dependence of indirect exciton (IX) and trion (T) from CdTe QWs on the excitation photon.



Fig.1 (a) Dependence of the energies of IX, DX from DMS QW and T energies on the magnetic field at different excitation powers; (b) PL spectra at different excitation powers at B=6.5 T; (c) Dependence of the energies of IX and T on the excitation power. The values of temperature, magnetic field and excitation power density are shown in the inset.

Anomalous non-monotonic behavior of the IX emission spectrum is observed near the energy of a (Cd,Mg)Te barrier band edge, while direct exciton from DMS WQ (DX) and trions (T) do not depend on the optical excitation energy. The energy difference between IX and T, i.e. the magnitude of the effect, is generally dependent on the optical excitation power density, but at low excitation power densities, this dependence disappears. We propose that the main mechanism responsible for this effect is due to the accumulation of photo-induced carriers in different QWs, which induces the appearance of an intrinsic electrostatic field that strongly modifies the energy spectrum of IX.
Properties of Infrared Detectors Made of Pb_{1-x}Mn_xTe/CdTe Multilayer Composite

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Infrared detectors are widely used for various purposes, such as chemical gas analysis, gas leak detection, IR imaging, remote temperature measurements, etc. In particular, there is continued interest in applications and consequently in the development of IR detectors operating at room temperature. We have shown experimentally that photoresistors made of the PbTe/CdTe multilayer composite are highly sensitive to infrared light, and their detectability at room temperatures is comparable to that of commercially available infrared detectors [1]. The high performance of PbTe/CdTe detectors is due to three factors. The first factor is a significant reduction in the concentration of free carriers in conductive PbTe layers due to their capture by broken bonds located at PbTe/CdTe interfaces. The atomic bonds at the interfaces are broken because of the different crystal structures of PbTe and CdTe, the rock salt and zinc blende, respectively. The second factor is a high mobility of carriers present in the PbTe layers. Despite the huge number of defects at the interfaces, the high mobility is preserved due to the high dielectric constant of PbTe, which effectively screens the scattering centers at the interfaces.

The main objective of the present work is to check how manganese atoms introduced into PbTe layers affect the performance of IR photoresistors made of Pb_{1-x}Mn_xTe/CdTe multilayer composite. It is known, that Mn effectively increases the energy gap of $Pb_{1-x}Mn_xTe$ [2], which leads to the increase of carrier effective masses and thus of the density of states. As a result, the concentration of carriers is expected to decrease with the increasing manganese content, x, which in turn should made the of Pb_{1-x}Mn_xTe/CdTe photoresistors more sensitive to optical excitation. Properties of Pb_{1-x}Mn_xTe/CdTe multilayer composite grown by molecular beam epitaxy on GaAs substrates have been studied. The study included morphological characterization by X-ray diffraction, scanning electron microscopy, secondary ion mass spectroscopy as well as transport and optical measurements. The main focus was on sensing properties of photoresistors made of Pb_{1-x}Mn_xTe/CdTe in the infrared spectral region. It was shown that the presence of Mn in Pb_{1-x}Mn_xTe conductive layers shifts the cut-off wavelength toward blue and weakens the spectral sensitivity of the photoresistors. The first effect is due to an increase in the energy gap Pb_{1-x}Mn_xTe with an increase in Mn concentration, and the second is due to a pronounced deterioration in the crystal quality of the multilayer by the presence of Mn atoms as shown by the morphological analysis.

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Terahertz Spectroscopy of Hyperbolic Metamaterial Based on $CdTe/Cd_{1-x}Mg_xTe$ Multiple Quantum Wells

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The tensor of the dielectric function of hyperbolic materials (HM) is defined by components which real parts are of opposite signs. Then, the surface of the constant frequency in the $\omega - k$ space is a hyperboloid which supports (theoretically) infinite number of wave vectors k for given value of ω . This drastically changes propagation of the electromagnetic wave, leading, e.g., to sub-diffraction imaging or negative refraction. Although some of naurally existing materials show such properties in a certain range of frequency (e.g., hexagonal Boron Nitride in the range 6.3 - 7.3 μ m) HM are usually artificially constructed with the help of advanced growth or processing technology. One of such structures, falling into the category of artificial hyperbolic metamaterial (HMM), is an alternating sequence of conducting and isolating layers showing hyperbolic properties for a limited range of wavelenghts λ of electromagnetic radiation (under the condition $\lambda \gg d$, where d is the period of the structure). Then, HMM can be treated as a medium characterized by effective constants $\epsilon_{\parallel}\epsilon_{\perp} < 0$, where \parallel and \perp are defined with respect to the plane of the HMM and in such a HMM the hyperbolic properties are exhibited for the Tranverse Magnetic modes of radiation.

Samples of molecular-beam-epitaxy-grown modulation-doped multiple quantum wells can be considered as appropriate for such studies because alternating doped barriers and quantum wells supplied with free electrons consitite the required sequence of isolating and conducting layers. In the present work, we studied response of such structures, grown in the form of a series of ten CdTe quantum wells, modulation doped in each barrier separating the wells. The total electron concentration estimated by transport measurements was equal to about 10^{12} cm⁻². Measurements of transmission of linearly polarized THz radiation through the samples were carried out at 4.2 K (the sample was cooled with an exchange helium gas) as a function of magnetic field up to 9 T and as a function of the angle between the plane of the sample and the magnetic field. Tilting the sample allowed to generate TM modes of the radiation while the magnetic field allowed to tune the effective dielectric tensor of the structure.

We observed a rich response of investigated samples showing an interplay of the tilt, magnetic field and frequncy of radiation. These factors conributed both to the shape as well as to position of spectral features, typically built aroung the magnetic field of the cyclotron resonance. By studying the THz photoresistance we showed that investigated samples could be used as sensitive detectors of THz radiation in magnetic fields below 1 T when Shubnikov - de Haas oscillations become a dominant feature of spectra.

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Near-band-gap Optical Properties of Modulation-doped $CdTe/Cd_{1-x}Mg_xTe$ Multiple Quantum Well Structures

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Modulation-doped quantum wells (QWs) constitute a system exhibiting properties which may essentially differ from properties of their undoped counterparts. The source of differences is the presence of free carriers in the quantum wells which strongly influence optical response of the system to electromagnetic radiation and, of course, also change the low-frequency conductivity. In general, doping with donors or acceptors leads to, respectively, n-type or p-type gas of charged carriers confined by a two-dimensional potential. In the present case we consider only n-type modulation-doped CdTe-based quantum well structures. At low temperatures, the near-band-gap optical properties of low-doped structures are dominated by excitation or recombination of neutral excitons and negatively charged excitons (trions). In the case of a single quantum well, by increasing doping level, which results in increasing the concentration of a two-dimensional electron gas (2DEG) in a QW, one observes a gradual change of the shape of the luminescence spectrum from that showing sharp lines resulting from recombination of excitions and trions to a broad structure defined rather by a Fermi-edge singularity. Application of magnetic field, perpendicular to the plane of the 2DEG further modifies spectra. In the case of undoped QWs or a low density 2DEG, one observes splitting of excitons or trions levels and optical transitions between split levels according to the relevant polarization selection rules. When the concentration of free electrons is high enough, transitions between Landau levels in the conduction and the valence band dominate the spectra.

In the present work, we studied of photoluminescence and reflectivity of modulationdoped $CdTe/Cd_{1-x}Mg_xTe$ quantum well structures grown by a molecular beam epitaxy. Each of four samples studied comprised ten QWs. Within each sample, parametrs of QWs and the doping of each of them were nominally identical. The samples differed by the doping concentration and the width of the spacer which resulted in differences in the concentration of a two-dimesional electron gas (2DEG). The samples were studied at liquid-helium temperatures and in magnetic fields up to 9 T. A near-band-gap (at about 1.6 eV) photoluminescence was excited with a green laser light while the reflectivity (a halogen lamp was used) was observed in a broader spectral range between 1.6 and 2.5 eV, clearly showing also the spectrum of higher-energy transitions, absent in the luminescence spectra. The measurements allow us to discuss the evolution of the optical response (in the near-band-gap energies) of modulation-doped CdTe-based multiple quantum well structures as a function of 2DEG concentration which seem to be the first result for this kind of II-VI quantum structures.

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Terahertz Emission from CdTe-based Multiple Quantum Wells

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Terahertz radiation is a natural candidate to be a game changer in the future electronics and communication systems. The roads towards creation of 6G and 7G systems begun to be explored and it is only the matter of time before full-scale THz systems become a reality. One of the directions leading to these advanced ways of exploring the THz part of the electromagnetic spectrum is defined by the need to construct compact detector emitter systems that should fulfill challenging requirements of a low-energy consumption, tunability, high spectral resolution and sensitivity. Despite a tremendous progress in THz science and technology over the last two dacades, there are still many open problems to be solved. In particular, new solutions of low-energy consumption emission – detection semiconductor systems are required to fulfill market demands. While the detection part of future systems is well developed, the efficiency of semiconductor emitters is low. That is why, research that can lead to proposing new types of emitter designs is so imprtant.

We have recently shown that low-photon-energy emission, with a maximum at about 1.2 meV was obtained from electrically pumped modulation-doped single GaAs/AlGaAs heterostructure at zero magnetic field. In the present paper, we address the problem of THz emission using electrically pumped semiconductor quantum system at low temperatures and in magnetic fields. The systems under study are multiple CdTe quantum wells modulation-doped with Ioidne. The structures were grown by molecular beam epitaxy on semi-insulating GaAs substrates. Each sample contained ten quantum wells that were 20 nm wide. A series of samples was grown with different doping levels and different width of barriers separating the wells. The presence of free electrons with a total concentration of about 10^{12} cm⁻² was verified by the cyclotron resonance and magnetotransport (classical and quantum) studies. For emission experiments, the samples were placed at the magnetic field and biased with pulses of the electric power. The emitted radiation was coupled to a Michelson interferometer with a Golay cell as a detector. We show that the emission of THz radiation appears only at a certain magnetic field strongly suggesting that the mechanism responsible for the emission is so called Landau emission, a process of a radiative recombination of electrons from upper to lower Landau levels which can be thought of as an inverse of cyclotron resonance. The magnetic field at which the emission appears is around 1 T, and the spectrum is peaked at energy close to the cyclotron resonance of electrons with the effective mass of CdTe $(0.102m_0)$. The present study allows us to determine the design of the multiple CdTe-based quantum structure (doping, spacer and barrier width) to maximize the power of emitted radiation.

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Simulation of Optical Properties of Terahertz Hyperbolic Metamaterials Based on Multiple Quantum Wells

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One of types of hyperbolic metamaterials is a stack of alternating dielectric and metallic layers (thickness d_d , dielectric function $\epsilon_d(\omega)$ and d_m , $\epsilon_m(\omega)$, respectively), where $\omega \sim 1/\lambda$ is the frequency of the electromagnetic wave. Within an effective medium approach, i.e., when the wavelenght $\lambda \gg (d_d + d_m)$ the material behaves as an optically uniaxial medium with the effective dielectric constants $\epsilon_{\parallel} = p\epsilon_m + (1-p)\epsilon_d$ and $\epsilon_{\perp} = \epsilon_m \epsilon_d / [p\epsilon_d + (1-p)\epsilon_m]$, where ϵ_{\parallel} and ϵ_{\perp} describe the dielectric polarization of the effective medium material for the electric field with zero and non-zero component along the optical axis (which is perpendicular to the surface of the layers) and $p = d_m / (d_m + d_d)$.

Considering the electric field of the wave propagating in this medium, one gets from the Maxwell's equations

$$\left(k_{\perp}^{2}+k_{\parallel}^{2}-\epsilon_{\parallel}\frac{\omega^{2}}{c^{2}}\right)\left(k_{\perp}^{2}\epsilon_{\perp}+k_{\parallel}^{2}\epsilon_{\parallel}-\epsilon_{\perp}\epsilon_{\parallel}\frac{\omega^{2}}{c^{2}}\right)=0$$
(1)

where c is the speed of light, and k_{\perp} and k_{\parallel} are components of the wave vector perpendicular and parallel to the layers, respectively.

By equating to zero the first and second term in Eq. 1 one gets equations describing a spherical and ellipsoidal isofrequency surfaces, if $\epsilon_{\perp}\epsilon_{\parallel} > 0$. These solutions correspond to the Transverse Electric (TE) and Transverse Magnetic (TM) modes, respectively. However, if $\epsilon_{\perp}\epsilon_{\parallel} < 0$, TM waves (these with a non-zero component of the electric field of the wave in the direction of the optical axis) are supported by a hyperboloidal isofrequency surface with allowing k-vectors to stretch to infinity (in fact, to $2\pi/a$, where a is the period of the lattice).

The aim of the present paper is to numerically simulate THz optical properties of structures in which conductive and dielectric layers are CdTe quantum wells containing electrons and $Cd_{1-x}Mg_x$ Te barriers, respectively, situated on a SI-GaAs substrate (thickness d_s). The essential difference between our study and the work of other groups is introduction of magnetic field which leads to an anisotropy of the conductive layers and introduces a cyclotron resonance. Numerical simulations are carried out according to a transfer matrix method for anisotropic system. The results are coefficients of energy transfer for both TE and TM modes and allow to determine the state of polarization of transmitted and reflected waves. Simulations are carried out as a function of geometrical parameters of structures (the number of quantum wells N, d_d , d_m , d_s), the electron concentration in the wells, the magnetic field, the frequency of the wave and the angle of incidence.

We applied the numerical procedure treating the system as composed of N periods of layers characterized by ϵ_m , ϵ_d on a substrate or as a system of the thickness $N(d_d + d_m)$ characterized by effective parameters ϵ_{\perp} , ϵ_{\parallel} . This allows us to establish the validity of the effective medium approach in the presence of magnetic field, which is a new result.

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Diffusional dependent structures on the crystal surface

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Different patterns can be created on the surface of growing crystals, among which the step bunches and/or step meanders are two of the most studied. The Ehrlich–Schwoebel effect at the surface steps is considered one of the "usual suspects" of such patterning. The combination of a direct and inverse step barrier and the proper selection of the potential of the well between them or changing the height of the direct step barrier leads to the growth of nanocolumns, nanowires, and nanopyramids or meanders, in the same system [1]. Based on our (2 + 1)D vicinal Cellular Automaton model [2,3] we show that not only the combination of step barriers is crucial in the formation of surface structures. In particular, we show that changes only in the diffusion process can lead to different patterns (Fig).



Fig. Structures obtained for initial concentration of particles $c_0=0.02$, Inverse Ehrlich-Schwoebel barrier with jump probability given by $P_{ies}=0.2$, direct Ehrlich-Schwoebel barrier with jump probability given by $P_{des}=0.4$, the energy of particle that stays at the bottom of the step $p_w=2.5$, length of terrace $l_0=2$ and a) number of diffusional steps $n_{DS}=10$ which leads to nanopillars b) $n_{DS}=40$ which leads to nanowires. System size 200 x 200.

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Fitting parameters of atomic orbitals to reproduce optical properties of III-V systems

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The multiband $\mathbf{k} \cdot \mathbf{p}$ models in the envelope function approximation are commonly used to describe electronic and optical properties of semiconductor nanostructures [1]. However, in some applications, like the modeling of fine structure splitting [2], the precise knowledge of the wavefunction at atomic scale is needed. In consequence, it is crucial to gather some atomic-scale information on Bloch functions calculated within the $\mathbf{k} \cdot \mathbf{p}$ approach. In particular, it would be beneficial to bridge between the empirical parameters and microscopic properties of the wave functions.

In our work, we model the Bloch states around the single atoms of the crystal. We study the commonly used III-V materials of the zinc-blende structure. The zone-center Bloch states are expressed as combinations of Slater-type atomic orbitals [3], where we take into account the contributions allowed by the irreducible representation related to a given band [4]. Then, we fit the orbital-dependent coefficients and exponential decays to reproduce the optical properties of the considered materials.

The utilized theoretical framework is similar to Ref. [3]. However, we aim to fit a larger set of interband momentum matrix element parameters, which are given in the 30-band $\mathbf{k} \cdot \mathbf{p}$ Hamiltonian [5].

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Excited Electronic States of Sr₂: *Ab Initio* Predictions and Experimental Observation of the $2^{1}\Sigma_{u}^{+}$ State

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Despite its apparently simple nature with four valence electrons, the strontium dimer constitutes a challenge for modern electronic structure theory. Here we focus on excited electronic states of Sr₂, which we investigate theoretically up to 25000 cm⁻¹ above the ground state, to guide and explain new spectroscopic measurements. In particular, we focus on potential energy curves for the $1^{1}\Sigma_{u}^{+}$, $2^{1}\Sigma_{u}^{+}$, $1^{1}\Pi_{u}$, $2^{1}\Pi_{u}$, and $1^{1}\Delta_{u}$ states computed using several variants of *ab initio* coupled-cluster and configuration-interaction methods to benchmark them. In addition, a new experimental study of the excited $2^{1}\Sigma_{u}^{+}$ state using polarisation labelling spectroscopy is presented, which extends knowledge of this state to high vibrational levels, where perturbation by higher electronic states is observed. The available experimental observations are compared with the theoretical predictions and help to assess the accuracy and limitations of employed theoretical models.



Figure 1: Potential energy curves for the ground and excited electronic states of Sr_2 obtained using the non-relativistic spin-free sMRCI+Q/5Z method with the scalar-relativistic small-core pseudopotential.

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Excellent excitonic properties of novel hexagonal MA₂Z₄ monolayers

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 MA_2Z_4 monolayers form a new class of hexagonal non-centrosymmetric materials hosting extraordinary spin-valley physics. While only two compounds ($MoSi_2N_4$ and WSi_2N_4) have been synthesized so far [1], theory predicts interesting (opto)electronic properties of a whole new family of such two-dimensional materials, as analogs of transition metal dichalcogenides [2]. We study within Density Functional Theory the electronic structure of selected MSi_2Z_4 (M = Mo, W; Z = N, P, As, Sb) monolayers. They exhibit direct band gaps and significant spin-orbit splittings of bands at K valleys. Effective Bethe–Salpeter-equationbased calculations reveal exciton binding energies up to 450 meV. The wave functions of electrons and holes are confined to the inner Z-M-Z layer. Evolution of excitonic energies under external magnetic field is predicted by providing their effective g-factors and diamagnetic coefficients, which can be directly compared to experimental values. In particular, large positive g-factors are predicted for excitons involving higher conduction bands. Additionally, a spin-orbit induced bands inversion is observed in the heaviest studied compound, WSi₂Sb₄, a hallmark of its topological nature.

In view of these predictions, MA_2Z_4 monolayers yield a new platform to study excitons and are attractive for optoelectronic devices, also as new building blocks of van der Waals heterostructures.



Figure 1. Top and side view of atomic structure of the studied MA₂Z₄ monolayers. Circularly polarized excitonic transitions at +K valley.

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Macroscopic Quantum Tunneling of a Topological Ferromagnet

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Since the early days of quantum mechanics, physicists have been trying to understand how the laws governing the microscopic world merge into those describing the classical mechanics at macroscopic size scales. A trademark quantum mechanical phenomenon that has no direct analog in classical physics is quantum tunnelling between different eigenstates of the system. This posed an intriguing question regarding the possibility of existence of quantum tunnelling in systems that can be regarded as macroscopic.

Only in the recent couple of decades the experimental techniques have reached a level of advancement that allowed to directly address that question experimentally. This was done in the context of macroscopic quantum tunnelling between different states in the Josephson junctions [1], as well as quantum tunnelling of magnetic domain walls [2], and magnetization [3] in magnetic systems.

Here, we investigate the electronic transport in a $(V,Bi,Sb)_2Te_3$ ferromagnetic topological insulator nanostructure in the quantum anomalous Hall regime. This provides access to the dynamics of an individual ferromagnetic domain. Telegraph noise resulting from the magnetization fluctuations of this domain is observed in the Hall signal. Careful analysis of the domain switching statistics provides evidence for macroscopic quantum tunneling of magnetization [4]. This ferromagnetic macrospin is not only the largest magnetic object in which quantum tunneling has been observed, but also the first observation of the effect in a topological state of matter.

Figure 1: (a)High magnification false color SEM image of the device together with a circuit diagram schematic. (b) Evolution of lifetime τ for each state with temperature, demonstrating a low temperature saturation caused by macroscopic quantum tunneling of magnetization [4].



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Anomalous transverse response in topological magnet CeAlSi

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A non-zero Berry curvature, which is a characteristic feature of topological materials, especially in the electronic structure, can be accessed through measurements of the anomalous Hall and Nernst effect [1]. A good candidate for such studies is CeAlSi, a ferromagnetic Weyl semimetal in which Weyl nodes can be tunable in the time reversal breaking regime [2]. We investigated the anomalous Hall conductivity for two different orientations of the magnetic field (B), namely σ_{yz}^A for $B \parallel a$, and σ_{xy}^A for $B \parallel c$ (a and c are crystallographic axes). In the low temperature limit σ_{xy}^A and σ_{yz}^A turn out to be of opposite sign, which was attributed to shifting of the Weyl point due to reconstruction of the band structure driven by spin reorientation. The anomalous contribution has been also detected in the Nernst conductivity (α_{xy}^A/T) measured with B oriented along c - axis. α_{xy}^A/T turns out to be large in the low temperature phase and slowly decreases at high temperatures. The ratio of σ_{xy}^A and α_{xy}^A is a sizable fraction of k_B/e at ~ 52 K indicating that the anomalous transverse response is fundamentally associated with the nontrivial band structure [3]. Using a single band toy-model assuming a non-zero Berry curvature in the vicinity of the Weyl node, we were able to recreate the temperature dependences of σ_{xy}^A and α_{xy}^A/T at high temperatures. In this region, large σ_{xy}^A and non-vanishing α_{xy}^A appear to be consequences of the fact that the Fermi level lies close to the band crossing point.

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ThO4

Superconductivity in PbTe/SnTe semiconductor heterostructure: a candidate for spin-triplet superconductor

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The superconductivity of IV-VI semiconductor heterostructures has been a puzzling phenomenon for many years [1,2]. The underlying mechanism seems to be based on the inversion of the band structure induced by compression due to periodic dislocation grids on the interfaces, which gives rise to topological crystalline insulator surface states [3]. At the same time, the resulting periodically varying strain acting on these states creates topological flat band, which promotes superconducting phase transition [4]. However, the pairing mechanism behind this superconductivity has not yet been fully demonstrated, to the best of our knowledge.

Here we present the results of soft point contact spectroscopy (PCS) of superconducting PbTe/SnTe heterostructures, in which the presence of dislocations was revealed using transmission electron microscopy. The PCS experiments exhibit a distinct zerobias conductance peak (ZBCP). It can be fitted within a theoretical model involving an Anderson-Brinkman-Morel electron pairing potential [5]. This superconducting mechanism realizes p-wave pairing symmetry and was initially introduced to explain the spintriplet phase of the superfluid ³He. Recent works have expanded this formalism to explain flatband-induced superconductivity [6].

In the PCS experiments, we used both silver and nickel contacts (normal metal and highly spin-polarized metal). The observed spectra do not depend on the spin polarization, which is a signature of p-wave pairing potential [5]. We will present a discussion that sheds light on the mechanism of the unconventional superconductivity observed in this class of narrow-gap semiconductor heterostructures.

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Peculiar magnetic phase in antiferromagnet MnTe

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Antiferromagnets get increasing attention nowadays as their magnetic state is coupled to its resistivity state and importantly, it may be influenced by electric current [1, 2]. A new puzzling chapter has opened with the developments of topology - in particular, some antiferromagnets have been predicted to exhibit exceptional properties, induced just by the crystal symmetry and not by relativistic effects [3], e.g. existence of anomalous Hall effect (even in a magnetically-compensated system). Recently, experimental evidence was made for RuO_2 [4] and thin MnTe films [5].

Hexagonal MnTe is a semiconductor with a moderate band gap (about 1.3 eV), room temperature resistivity of about a few Ω -cm and the Néel temperature T_N of 308 K. We performed structural, optical, transport and magnetic studies of state-of-the-art bulk samples in wide temperature and magnetic field range. We have traced the Hall voltage behaviour revealing nonlinearities in $\rho_{xy}(B)$ dependence. The observed data were fitted in multi-carrier model, providing two hole and one electron channel. The results will be referred to the recent band structure calculations [6]. Moreover, similarly to [5] we have observed a clear hysteresis loop in $\rho_{xy}(B)$ seen exclusively below T_N . However, the loop is flipped with respect to the one observed in epitaxial MnTe [5]. The presence of hysteresis in the Hall resistivity coincides with a weak ferromagnetic signal, resolved in SQUID magnetometry. A discussion about the origin of the observed phenomena will be provided.

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Photoluminescence studies of the futuristic SnSe semiconductor

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SnSe semiconductor belongs to the family of binary IV-VI chalcogenide compounds (Sn-X; X=Se,S), which owe the interest of researchers to the similarity of their layered crystal structure to molybdenum dichalcogenides - a prototype 2D systems. In particular, SnX_2 chalcogenides have been extensively studied as they exhibit high optical absorption coefficient and strong photo-responsivity in the near infrared spectral range allowing their wide application in a new generation of optoelectronic and photonic systems. Unlike rock salt lead monochalcogenides tin-based chalcogenides crystallize in hexagonal and monoclinic phases for SnX₂ or orthorhombic phase for SnX compounds. The layered orthorhombic structure (distorted rock salt structure) of SnSe leads to anisotropy and anharmonicity of chemical bonds responsible for extremely low thermal conductivity along the plane with zigzag ordering of ions and finally for the very high value of the thermoelectric figure of merit parameter ZT reaching 2.6 [1]. The excellent thermoelectric properties of monocrystalline SnSe have become a new inspiration for research on SnX materials. Therefore, tin chalcogenides appear as a family of semiconductors with the possibility of dual use in the field of photovoltaic and thermoelectric energy conversion, important from the point of view of environmental protection. In addition, the exciton binding energy and the energy gap of SnSe significantly depend on dimensionality of the structures (due to strong quantum confinement) and on the method of material preparation. For SnSe bulk samples an indirect bandgap of about 0.9 eV is expected, while for single-layer material the indirect and minimum direct gap differs slightly and amount to 1.63 eV and 1.66 eV, respectively [2]. Since SnX chalcogenides are generally less investigated than SnX₂ compounds, systematic study of their optical properties is needed. In particular, the results of luminescence (PL) studies of these compounds are rarely reported.

In this paper we discuss the results of PL measurements of monocrystalline SnSe synthesized in a direct reaction between the elements in a vacuum-sealed quartz ampoule at 950 °C. Orthorhombic structure of the resulting crystal was confirmed by XRD analysis. For these crystals excited with a 405 nm laser at cryogenic temperatures we observed several PL lines with energies of 0.9 eV, 0.94 eV and 1 eV, the position of which is weakly dependent on temperature. We assign the weakest line with the lowest energy to the indirect bandgap transition expected for bulk SnSe material. Higher energy lines, which are 5 to 10 times stronger, we associate with direct transitions in layered terraces-like structures (revealed on the surface of our SnSe crystals in SEM microscopic measurements), because the energy positions and amplitudes of these lines depend on the spatial position at the sample in which measurement has been taken. Our interpretation of the obtained PL spectra will be discussed in detail, taking into account the results of micro- and cathodo-luminescence measurements.

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Investigating of the topological phase transition in $Pb_{1-x}Sn_xTe$ topological crystalline insulators by using nonlinear Hall effect measurements

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The linear Hall conductivity occurs in the systems, which have broken time-reversal symmetry either by the intrinsic or external magnetic field. However, recently predicted nonlinear Hall effect (NLHE) preserves time-reversal symmetry but it breaks inversion symmetry [1]. NLHE appears from the anomalous velocity of the Bloch electrons due to Berry curvature. Therefore, topological crystalline insulators and the Weyl semimetals are the potential candidates for NLHE due to a Berry curvature dipole [1-3]. The surface of topological crystalline insulators hosts massless Dirac fermions protected by mirror symmetries [2]. At low temperatures, ferroelectric transition caused one of the mirror symmetries to be broken. In the case of thin film, the small mismatch of the lattice parameter can also break the mirror symmetry. The present study reports a NLHE in the $Pb_{1-x}Sn_xTe$ topological crystalline insulator thin films with varying Sn composition. The thin film samples of $Pb_{1-x}Sn_xTe$ with compositions x=0.35 and x=0.42 were grown by molecular beam epitaxy (MBE) on (100) oriented CdTe (4 μ m) // GaAs substrates. The nonlinear Hall signal appeared only when the AC excitation was applied along the mirror axis i.e. [110] in our case. The measurements of a nonlinear Hall voltage follows the same geometry as used in ordinary Hall effect, but the transverse voltage is measured at double-frequency, $V_2\omega$ and zero-frequency (DC) V_0 by using lock-in amplifiers. Both voltages quadratically depend on the perpendicular driving current and decrease when reaching transition from topological to trivial band ordering with rising temperature. The observed phenomenon opens the possibility of exploring topological phase transition as a function of temperature, composition and hydrostatic pressure in topological crystalline insulators.

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Electrical Properties of Anisotype ZnO:Al/ZnSe/p-CdTe Heterostructures

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Solar cells based on CdTe, fabricated in the form of ITO/CdS/CdTe heterostructures, achieved an efficiency of more than 22% when the buffer layer from the CdS compound was replaced by a wide-gap MgZnO solid solution. The wider band gap of zinc selenide (2.7 eV), higher transparency, and a small difference in electron affinities (0.19 eV) contribute to the use of ZnSe as an alternative to CdS buffer layer in these solar cells to increase their efficiency [1].

The investigated ZnO:Al/ZnSe/*p*-CdTe heterostructures were produced by sequentially deposition of ZnSe and ZnO:Al thin films by high-frequency magnetron sputtering on the surface of freshly cleaved plates from the crystalline parts of a CdTe ingot. Cadmium telluride crystals were grown by the vertical Bridgman method with a low pressure of cadmium vapor in an ampoule to form acceptor defects of cadmium vacancies (V_{Cd} , V_{Cd} ⁻²), which determine the hole conductivity of the semiconductor. The kinetic parameters of *p*-CdTe at *T*= 295 K were: specific electrical conductivity $\sigma = 3 \cdot 10^{-3} \Omega^{-1} \cdot \text{cm}^{-1}$, hole concentration and mobility $p= 3.5 \cdot 10^{14} \text{ cm}^{-3}$ and $\mu_{\rm H} = 56.0 \text{ cm}^2 \cdot \text{V}^{-1} \cdot \text{s}^{-1}$, respectively. The problem of making ohmic contacts to *p*-CdTe was solved by creating a p^+ -region in the near-surface layer of the semiconductor when irradiated with ruby laser pulses (wavelength $l = 0.694 \, \mu\text{m}$, absorption coefficient $a = 6 \cdot 10^4 \text{ cm}^{-1}$).

The rectifying properties of ZnO:Al/ZnSe/*p*-CdTe heterojunctions are determined by a potential barrier of ~1.0 eV formed in the near-contact region of *p*-CdTe. Current rectification coefficient: ~10². The *I-V*-characteristics of the structures are well explained within the framework of the theory of currents limited by the space charge (which is caused by the formation of an inverse layer in the near-surface region of *p*-CdTe), both in the region of small forward biases (3kT/q < V < 0.45 V) and in the region reverse biases (-0.6 V < *V* < -3*kT*/*q*). As the forward voltage increases, the main mechanism of current transfer is electron tunneling through a thin potential barrier from the conduction band of the film to the conduction band of *p*-CdTe. In the region of higher reverse biases (*V* < -0.6 V), the current is formed by the tunneling of electrons from deep energy levels $\Delta E = 0.5$ eV located in the band gap of cadmium telluride in the region of the inversion layer.

The *C-V*-characteristics of ZnO:Al/ZnSe/*p*-CdTe heterojunctions in the region of reverse biases at high frequencies (f = 600 - 1000 kHz) correspond to sharp surface-barrier structures. The dependence of the capacitance on the frequency of the alternating signal observed at low frequencies (f = 10 - 100 kHz) is explained by the influence of the charge of deep surface levels at the ZnSe/*p*-CdTe interface.

The constructed energy diagram of the ZnO:Al/ZnSe/*p*-CdTe heterostructure (taking into account the energy parameters of the materials) well explains the obtained experimental results.

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Current Transfer Mechanisms in CuMoO₄/*n*-Si Heterostructure

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Delafossite (CuMoO₄), together with other minerals of its group, is known for a wide range of electrical properties. Its conductivity can vary from insulating to metallic materials. CuMoO₄ nanocomposites can be used for the manufacture of electroluminescent devices as a photocatalyst. This compound also refers to transparent conducting oxides (TCO) – these are semiconductors that combine transparency to visible light and high electrical conductivity in one material. The development of *p*-type TCO will give us opportunities that cannot be used only with *n*-type materials, such as transparent *p*-*n* heterojunction, diodes, transistors.

Thin films of CuMoO₄ (thickness ~ 100 nm) were obtained by RF magnetron sputtering on glass substrates (for optical studies) and on plane-parallel *n*-Si plates (for obtaining heterostructures). A stoichiometric mixture of CuO and MoO₂ was used to manufacture the target. The temperature of the substrate $T_S = 613$ K, sputtering was carried out for 30 min, with a magnetron power of 180 W.

To determine optical coefficients, a method based on independent measurement of transmission and reflection coefficients was used. For the obtained films, transmission coefficient ~ 30 - 40% in the visible region. The presence of a straight section near the region of the own absorption edge on the dependences $(\alpha hv)^2 = f(hv)$ confirms the fact that the process of absorption of light photons takes place with the help of direct optical transitions. For the studied films, by extrapolation, the optical band gap $E_g^{op} = 4.1$ eV was determined.

On the basis of the *I-V*-characteristics obtained at different temperatures (Fig. 1) by extrapolation of linear sections of forward branches, the value of the height of the potential barrier was estimated and its temperature dependence was constructed (Fig. 1, inset), from which the temperature coefficient of change in the height of the potential barrier was determined and its values at 0 K, which are equal to $d(q\varphi_k)/dT = -1.9 \cdot 10^{-3} \text{ eV/K}$ and $q\varphi_k(0 \text{ K}) = 0.75 \text{ eV}$, respectively.

Fig. 1 shows that the *p*-CuMoO₄/*n*-Si heterostructure has rectification properties, the rectification ratio $RR \approx 120$ for |V| = 0.4 V and T = 294 K. According to the slope of the temperature dependence $ln(R_s) = f(10^3/T)$, the activation energy of charge carriers is determined, which is $E_a = 0.25$ eV.

At forward biases of 3kT/q V < V < 0.2 V, the generation-recombination mechanism of current transfer prevails in the *p*-CuMoO₄/*n*-Si structure. The reverse current at



Fig. 1. *I-V*-characteristics of the *p*-CuMoO₄/*n*-Si heterostructure at different temperatures (inset: $q\varphi_k = f(T)$)

biases -2 V < V < -3kT/q V is determined by tunneling processes involving surface states.

The *p*-CuMoO₄/*n*-Si heterostructure is photosensitive under reverse bias under AM1.5 radiation conditions.

Kinetics of Fe Island Films degradation in Air

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Iron island films are a promising material for nanotechnologies and spintronic. However, due to their high reactivity, they quickly oxidize in air and can only be used under conditions of ultrahigh (< 10^{-9} Torr) vacuum. The aim of this work is to study the degradation kinetics of Fe island films in air and the effect of this process on such properties as resistivity ρ , optical transmission T, and thickness.



Figure 1 shows the growth kinetics ρ of the Fe film depending on the time τ of its exposure to air. The initial film thickness is 20 nm, which, according to [1], corresponds to island morphology and the tunneling mechanism of charge transfer between islands. It can be seen from this figure that due to air oxidation, the resistance ρ of the Fe island film rapidly increases in a power-law dependence on the exposure time: $\rho = a\tau^{\kappa}$, where $a = 7.5 \ 10^{-3} \Omega/h$, $\kappa = 0.08$. The growth constant κ of the oxide layer is much less than the value of 0.5 observed during reaction diffusion in continuous films

[2]. This may be due to the specific morphology of the island film, in which the empty spaces between the islands do not take part in the oxidation process. In addition, at room temperature of the experiment, the low-temperature Mott oxidation mechanism should also take place under the influence of electric field that arises in interface by sorbet oxygen atoms and Fe atoms. This mechanism has a lower rate of oxidation.

After 14 hours of exposure, ρ of the Fe film begins to correspond to Fe₃O₄. And then it increases at a slower rate, approaching Fe₂O₃.

Figure 2 shows the spectral dependence of the optical transmission $T(\lambda)$ of the Fe island film before (1) and after (2) exposure to air for 48 hours.

As can be seen from this figure, the process of Fe oxidation is accompanied by film bleaching, which manifests itself to the greatest extent in the long-wavelength region of the spectrum. It is obvious that the transparency of the film is associated with a decrease in its reflection coefficient as the composition changes from metal (Fe) to semimetal (Fe₃O₄) and then to semiconductor (Fe₂O₃).

The thickness of the Fe film in the process of



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oxidation increases from 20 nm to 40 nm in 48 hours of exposure to air. The increase in thickness caused by the incorporation of oxygen atoms should change the configuration of the islands and additionally affect the change in ρ and T shown in Figures 1 and 2.

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Isolation of responsive elements of coordinate *p-i-n* photodiodes by *p*⁺-layer

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An urgent task of modern optoelectronics and photoelectronics is the development of systems for detecting the coordinates of various objects. Coordinate multi-element or matrix *p-i-n* photodiodes (PD) are most often used in such systems. The coordinate PD is usually a two- or four-element (sometimes more) photodiode on one semiconductor plate, and the responsive elements (RE) of photodiodes are separated by gaps smaller than the size of the light probe [1]. An important task in the manufacture of multi-element PDs is to ensure proper insulation of the REs among themselves, since the formation of inversion layers at the Si-SiO₂ interface can significantly deteriorate the insulation resistance of the active elements. There are various methods of isolation of REs. A classic option in planar technology is insulation with a base material and a dielectric gap on the surface, most often SiO₂ (Fig. 1a). This method provides insulation resistance between elements R_{con} =1-15 M Ω . But with an increase in the level of positive charge in the oxide, a sharp deterioration of this parameter is possible. A common method of isolation is the formation of a mesastructure, where the dielectric is air $(R_{con}=6.7-20 \text{ M}\Omega)$ (Fig. 1b). But this method does not exclude the formation of inversion channels at the Si-SiO₂ interface. An effective method of isolation of the REs is the formation in the gaps between the active elements of areas of restriction of channels of leakage of charge carriers isotypic with the material of the substrate (in this case, p^+ -type, R_{con} =18-50 MΩ) (Fig. 1c). In the case of *p-i-n* PD, this is embodied by the introduction of additional thermal operations - diffusion of boron into the gaps between the REs and oxidation, as well as one photolithography, which significantly increases the cost of the products.



Fig. 1. Image of crystals of PDs: a) classic PD; b) PD with mesastructure; c) PD with p^+ - area between REs.

We proposed to first carry out boron diffusion in the entire front surface of the substrate with a low concentration, and then the sequence of operations will correspond to the classical route [1]. Boron diffusion in this case was carried out at a temperature of 1073-1123 K for 1-2 min, the surface resistance after diffusion reached $R_S=150-200 \ \Omega/\Box$. Accordingly, the formation of p^+ -type areas was ensured by the introduction of one "low-temperature" thermal operation and without the introduction of additional photolithography. The insulation resistance between REs with such a crystal design reached $R_{con}=15-30 \ M\Omega$. It was noted that the introduction of a small concentration of boron significantly reduces the density of dislocations that are formed during the diffusion of phosphorus. This is caused by the compensation of mechanical stresses introduced by phosphorus atoms, which have a larger diameter than silicon, since the diameter of boron atoms is lower than in silicon.

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Damping of open circuit photovoltage in a Si homojunction

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The paper continues the study of interaction of free electrons and holes with defect states in the illuminated photo junction [1]. The laser beam of selected energy photons was used to measure open circuit voltage V_{oc} versus illumination intensity dependence. Obtained results are compared with the proposed theoretical model [2,3].

In general, photo junctions are build of two different semiconducting materials with different crystal structure, electronic band structure as well as different minority and majority carriers. Free and trapped electrons and holes interact in the junction. In such a case the junction has different electrons and holes concentrations, n_{10} and p_{10} on side 1, and n_{20} and p_{20} on side 2. Under illumination in steady state conditions they change to new concentrations of n_{11} , p_{11} and n_{21} , p_{21} . The achieved steady state conditions will be approximately (Maxwell-Boltzmann approximation) described by the energy shifts corresponding to four values of quasi Fermi $F_{2n1} = kTln(n_{21}/n_{20})$ levels $F_{1n1} = kTln(n_{11}/n_{10}),$ $F_{1p1} = kTln(p_{11}/p_{10}),$ energies: and $F_{2p1}=kTln(p_{21}/p_{20})$ different from the common energy of the thermal equilibrium Fermi level F=0. The created difference of chemical potential between the sides of the junction are equal to $(F_{1n1}-F_{2n1})/e$ for electrons and $(F_{1p1}-F_{2p1})/e$ for holes. This difference of energies contribute



Fig. 1. A comparison of the measured open-circuit voltage with predictions of the presented model for a homo- and heterojunction.

to an open circuit voltage value $V_{oc1} = (F_{1n1} - F_{2n1})/e + (F_{1p1} - F_{2p1})/e.$

We test the described model by comparing predictions derived from it with the results of the open circuit voltage measurements performed for two systems: a heterojunction n-type ZnO/p-type Si and a homojunction made of n-type and ptype Si. In the former, the ZnO top layer (Eg = 3.37 eV) is transparent to the photons of the energy hv = 1.91eV, so electrons and holes are generated only in the Si layer (the absorption edge at about 1.1eV). In the latter, the laser beam radiation is absorbed in both parts of the junction. The model satisfactorily describes the circuit voltage measured open in the heterojunction (Fig. 1), while for homojunction it fails to describe the experimental results for the high intensity illumination. Likely, stronger

influence of the interface imperfections reducing the efficiency of the junction has to be taken into account if both sides of it are populated with electron-hole pairs.

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Epitaxial Hexagonal Boron Nitride – Growth, Properties and Applications

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Hexagonal boron nitride (h-BN) is a two-dimensional (2D) material and a member of the IIInitride family that has recently attracted great interest due to its versatile range of applications. One of the most prominent examples is the use of h-BN as a substrate for other 2D materials. The lack of dangling bonds, the atomically flat surface and the homogenous dielectric environment lead to a significant improvement of the electrical and optical properties of adjacent 2D materials and render h-BN the prototype 2D-insulator for van der Waals (vdW) heterostructures that consist of stacks of many different 2D materials.

However, most of the work on vdW heterostructures is so far based on exfoliated flakes of h-BN. It is clear that the bottleneck for industrial applications of 2D materials will be the possibility to fabricate high-quality large-area layers.

In the first part of my presentation I will address this issue and show results on the growth of epitaxial h-BN on sapphire by metalorganic vapour-phase epitaxy (MOVPE) [1-3], which is currently regarded as one of the most promising growth techniques. I will show that the growth of a 2D material on a conventional substrate at high temperatures followed by a cool down to room temperature will lead to wrinkle formation [1,4], which can be used to assess the quality of the grown layer.

In the second part, I will focus on the properties of h-BN in general and of our epitaxial h-BN in particular, with a detailed characterization and discussion of the properties.

The last part will be dedicated to applications of our MOVPE-grown h-BN ranging from large-area growth of vdW heterostructures [5], single photon emitting defects [6], photonic applications like the growth of Bragg mirrors [7] to hydrogen generation and storage applications [1].

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h-BN bubbles – a step towards deterministic activation of single photon emission

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Hexagonal boron nitride (h-BN) is a wide bandgap (~6 eV) two-dimensional semiconductor which can be easily integrated with other materials. Point defects hosted by h-BN were found to act as single photon emitters even at the room temperature [1]. Therefore, h-BN is regarded as a promising material in terms of quantum optoelectronics. However, it was shown that single photon emission needs to be activated e.g. by annealing or by bending the material [2]. We address this issue by the deterministic creation of h-BN bubbles. Via electron irradiation [3] of our MOVPE material grown on sapphire substrates [4], we initialize the radiolysis of interfacial water. This leads to the creation of stable hydrogen-filled h-BN bubbles (Fig. 1a).

Here, we discuss the optical properties of the obtained h-BN bubbles. To study the impact of created deformation on defect-related emission in different spectral ranges, we performed photoluminescence (PL) measurements using different excitation wavelengths. PL mapping measurements show that bubbles enhance the total light emission (Fig. 1b) and prove that they can activate additional spectral lines impossible to observe on flat h-BN (Fig. 1c), which are candidates for single-photon emitters. Since bubbles are non uniform, they create a inhomogeneous strain distribution that can shift the energy. Therefore they can also tune the energy of emitted light. The key point is that bubbles are created only in irradiated precisely selected locations. Therefore, h-BN bubbles seem to be a useful tool to deterministically locate and activate quantum emitters on demand.

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Fig. 1. PL measurements of one single h-BN bubble with 532 nm laser excitation at 4.2 K: a) Optical image of the h-BN bubble, b) total integrated intensity (2.06-2.30 eV), c) results for specific points marked at b).

Nature of UV (3 – 5 eV) Emission from Hexagonal Boron Nitride

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Boron nitride (BN) is a semiconductor with energy gap of about 6 eV. In spite of wide interest in BN properties, there are still many discrepancies in reports on its photoluminescence (PL) spectra. In the present work we aim to analyze nature of bright UV PL emission from defects in hexagonal BN (h-BN) by means of a joint experimental and theoretical study.

The experimental part was performed on the boron nitride layers grown on sapphire substrates using the Metal Organic Vapor Phase Epitaxy (MOVPE) [1]. The growth temperature, t_{gr} , for a series of samples was between 1050°C and 1300°C.

The photoluminescence was excited with the 215 nm (5.8 eV) light generated as fourth harmonic of the Ti:Sapphire laser line. Micro-PL spectroscopy was performed with the mirror objective. Time-resolved PL (TRPL) was measured with a UV-designed streak camera equipped with quartz optics.

In the PL a high-energy emission was observed at 230 nm (5.4 eV) and several bright lines were at range of 300 nm (4.1 eV) and 380 nm (3.2 eV). The C300 emission spectra consisted of a few sharp lines with nearly identical lifetimes (about 0.9 ns). The identical lifetimes suggest that these are phonon replicas what was confirmed by comparison with phonon energies. The lifetimes of the C380 set of lines were about 1.8 ns with a similar phonon replica structure. The lines were narrow (0.6 nm width) showing that the C300 and C380 emissions have a character of color centers. The lifetimes were nearly constant in temperature range from 6 to 300 K, what means that their PL efficiency was high. A strong dependence between the C300 and C380 intensity and $t_{\rm gr}$ has been observed. The color center lines vanished for $t_{\rm gr} > 1200^{\circ}$ C, and were replaced by broad bands (D330 and D400) centered at similar wavelengths.

Based on the theoretical results [2], we postulate that the 300-nm and 380-nm lines can be assigned to pairs of carbon atoms, which is expected in our samples due to the use of organic precursor (C_2H_5)₃B during the growth. In the theoretical calculations the defects of h-BN were simulated by the BN clusters modified with various carbon-related defects. Their electronic spectra were calculated by time-dependent density-functional theory. The introduced defects gave electronic excitations of lower energies, 4.8 eV and 3.7 eV for two simple configurations of C_2 (1,2- C_BC_N and 1,4- C_BC_N), what is in a good agreement with energies of the observed C300 and C380 defects.

The observed temperature dependence enables us to control number of the color centers. Such intentionally introduced centers are important for applications like deep UV emission or quantum information.

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Optically Detected Magnetic Resonance of p- and n-type Doped (Cd,Mn)Te QWs

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In this work, we combine magnetooptical measurements and optically detected magnetic resonance (ODMR) technique to study magnetic system composed of Mn^{2+} ions in (Cd,Mg)Te/(Cd, Mn)Te QWs with carrier gas.

The advantage of the ODMR technique is the possibility to study local properties of magnetic ions incorporated in well-defined position of nanostructure. The basic information extracted from the ODMR spectra is the energy level structure of the Mn^{2+} ion, which depends, e.g., on the local strain [1]. Although the ODMR technique in diluted magnetic semiconductors is sensitive selectively to the magnetic ions, the detailed analysis of the measured signal reveals interactions within the magnetic ion system or between ions and charge carriers [2].

The nominally undoped (Cd,Mn)Te/(Cd,Mg)Te quantum wells are typically p-type [3]. The hole gas originates from the background doping of the (Cd,Mg)Te barrier material or from the surface states. By covering the (Cd,Mn)Te/(Cd,Mg)Te QW structure with a nickel metallic layer, we produced a sample with different carrier gas properties. As we observe by magnetooptical measurements, the hole gas is replaced by electron gas in the QW. Additionally, the application of the voltage to the nickel gate gives us an opportunity to tune the electron density.

Depending on the conditions, we have observed that the ODMR signal is affected by the carriers present in the sample in two ways. The first effect is the shift between the ODMR signals obtained on neutral and charged exciton (Knight shift). The second one is a change of the spin-lattice relaxation (SLR) rate in the presence of the carriers.

Our results clearly highlight the asymmetry between electron and hole systems. The observed change of effective g-factor in the Knight shift is positive (shift towards lower magnetic fields) and much more pronounced for p-type than for n-type case. Interestingly, for both types of carrier gas we observe significant acceleration of the SLR.

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Magnetic signatures in structural, electronic and optical properties of 2D MPX₃ crystals

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Since recent years the magnetism in two-dimensional (2D) materials has emerged as one of rapidly growing research areas, with many exciting properties and potential applications in low-power spintronics, and optical communication [1]. The current study focuses on the transition metal phosphorous trichalcogenides semiconductors (MPX₃, M=Mn, Ni, Fe, X=S, Se), which are layered antiferromagnetic semiconductors. These materials are stable in air and exhibit facinating properties, such as recently predicted giant excitionic binding energies sensitive to the polarization of light [2].

In this report, we adress a following scientific questions: what mechanism sustains the long-range AFM ordering, and whether the type of magnetic arrangement can be manipulated? Here, we report a comprehensive theoretical *ab initio* results of the structural, electronic and optical properties of the series of MPX₃ monolayers (M=Mn, Ni, Fe, and X=S,Se). We also present the results for alloy systems with magnetic [3] and nonmagnetic substitution [4]. In particular, for the AFM-Neel magnetic ordering, the inclusion of the spinorbit interaction (SOI) causes an in-equivalency of the pair of valleys (K+,K-), resulting in sizeable valley splitting, which can be tuned by the rotation angle of the spins. In the case of MnPS₃, MnPSe₃ and FePS₃ monolayers, we have demonstrated that the band edge direct transitions are optically active and sensitive to the polarization of light. In addition, our results reveal an effective tuning of magnetic interactions and anisotropies in both MnPS₃ and NiPS₃ upon nonmagnetic substitution [4]. Finally, we highlight the importance of the structural anisotropy in monolayer of FesPS₃, resulting in local inversion symmetry breaking. leading to lifted spin degeneracy of K+/K- valleys, and two optically active transitions visible in experiments [5]. Such efficient engineering of the magnetism in MPX₃ materials provides a suitable platform to understand the magnetism in thin samples

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Magneto-spectroscopy of excitons in two-dimensional layered perovskites

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Over the past few years we have witnessed a booming interest in the field of twodimensional (2D) layered metal halide perovskite. The revival of this class of materials is driven in part by their increased environmental stability with respect to the 3D perovskites as well as their outstanding performances in photovoltaic and light-emitting devices [1]. The reduction of dimensionality allows material engineering to go beyond the mere control of the thickness of the inorganic quantum wells. In such a hybrid organic-inorganic materials, the plethora of available organic molecules not only allows to tailor various material functionalities, but also gives rise to a considerable structural diversity. Due to the high contrast of dielectric constant of organic/inorganic sublattices, the 2D perovskites are characterized by very large exciton binding energies with low effective Bohr radius. As a result of large binding energies, the exchange interaction between the electron and hole spins is greatly enhanced, which in turn increases considerably the exciton fine structure. On top of that, the ionic nature and the softness of the perovskite lattice results in a significant coupling of electronic excitations to the lattice vibration (i.e., electron-phononinteractions). Such coupling has a pronounced impact on the optoelectronic properties of 2D perovskites and is manifested in a complex absorption and emission spectra [2]. which usually consist of multiple, equally spaced (in energy scale) resonances [3]. As the result, the mutual understanding of electronic and lattice excitations and their correlation - polarons - is crucial to adequately describe this material system.

In this work, we highlight our recent results on the determination of excitonic and polaronic properties of various 2D layered perovskites. In order to access such parameters, spectroscopy in high magnetic field, up to 160 T was employed. By these experiments, we not only determine important material parameters such as effective mass, exciton binding energy and exciton Bohr radius [4], but we also address the longstanding open question about the exciton fine structure in 2D perovskites. We elucidate the order of excitonic states and determine the bright–dark splitting energy [5]. Finally, by correlating the magneto-spectroscopy data with resonant Raman scattering we explain the complex multi-peak absorption response of 2D layered perovskites as being dominated by exciton-polarons.

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Temperature Dependence of Effective Masses in Three-dimensional Methylammonium Lead Trihalides

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Perovskite methylammonium lead trihalides (CH₃NH₃PbX₃ with X=Cl, Br, I) have gained particular research interest as promising hybrid compounds with excellent sunlight-toenergy conversion abilities [1]. They constitute a particular semiconducting materials system, in some aspects very different from the well-known family of epitaxial semiconductors. The soft, ionic lattice which mixes crystalline, liquid and glassy behaviour is often invoked to explain the superior characteristic of perovskite materials in applications. At the same time particular mechanical properties of perovskites create a complex background for electronic excitation where polaronic effects cannot be neglected and often challenge our understanding of fundamental parameters describing any semiconductors, for example, carrier mobility or effective mass μ [2,3].

For instance, theoretical prediction points that the soft, ionic lattice together with the unusual temperature dependence of the bandgap characteristic for these materials can lead to a significant rise of the carriers' effective mass with temperature [4,5]. Unfortunately, the temperature-induced rise of the carrier's effective mass in $CH_3NH_3PbX_3$ has never been confirmed experimentally so far. Understanding temperature dependence of reduced effective mass μ is particularly relevant in the context of solar cell applications as it affects carrier mobility.

Here we address the problem of temperature evolution of carrier effective mass by means of magneto-optical studies as μ can be directly extracted from the quantization of the free carrier states into Landau levels. We investigate three crystals methylammonium lead iodide (MAPbI₃), bromide (MAPbBr₃) and their mixed bromide-chloride analogue (MAPbBr_{1.5}Cl_{1.5}) under various temperatures. Using an extreme pulsed magnetic field up to 90 T we observe well-resolved Landau levels in each of the crystals even at temperatures exceeding 100 K. The temperature evolution of the Landau levels clearly points to the rise of the carrier effective mass as a function of temperature corroborating theoretical predictions. Importantly the observed rise of carriers' effective mass cannot be simply attributed only to the bandgap opening with temperature highlighting the importance of polaronic effects for metal-halide perovskites semiconductors.

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Spin dimensionality in chromium trihalides

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Ferromagnetism in layered van der Waals materials that preserve up to the monolayer regime attracts the exceptional attention of scientists [1]. Intensively examined chromium trihalide compounds (CrX_3 , where X=I, Br, or Cl) are layered magnetic materials of ferromagnetic order within a layer. $CrBr_3$ layers are ferromagnetically coupled, while CrI_3 and $CrCl_3$ possess antiferromagnetic coupling. Furthermore, the spins in $CrBr_3$ and CrI_3 are preferentially aligned in the out-of-layer direction, whereas in $CrCl_3$ they are characterized by the in-plane orientation [2].

In this work, we study the spin dimensionality of CrX_3 compounds by applying the magnetic field in the out-of-plane (Faraday) and in-plane (Voigt) configurations in reference to the sample, see Figure (a). Magnetization was studied by the polarization-sensitive photoluminescence technique. Circularly polarized excitation and detection were used to evaluate circular dichroism in CrX_3 arising from unequal absorption coefficients for photons characterized by opposite circular polarization.

The inspection of the PL intensity as a function of applied out-of-plane magnetic field sweeping in both directions for CrI_3 and $CrBr_3$ revealed the hysteresis as seen in Fig. (b,c). Hysteresis indicates intrinsic ferromagnetic ordering. $CrCl_3$ not present hysteresis due to strong in-plane orientation of spins, see Fig. (d). Magneto-PL intensity in Voigt configuration reveals complex hysteresis with unusual shape, see Fig. (e)-(g). The comparison of coercive fields in both configurations of the magnetic field provides information about spin dimensionality, which is consistent with theory [2]. $CrBr_3$ can be described by an isotropic Heisenberg model with 3D spin dimensionality. CrI_3 has a greater out-ofplane anizotropy than $CrBr_3$, which suggests an 1D spin dimensionality described by an Ising model. $CrCl_3$ presents the 2D spin dimensionality described by an XY model.



Figure: (a) The schemes of Faraday and Voigt configurations of placement sample in magnetic field. The magneto-PL intensity in Faraday (b,c,d) and Voigt (e,f,g) configurations.

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